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Coupling moisture diffusion and internal mechanical states in composite materials

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Coupling moisture diffusion and internal mechanical states in composite materials

OBJECTIVE

Time-dependent environmental conditions
temperature and relative humidity

Coupled modelling:

- stress dependent moisture diffusion
- nonlinear hygrothermoelasticity

Decoupled modelling:

- Fourier and **Fick's laws**
- linear hygrothermoelasticity

"weak" coupling : plasticization

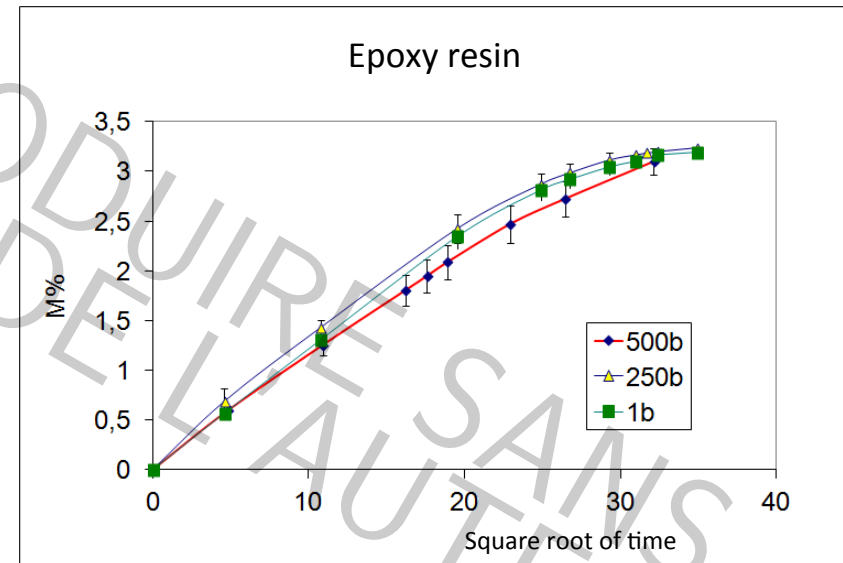
"strong" stress-diffusion coupling

Time-dependent moisture content and
hygrothermal stress at the micro and macro levels

Durability of polymer matrix composites

Coupling moisture diffusion and internal mechanical states in composite materials

Stress dependent moisture diffusion



hyperbaric testing tanks – IFREMER Brest



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The free volume theory

The thermodynamical approach

Multi-scale stresses estimations in composite structures submitted to environmental conditions

...and the biocomposites

DECOUPLED-DIFFUSION MODELS

Fick's law

$$\frac{\partial C}{\partial t} = D \frac{\partial^2 C}{\partial x^2}$$

- C , moisture content
- D , diffusion coefficient
- x , position
- t , time

Langmuir model

$$\frac{\partial n}{\partial t} = D \frac{\partial^2 n}{\partial x^2} - \frac{\partial N}{\partial t}$$

$$\frac{\partial N}{\partial t} = \gamma n - \beta N$$

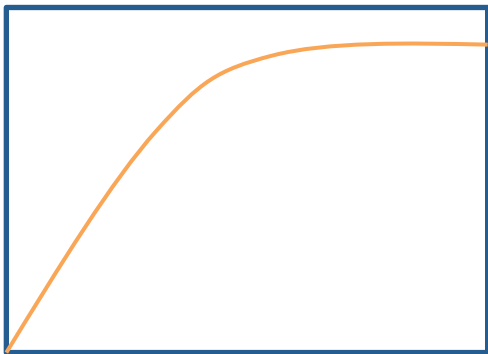
$$\gamma n_{\infty} = \beta N_{\infty}$$

- ✧ γ and β , probability per unit time that a mobile molecule becomes bound, and probability per unit time that a bound molecule becomes mobile.
- ✧ n and N , number of mobile molecules per unit volume, and number of bound molecules per unit volume.

DECOUPLED-DIFFUSION MODELS

Fick's law

$$\frac{\partial C}{\partial t} = D \frac{\partial^2 C}{\partial x^2}$$

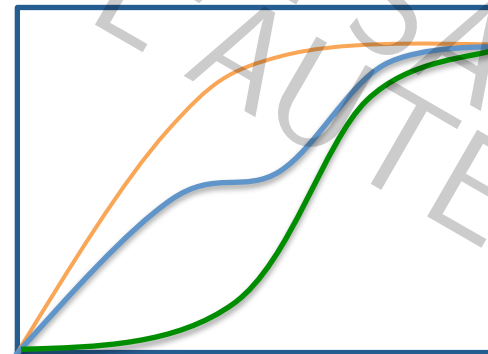


Langmuir model

$$\frac{\partial n}{\partial t} = D \frac{\partial^2 n}{\partial x^2} - \frac{\partial N}{\partial t}$$

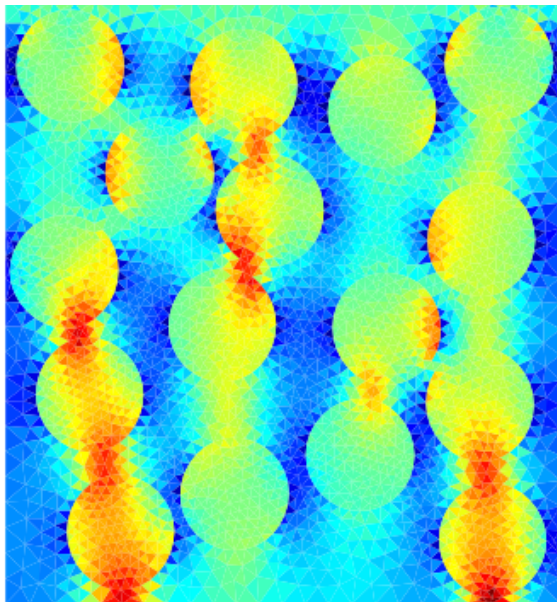
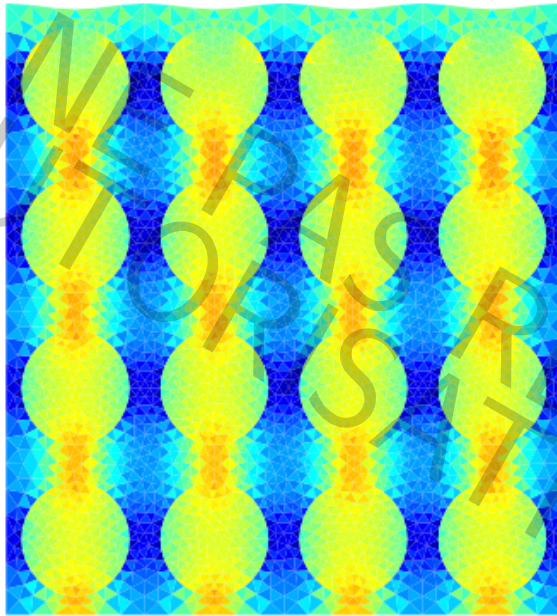
$$\frac{\partial N}{\partial t} = \gamma n - \beta N$$

$$\gamma n_{\infty} = \beta N_{\infty}$$

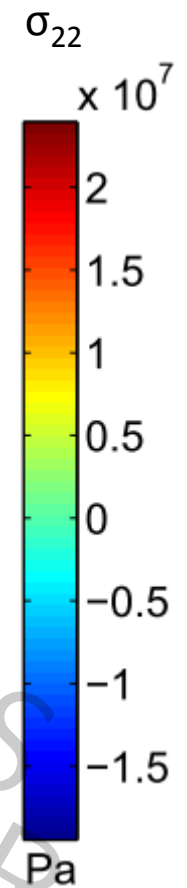
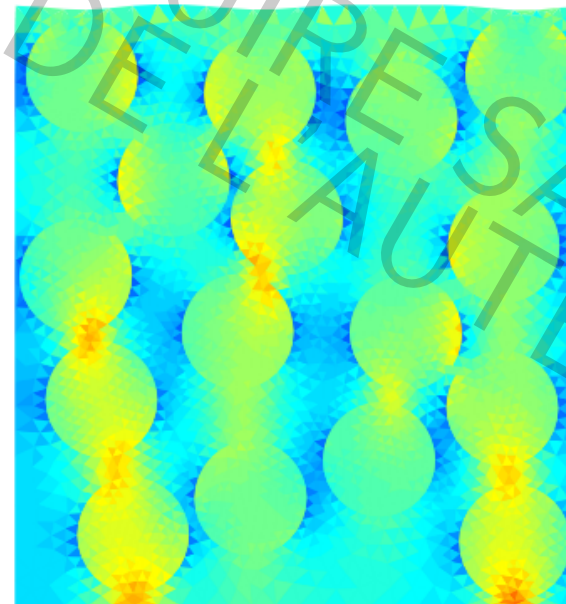
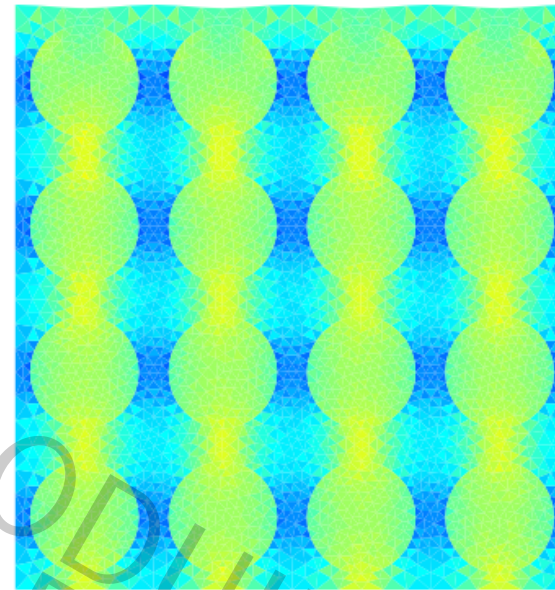


DECOUPLED-Modelling (linear hydroelasticity)

Fick's law



Langmuir model



The free volume theory

Moisture diffusion coefficient

Neumann et Marom¹ have developed a theoretical approach based on the calculation of the free-volume change in the stressed state:

$$\ln\left(\frac{D_{\varepsilon}^m}{D_0^m}\right) = \frac{a}{v^m} \left(\frac{1}{v_{f0}^m} - \frac{1}{v_{f\varepsilon}^m} \right)$$

The free-volume fraction for a strained epoxy is related to the corresponding value for the unstrained resin through:

$$v_{f\varepsilon}^m = v_{f0}^m + \frac{\Delta V^m}{V_0^m} = v_{f0}^m + \frac{V_{\varepsilon}^m - V_0^m}{V_0^m} = v_{f0}^m + \text{Tr } \varepsilon^m$$

The moisture diffusion coefficients of the strained/unstrained resins satisfy:

$$\ln\left(\frac{D_{\varepsilon}^m}{D_0^m}\right) = \frac{a}{v^m} \text{Tr } \varepsilon^m \left\{ v_{f0}^m \left[v_{f0}^m + \text{Tr } \varepsilon^m \right] \right\}^{-1}$$

¹ **Neumann, S., Marom, G.**, (1986). Free-volume dependent moisture diffusion under stress in composite materials. Journal of Materials Science, 21, 26-30.

The free volume theory

The general expression enabling to determine the moisture content C according to Fick's diffusion model writes as follows:

$$\frac{\partial C}{\partial t} = \text{div} \left(D \vec{\text{grad}} C \right) = D \text{div} \left(\vec{\text{grad}} C \right) + \vec{\text{grad}} D \vec{\text{grad}} C$$

Fick's law

Dependence of the diffusion coefficient upon the mechanical strains

$$\vec{\text{grad}} D = \vec{\text{grad}} \left(\text{Ln} \left(\frac{D_{\epsilon}^m}{D_0^m} \right) \right) = \frac{a}{v^m} \text{Tr} \epsilon^m \left\{ v_{f0}^m \left[v_{f0}^m + \text{Tr} \epsilon^m \right] \right\}^{-1}$$

The free volume theory

Maximum moisture absorption capacity

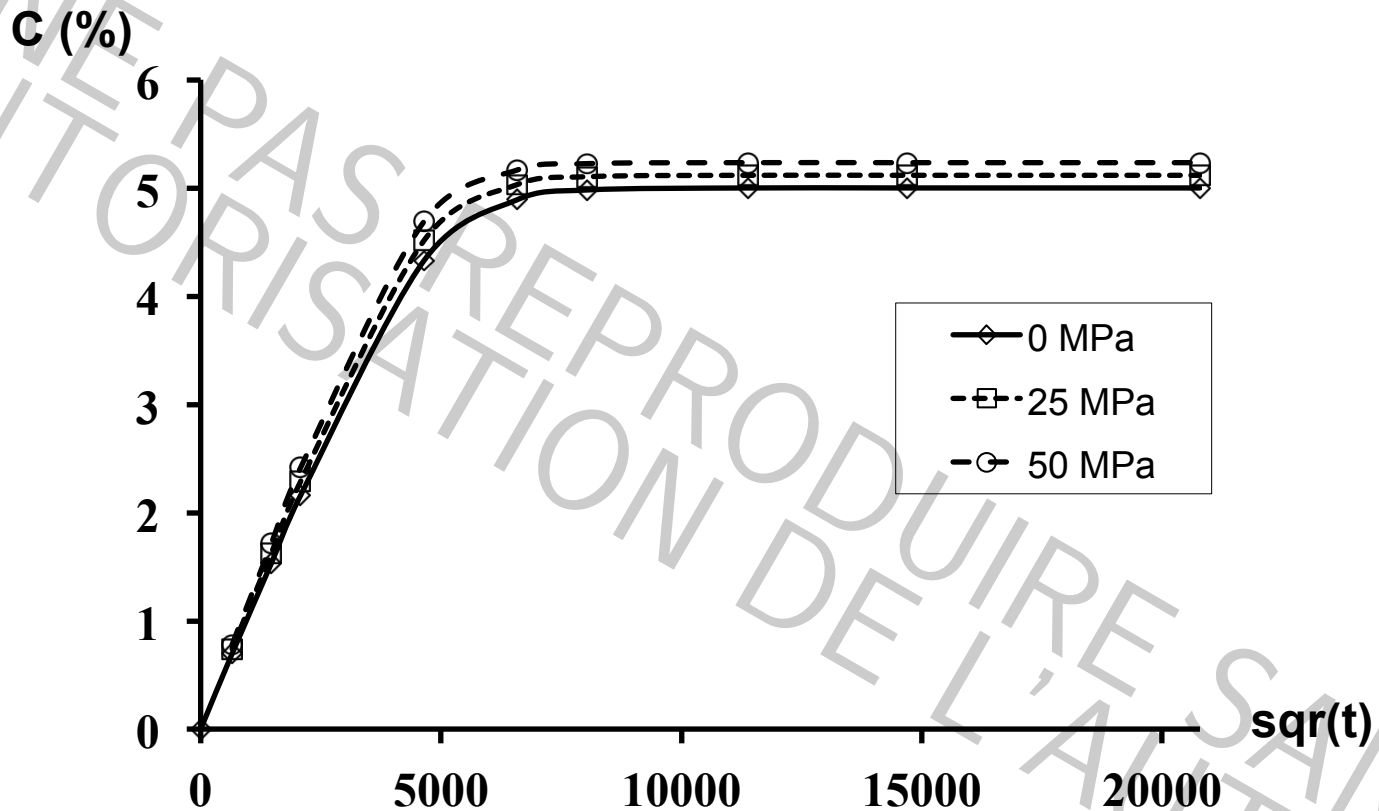
The maximum moisture absorption capacity is defined by:

$$M_{\infty 0}^m = v_{f0}^m \frac{\rho^w}{\rho^m} \quad \text{for the strain-free matrix} \quad M_{\infty \varepsilon}^m = v_{f\varepsilon}^m \frac{\rho^w}{\rho^m} \quad \text{for the strained matrix}$$

Thus, the equation of the maximum moisture absorption capacity becomes:

$$M_{\infty \varepsilon}^m = M_{\infty 0}^m + (v_{f\varepsilon}^m - v_{f0}^m) \frac{\rho^w}{\rho^m} = M_{\infty 0}^m + \text{Tr } \varepsilon^m \frac{\rho^w}{\rho^m}$$

The free volume theory



↗ UD tensile stress

→ ↗ Maximum moisture absorption capacity
→ ↗ Moisture diffusion coefficient



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The thermodynamical approach

❑ Chemical potential of water: $\tilde{\mu}_w(C) = \frac{\partial F}{\partial n_w} = \frac{\partial F}{\partial C} \frac{\partial C}{\partial n_w}$

❑ Free energy of Helmholtz: $F = F_0 + n f_w(C) + V_\varepsilon W$

❑ Hygro-elastic strain energy: $W = \frac{1}{2} \boldsymbol{\sigma} : \boldsymbol{\varepsilon}^{el} = \frac{k}{2} (\text{tr } \boldsymbol{\varepsilon} - 3 \eta C)^2 + G \boldsymbol{e} : \boldsymbol{e}$

where η , k , coefficient of moisture expansion, and bulk modulus.

❑ Variation of polymer density: $\frac{1}{\rho_p} = \frac{1}{\rho_0} (1 + \text{Tr} \boldsymbol{\varepsilon})$

where ρ_p , ρ_0 , density of polymer at present and initial states.

❑ Trace of strain tensor: $\text{Tr} \boldsymbol{\varepsilon} = -\frac{P_{ex} + P_{is}}{k} + 3\eta C$

The thermodynamical approach

- Chemical potential:

$$\tilde{\mu}_w(C, \text{tr } \epsilon) = \mu_0 + R T \ln \frac{C}{C_0} - \frac{3 \eta \omega_w k}{\rho_0} (\text{tr } \epsilon - 3 \eta C)(\text{tr } \epsilon + 1) + \frac{\eta \omega_w}{A_0 \rho_0} (3 A_0 k - \alpha)(\text{tr } \epsilon - 3 \eta C)(\text{tr } \epsilon + 1) + \frac{\eta \omega_w}{A_0 \rho_0} \frac{3 A_0 k - \alpha}{2} (\text{tr } \epsilon - 3 \eta C)^2$$

where C_0 is the reference moisture content.

- The moisture flux is related to the chemical potential:

$$J_i = - \frac{DC}{RT} \nabla \mu_w$$

where D is the diffusion coefficient in $[\text{mm}^2/\text{s}]$, C is the moisture content in $[\%]$, R is the ideal gas constant in $[\text{kJ}/(\text{mol.K})]$, T is the temperature in $[\text{K}]$.

- The law of mass conservation is given by: $\dot{C} + J_{i,i} = 0$

The thermodynamical approach

Moisture Flux + Mass Conservation Equation



$$\dot{C} = D \left[\left(1 + V_1 \eta^2 C + V_2 \eta^3 C^2 + \xi z_1 \right) \frac{\partial^2 C}{\partial x^2} + \eta^2 (V_3 + V_4 C + \xi z_2) \left(\frac{\partial C}{\partial x} \right)^2 \right]$$

where:

$$V_1 = -3 A_0 k \operatorname{tr} \boldsymbol{\varepsilon} + 2 \alpha \operatorname{tr} \boldsymbol{\varepsilon} + \alpha$$

$$V_2 = 9 A_0 k - 3 \alpha$$

$$V_3 = -3 A_0 k \operatorname{tr} \boldsymbol{\varepsilon} + 2 \alpha \operatorname{tr} \boldsymbol{\varepsilon} + \alpha$$

$$V_4 = \eta V_2 - \frac{2 \eta \alpha^2}{A_0 k}$$

$$\xi = \frac{3 A_0 k - \alpha}{3} \eta$$

The thermodynamical approach

□ Boundary conditions:

The boundary condition is obtained by equating the chemical potential of water in humid air:

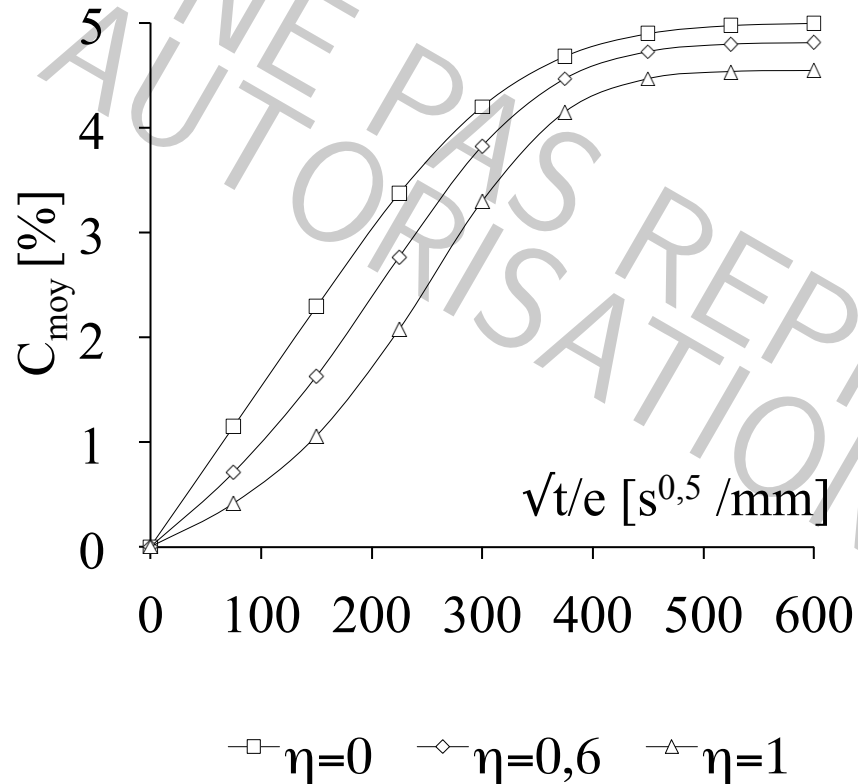
$$\hat{\mu}_w = \hat{\mu}_0 + RT \ln \frac{p_w}{p_0}$$

Where $\hat{\mu}_0$ is the chemical potential of water in humid air at the reference pressure p_0 , the partial pressure of water being p_w

$$C(x_b, t) = S p_w \exp \left[\left(\frac{\eta A_0}{k} - \frac{\eta}{3k^2} (3 A_0 k - \alpha) \right) (P^2 - 3 \eta C k P - k P) - \frac{\eta}{6k^2} (3 A_0 k - \alpha) P^2 \right]$$

Solubility (Henry's law)

The thermodynamical approach

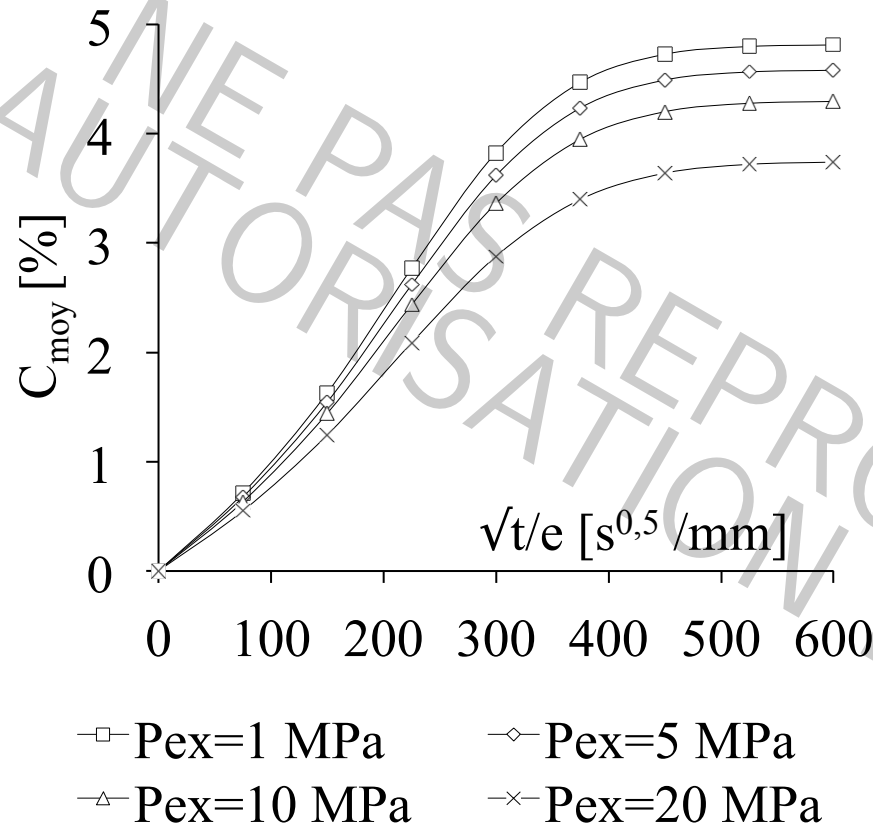


➤ of CME:

- $\eta = 0$: Fick behaviour
- ➤ Maximum moisture absorption capacity
- ➤ Moisture diffusion coefficient
- ➤ Non linear behaviour

Figure: Effect of CME on the average moisture content.

The thermodynamical approach



↗ of applied pressure:

- ↘ Maximum moisture absorption capacity
- ↘ Moisture diffusion coefficient
- ↗ Non linear behaviour

Figure: Effect of applied pressure on the average moisture content $\eta = 0.6$

The thermodynamical approach

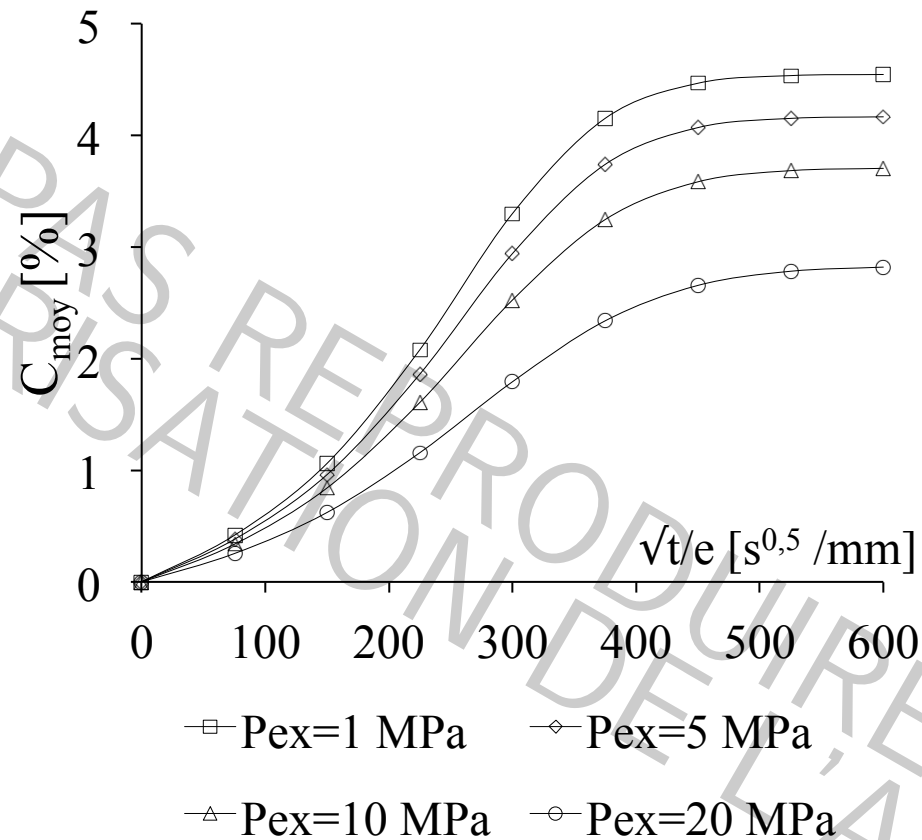


Figure: Effect of unsymmetrical applied pressure on the average moisture content $\eta = 0.6$ and $P_{int} = 1$ MPa

- Differential pressure has a significant effect on the average moisture content.

Coupled modelling for the polymers

The free volume theory

phenomenological laws

$$\frac{\partial C}{\partial t} = D \operatorname{div} \left(\vec{\operatorname{grad}} C \right) + \vec{\operatorname{grad}} D \vec{\operatorname{grad}} C$$

The thermodynamical approach

chemical potential of water

$$\dot{C} = D \left[\left(1 + V_1 \eta^2 C + V_2 \eta^3 C^2 + \xi z_1 \right) \frac{\partial^2 C}{\partial x^2} + \eta^2 (V_3 + V_4 C + \xi z_2) \left(\frac{\partial C}{\partial x} \right)^2 \right]$$

Composite materials ?



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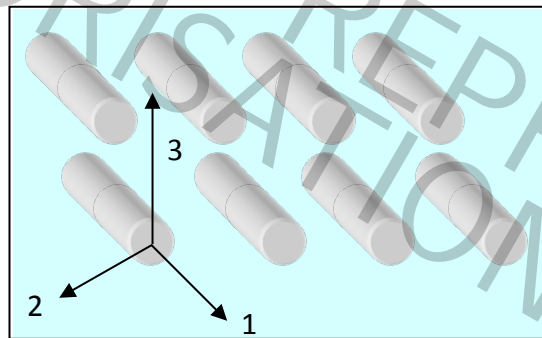
Multi-scale stresses estimations in composite structures submitted to environmental conditions

- **Hygro-elastic effective properties**
- **Plasticization effects**
- **Stress dependent moisture diffusion**

...and the biocomposites

Multi-scale stresses estimations in composite structures submitted to environmental conditions

Example of carbon/epoxy composites:



- Moisture contents of the epoxy matrix and carbon fibers are strongly different.
- Carbon fibers are considered not to absorb water.

≠ coefficients of moisture expansion (CME) + hygroscopic environment
⇒ Macroscopic (ply) and local (fiber and matrix) hygro-elastic stresses

Multi-scale stresses estimations in composite structures submitted to environmental conditions

Self-consistent estimates for the hygro-elastic properties

Multi-scale hygroelastic behaviours

$$\sigma^\alpha = \mathbf{L}^\alpha : (\boldsymbol{\varepsilon}^\alpha - \eta^\alpha \Delta C^\alpha)$$

Hill's relations on the volume average equations

$$\begin{aligned} \langle \sigma^\alpha \rangle_{\alpha=f,m} &= \sigma^I \\ \langle \boldsymbol{\varepsilon}^\alpha \rangle_{\alpha=f,m} &= \boldsymbol{\varepsilon}^I \end{aligned}$$

Scale transition relation from Eshelby's formalism

$$\sigma^\alpha - \sigma^I = -\mathbf{L}^I : \mathbf{R}^I : (\boldsymbol{\varepsilon}^\alpha - \boldsymbol{\varepsilon}^I)$$

$\mathbf{R}^I$ depends on elastic macroscopic stiffness and morphology of the constituents

Multi-scale stresses estimations in composite structures submitted to environmental conditions

Self-consistent estimates for the hygro-elastic properties

Classical self-consistent estimate for the macroscopic elastic stiffness

$$\mathbf{L}^I = \left\langle \left(\mathbf{L}^\alpha + \mathbf{L}^I : \mathbf{R}^I \right)^{-1} : \left(\mathbf{L}^I + \mathbf{L}^I : \mathbf{R}^I \right) : \mathbf{L}^\alpha \right\rangle_{\alpha=f,m}$$

Self-consistent estimate for the macroscopic CME

$$\boldsymbol{\eta}^I = v^m \frac{\Delta C^m}{\Delta C^I} \mathbf{L}^{I-1} \left\langle \left(\mathbf{L}^\alpha + \mathbf{L}^I : \mathbf{R}^I \right)^{-1} \right\rangle_{\alpha=f,m}^{-1} : \left(\mathbf{L}^m + \mathbf{L}^I : \mathbf{R}^I \right)^{-1} : \mathbf{L}^m : \boldsymbol{\eta}^m$$

Ratio depending on the constitutive materials

Multi-scale stresses estimations in composite structures submitted to environmental conditions

Macroscopic stresses

- Constitutive laws of hygro-elastic orthotropic materials
- Strain-displacement relations
- Compatibility and equilibrium equations and boundary conditions

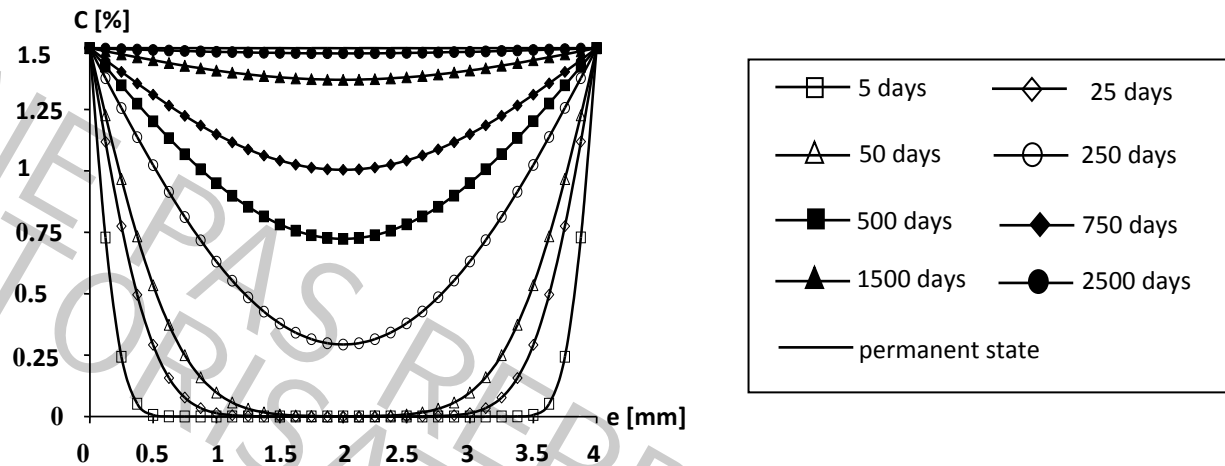
Local stresses

- Local stress-strain relation in constituents
- Scale transition relation for the strains in fibers (Eshelby for example)
- Local mechanical states in the matrix (Hill for example)

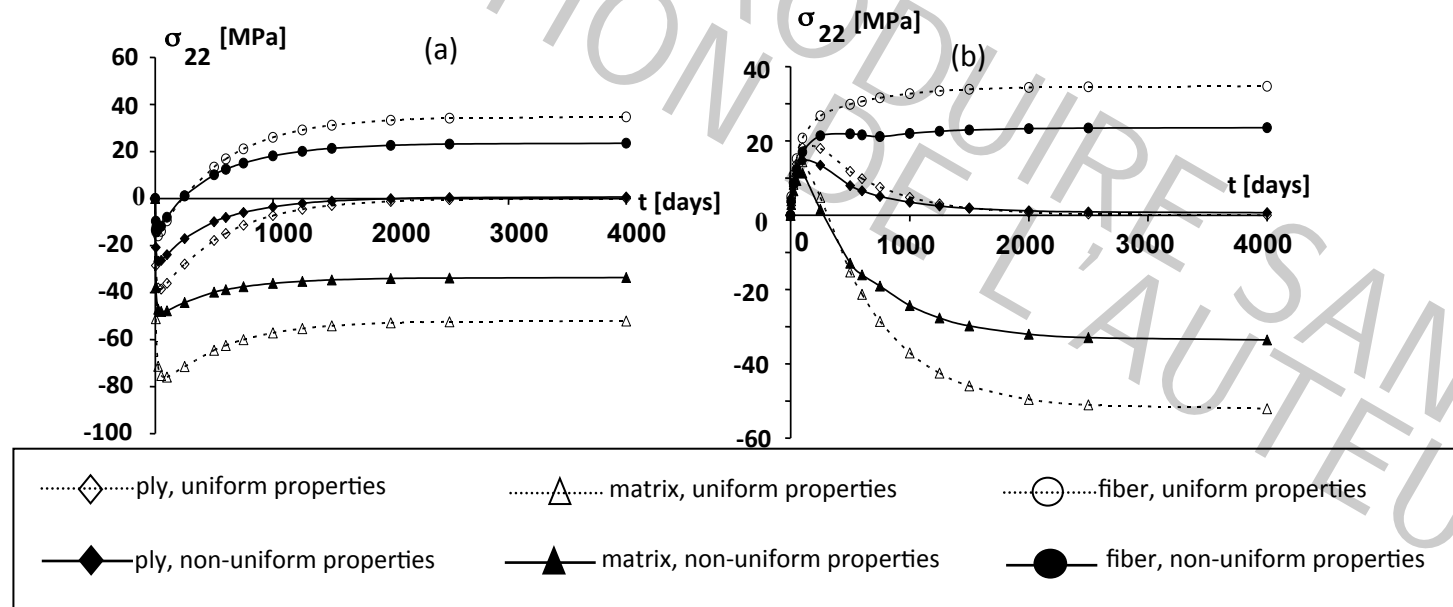
plasticization effects related to the evolution, as a function of the moisture content, of the hygro-elastic properties on the internal stress states



Plasticization effects on the multi-scale internal stresses



Time and space dependent moisture content profiles in the composite structure.



Effect of the plasticization on the multi-scale stress states in (a) the external ply / (b) the central ply of a uni-directional composite during the transient part of the moisture diffusion process.



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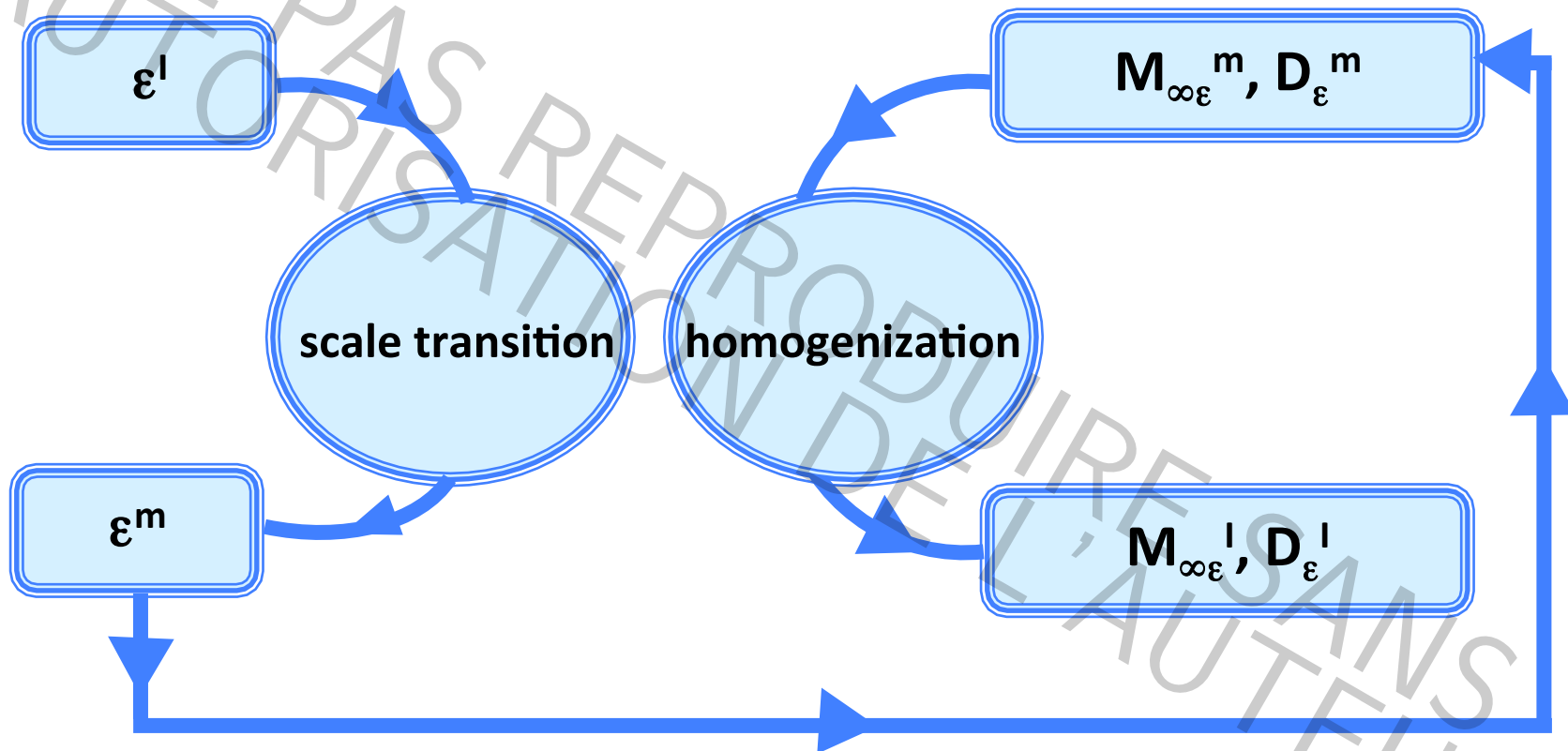
The thermodynamical approach

Multi-scale stresses estimations in composite structures submitted to environmental conditions

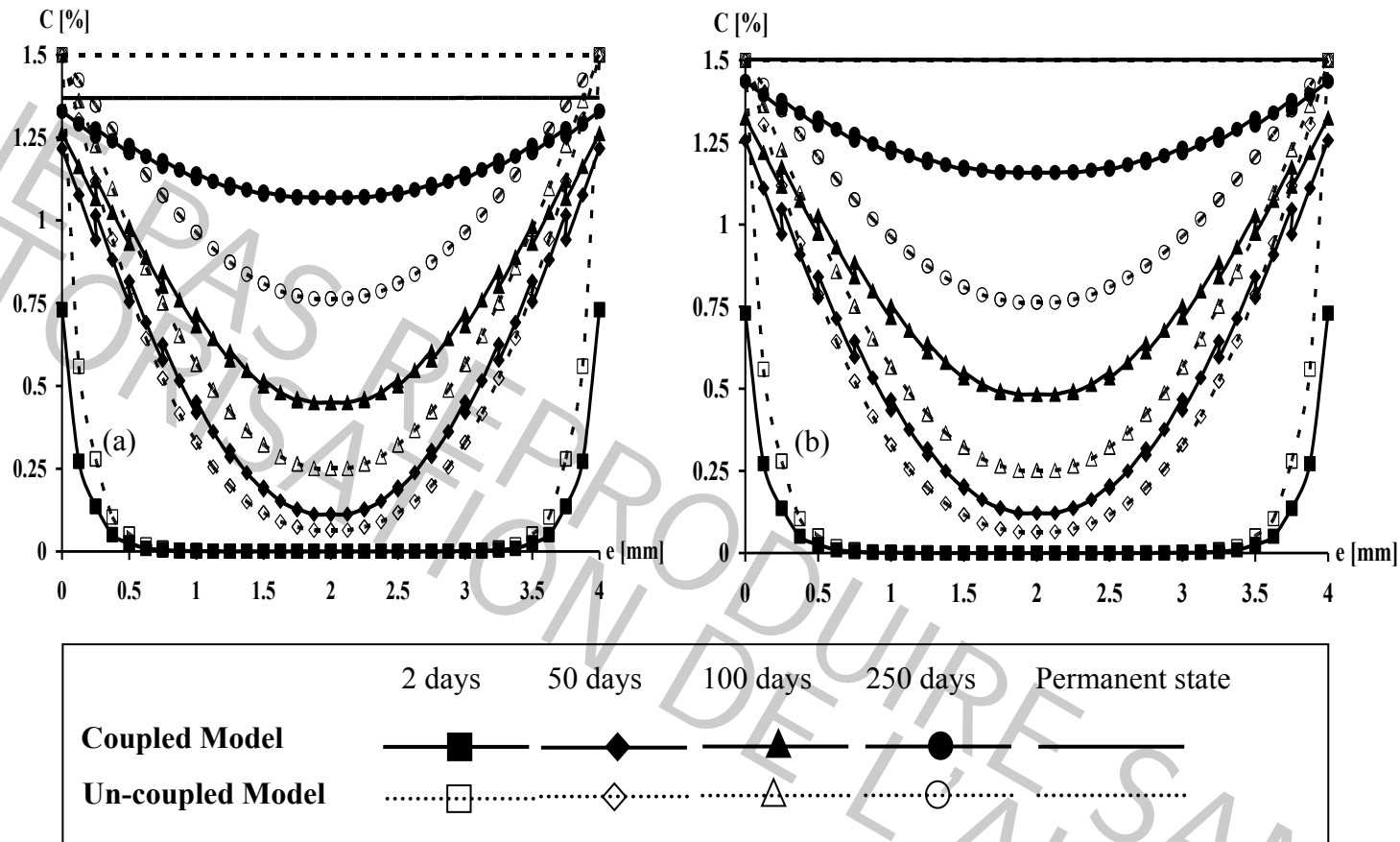
- Hygro-elastic effective properties
- Plasticization effects
- **Stress dependent moisture diffusion**

...and the biocomposites

The free volume theory



The free volume theory



Macroscopic moisture content profiles in the (a) $\pm 55^\circ$ composite / (b) unidirectional composite structure.

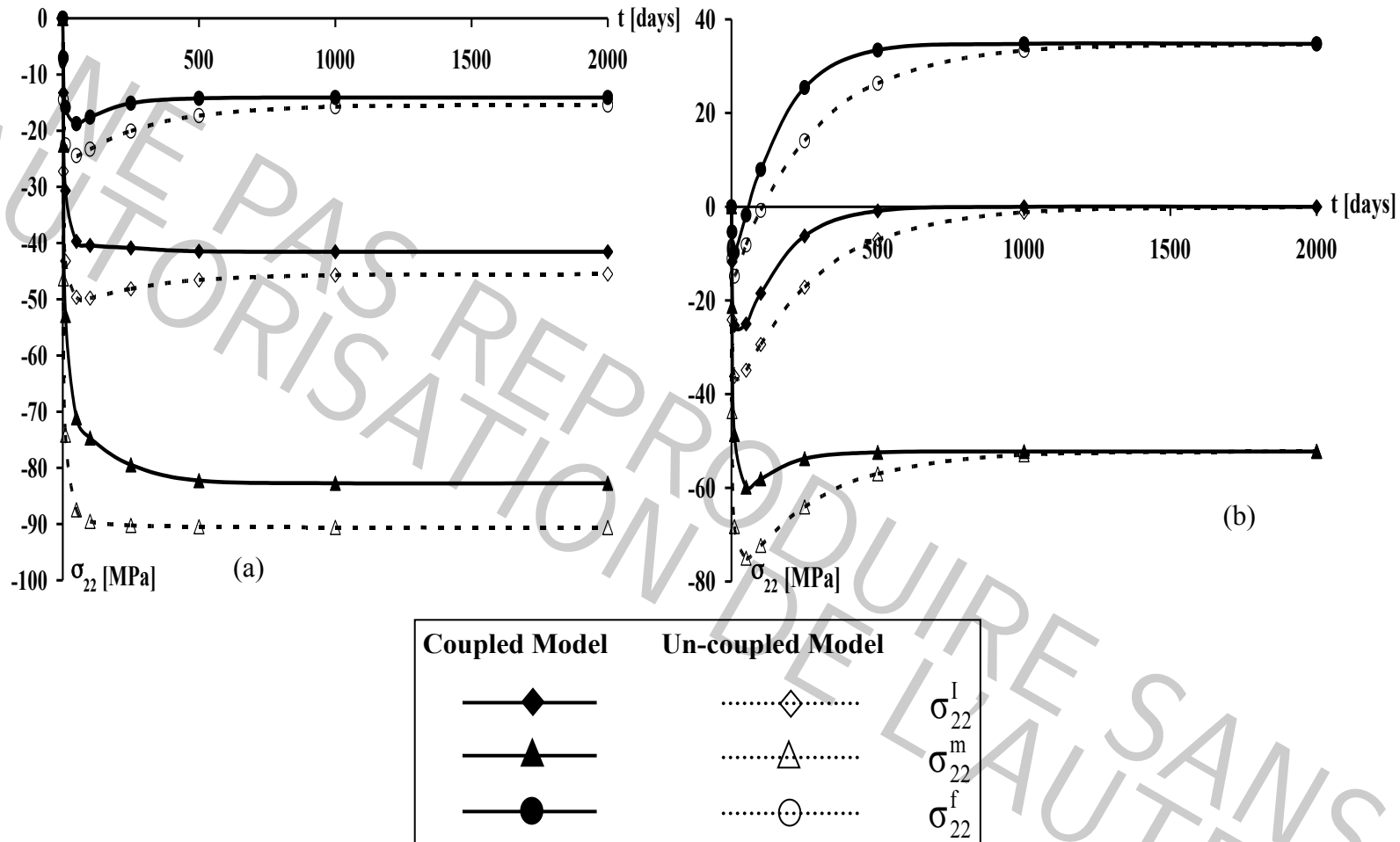
Uncoupled Model:

Identical moisture content profile evolution versus time in both the structures.

Coupled Model:

The strong evolution of the moisture content at the boundaries is attributed to the variation of the maximum moisture absorption capacity.

The free volume theory



Multi-scale transverse stress states in the internal ply of the (a) $\pm 55^\circ$ composite / (b) unidirectional composite structure.

The coupled model predicts transverse stresses states the absolute value of which is weaker than that predicted by the corresponding uncoupled approach.



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The free volume theory

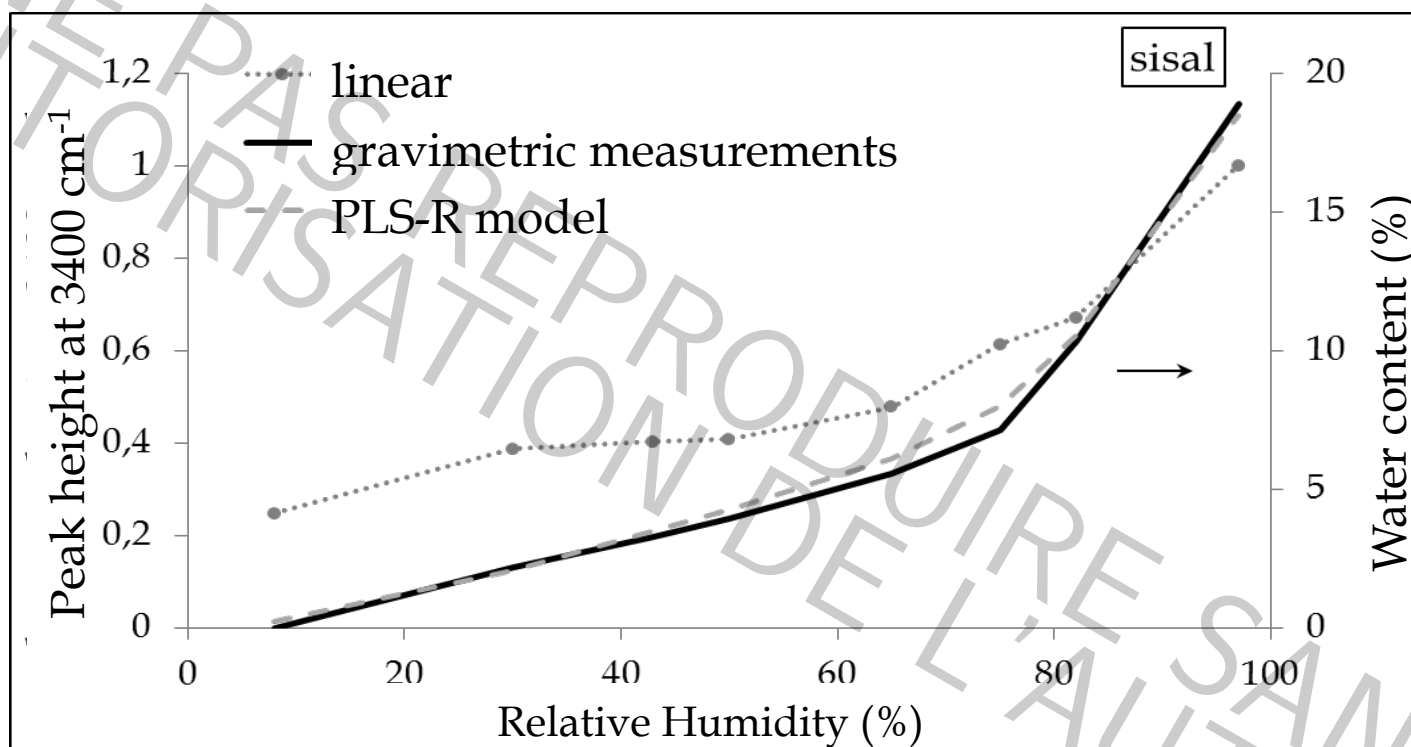
The thermodynamical approach

Multi-scale stresses estimations in composite structures submitted to environmental conditions

- Hygro-elastic effective properties
- Plasticization effects
- Stress dependent moisture diffusion

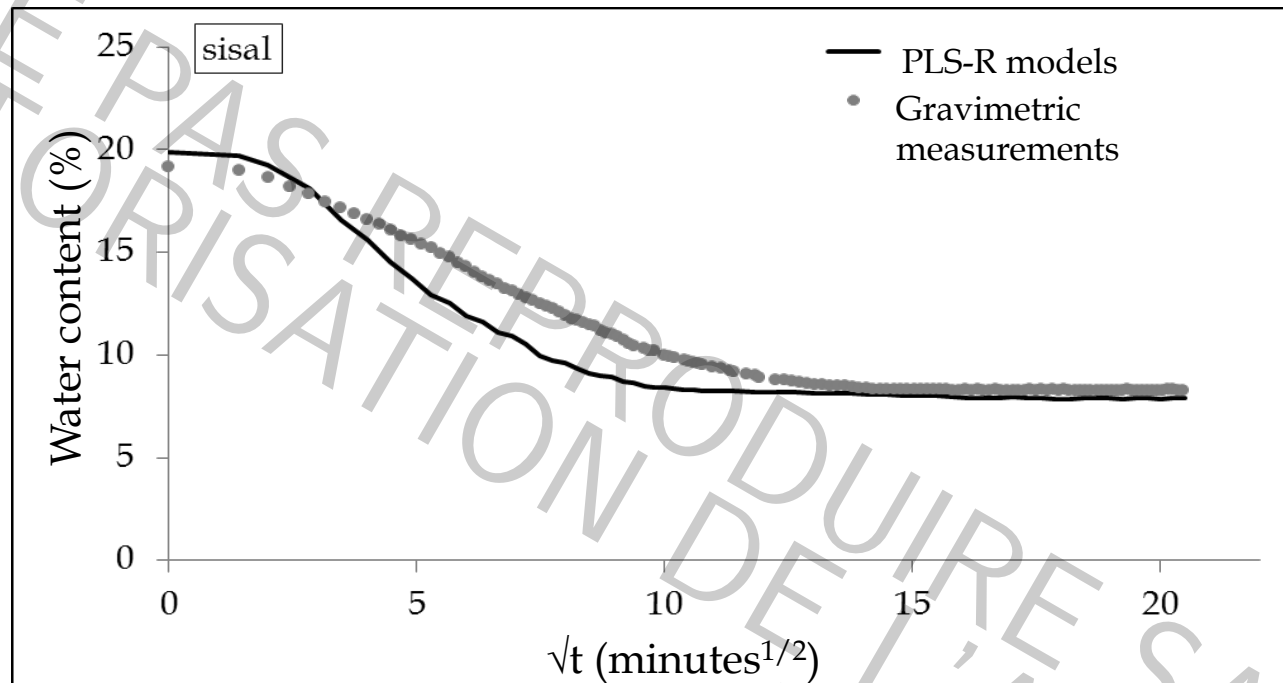
...and the biocomposites

Qualitative and quantitative assesment of water sorption in natural fibers using FTIR-ATR spectroscopy



- ➡ Sorption isotherm characteristic of natural fibres: **Sigmoidal shape**
- ➡ Univariate model doesn't fit the experimental curve $R^2 < 90 \%$
- ➡ Validation of multivariate model: **$R^2 = 99.79 \%$**

Qualitative and quantitative assesment of water sorption in natural fibers using FTIR-ATR spectroscopy

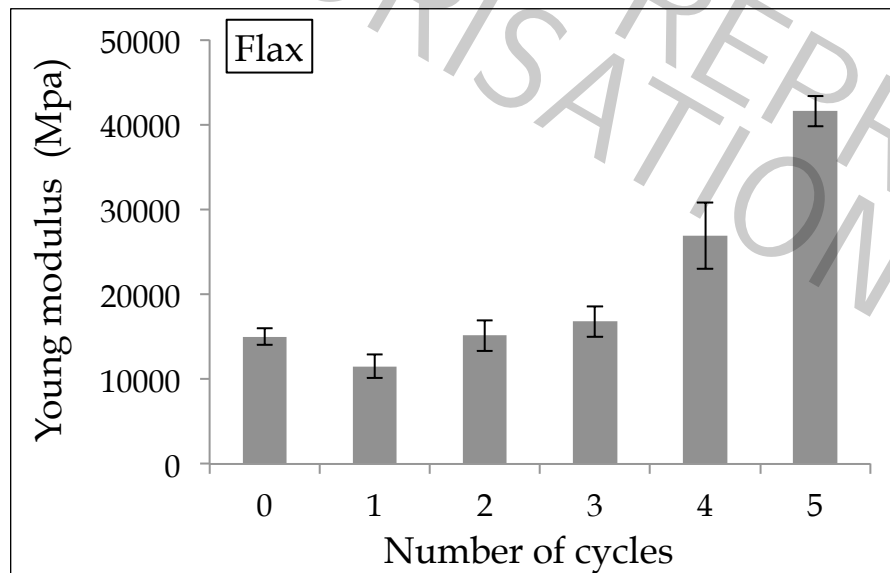


- Straightforward determination of the drying kinetics (comparison with the gravimetric measurements)
- Good correlation between the two methods: The slope deviation could be explained by the ATR method

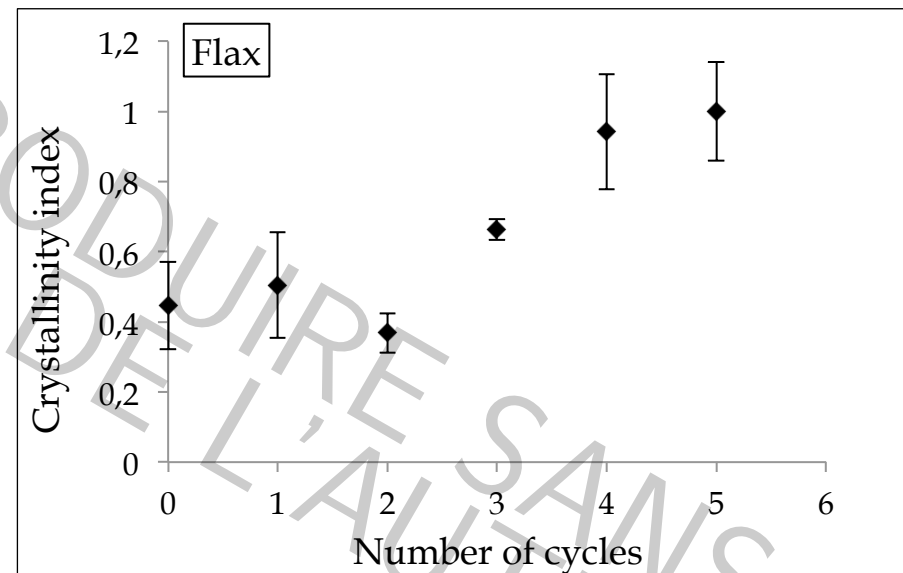
Advantages Informations at the first time of the kinetics / informations at the molecular scale

Qualitative and quantitative assesment of water sorption in natural fibers using FTIR-ATR spectroscopy

- Sorption/desorption cycles (80 % $\rightarrow$ 30 %)
- Tensile test and crystallinity index calculation every cycle



Young modulus versus number of sorption/desorption cycles



Crystallinity index evolution versus number of sorption/desorption cycles

- Young modulus increases with the number of cycles
- Relationship between Young modulus and crystallinity

Coupling moisture diffusion and internal mechanical states in composite materials

The aim of the present work is to propose models enabling to take into account:

- **water-mechanical property coupling** due to **plasticization** phenomenon
- coupling between moisture diffusive behavior law and internal mechanical states according to the **free volume theory** or the **thermodynamic approach**.

Such models can be used to represent several typical diffusive behaviors met experimentally.

Ongoing experimental studies should provide results to be compared with the numerical predictions.