



AUSSOIS 2019

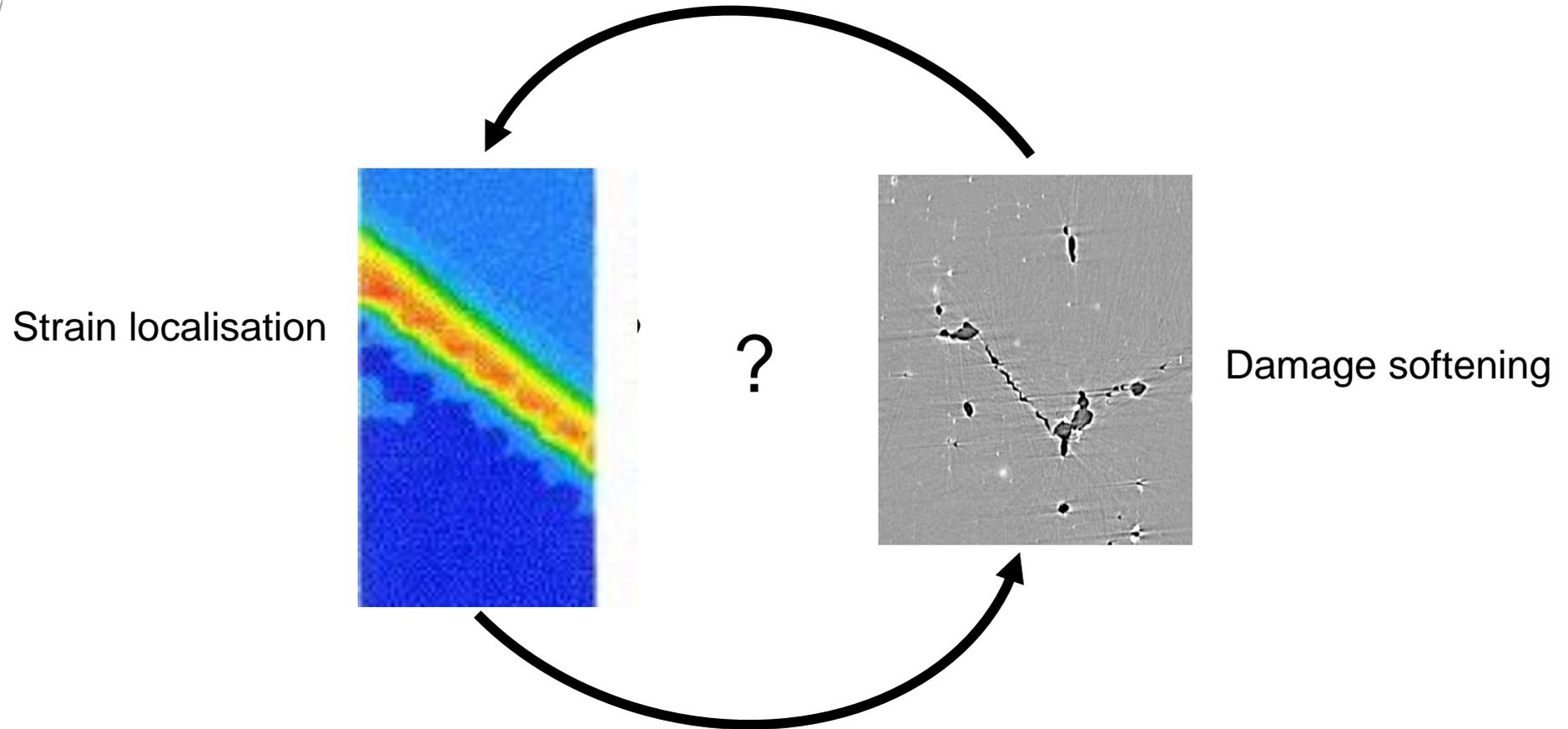


Interaction entre localisation et endommagement lors de la rupture ductile : mesures tridimensionnelles *in situ*

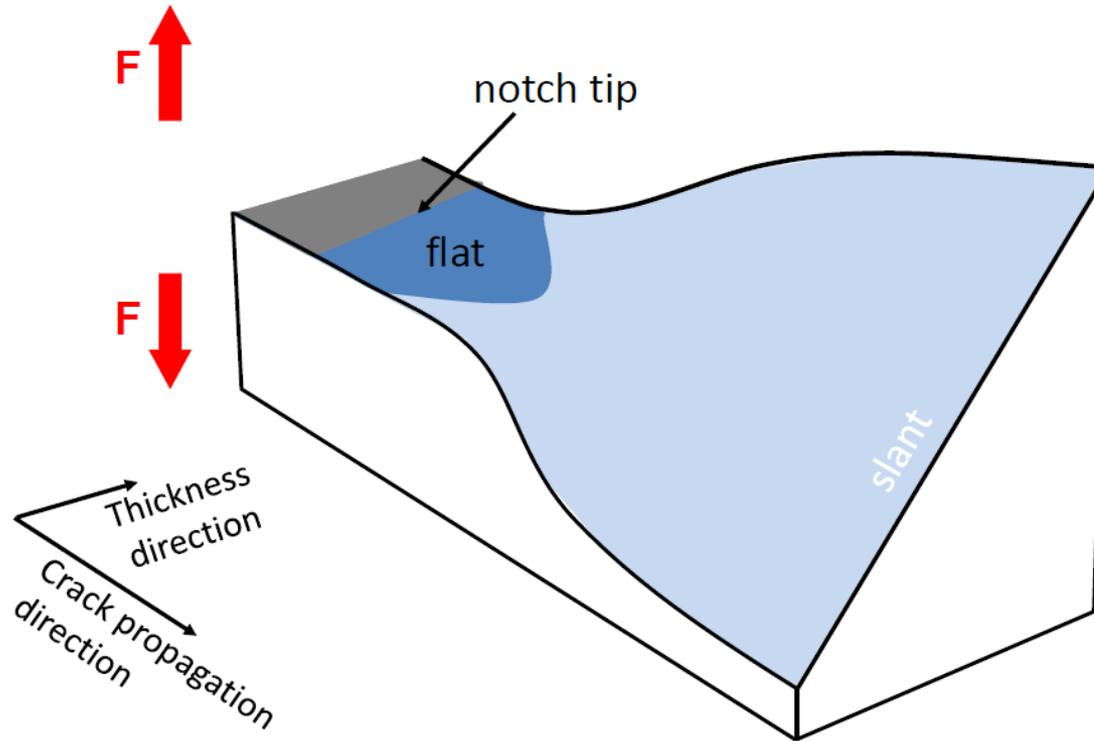
Thilo F. MORGENEYER

Centre des Matériaux (CdM) UMR CNRS 7633 / B.P. 87 91003 Evry cedex France

The chicken and egg question



What is the origin of slant fracture?



Approach:
Measure **damage** and **strain *in situ*** and simultaneously
via two novel techniques

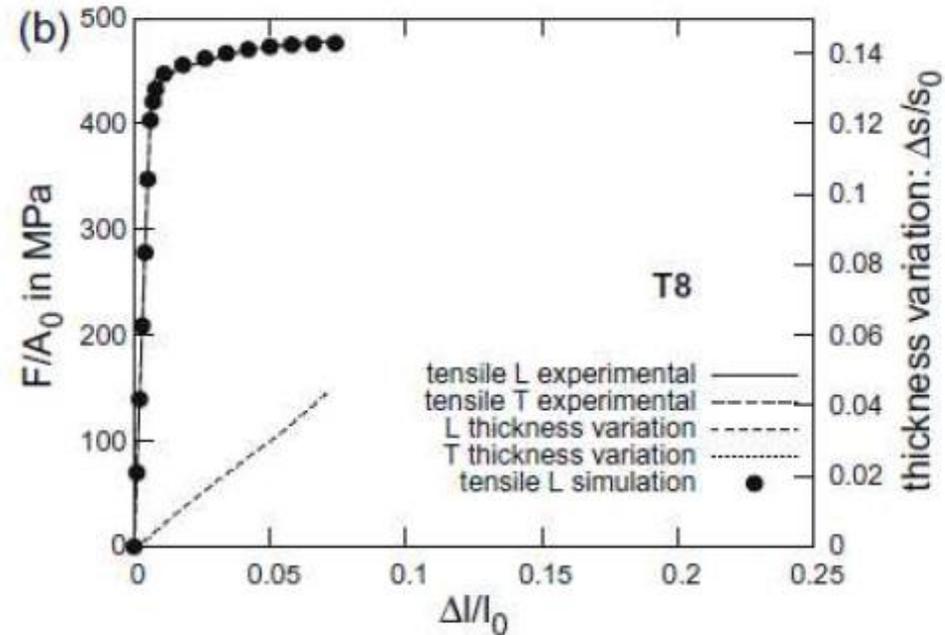
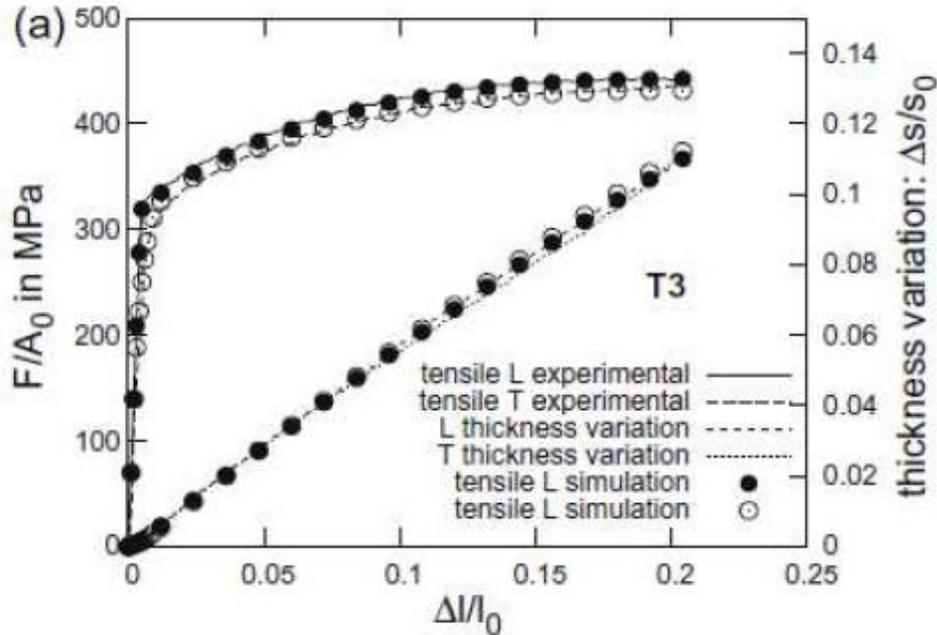
- Materials
- Synchrotron laminography:
 - imaging and damage analysis
- Digital volume correlation: volume strain measurement
- Strain and damage interactions ahead of a notch
 - Different materials
 - Origins?

- *In situ* damage evolution under shear loading
- Multiscale observations with enhanced resolution

2139 T3/T8 tensile stress stain curves

T3

T8



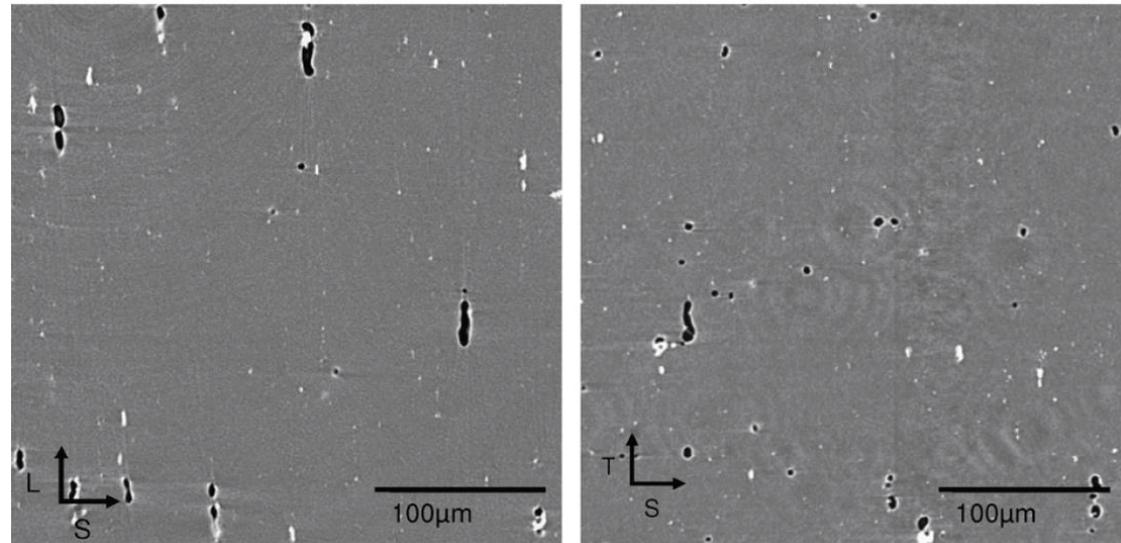
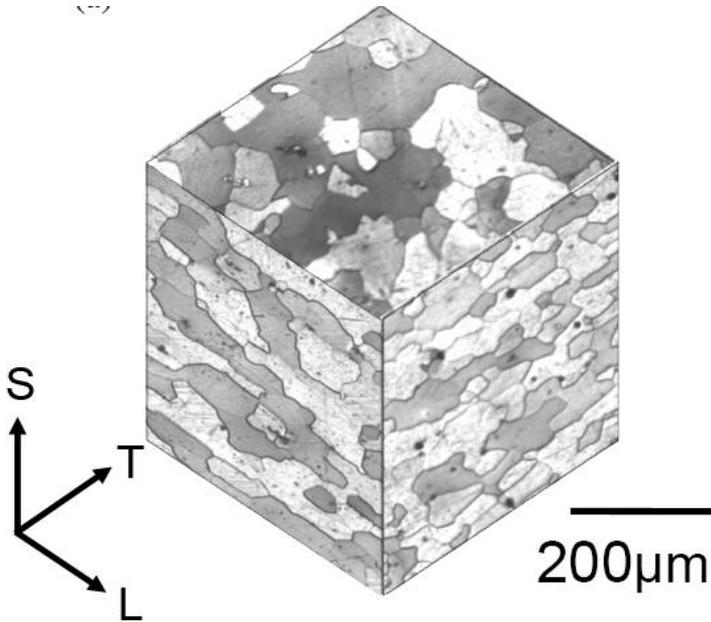
Application to AA2139-T3

Al-Cu alloy

► Grain size

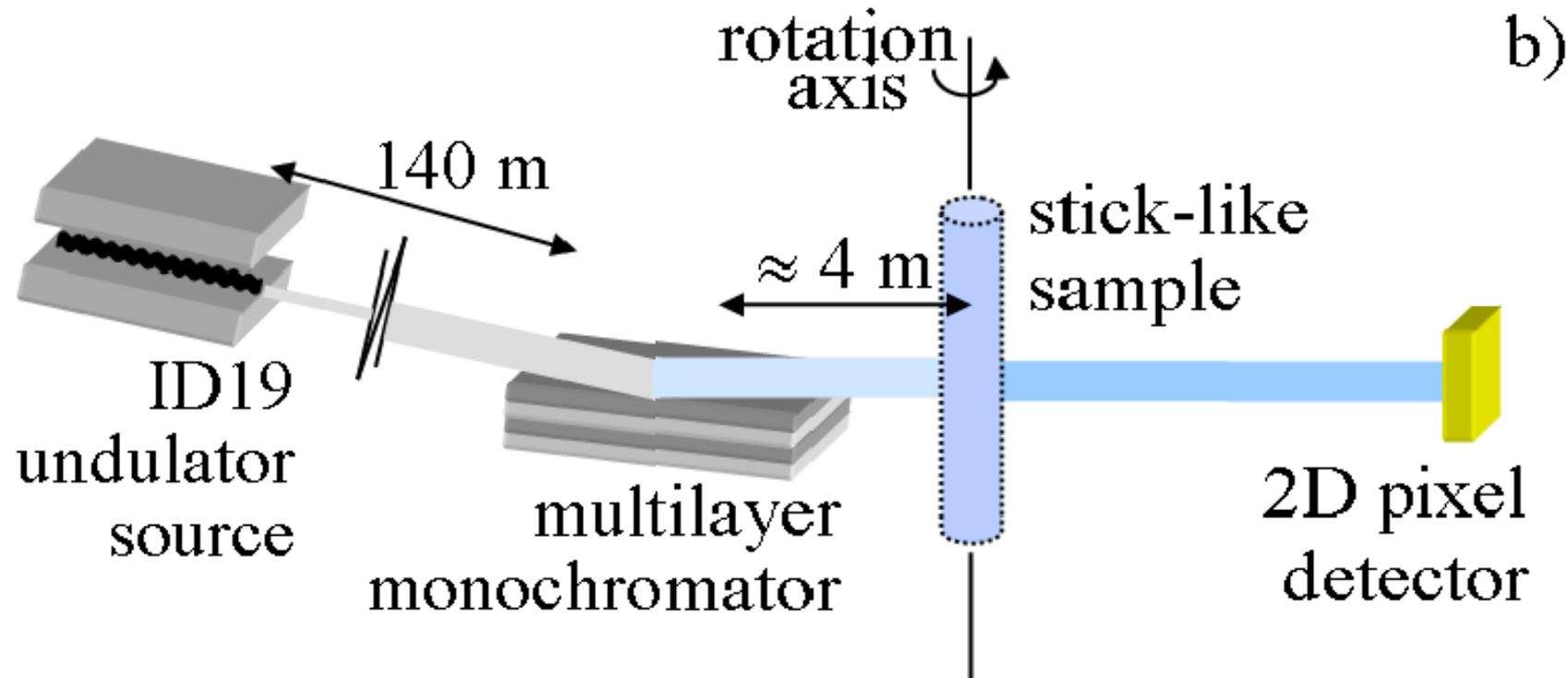
initial porosity: 0.3%

intermetallic content: 0.45%



Ductile fracture: tomography observation

○ Principle



- Stick like samples
- Typical diameter 1 mm for μm resolution
- Observation of sheets for ductile tearing not possible!

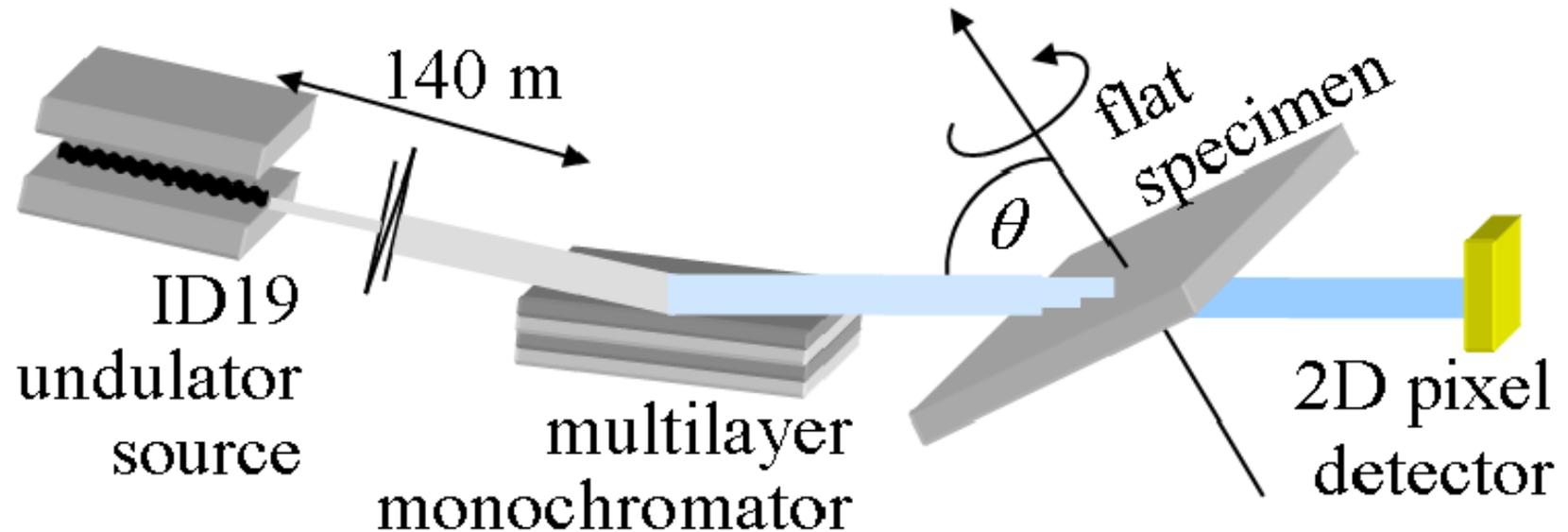
Ductile fracture: laminography observation



○ Principle

L. Helfen *et al.*, *Appl. Phys. Lett.* 2005

L. Helfen *et al.*, *Appl. Phys. Lett.* 2009

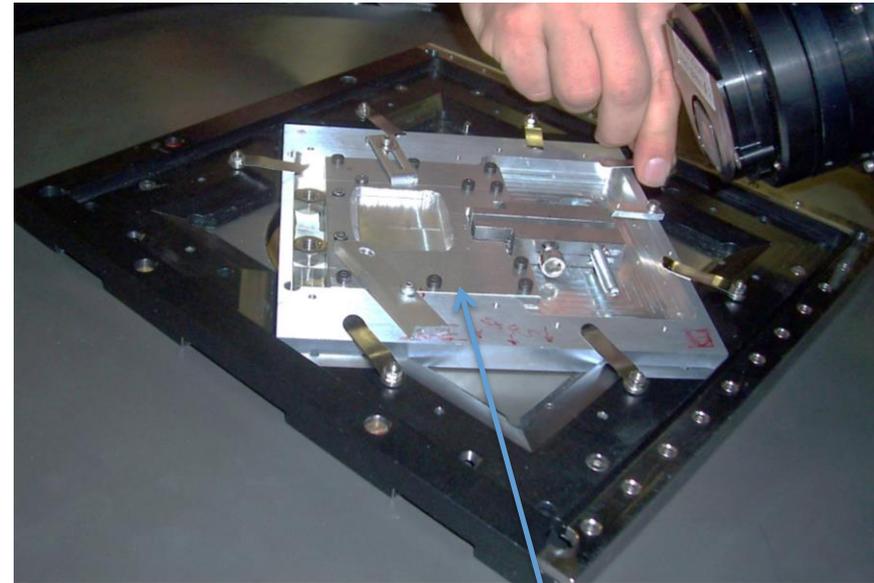
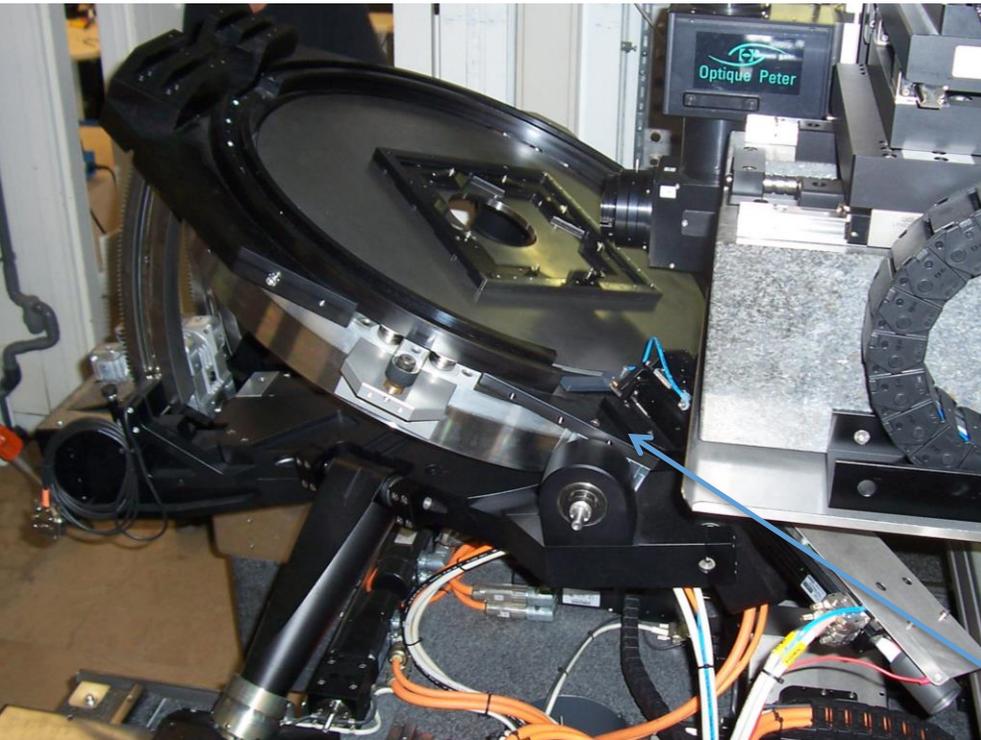


- Specimen can be much larger than lateral field of view of detector!
- Major benefit: Possibility to observe *in situ* large thin samples at realistic sizes/ length scales
- Additional artefacts due to missing information
- Available at ESRF (ID15; ID19), KARA (TOPOTOMO; IMAGE-beamline)

In situ laminography

○ end-station and loading device

- Dedicated *in situ* machines as there are specifications on:
 - geometry
 - weight



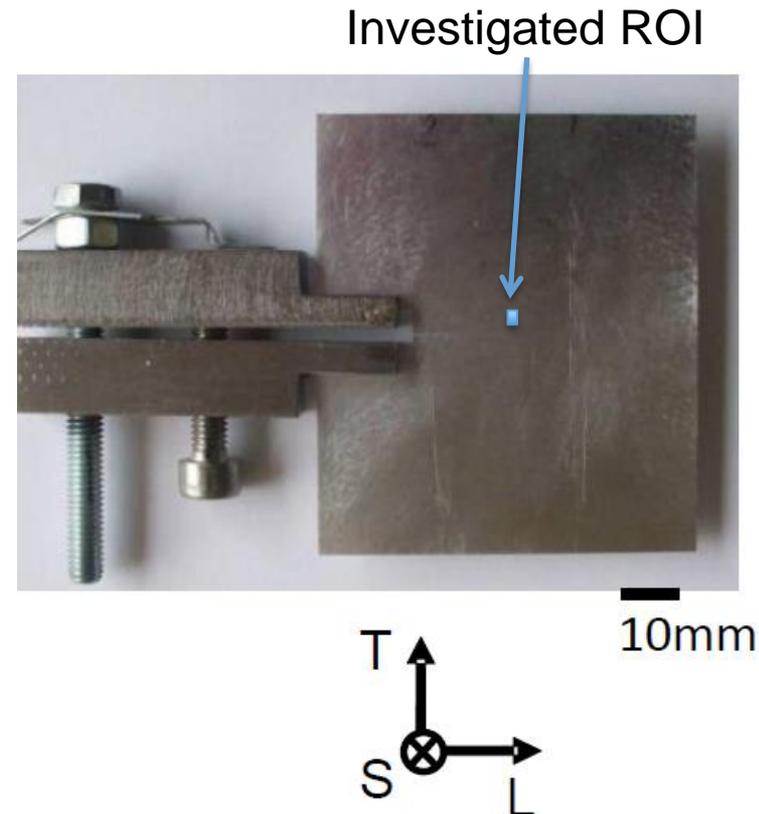
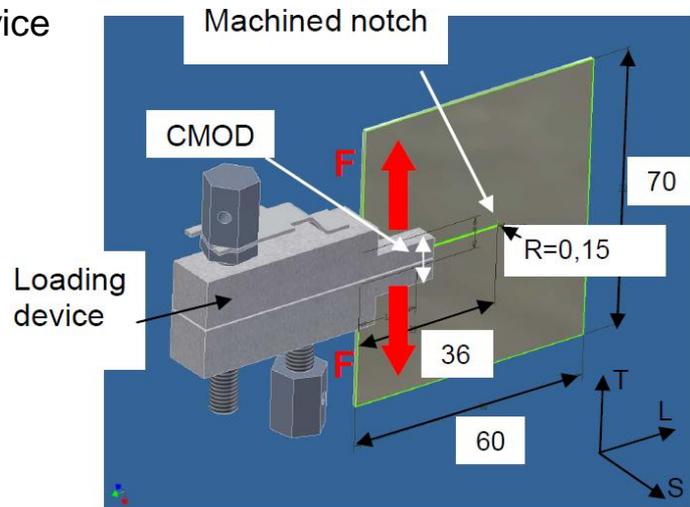
2 screw loading device
Anti-buckling frame
specimen

KIT- Laminography device at ESRF ID19

Aluminium sheet investigated by laminography

- 2139 Al alloy in T3 condition,
- Specimen thickness 1mm
- ~20 load steps via 2 load screws
 - T-L configuration

anti-buckling device
not shown here

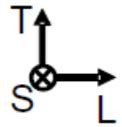


- Scan conditions, at ESRF ID19
 - Tilt angle 25 degree
 - Voxel size $0.7 \mu\text{m}$
 - Scan time ~12 mins

In-situ SRCLaminography

- 2139 T3: 2D section at mid-thickness

Crack progression
(and rolling) direction

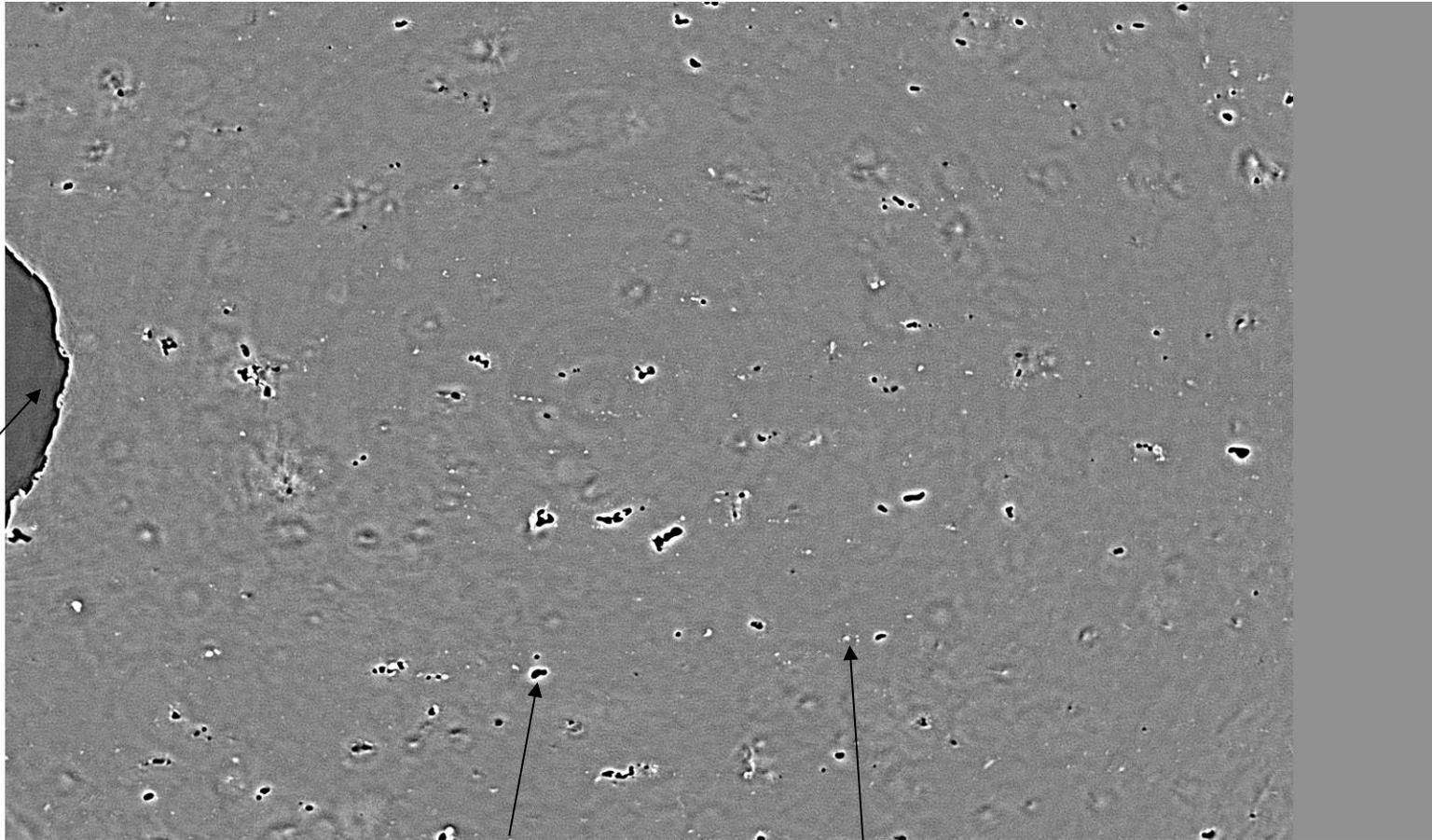


notch

200μm

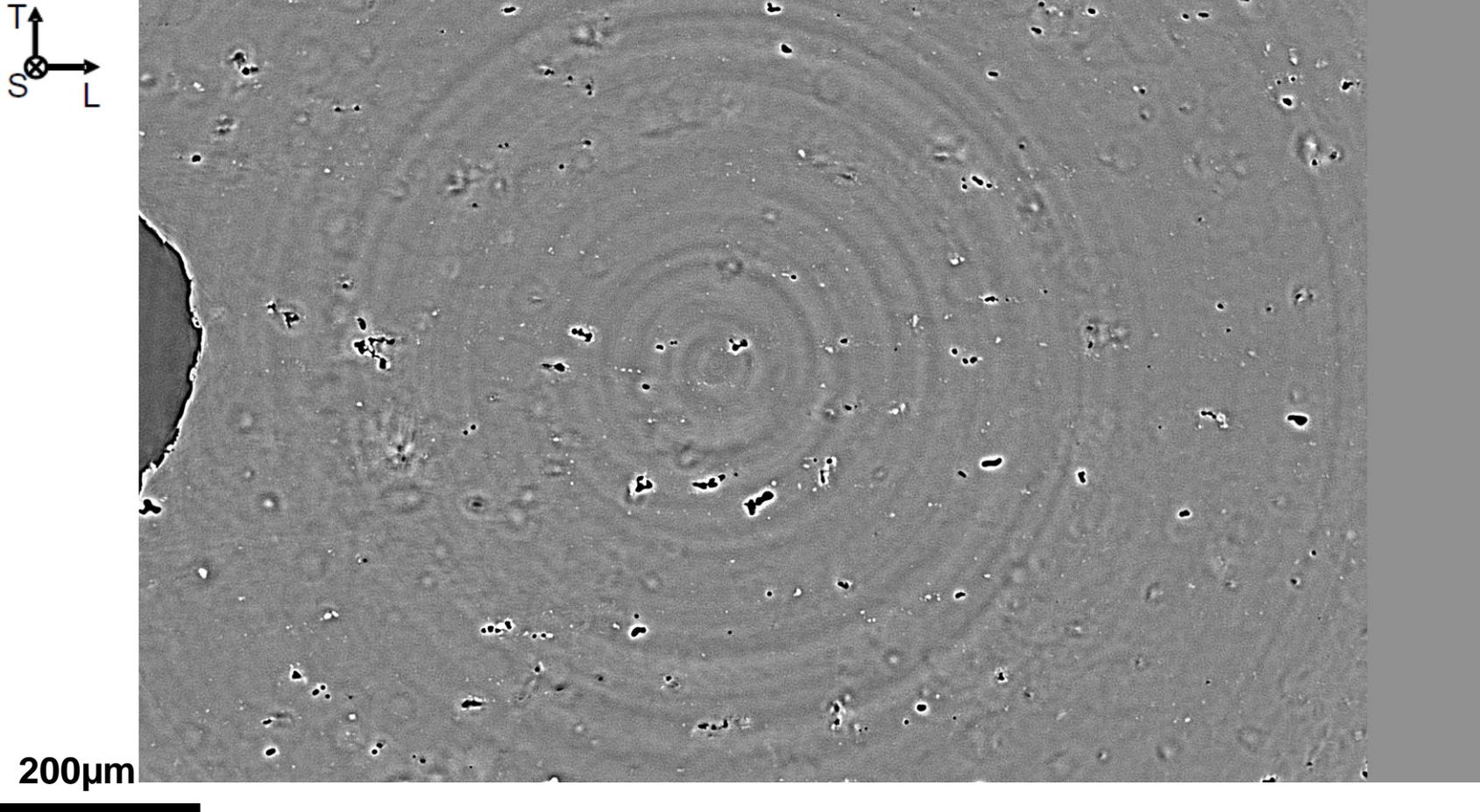
void

Intermetallic particle



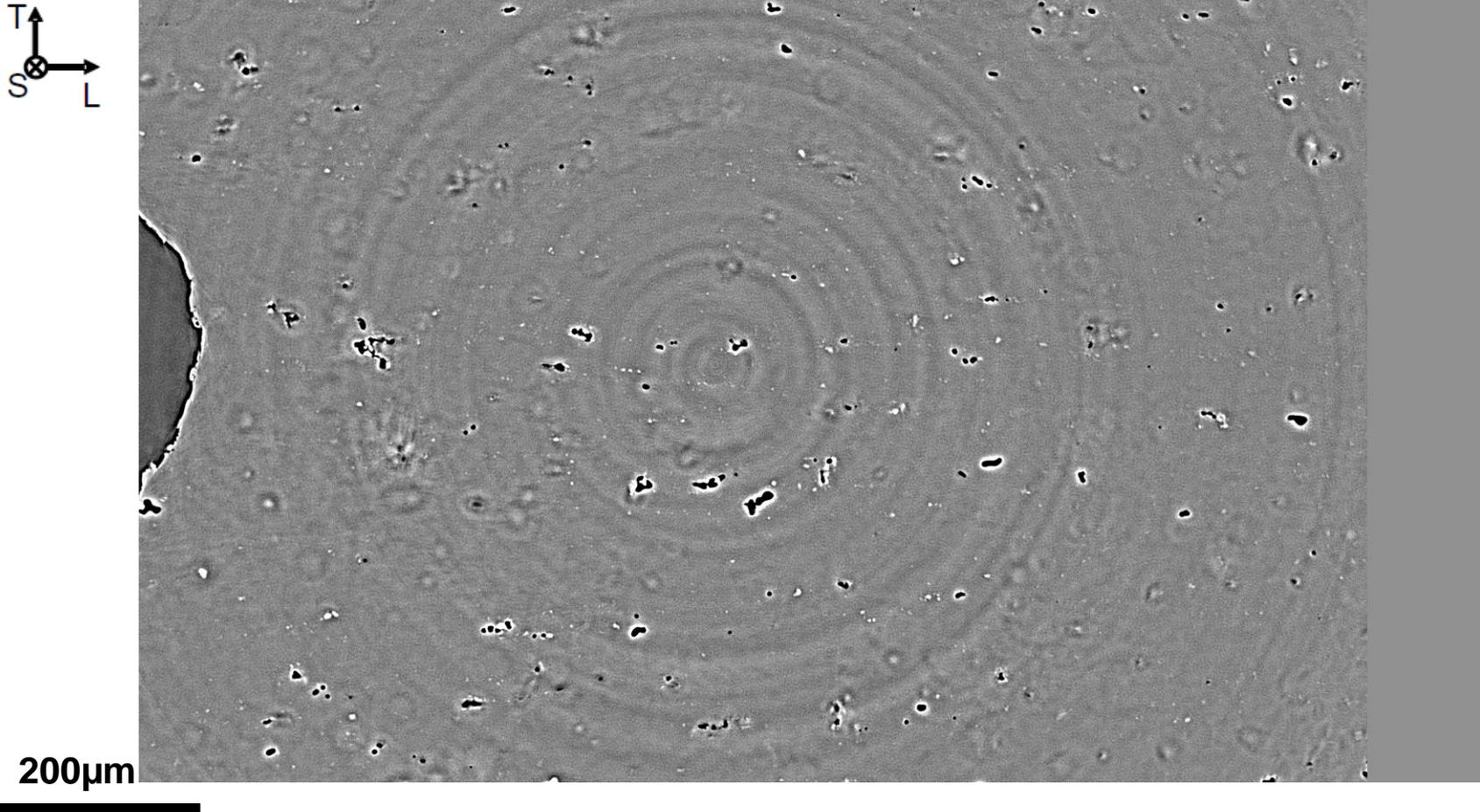
In-situ SRCLaminography

- 2139 T3: 2D section at mid-thickness



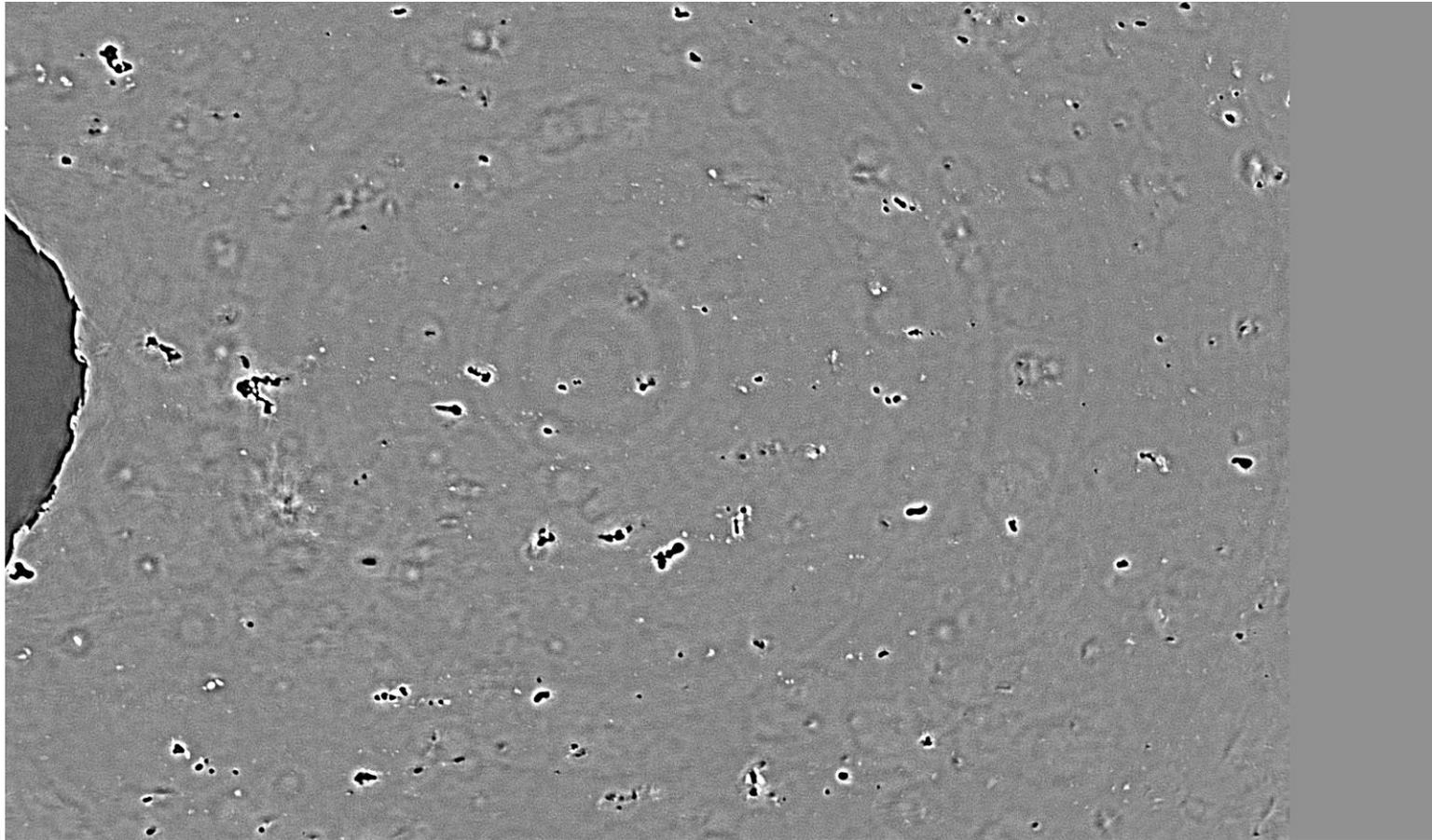
In-situ SRCLaminography

- 2139 T3: 2D section at mid-thickness



In-situ SRCLaminography

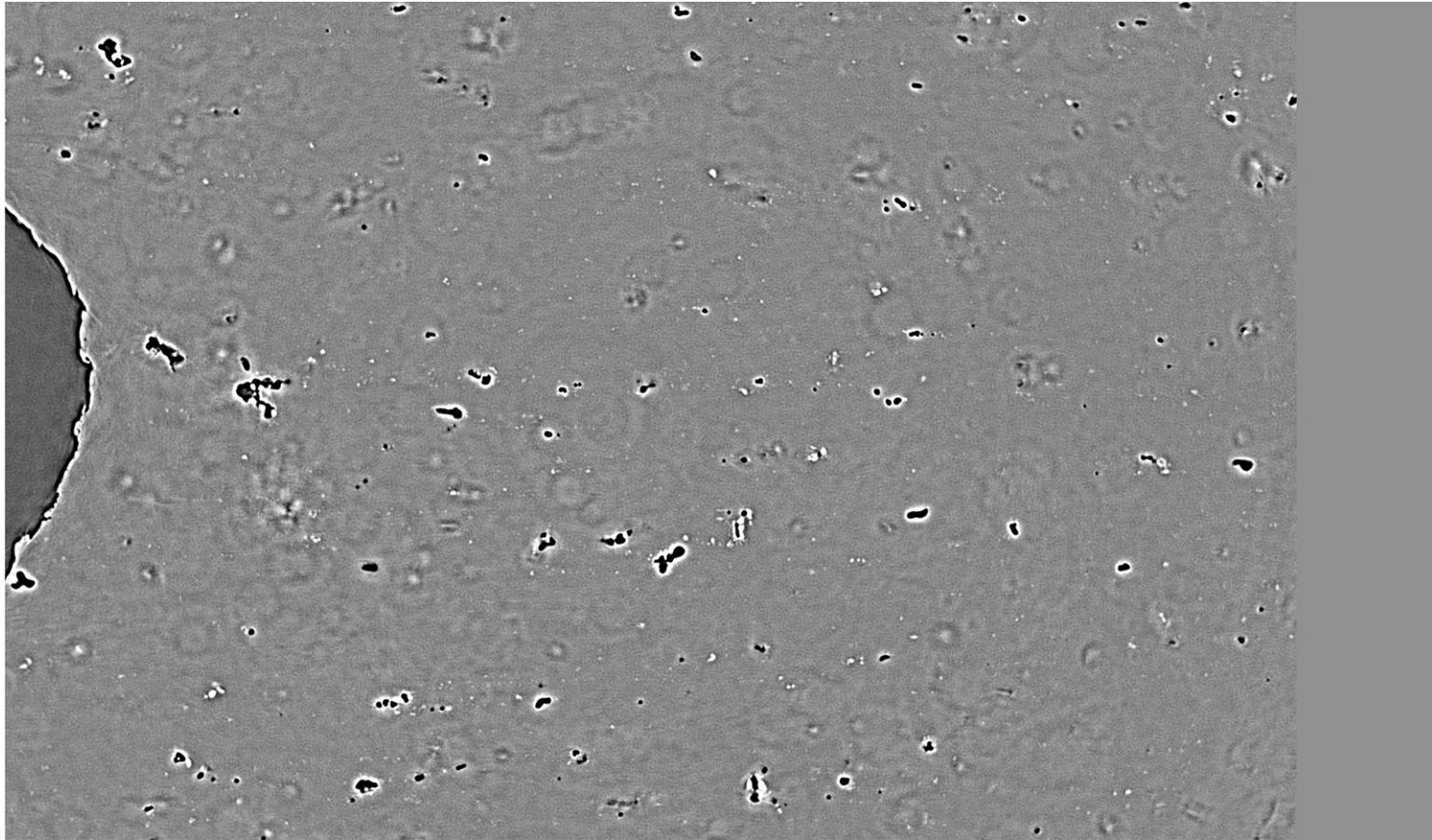
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

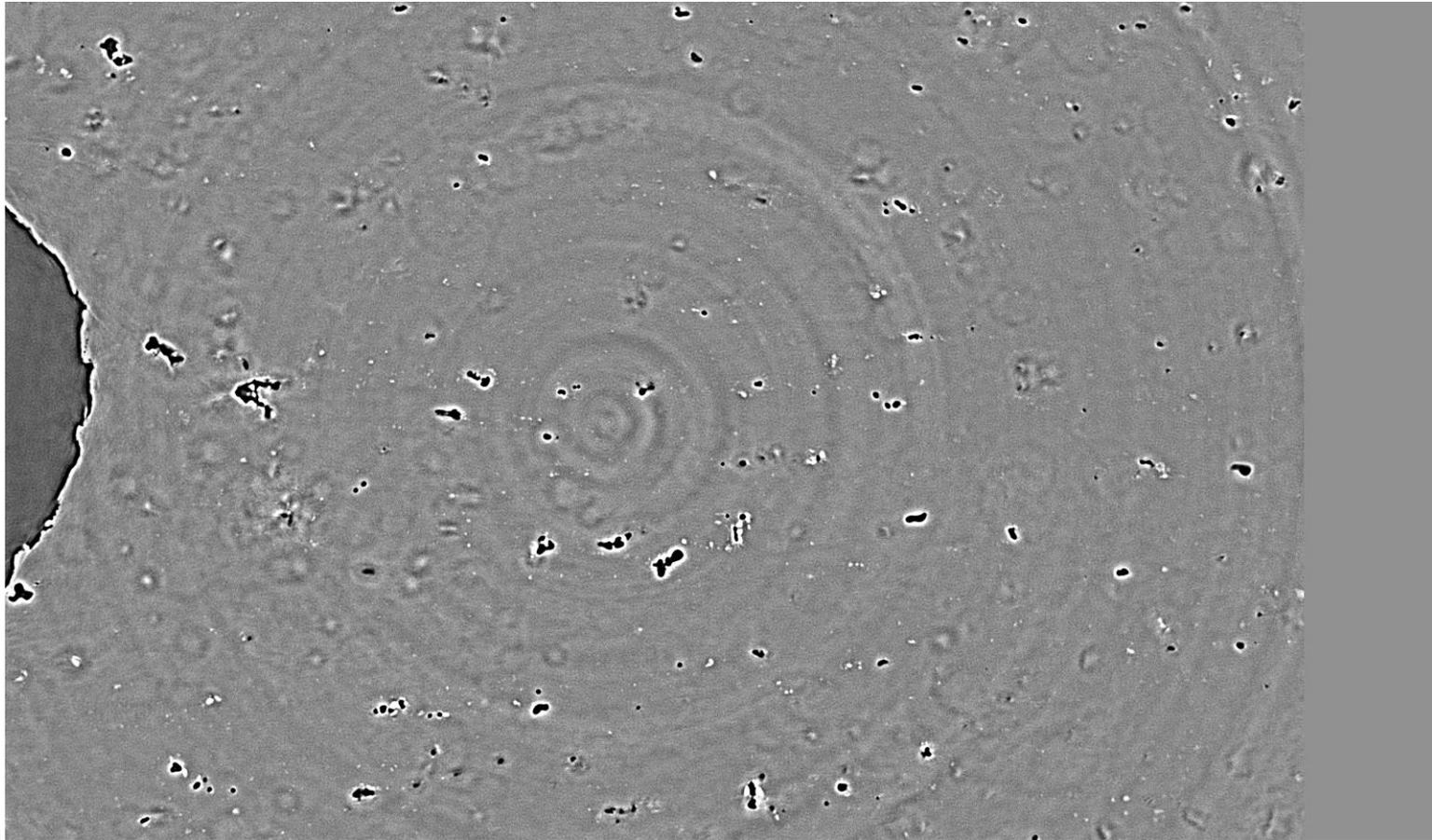
- 2139 T3: 2D section at mid-thickness



200μm

In-situ SRC Laminography

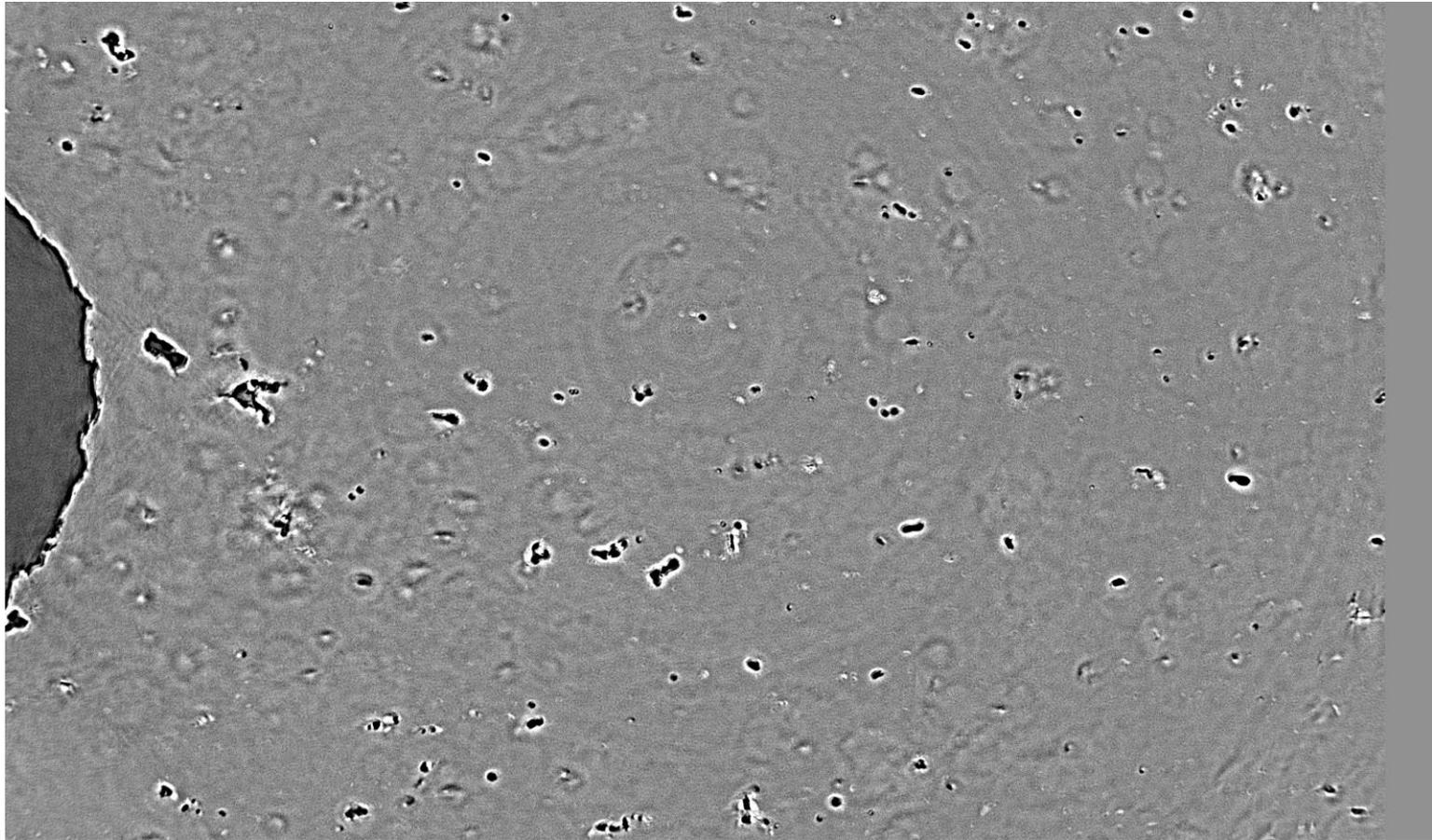
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

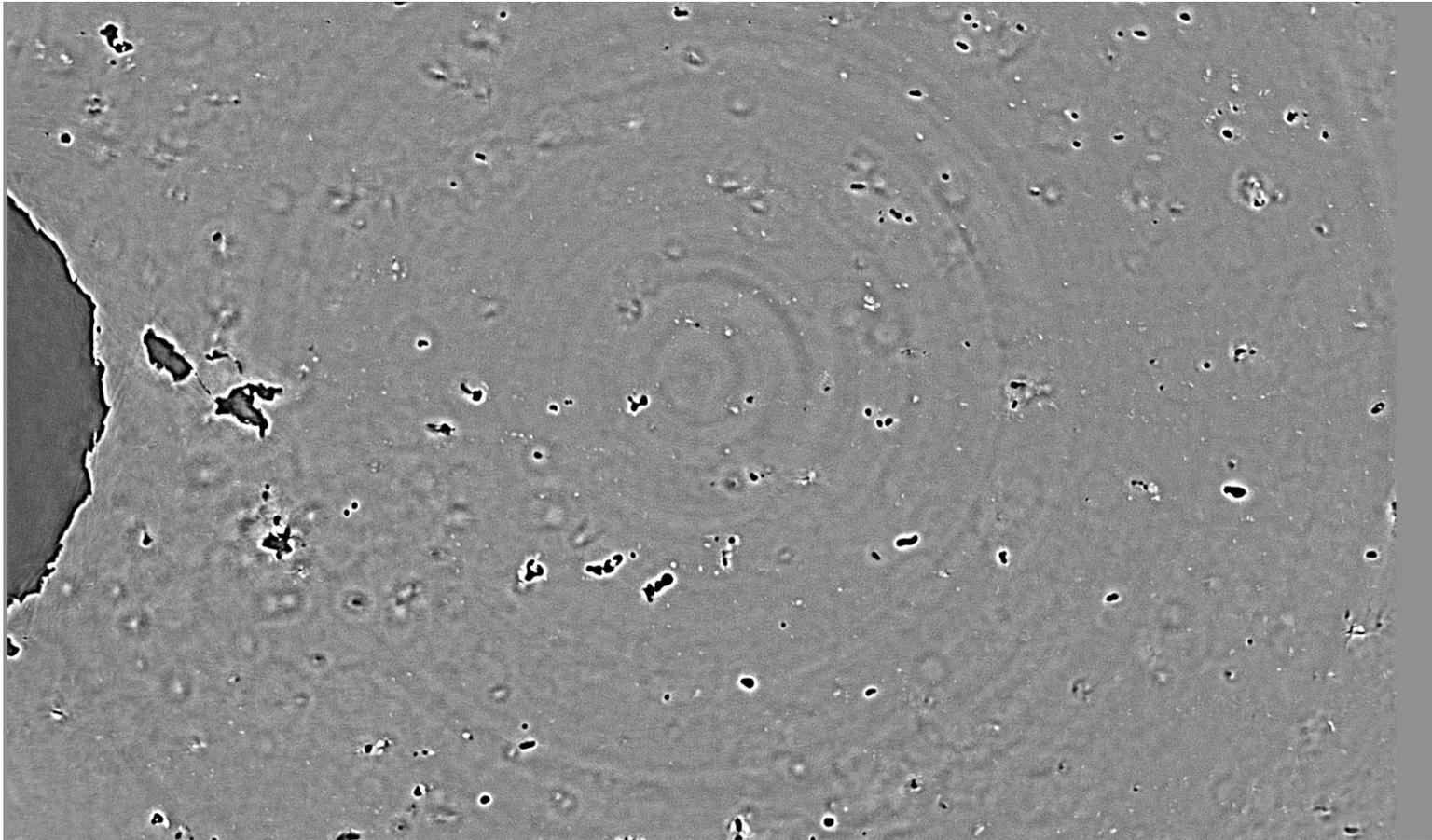
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

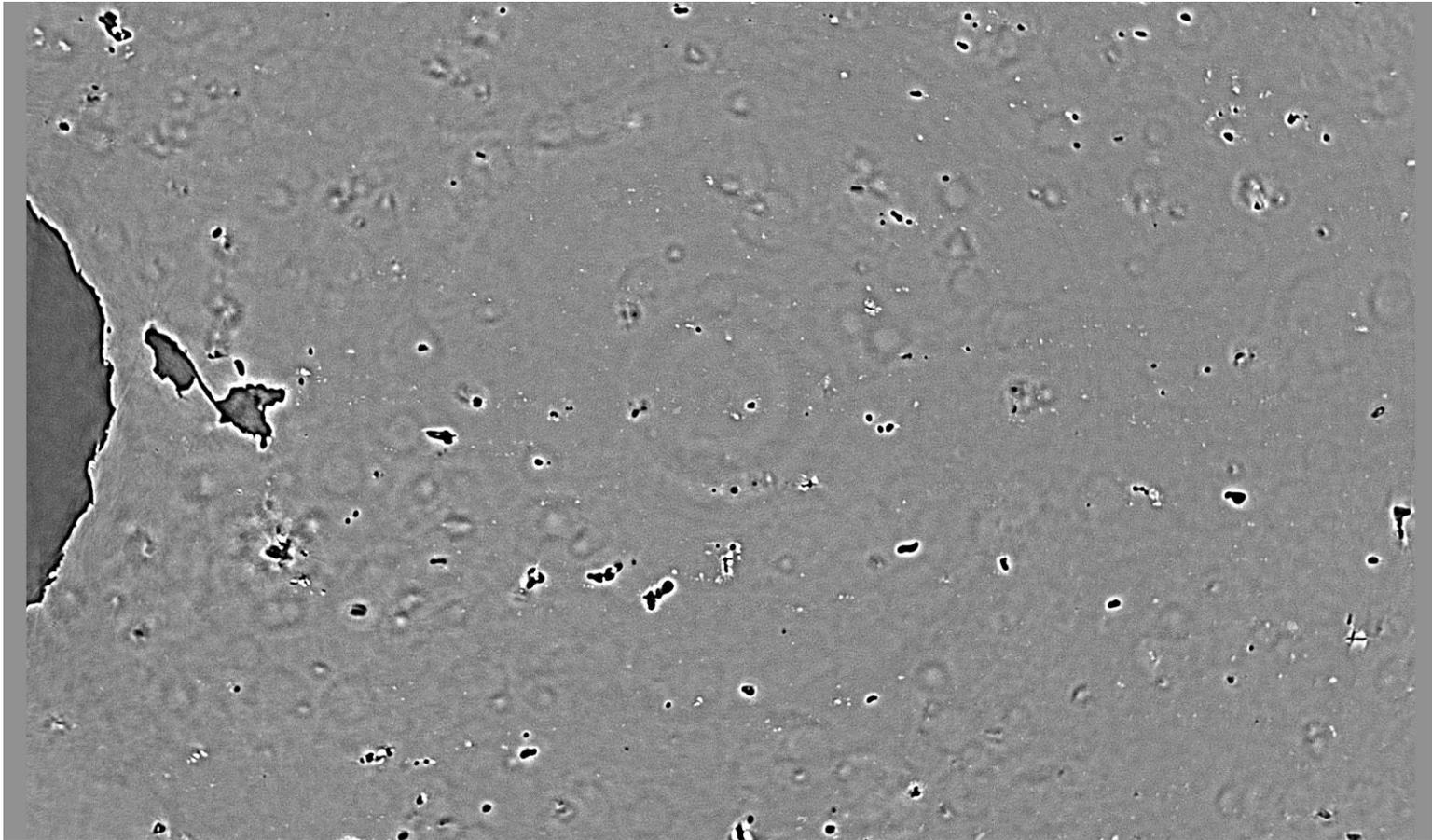
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

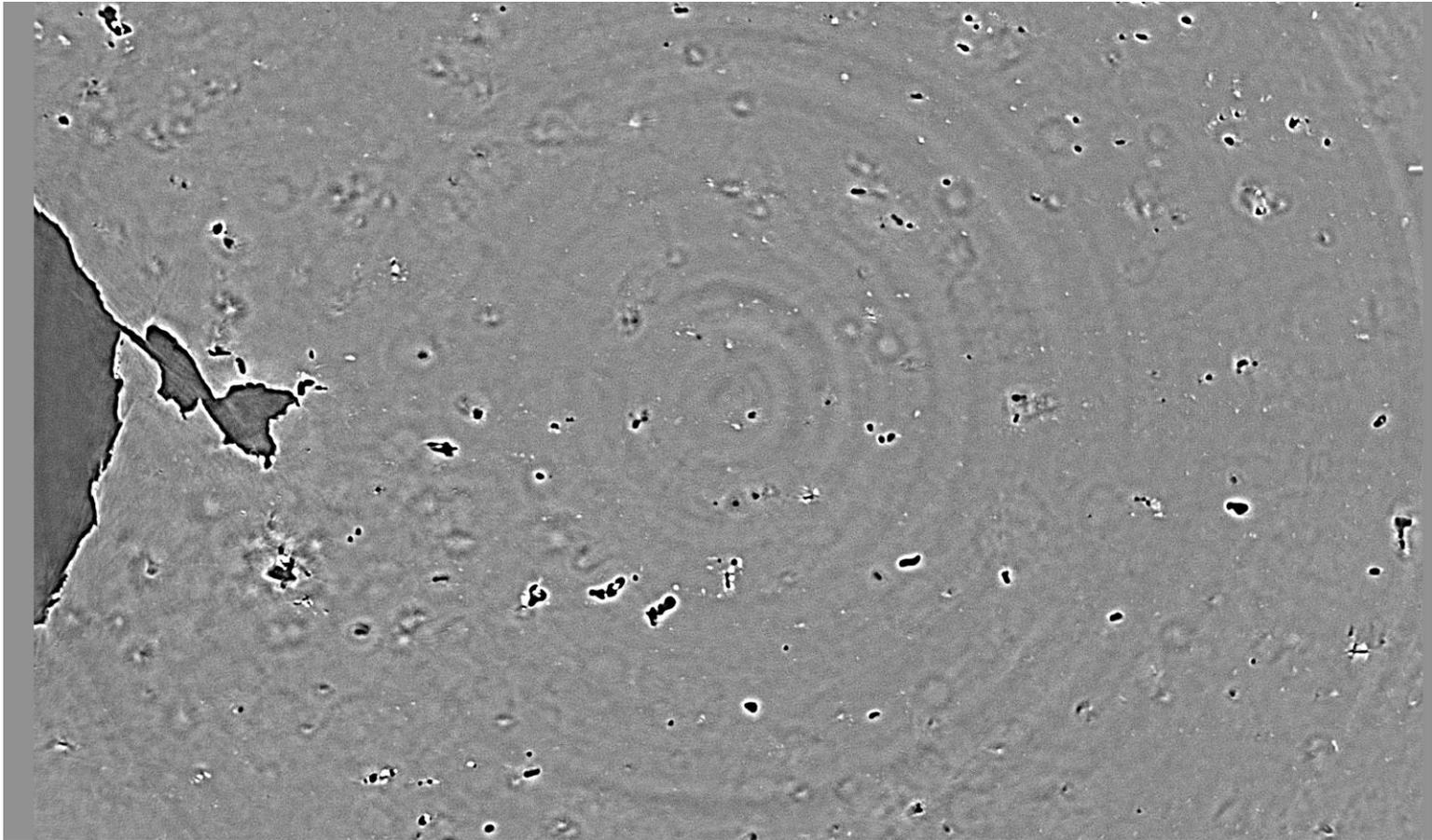
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

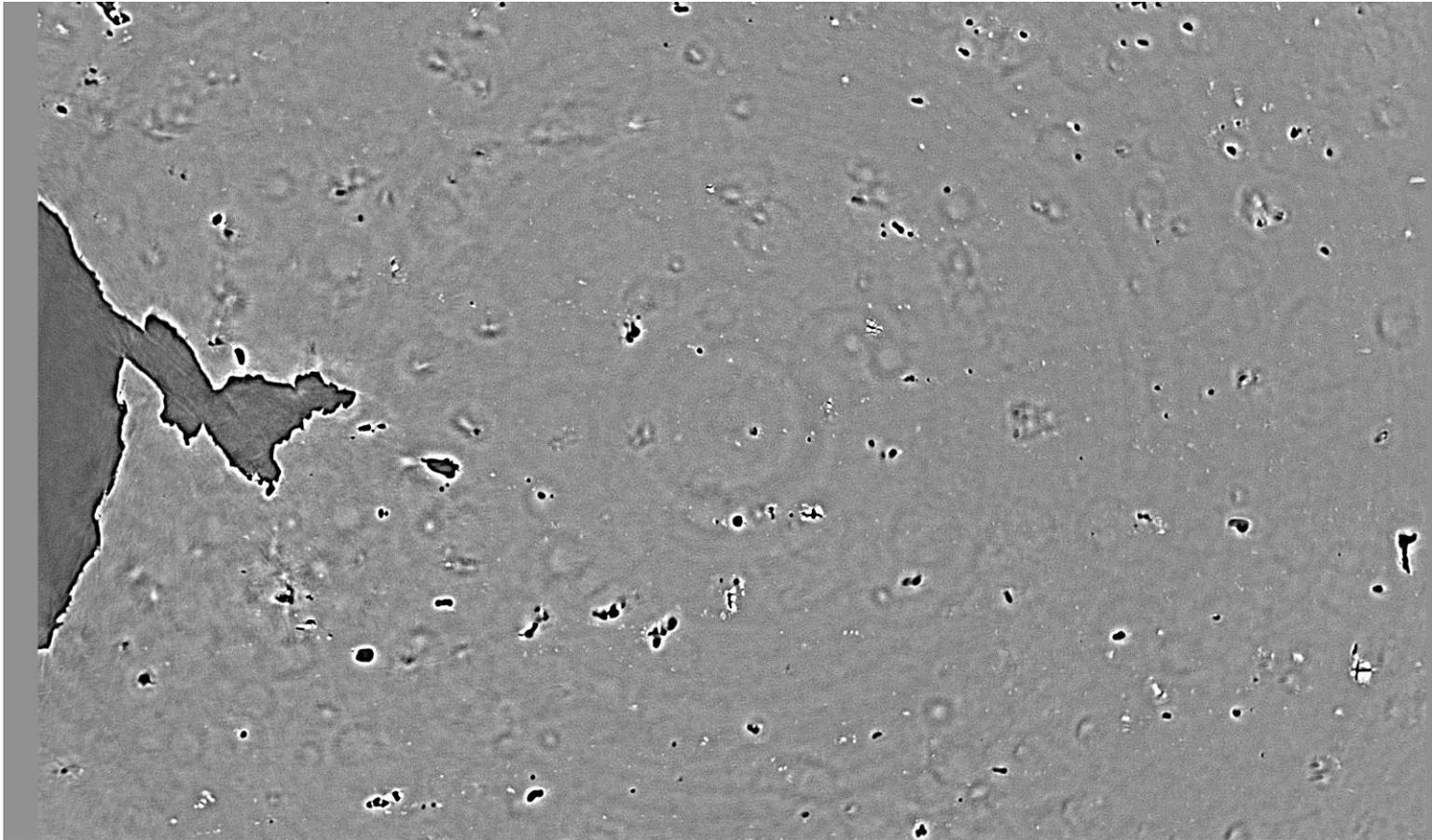
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

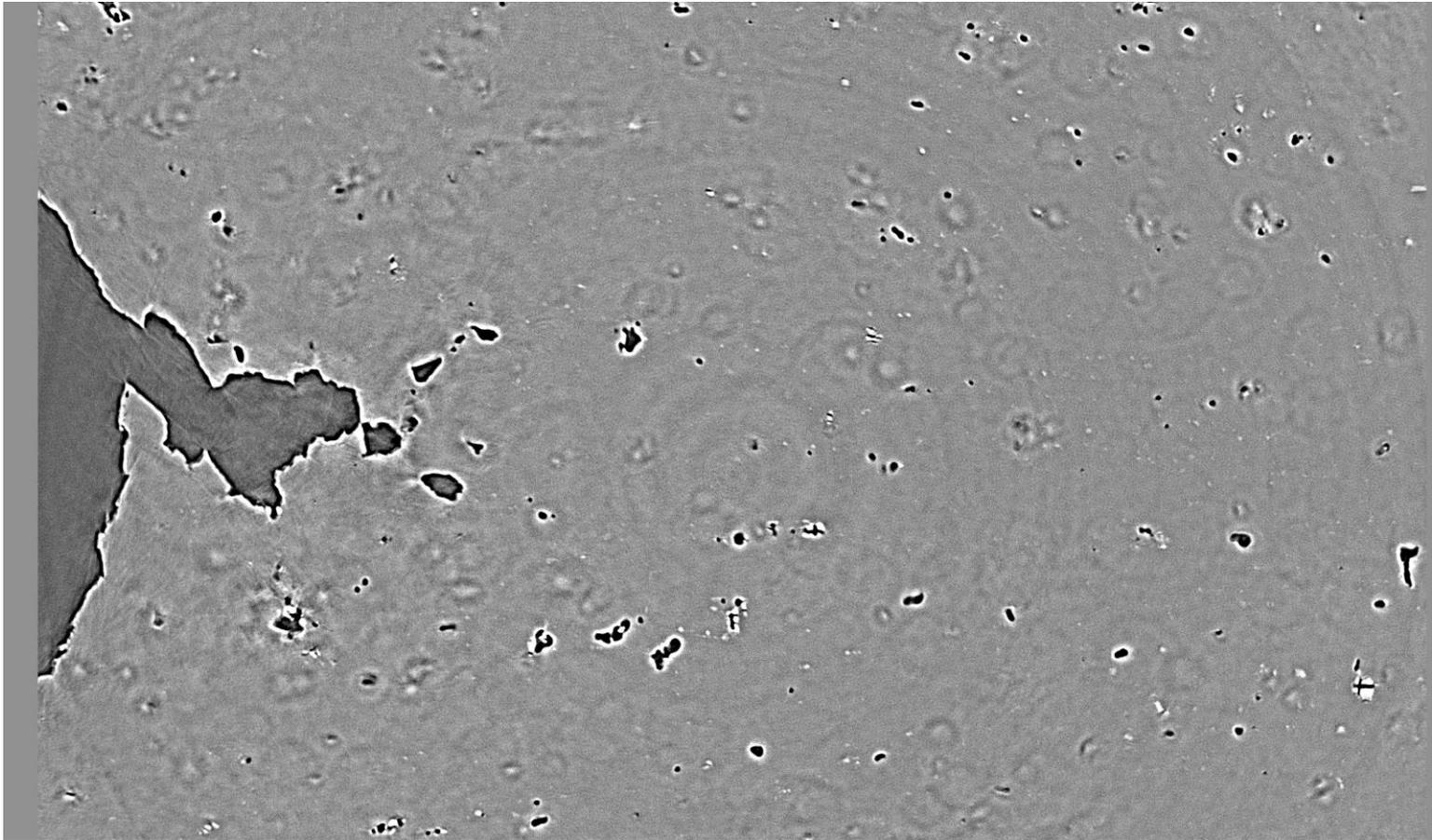
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

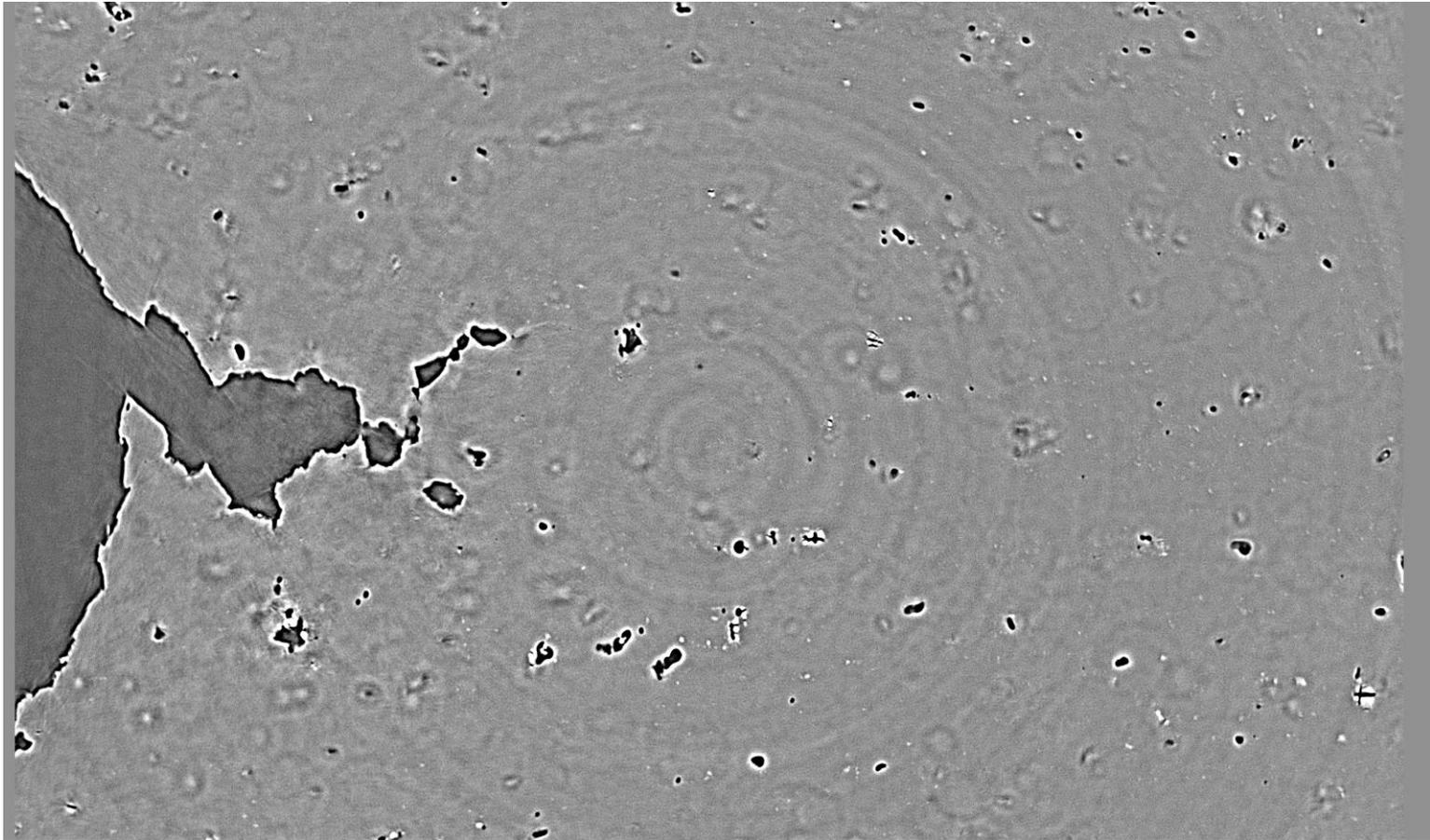
- 2139 T3: 2D section at mid-thickness



200µm

In-situ SRC Laminography

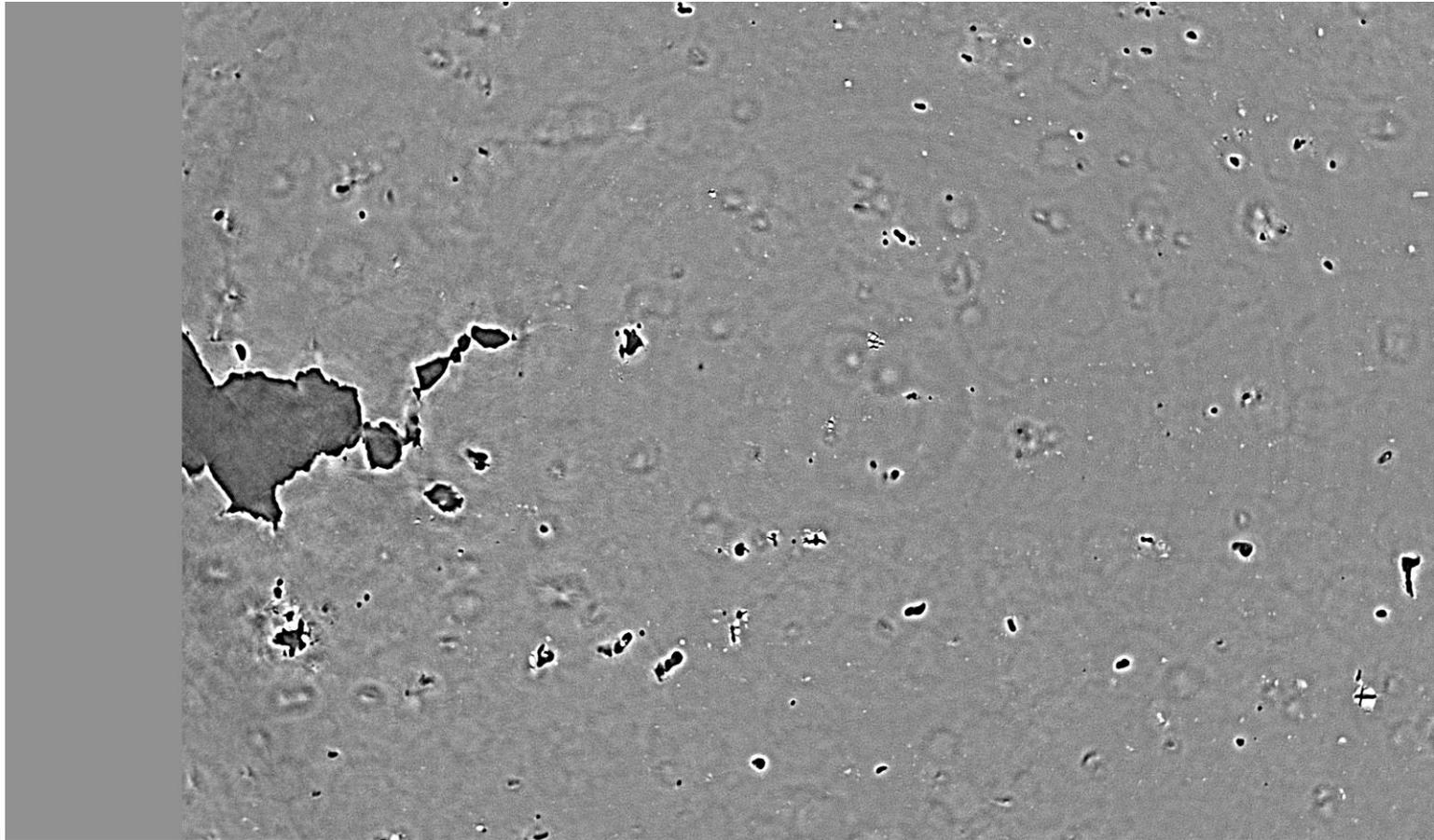
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

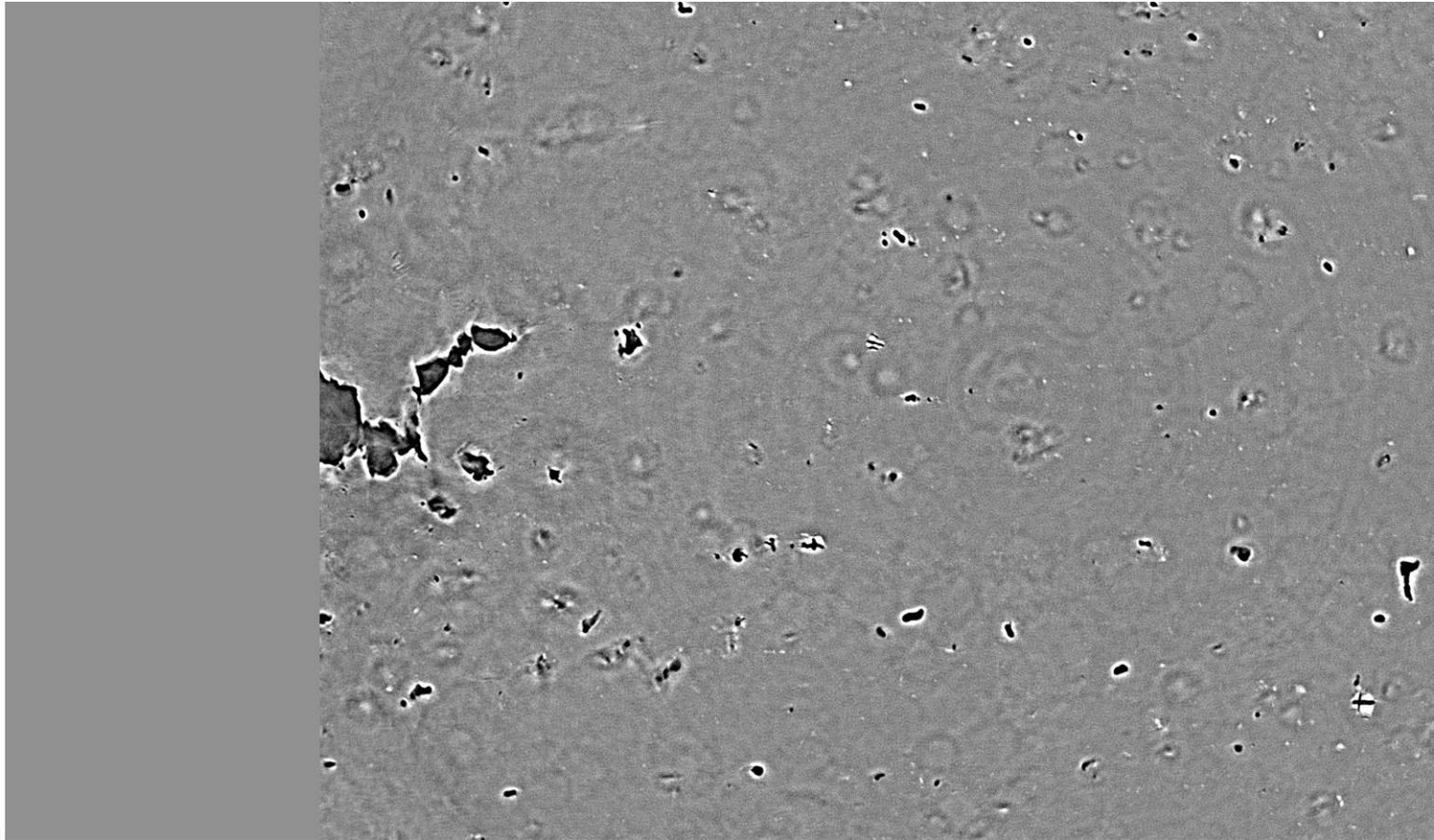
- 2139 T3: 2D section at mid-thickness



200μm

In-situ SRCLaminography

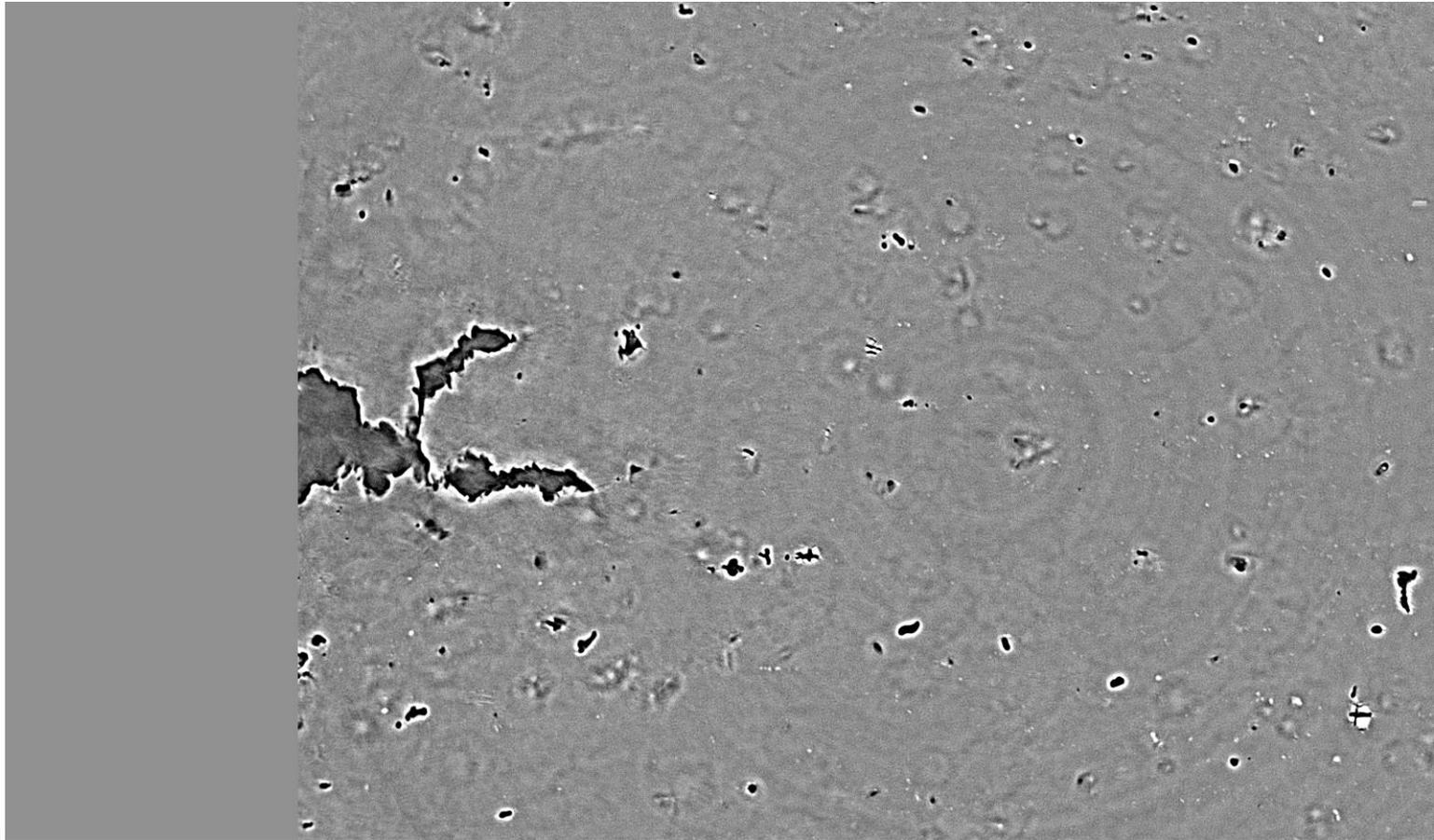
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

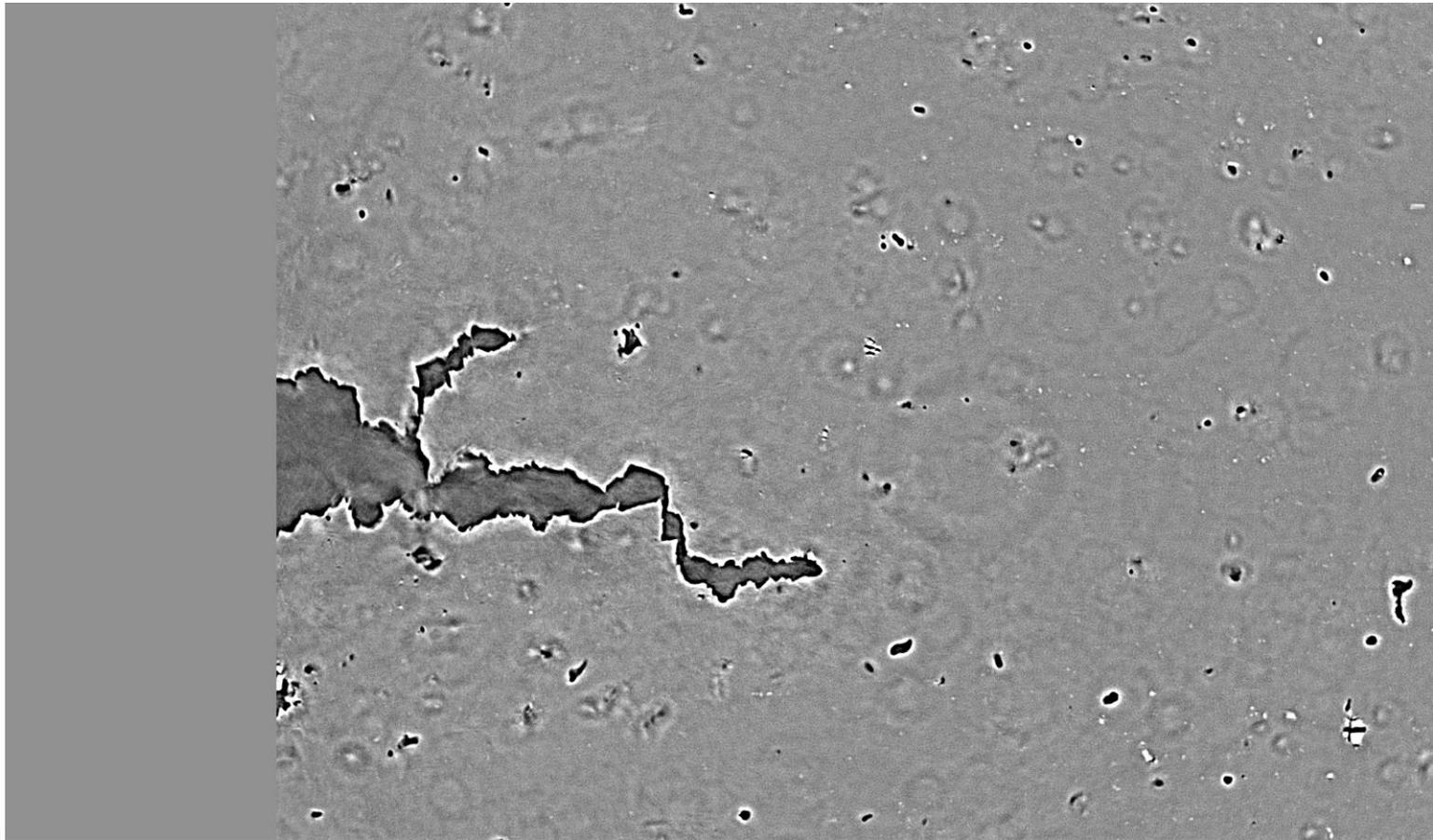
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

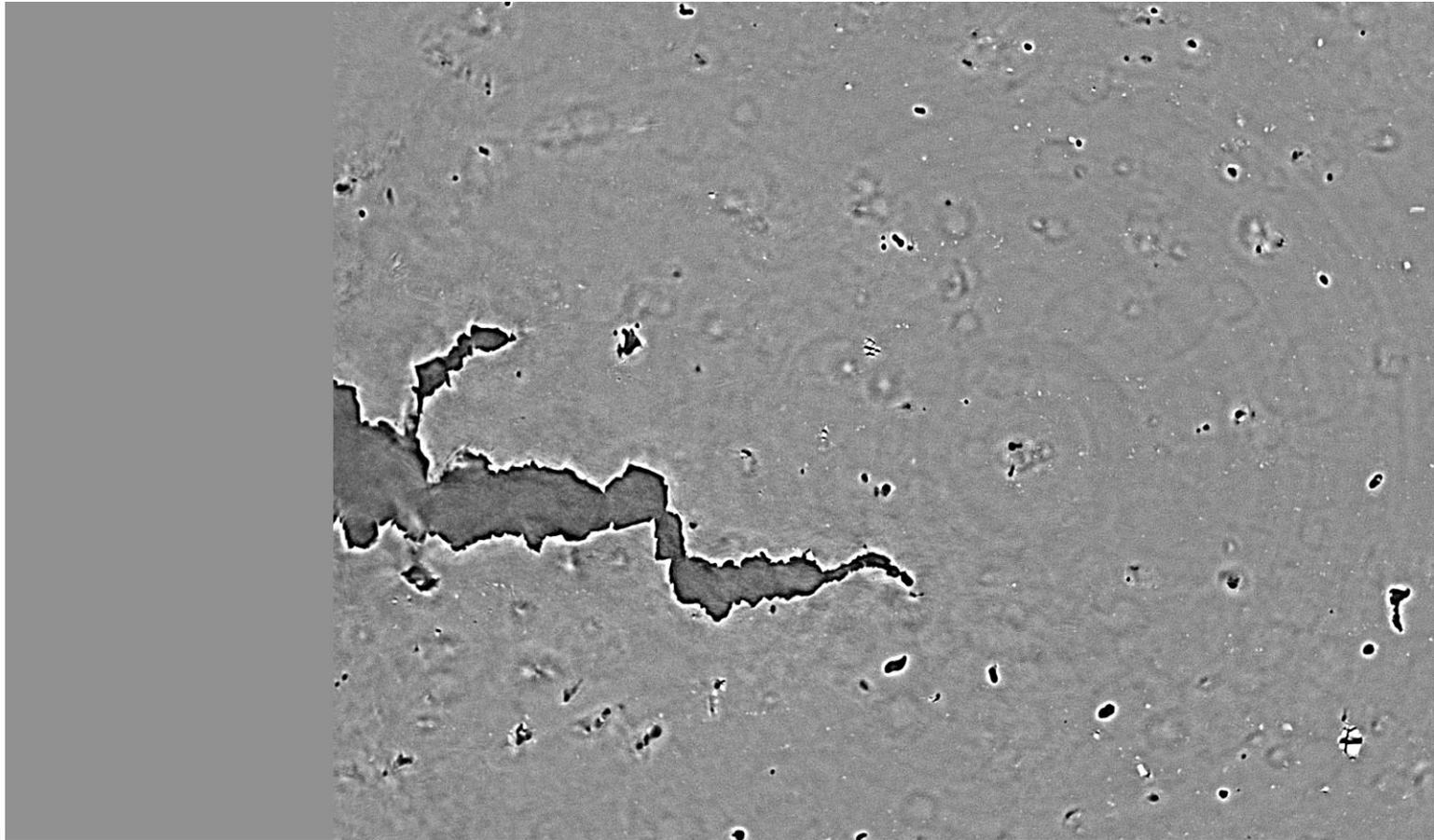
- 2139 T3: 2D section at mid-thickness



200μm

In-situ SRC Laminography

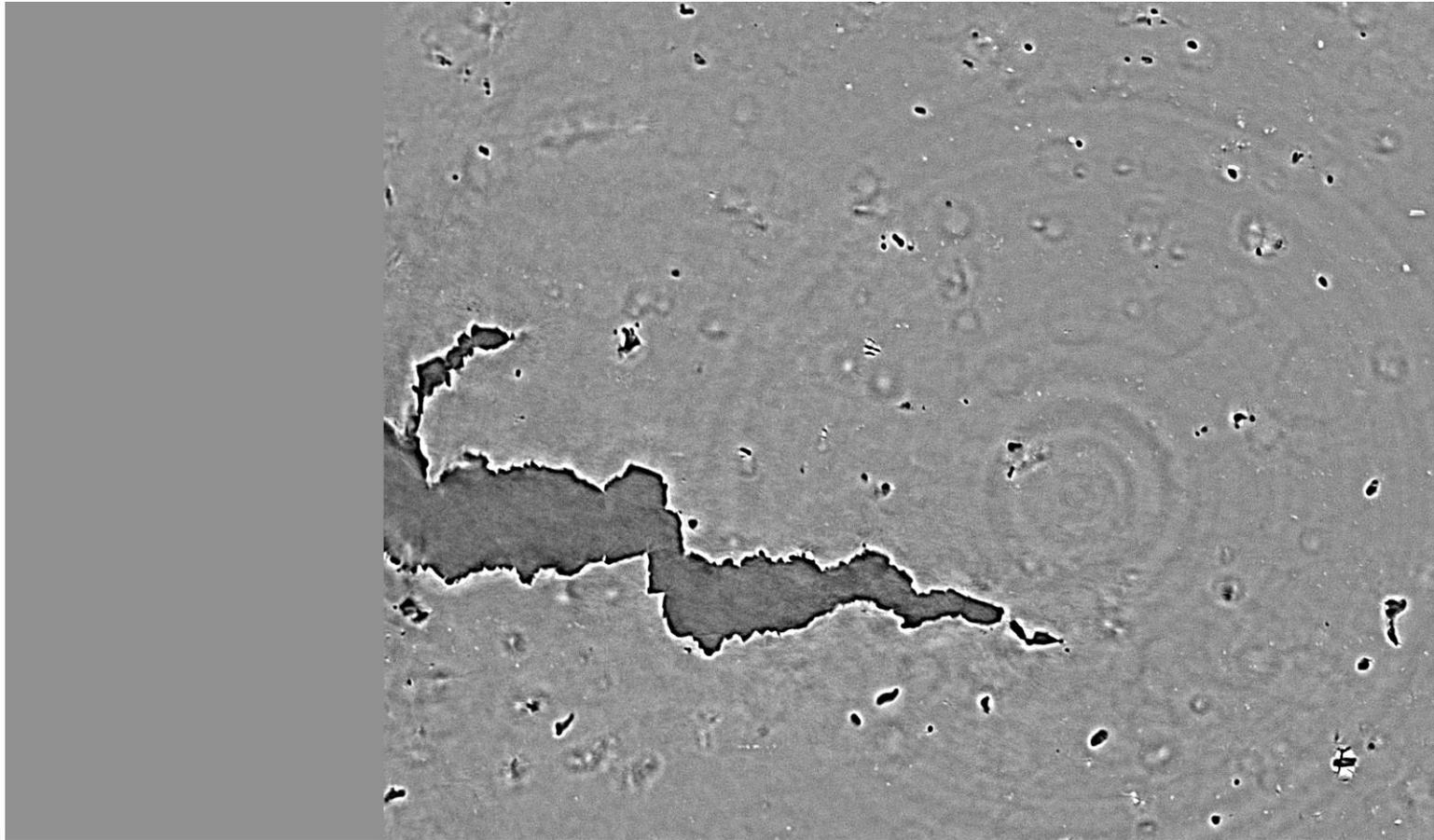
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

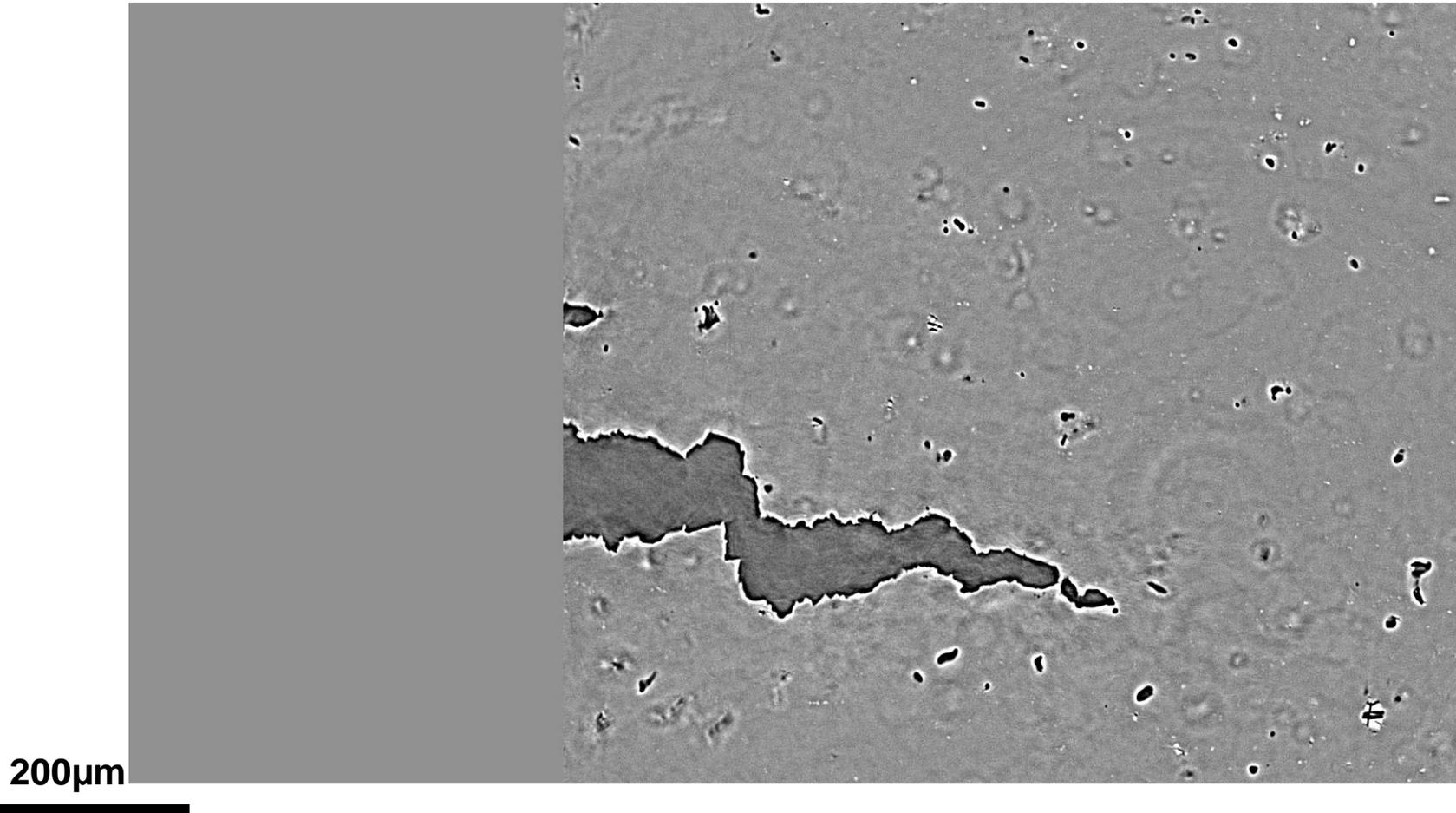
- 2139 T3: 2D section at mid-thickness



200 μ m

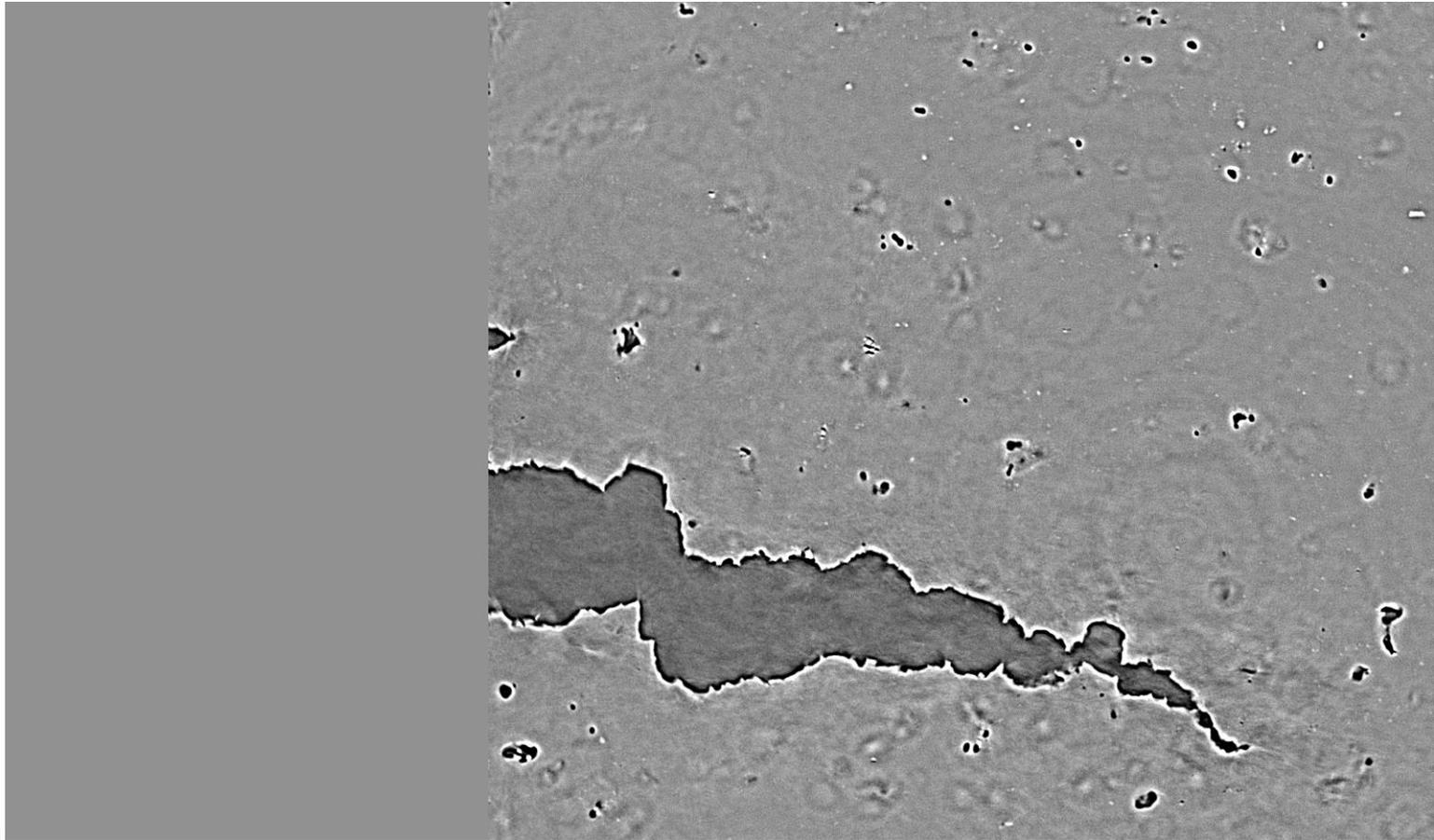
In-situ SRC Laminography

- 2139 T3: 2D section at mid-thickness



In-situ SRC Laminography

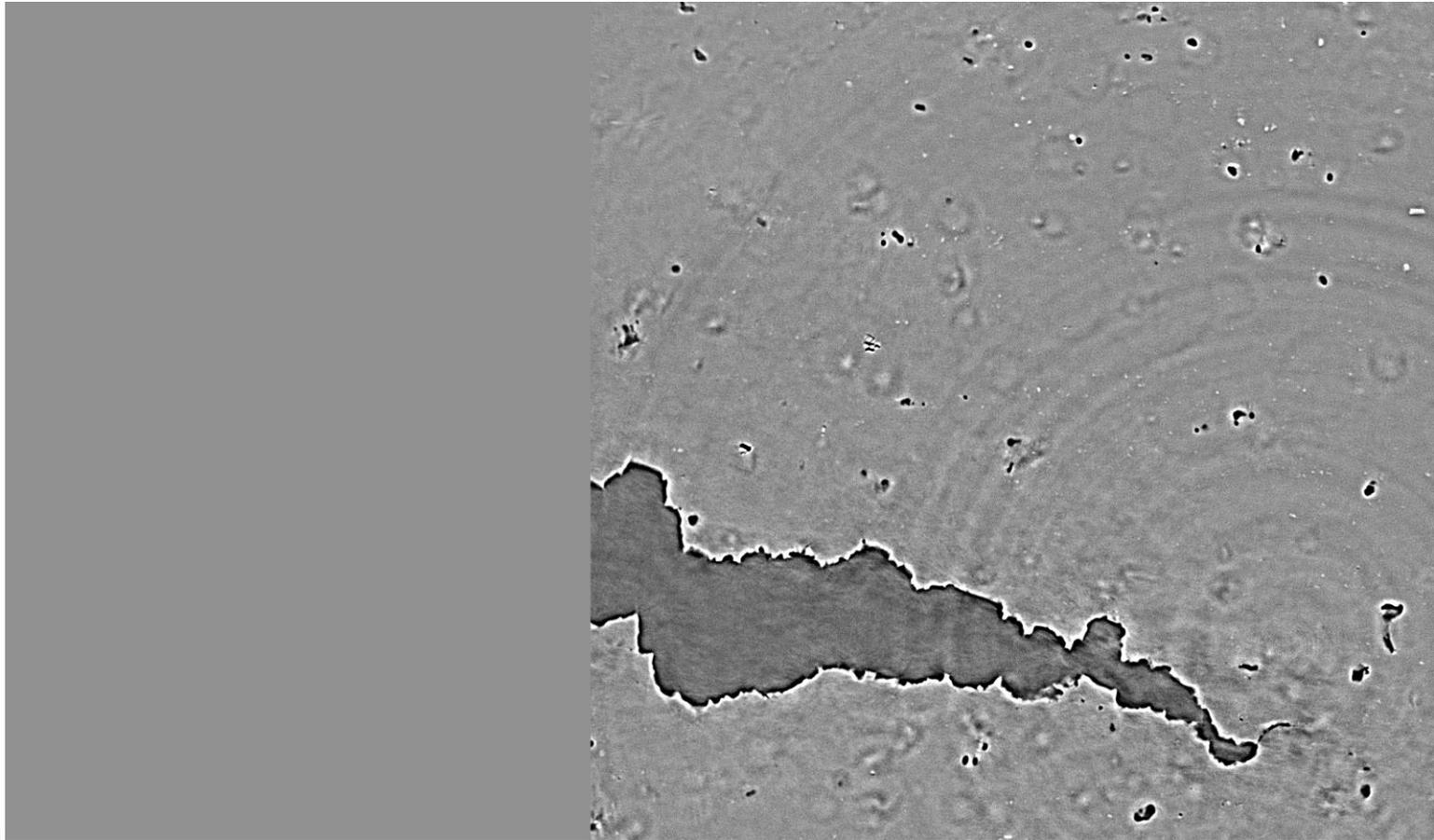
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRC Laminography

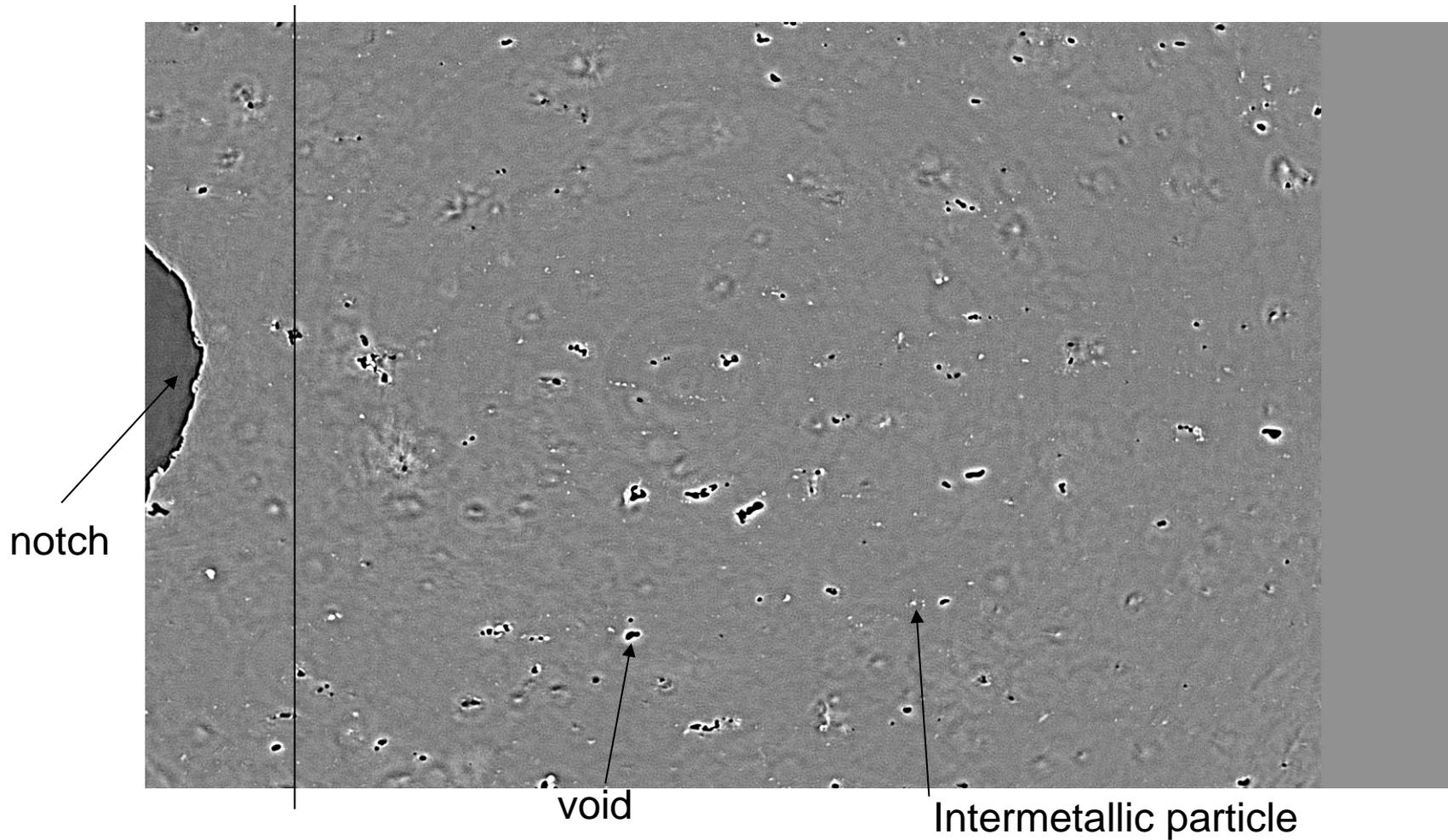
- 2139 T3: 2D section at mid-thickness



200 μ m

In-situ SRCLaminography

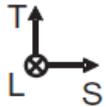
- 2D section at mid-thickness



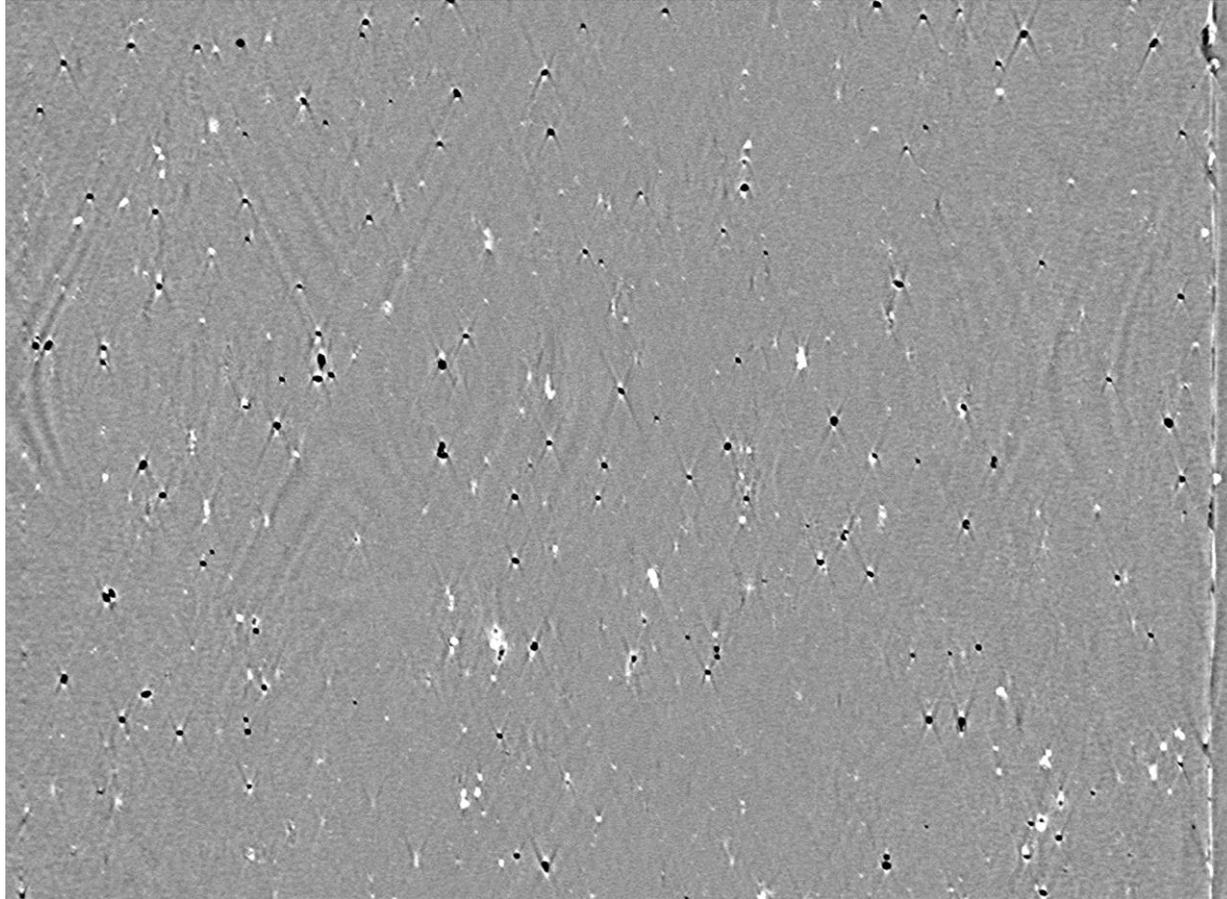
In-situ SRC Laminography

- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch
 - Laminography aretacts become visible

⊗ Crack
growth

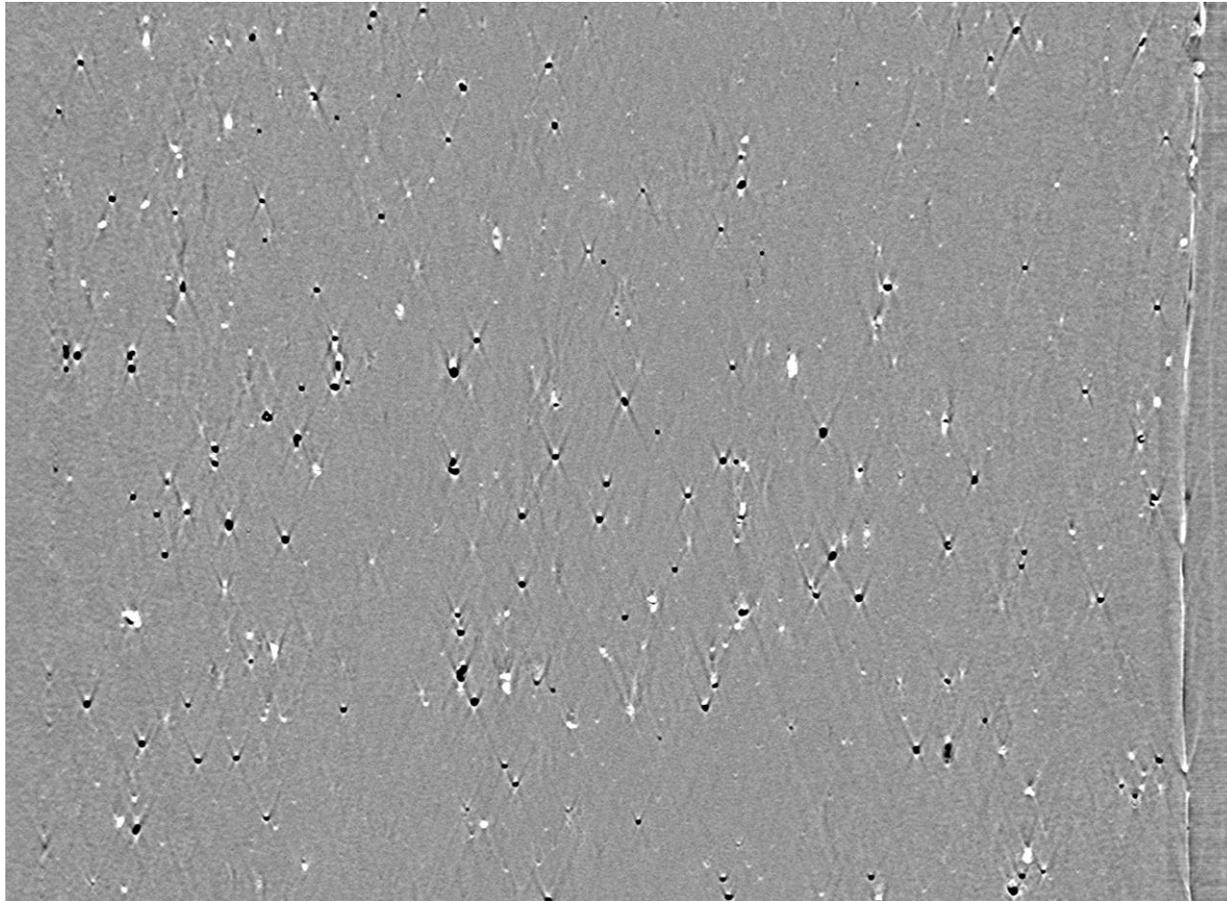


200 μm



In-situ SRC Laminography

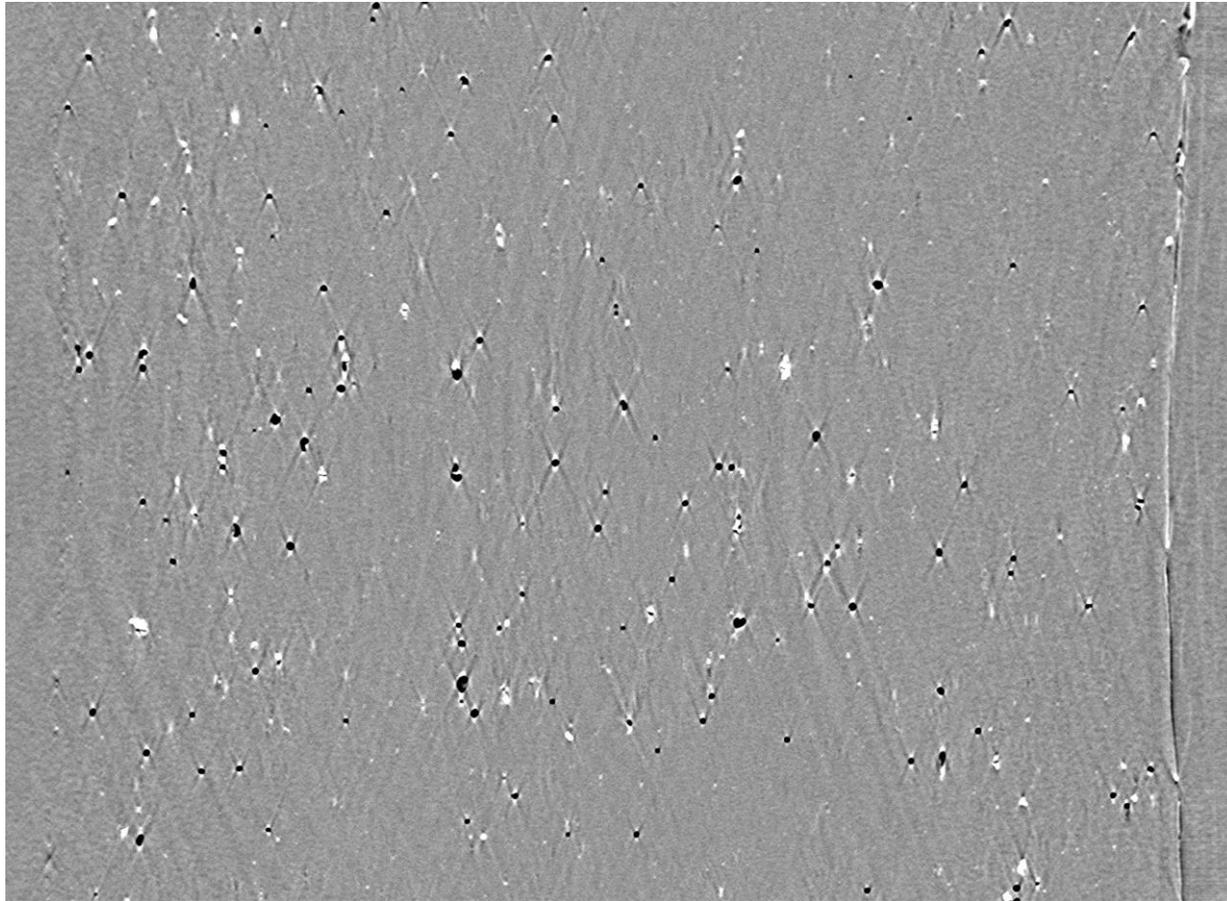
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

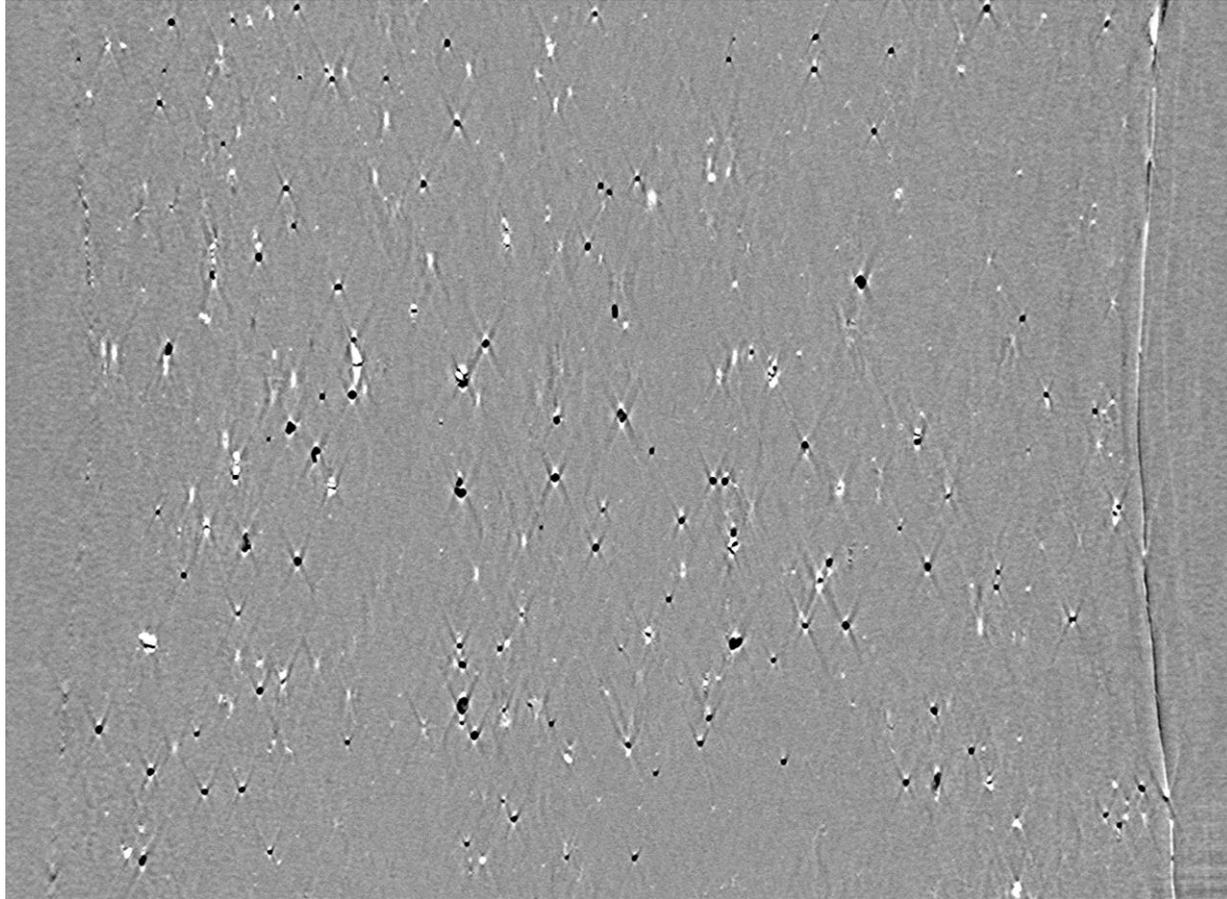
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

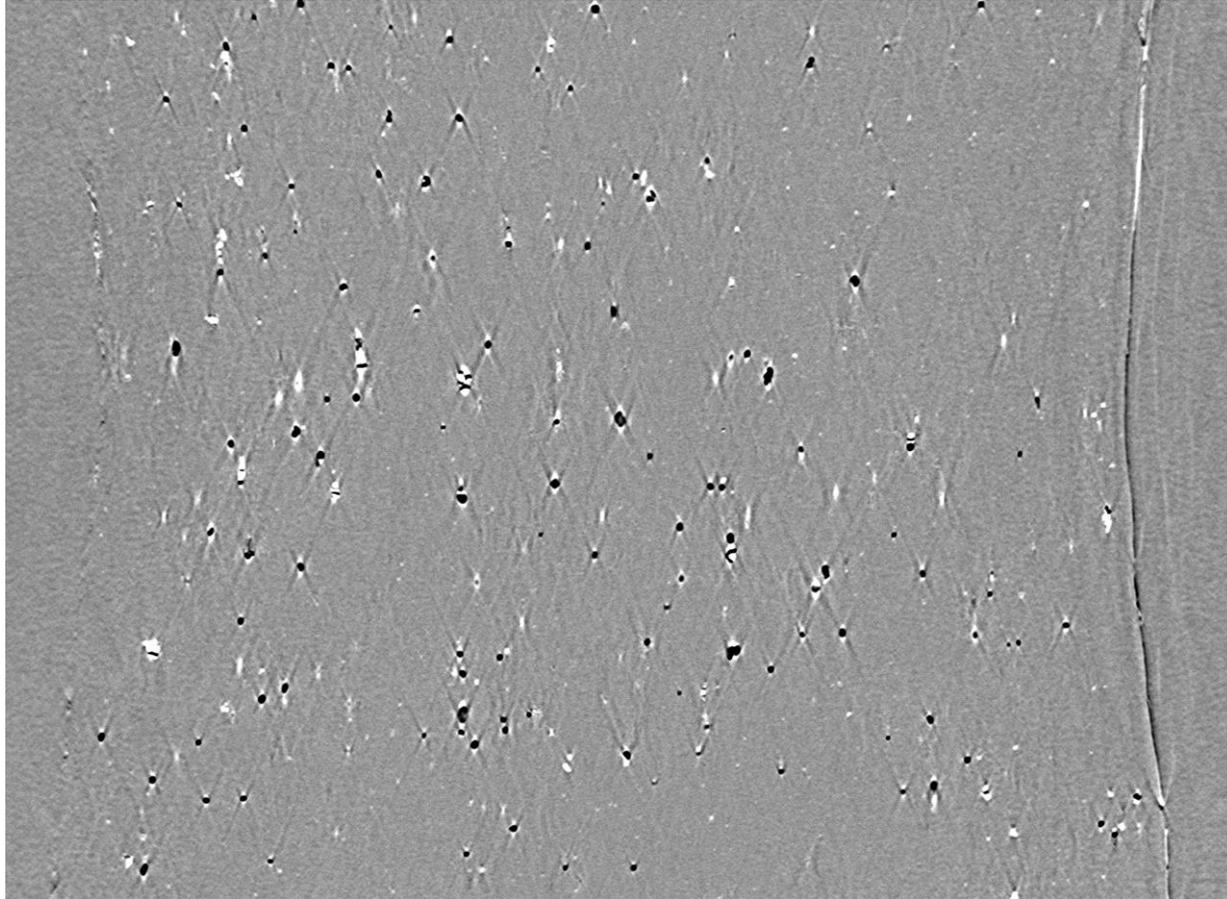
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

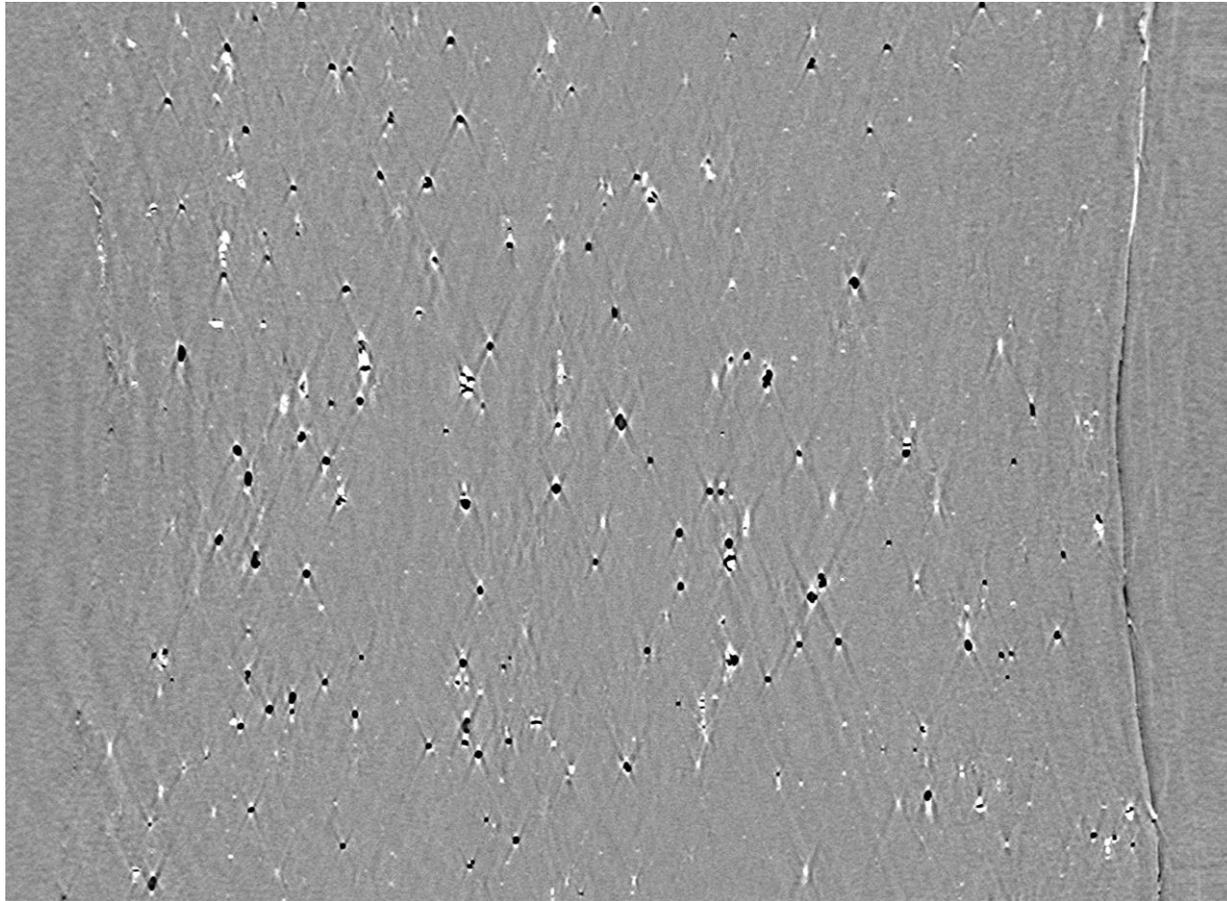
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

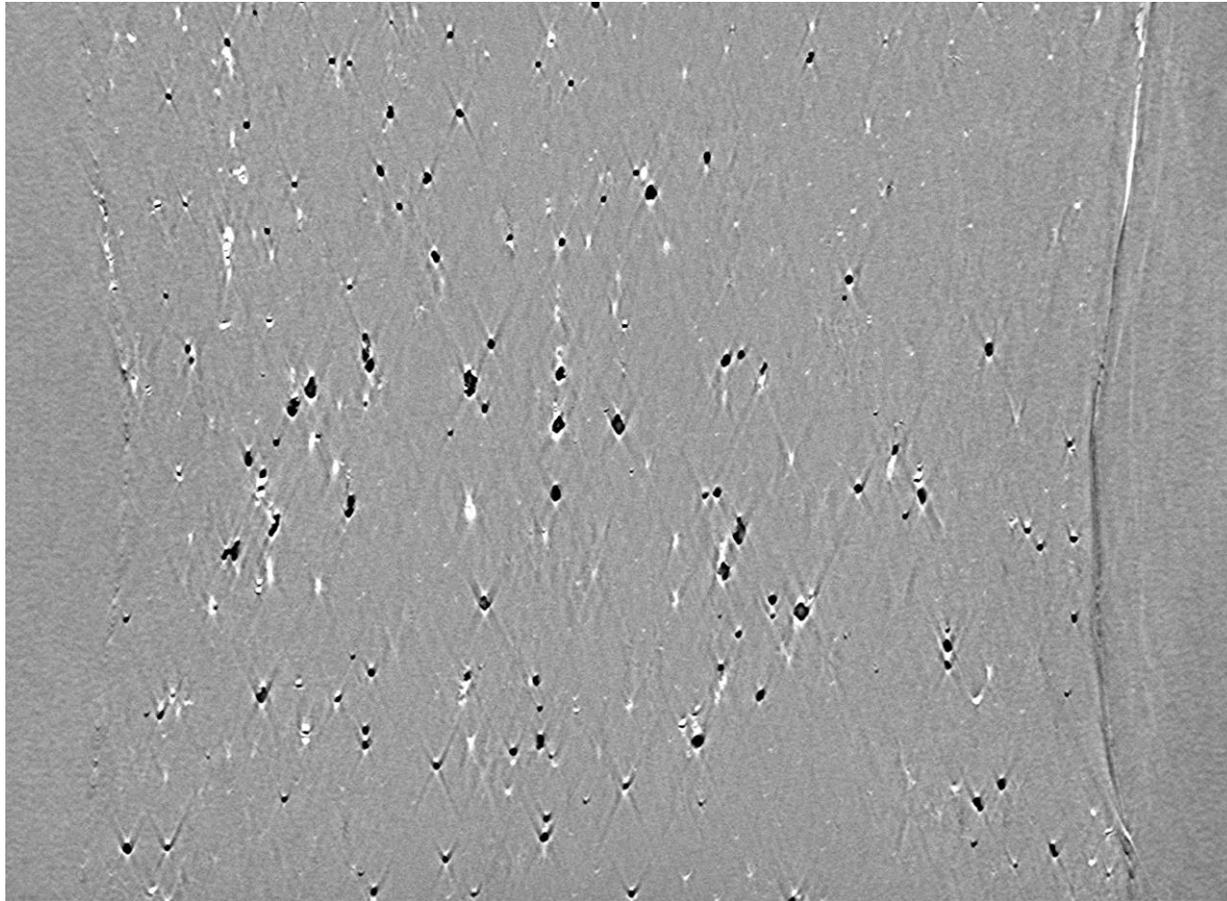
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

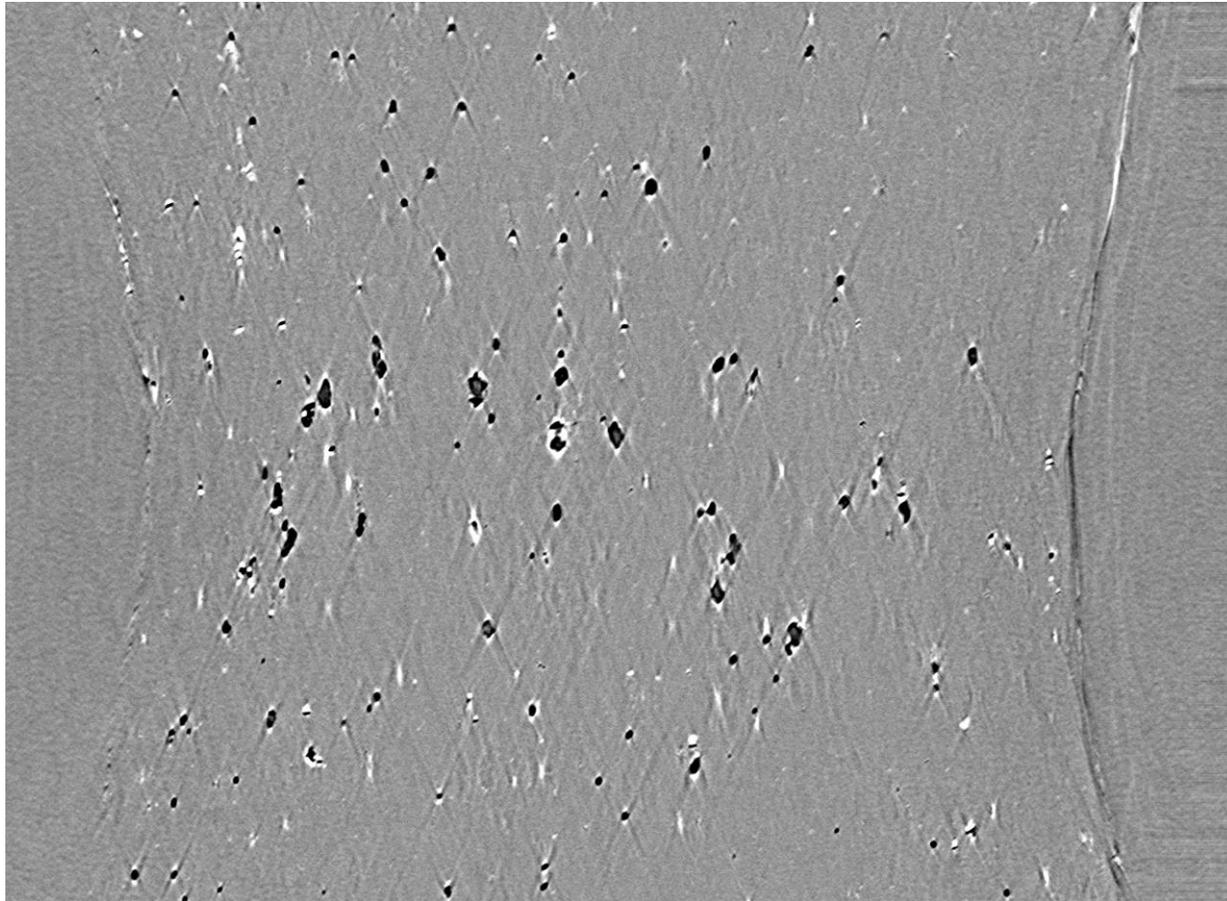
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

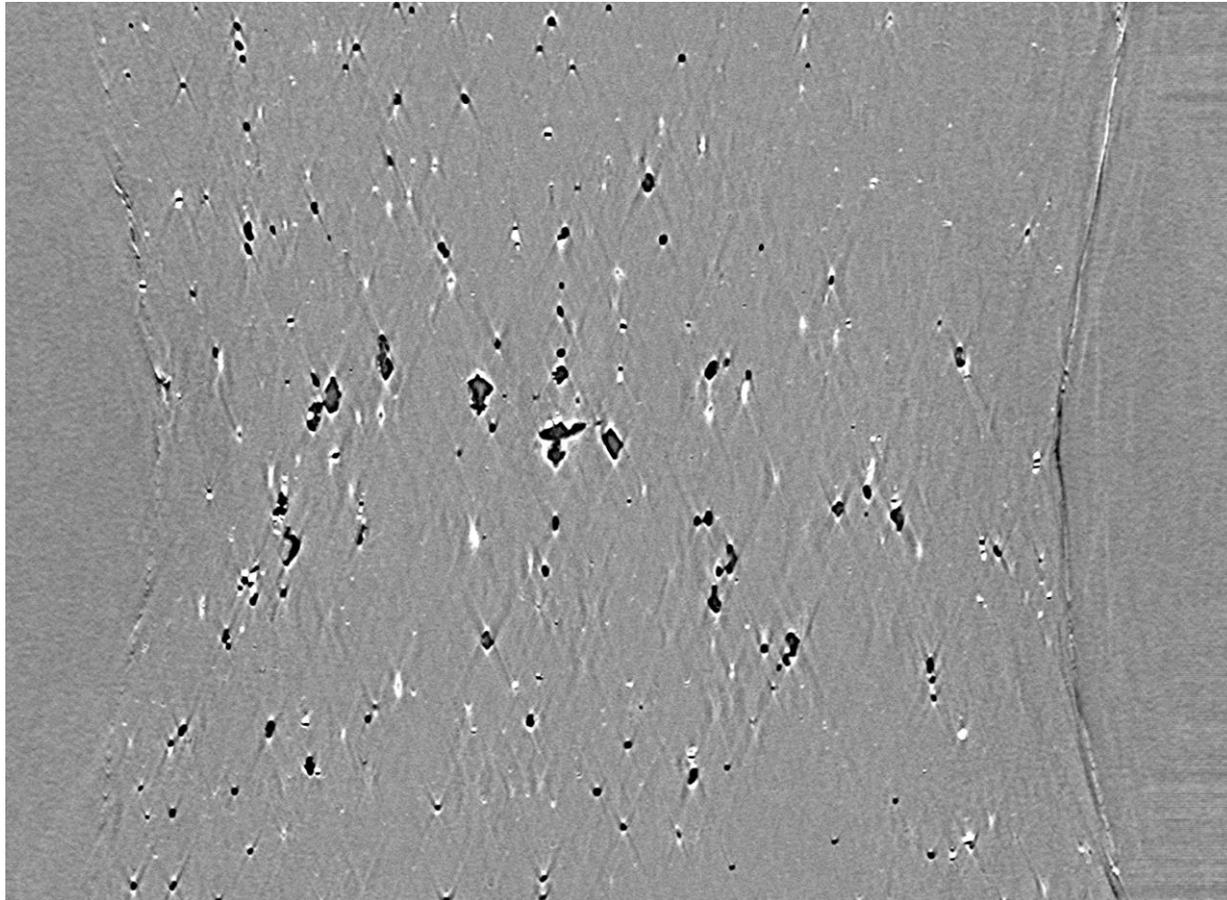
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

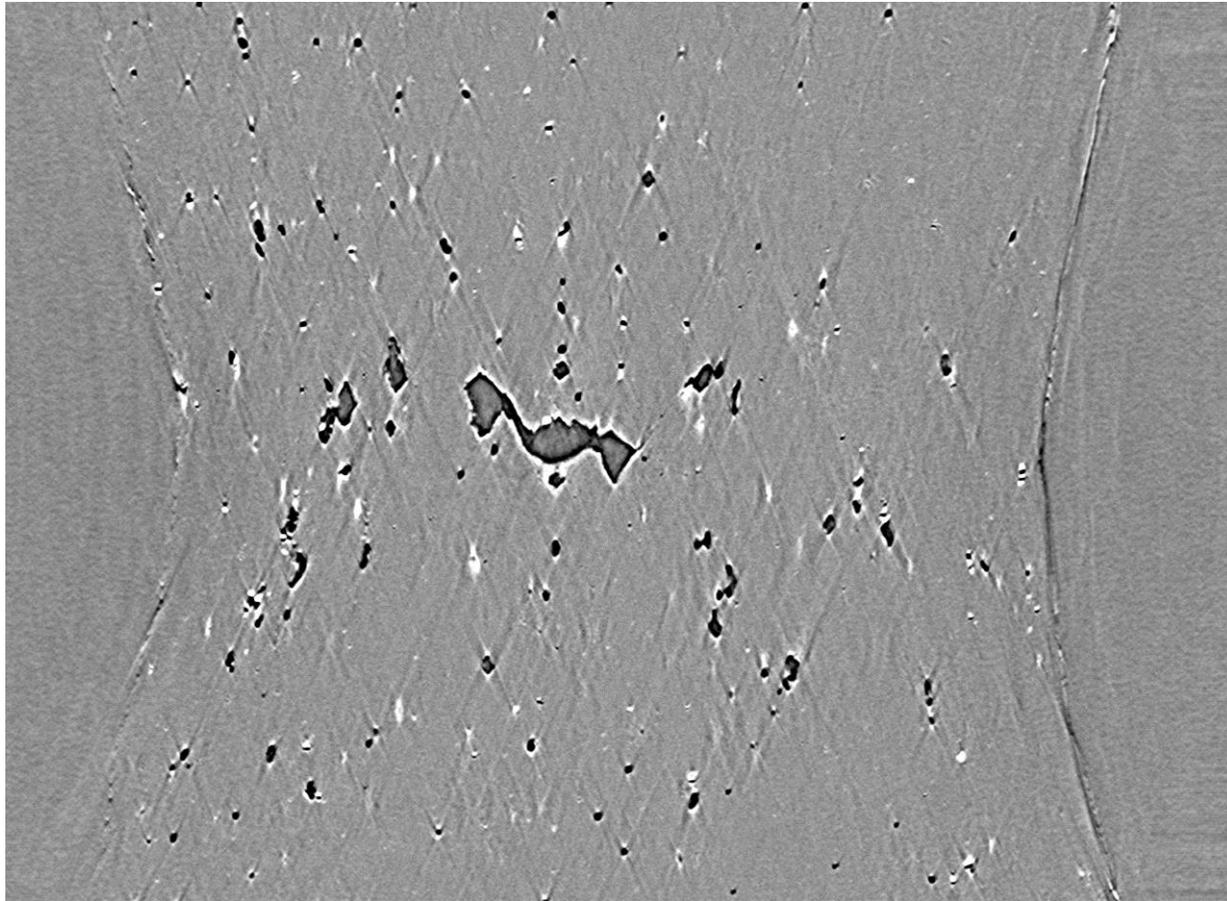
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

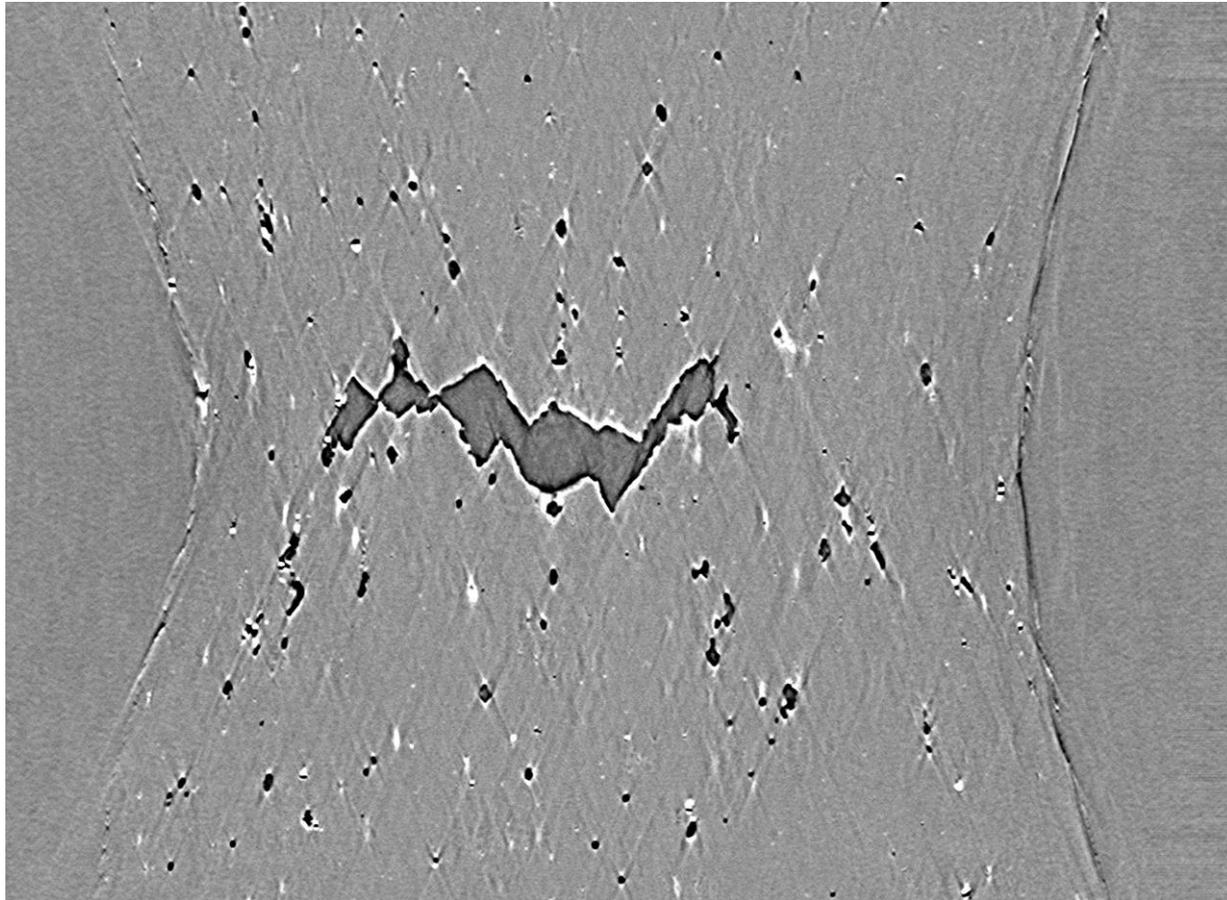
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

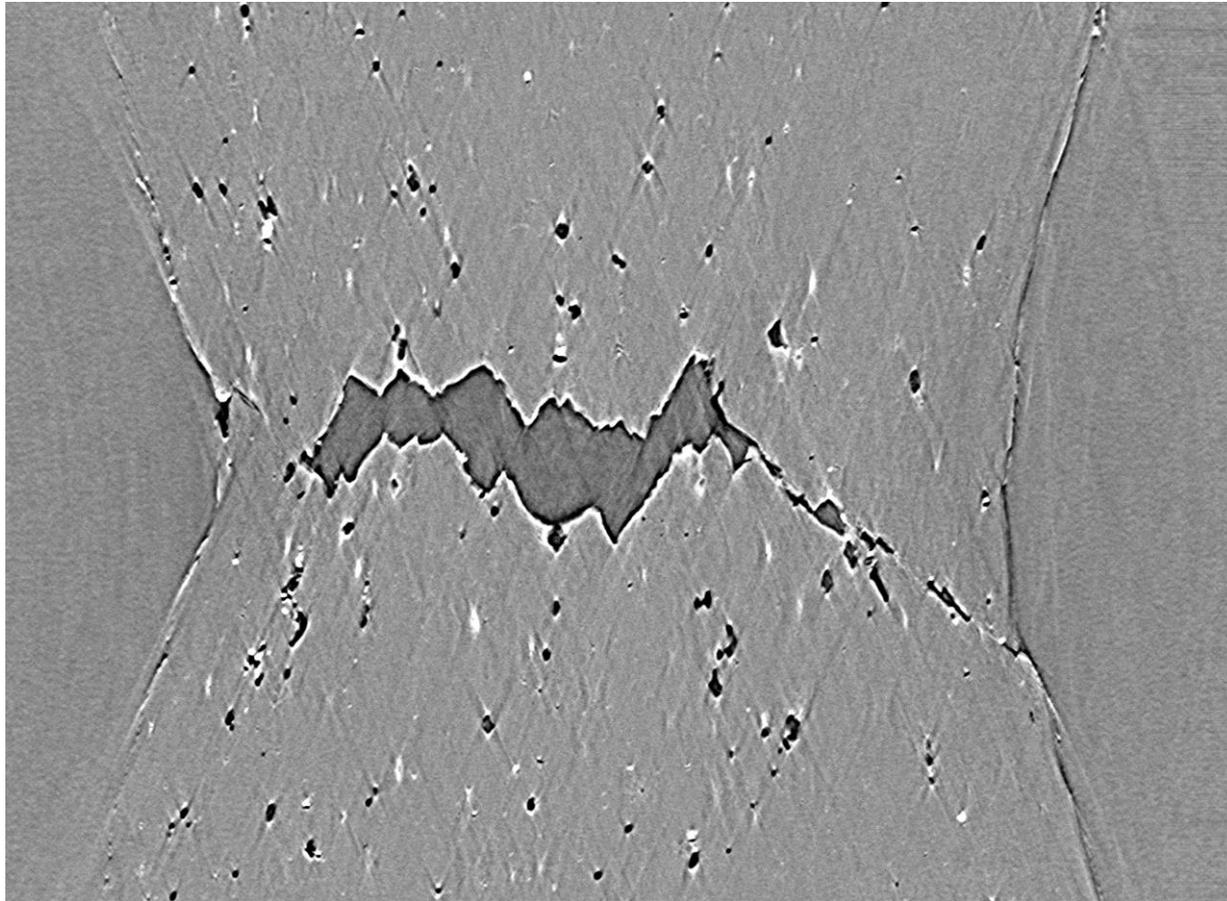
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200 μm

In-situ SRC Laminography

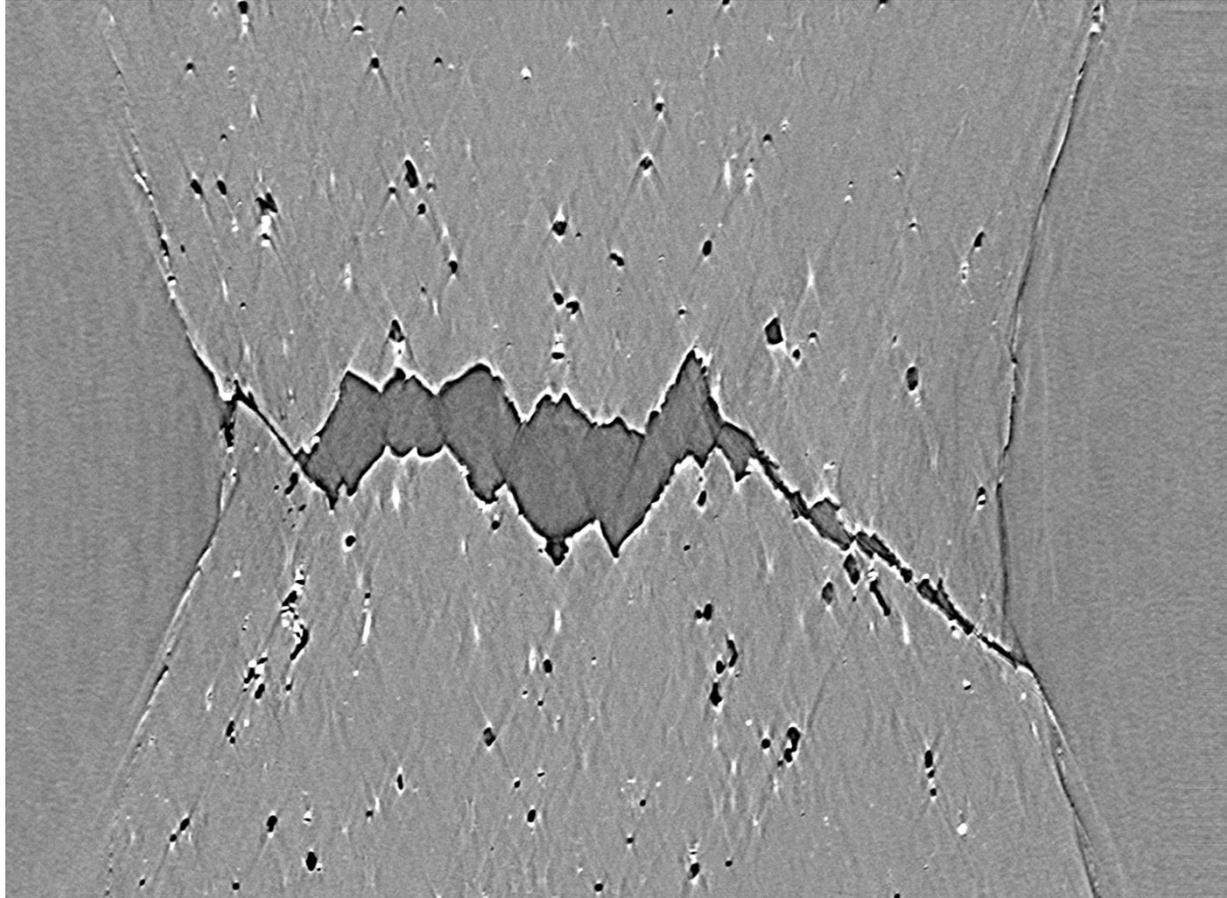
- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch



200μm

In-situ SRC Laminography

- 2D view in propagation direction $\sim 50\mu\text{m}$ ahead of notch

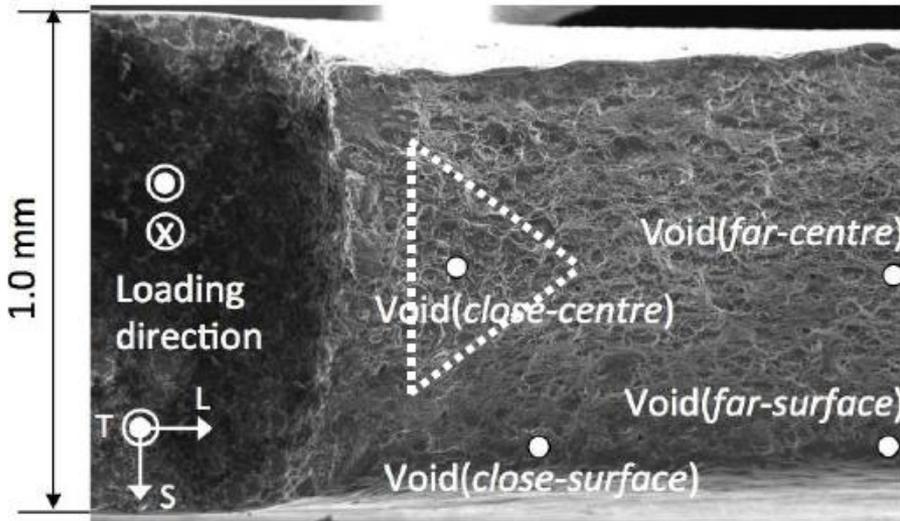


200 μm

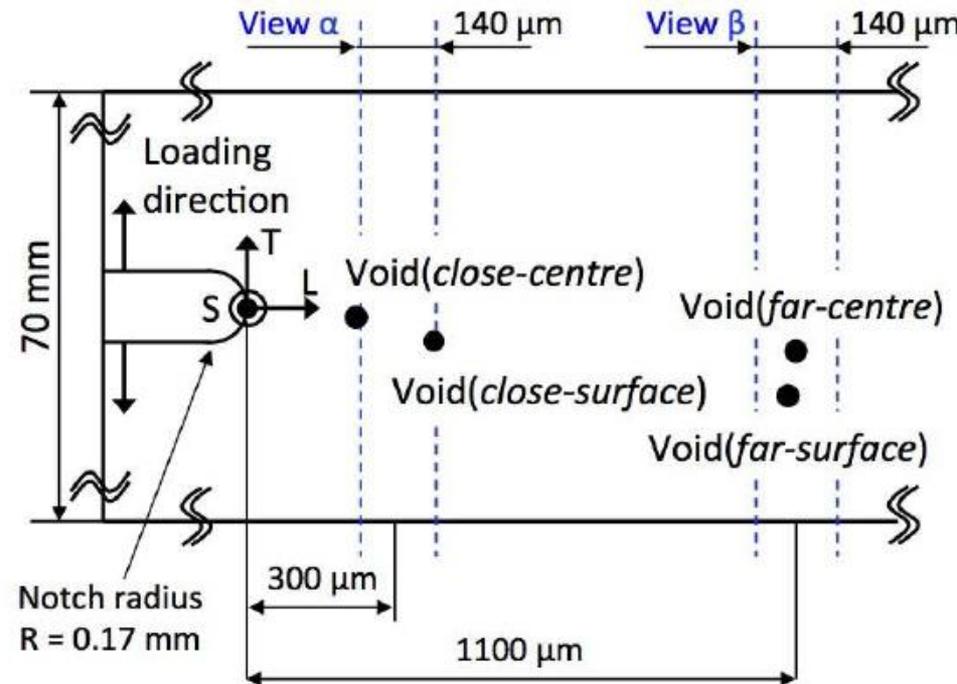


Zoom: Choice of voids

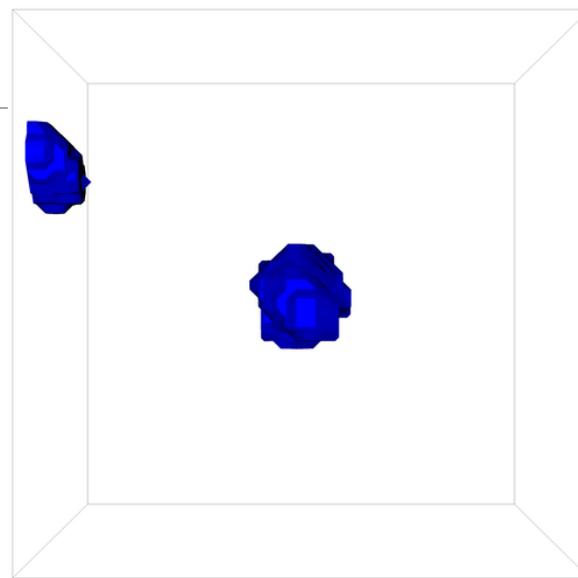
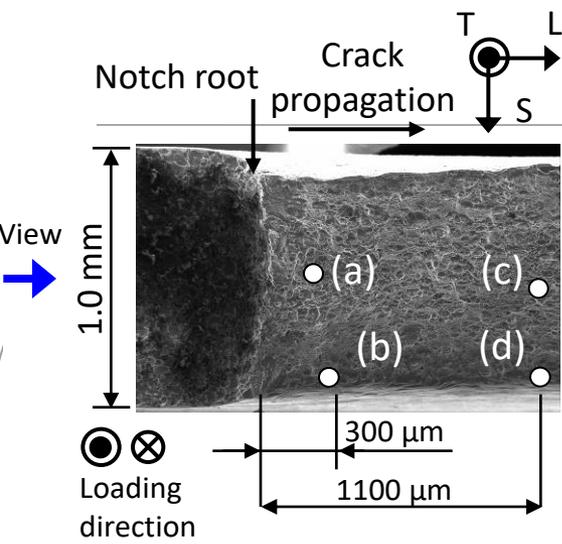
- 4 voids that will contribute to the final crack
- Investigate: shape change
orientation change
how they contribution to final crack



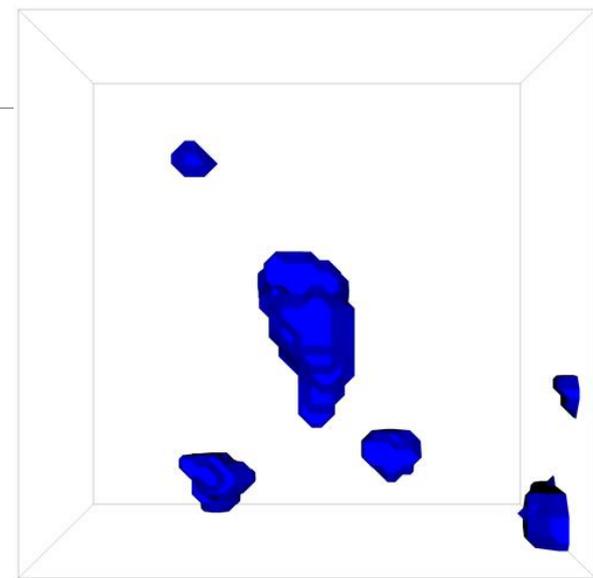
(a) L-S configuration



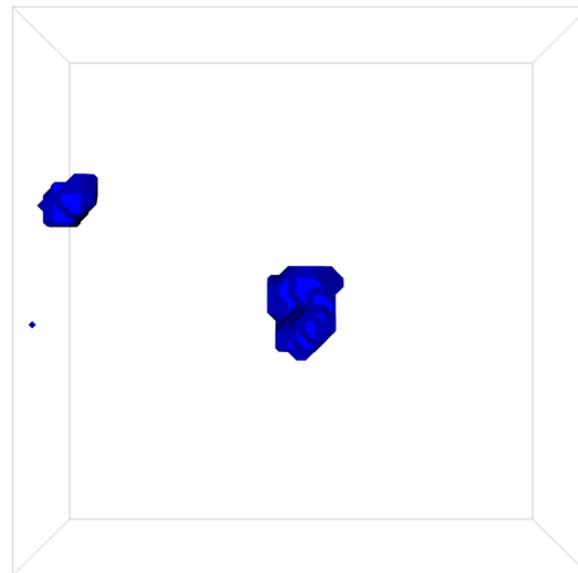
(b) L-T configuration



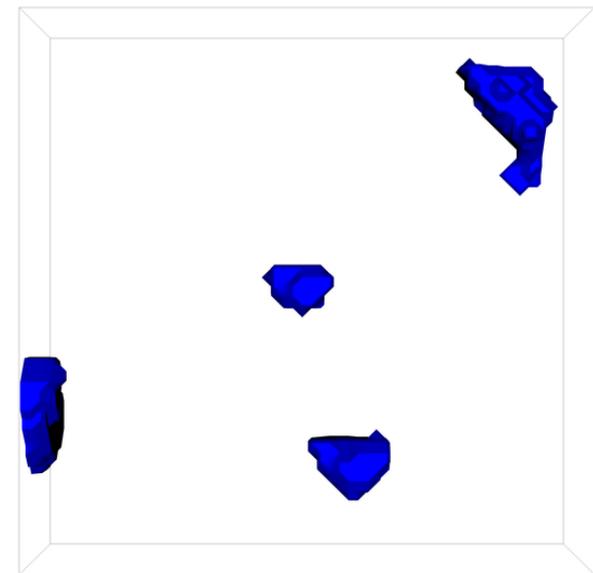
(a) Void(close-centre)



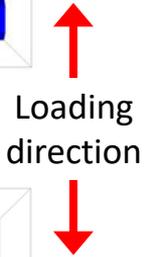
(c) Void(far-centre)



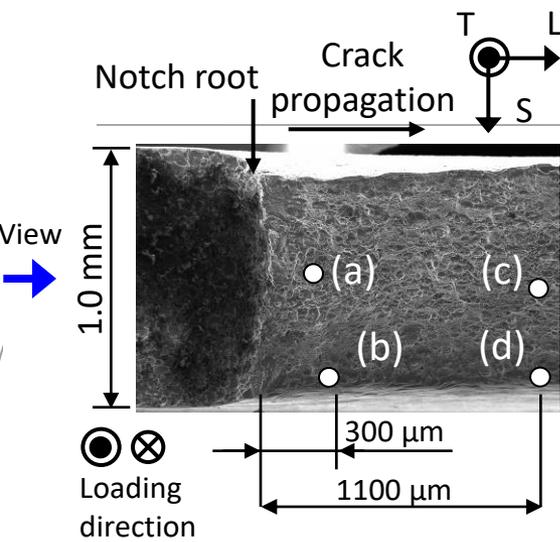
(b) Void(close-surface)



(d) Void(far-surface)

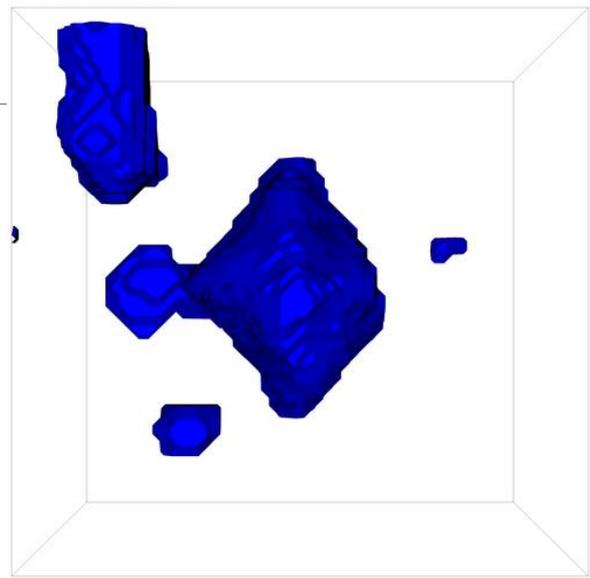


20 μm

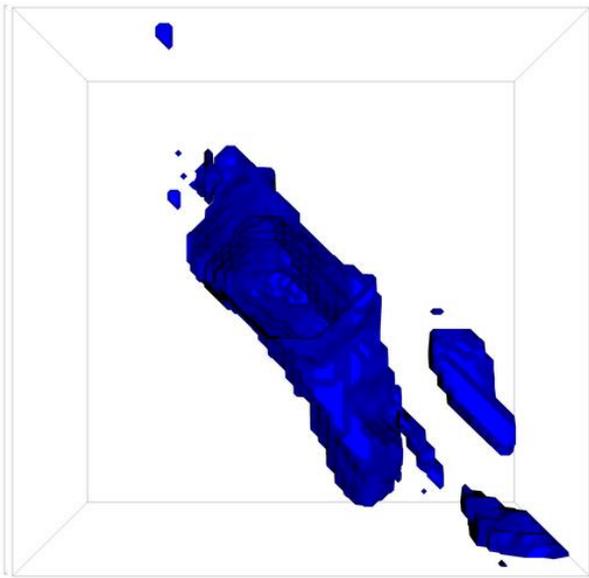


4 Voids (at different load steps) before coalescence

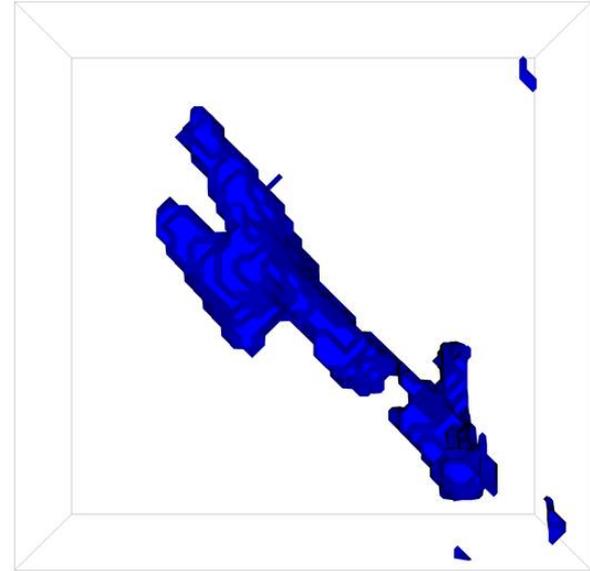
- Void in the flat region grows symmetrically
- Voids on slant region rotate and grow towards slant failure



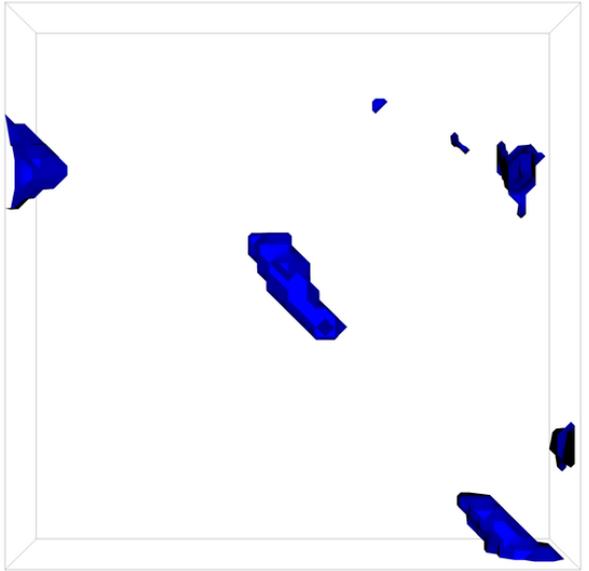
(a) Void(close-centre)



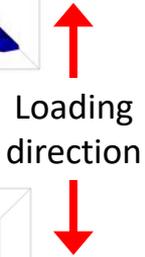
(c) Void(far-centre)



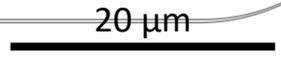
(b) Void(close-surface)



(d) Void(far-surface)



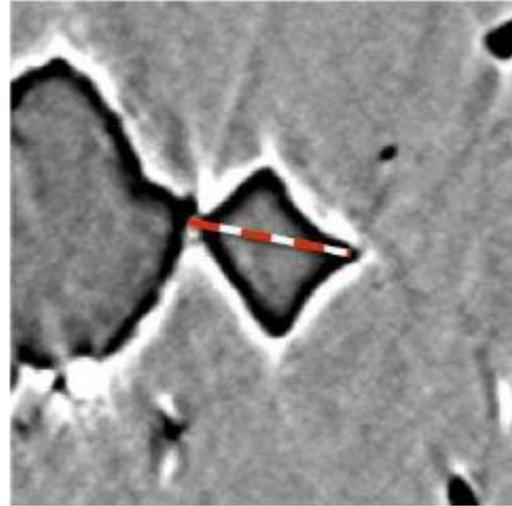
⊗ Crack Propagation



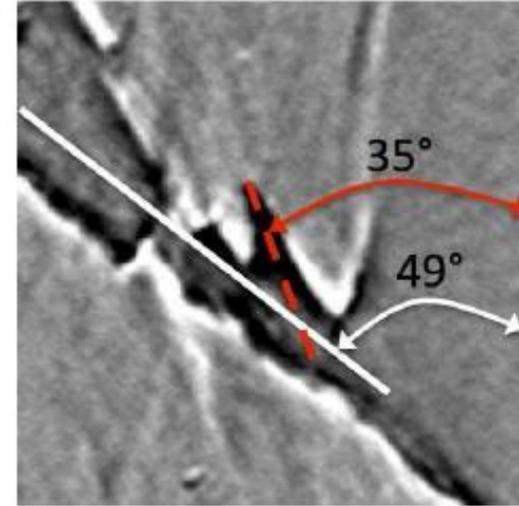
*[T. Ueda *et al.*, 2014, *Acta Mat*]

Void contribution to final fracture

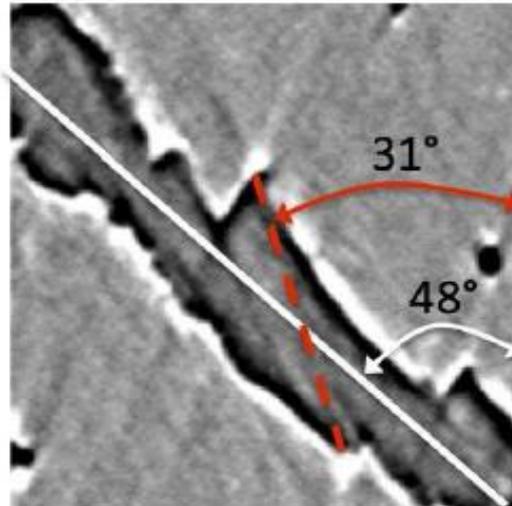
(a) Void(close-centre)



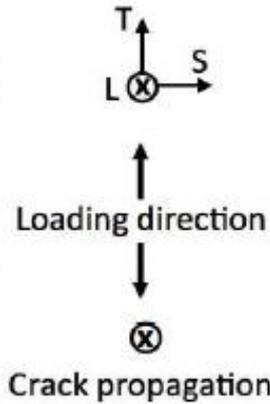
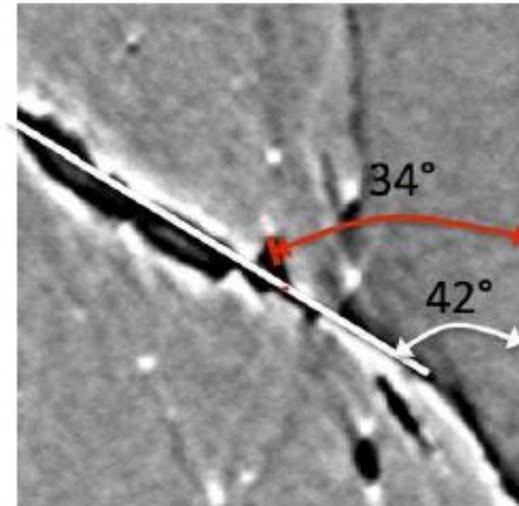
(b) Void(close-surface)



(c) Void(far-centre)



(d) Void(far-surface)



50 μ m

2D sections of laminography data

What does the strain field look like?

- Voids in the slant regions at the specimen surface do not fully contribute to the crack surface

*[T. Ueda *et al.*, 2014, *Acta Mat*]

Strain measurement

- Gray level (2D or) 3D images
- Image 1: $f(\underline{x})$ image 2: $g(\underline{x})$
- Conservation of grey levels

$$f(\underline{x}) \cong g(\underline{x} + \underline{u}(\underline{x}))$$

- Measure $\underline{u}(\underline{x})$?

- Relies on 'natural' image contrast
- What if damage sets in?

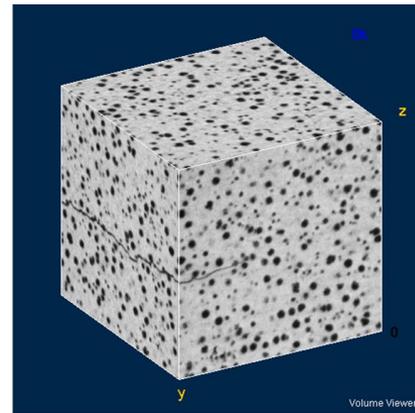
Global DVC (Correli C8)

F.Hild, S.Roux [Roux et al. Comp. Part A 2008]

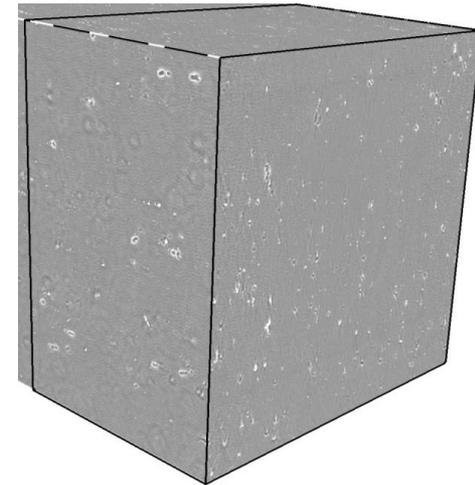
LMT Cachan



'COMINSIDE' project:
1. Laminography imaging +
2. strain measurement +
3. microstructural simulation
(CEMEF)



NCI

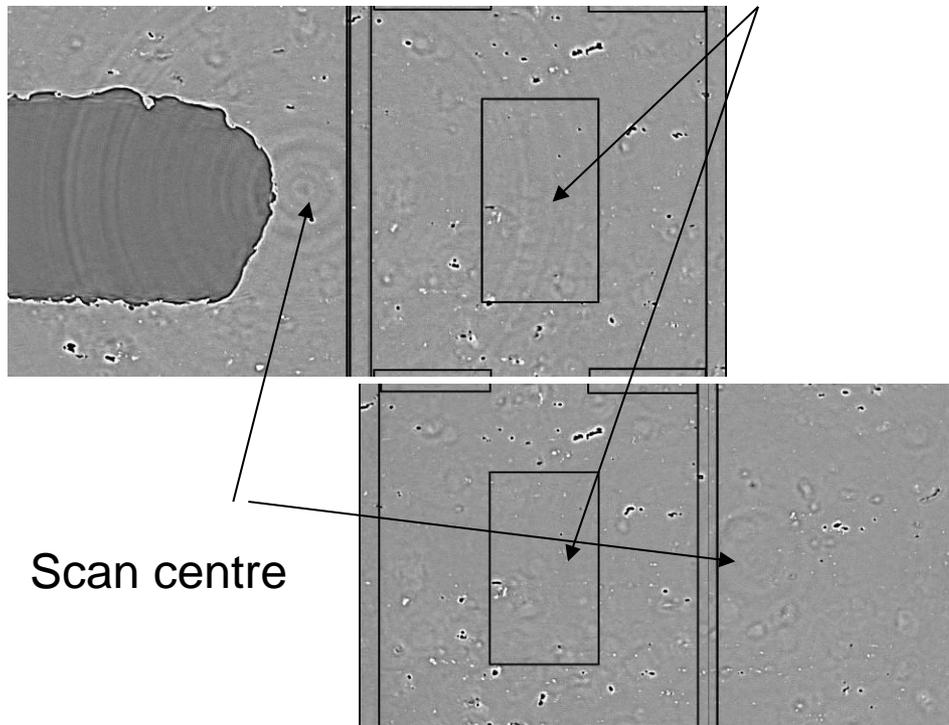


AA

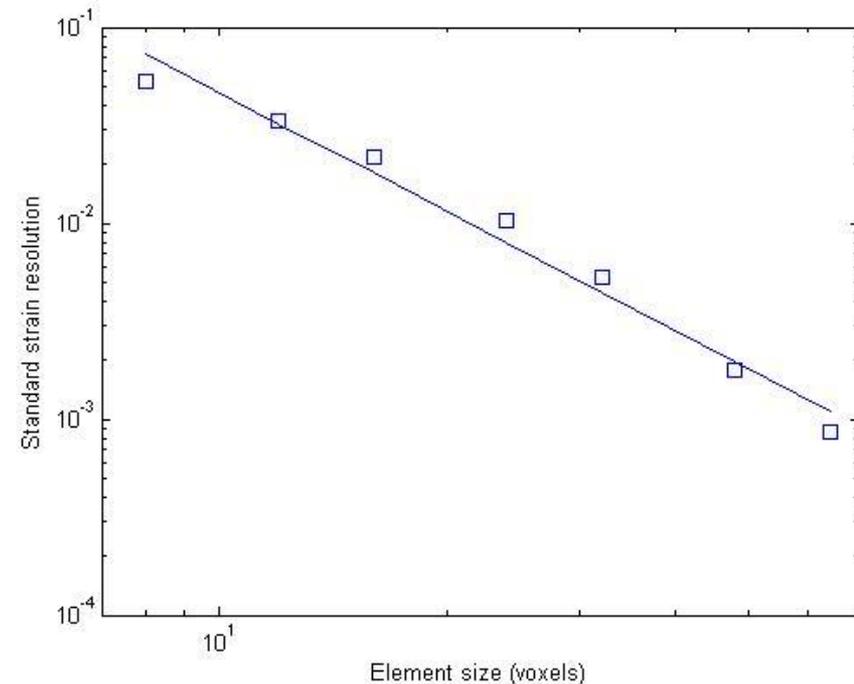
Digital Volume Correlation on *in-situ* SRCLaminography data

- Performance assessment: correlation of 2 scans of undeformed material of the same ROI: rigid body motion
- scan centre shifted

The same ROI

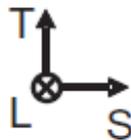
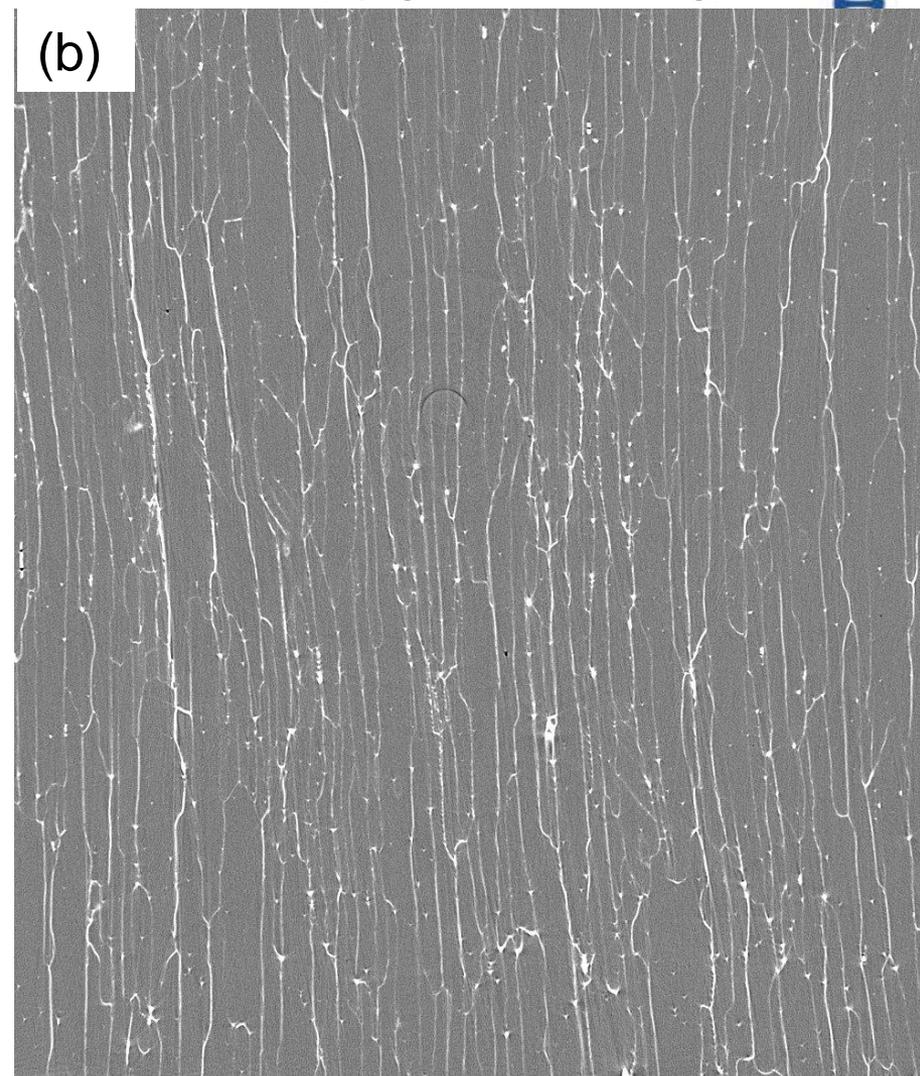
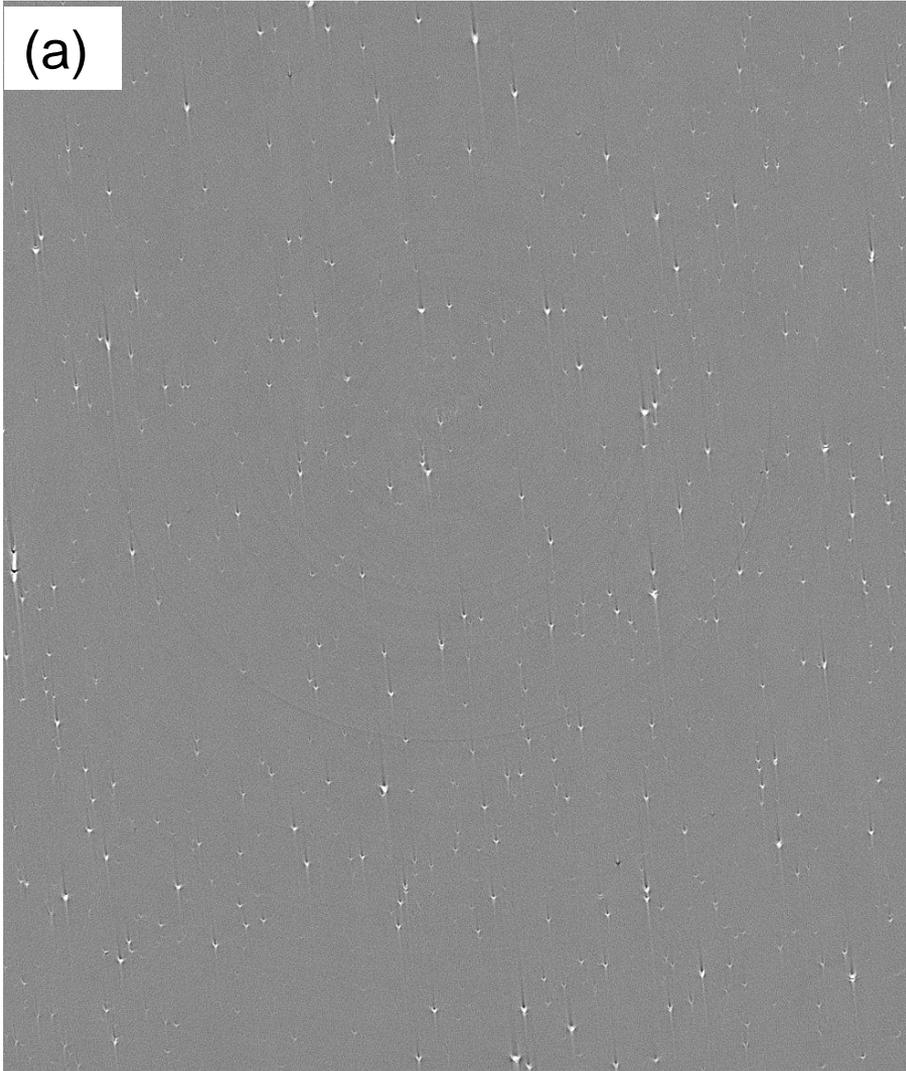


Measurement uncertainty below 1 %
For AA2139 elements 32 voxels



Microstructure: aluminium alloy AA 2198-T8

Grain structure (by gallium wetting):



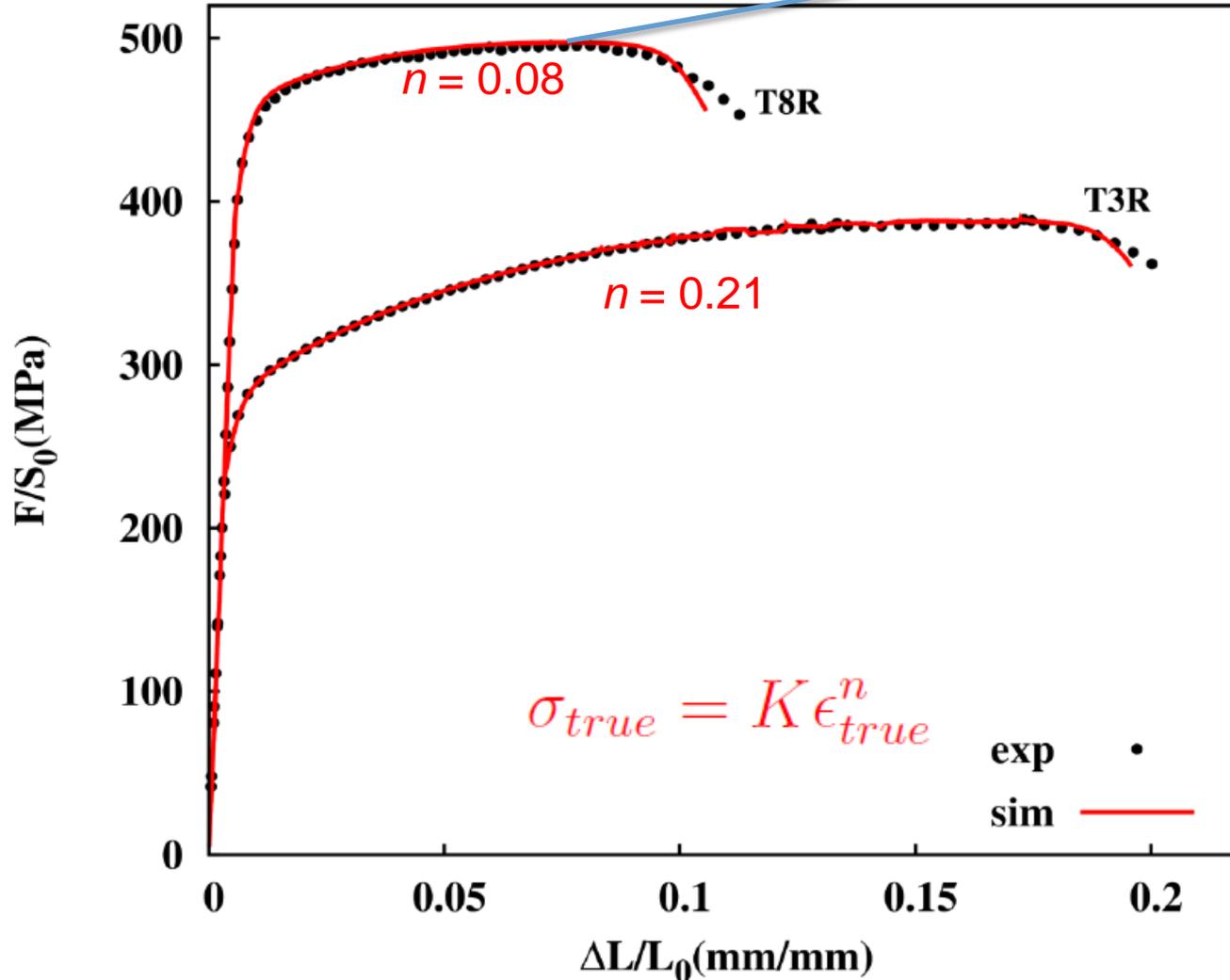
200 μ m

Particle volume fraction: 0.34%

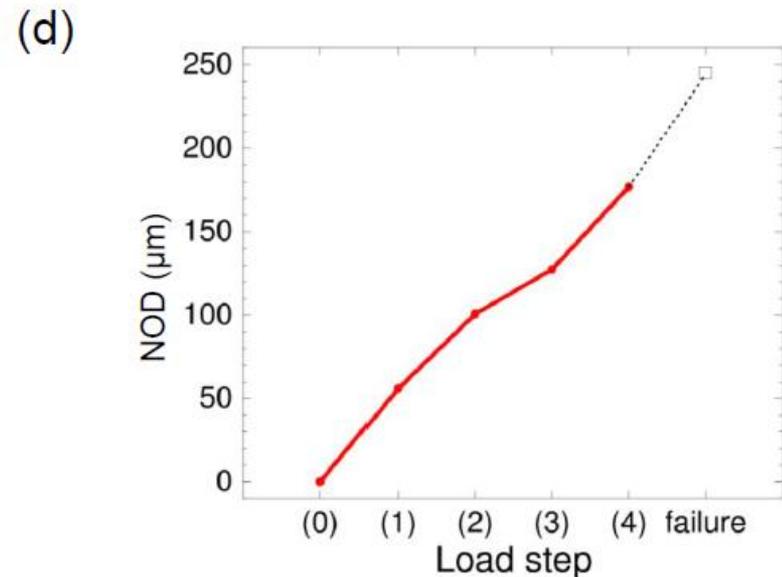
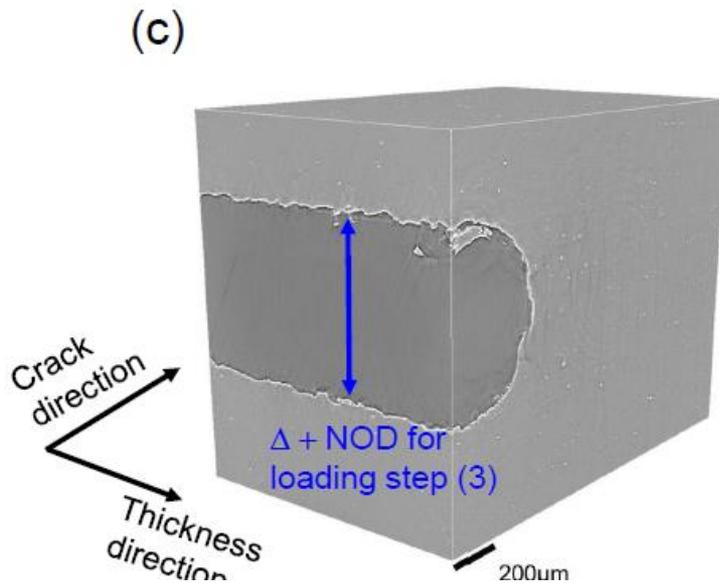
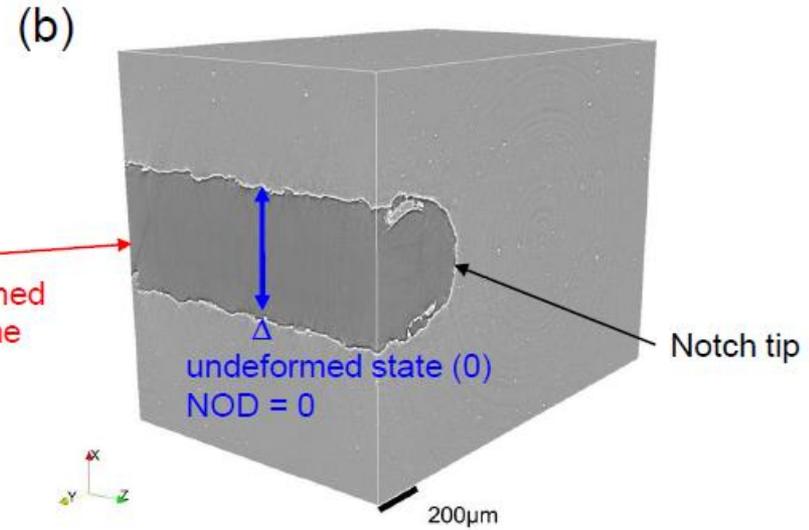
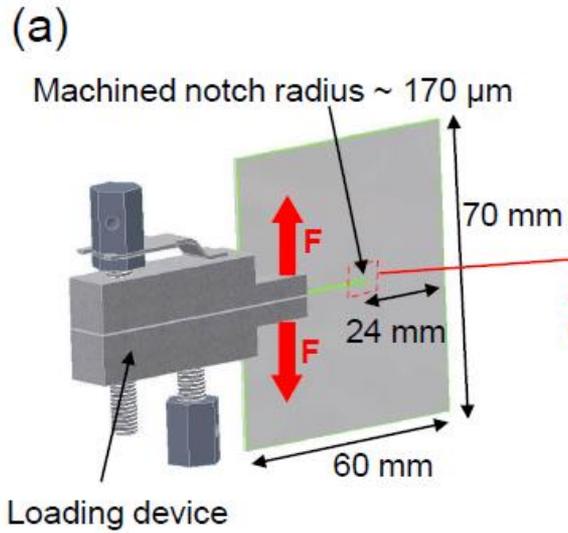
Void volume fraction: < 0.03%

- Al-Cu-Li alloy

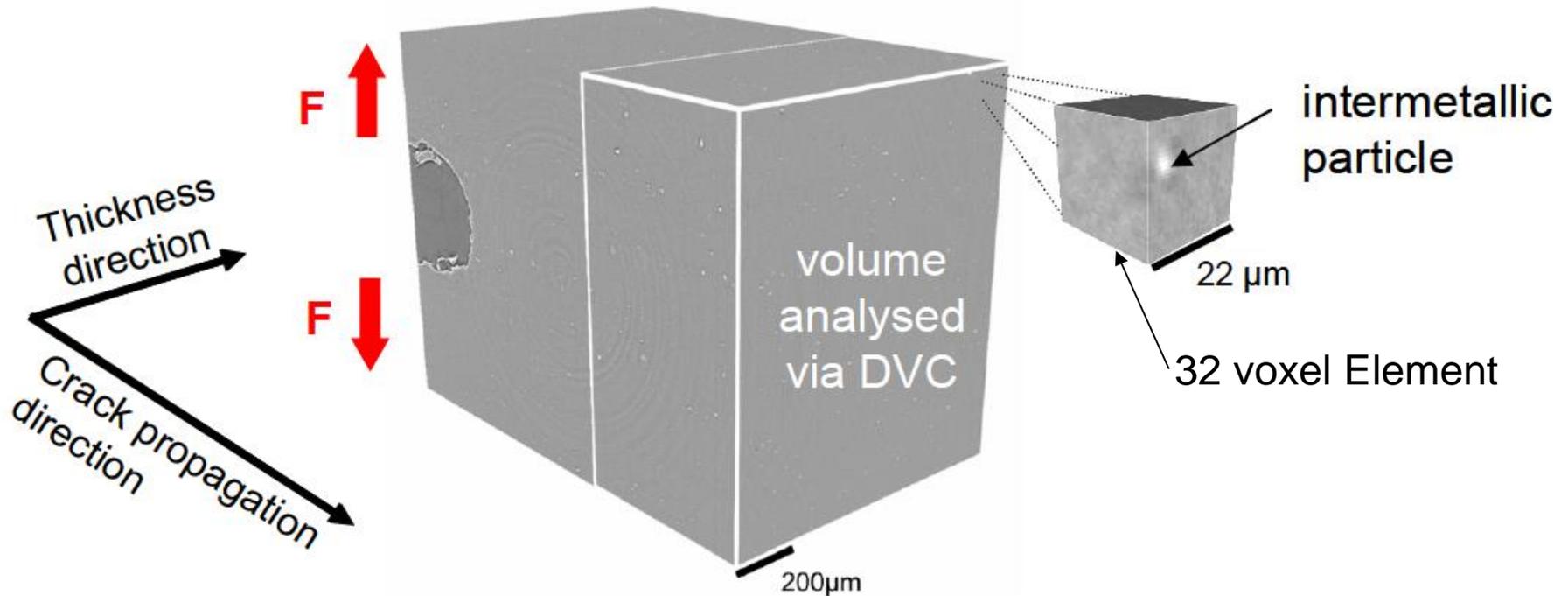
Next study



Mechanical experiment + laminography



Digital volume correlation



Size of the ROI: 1184 x 512 x 992 voxels (830 µm x 360 µm x 700 µm)

32-voxel elements:
Less than 0.1% of strain
uncertainty

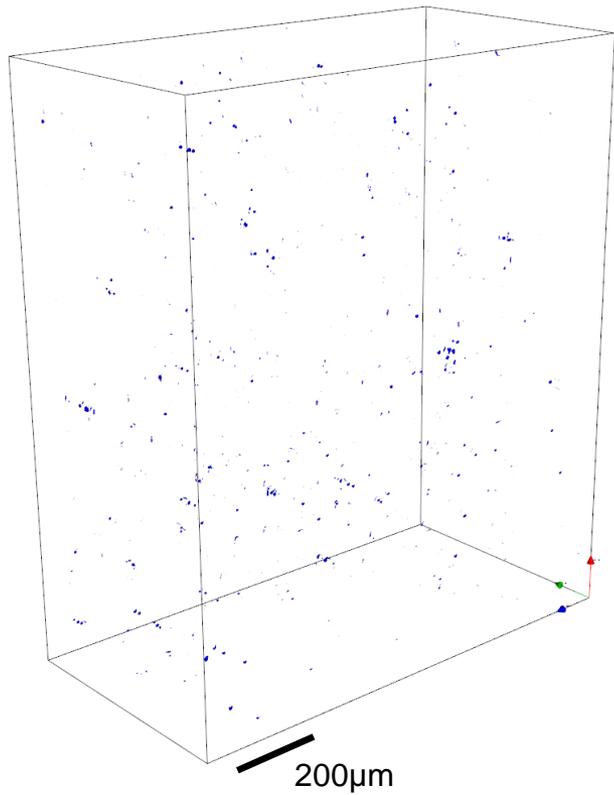
Spatial resolution:
Twice the element size:
 $2 \times 32 \text{ voxel} = 45 \text{ µm}$

Voxel length: 0.7 µm

Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

Master Thesis: T. Taillandier-Thomas

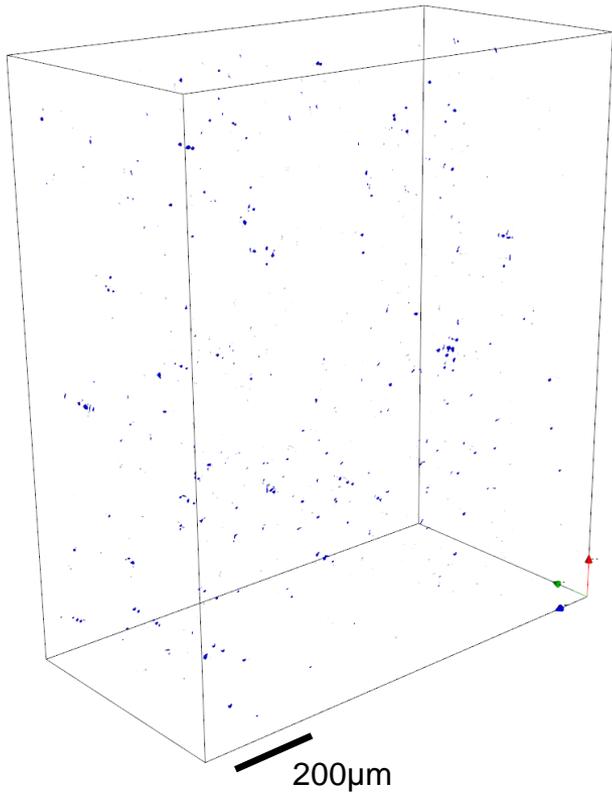


Rendered voids in:
Loadstep (0)

Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

Master Thesis: T. Taillandier-Thomas

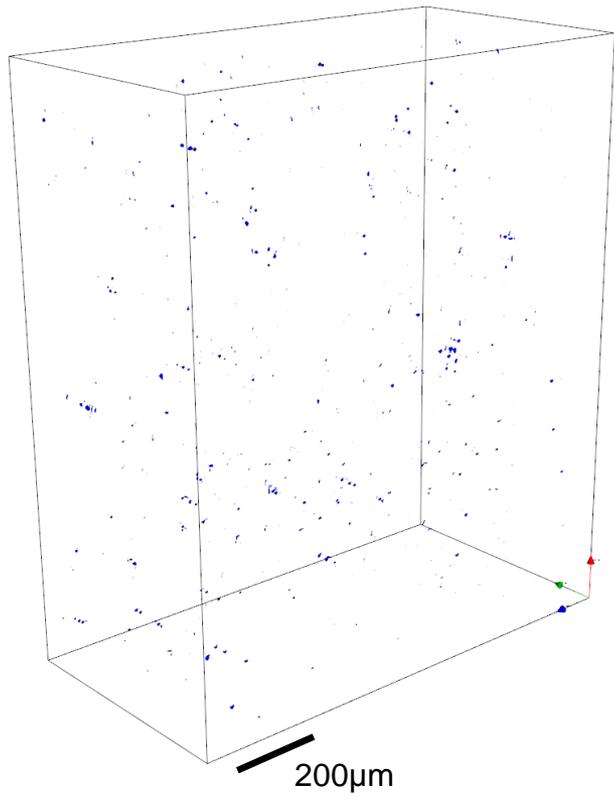


Rendered voids in:
Loadstep (1)

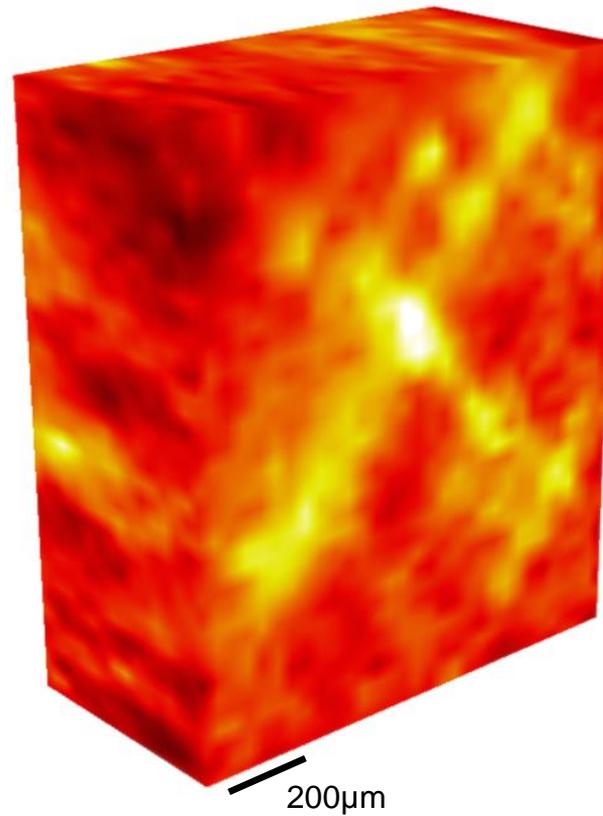
Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

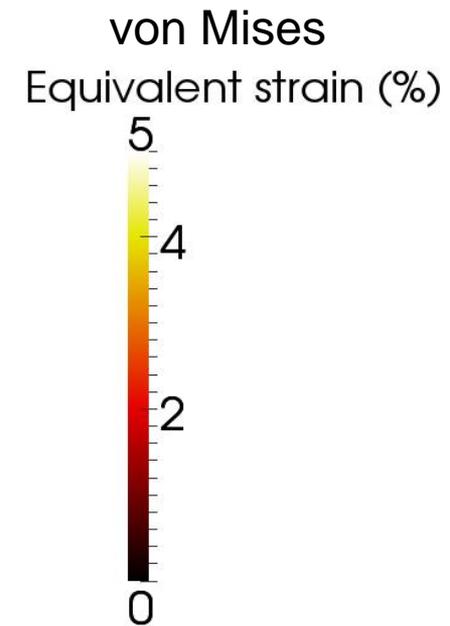
Master Thesis: T. Taillandier-Thomas



Rendered voids in:
Loadstep (1)



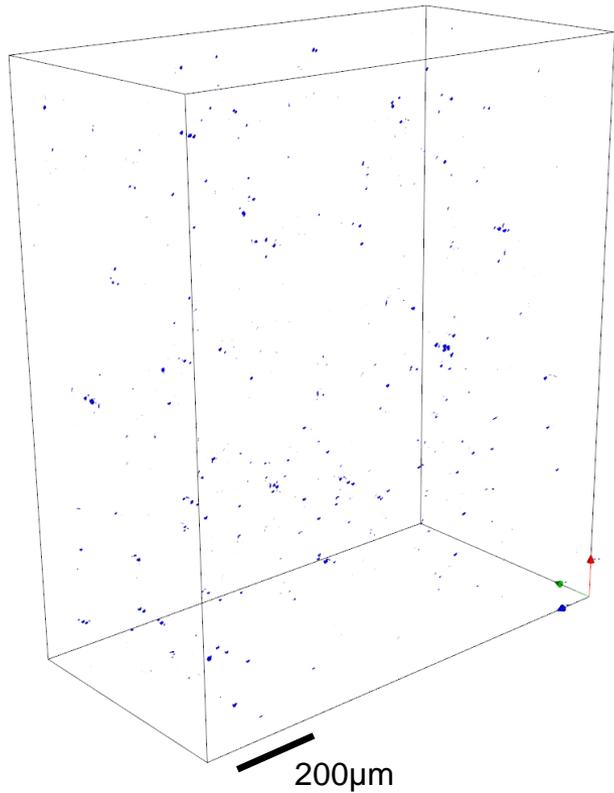
Equivalent strain between:
Loadstep (0) - loadstep (1)



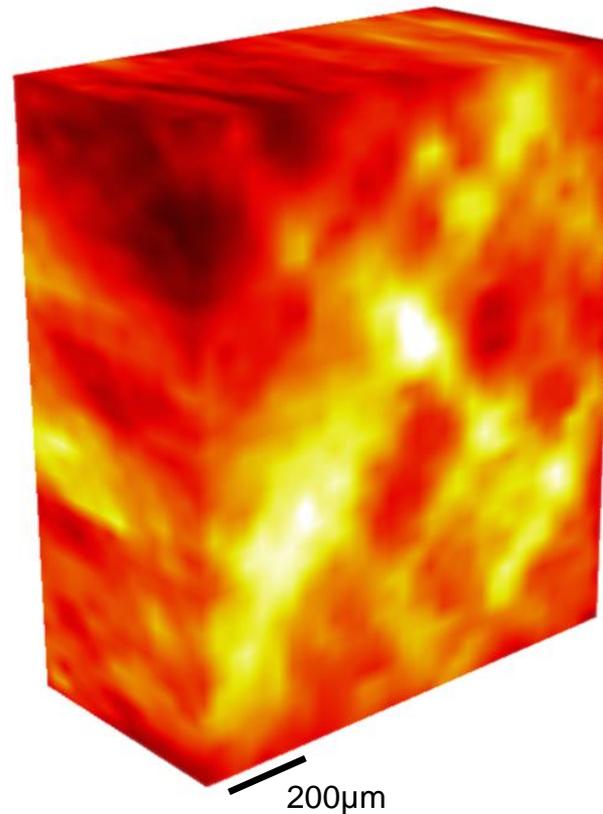
Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

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Rendered voids in:
Loadstep (2)



Equivalent strain between:
Loadstep (0) - loadstep (2)

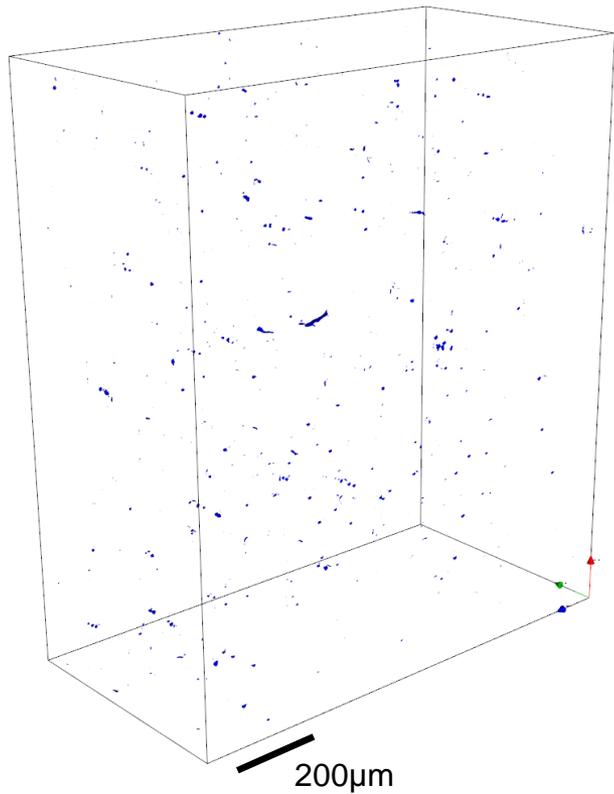
Equivalent strain (%)



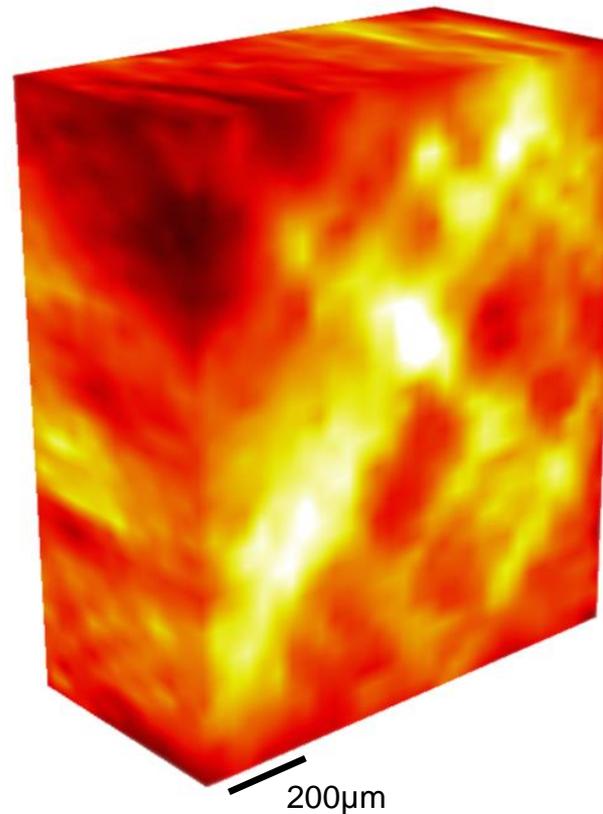
Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

Master Thesis: T. Taillandier-Thomas



Rendered voids in:
Loadstep (3)



Equivalent strain between:
Loadstep (0) - loadstep (3)

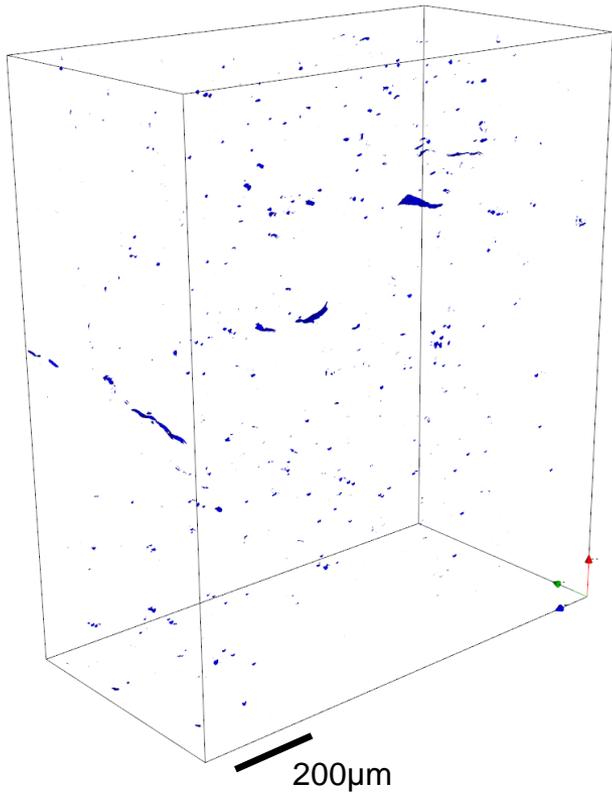
Equivalent strain (%)



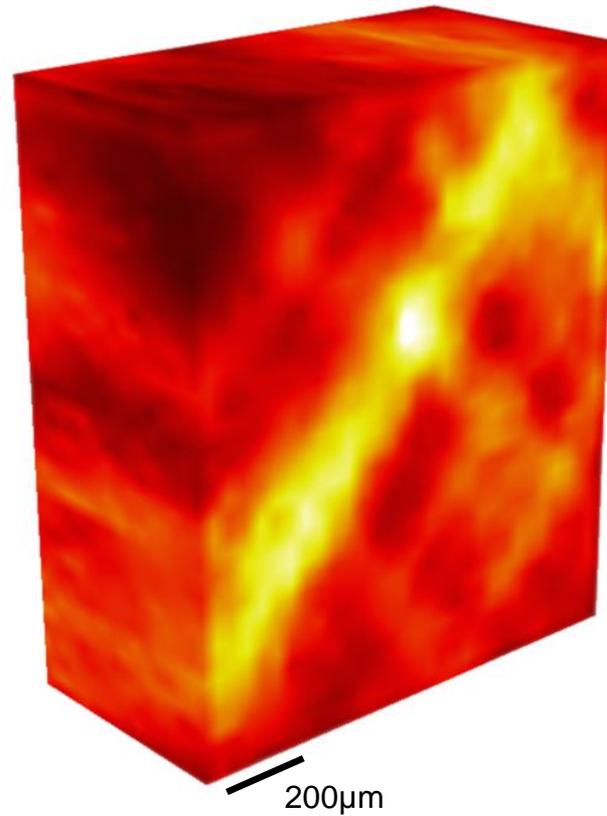
Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

Master Thesis: T. Taillandier-Thomas

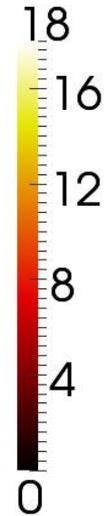


Rendered voids in:
Loadstep (4)



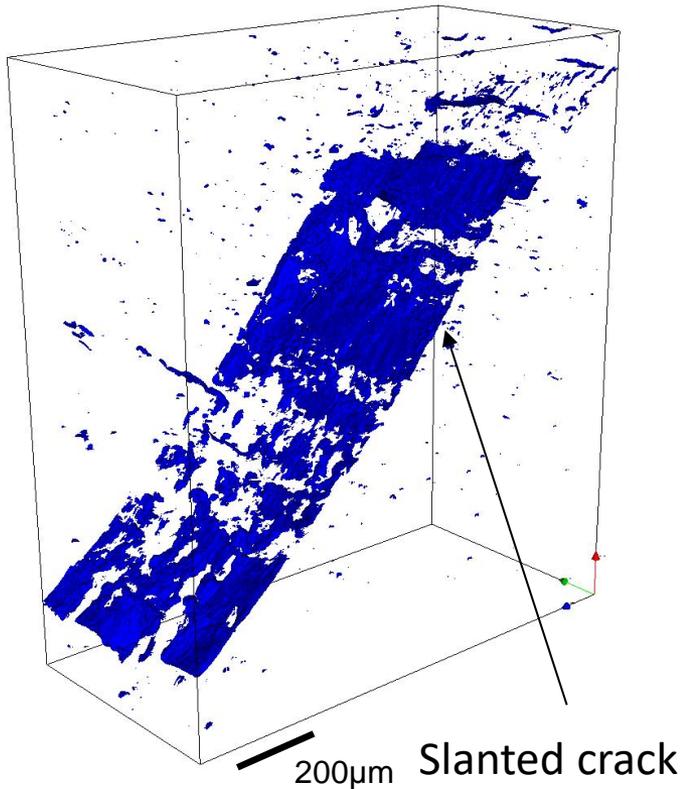
Equivalent strain between:
Loadstep (0) - loadstep (4)

Equivalent strain (%)

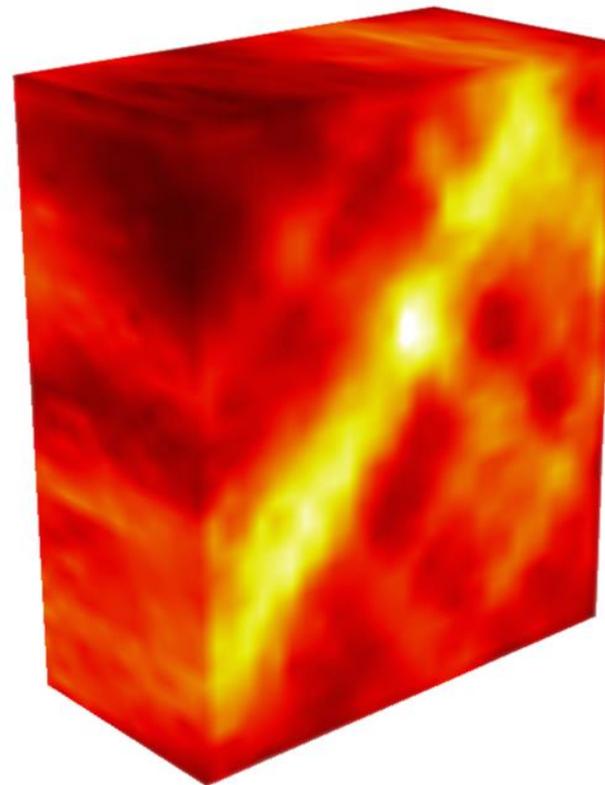


Comparison: damage and strain evolution

*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]
Master Thesis: T. Taillandier-Thomas

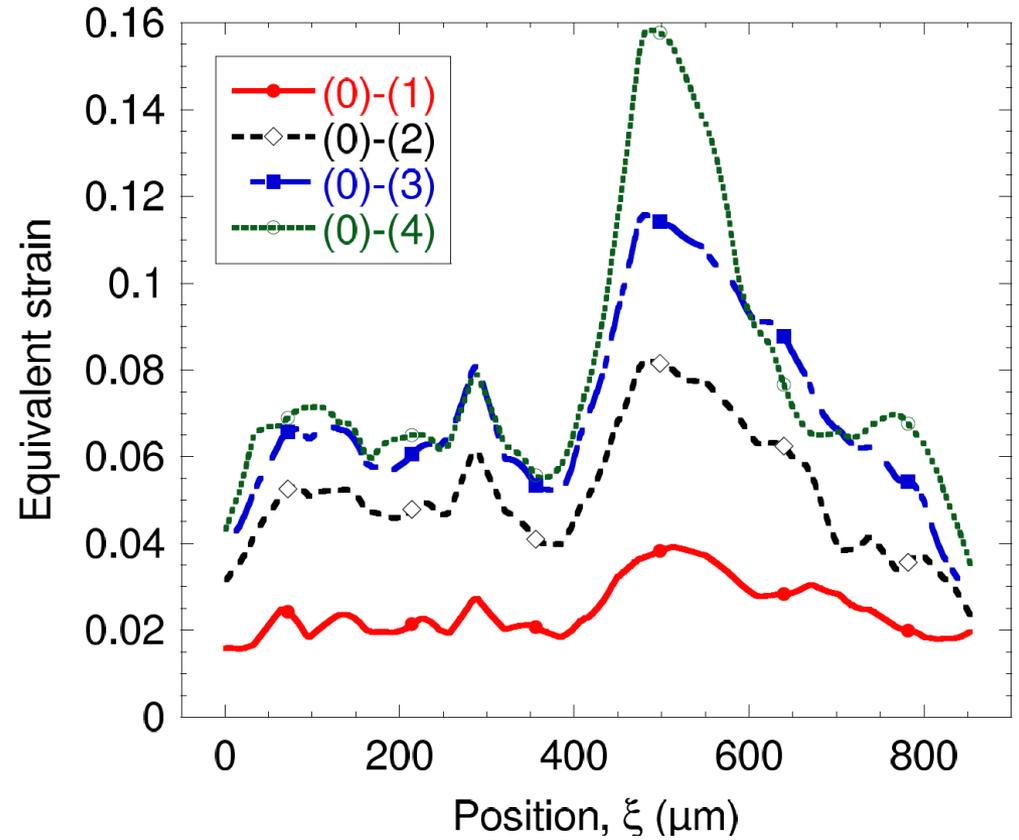
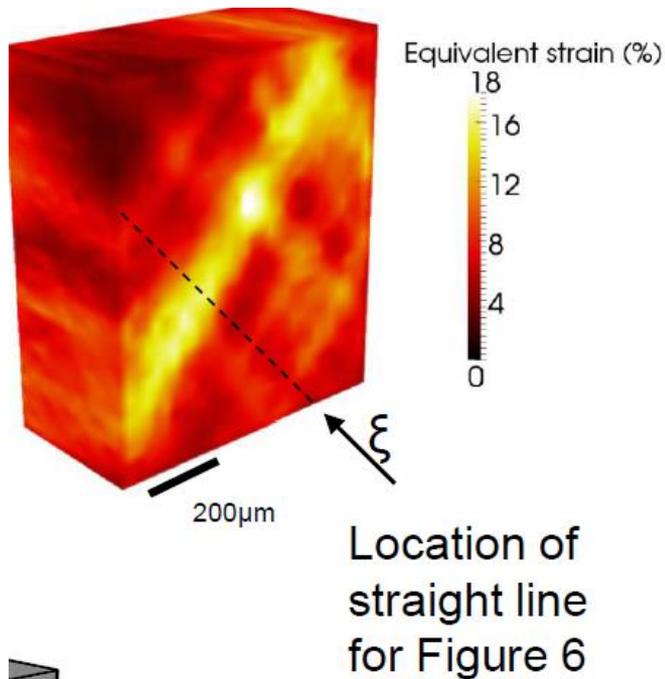


Rendered crack/damage
at failure



Equivalent strain between:
Loadstep (0) - loadstep (4)

Strain evolution



FE field?

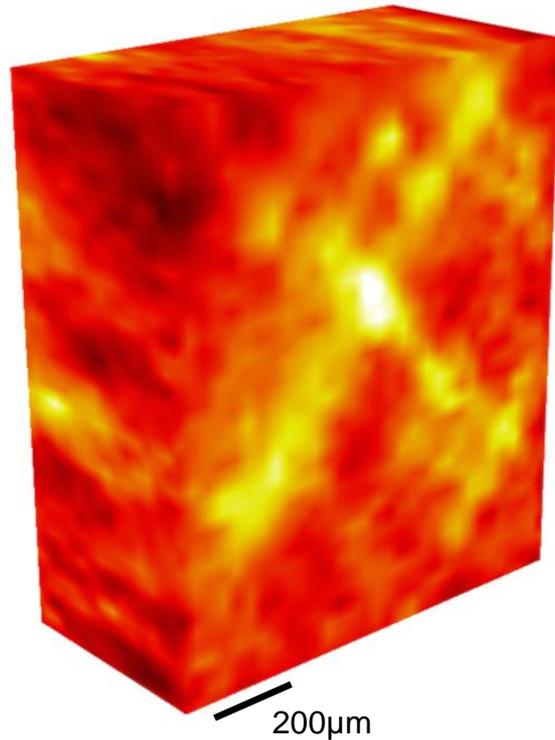
Measured vs. simulated field

○ AA2198

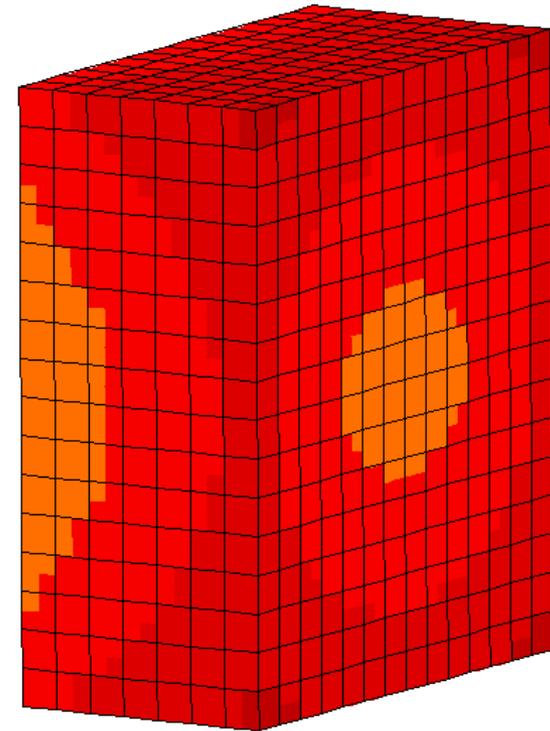
*[Morgeneyer *et al.*, 2014, *Acta Mat.* 69 pp. 78-91]

Master Thesis: T. Taillandier-Thomas

- Result from 3D FE simulation with ideal boundary conditions using:
 - 1. von Mises plasticity
 - or 2. GTN model
 - (or 3. *macroscopic anisotropic model*)



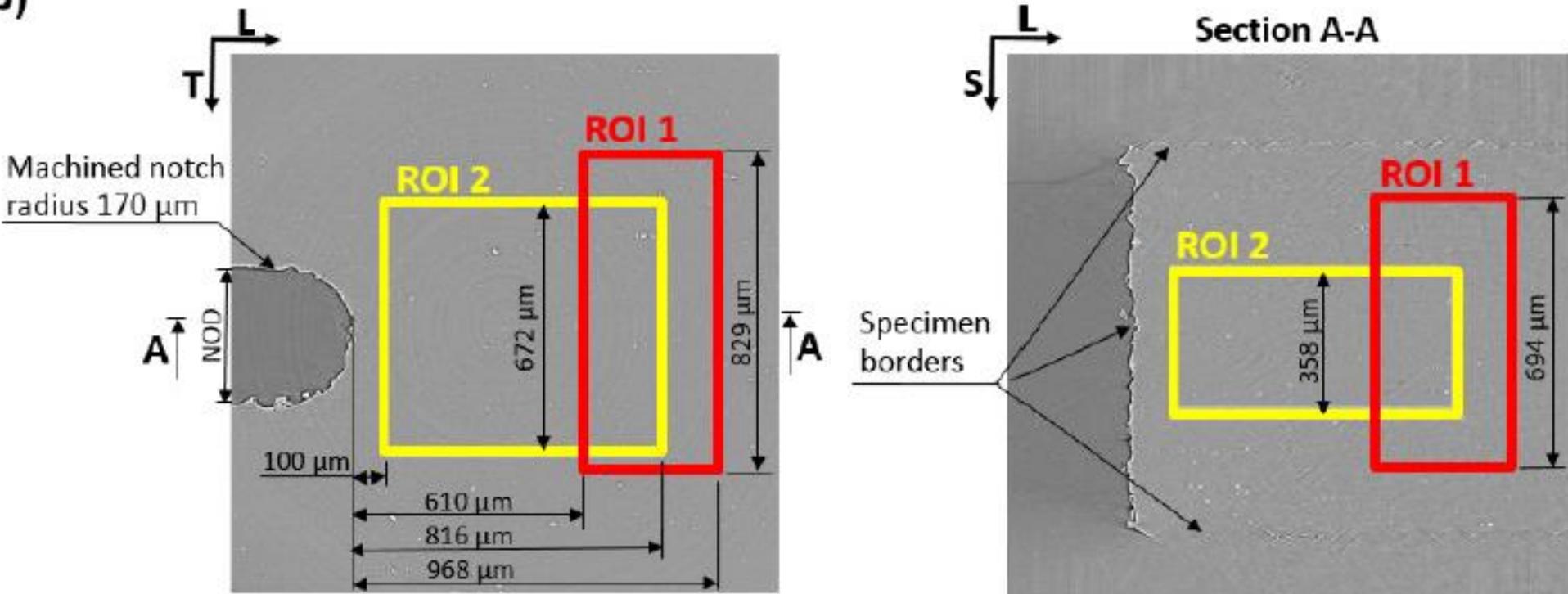
Equivalent strain (%)



Different ROI : closer to the notch

- Getting closer to the notch*
- Assess other strain components

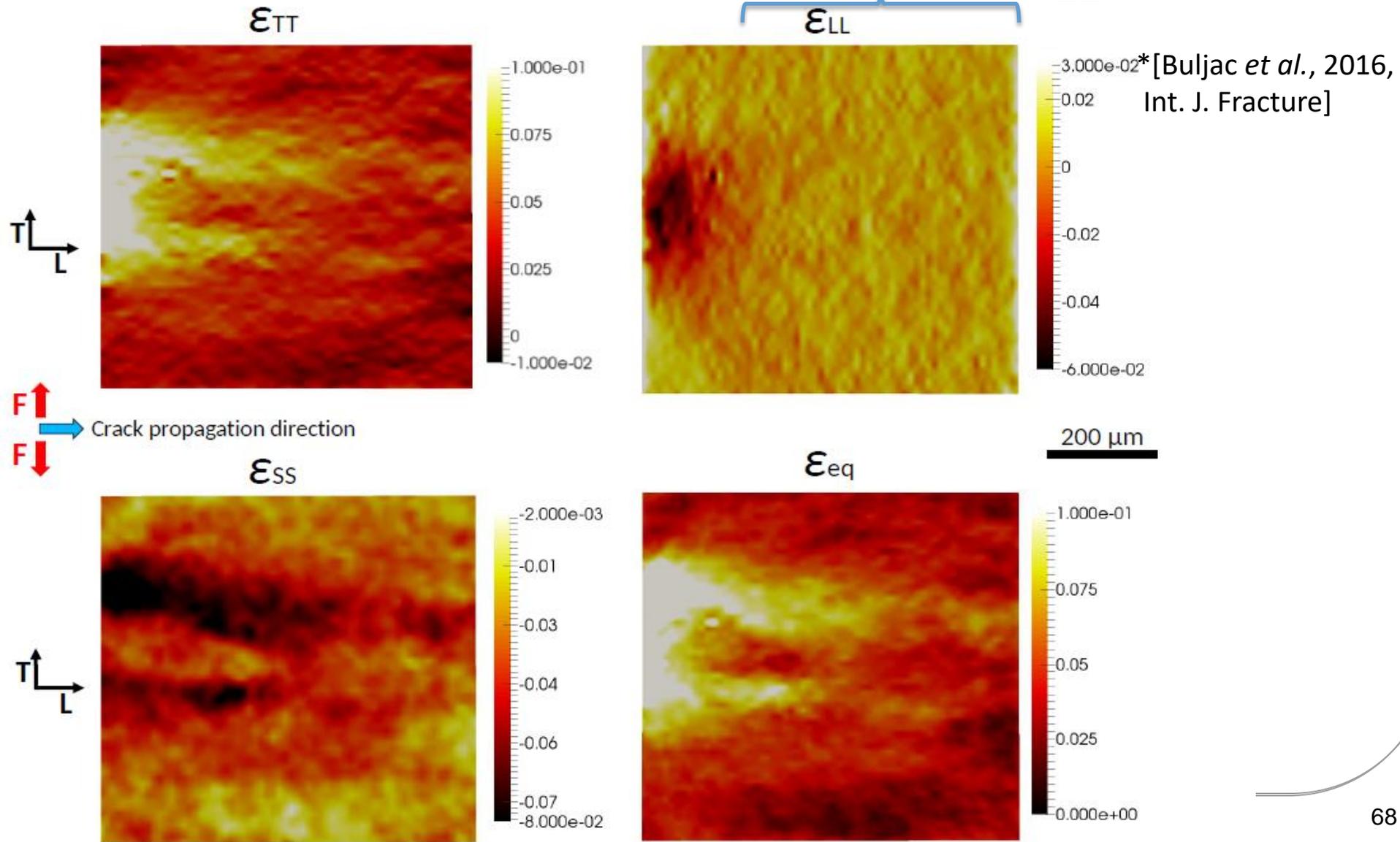
b)



Strains fields close to the notch

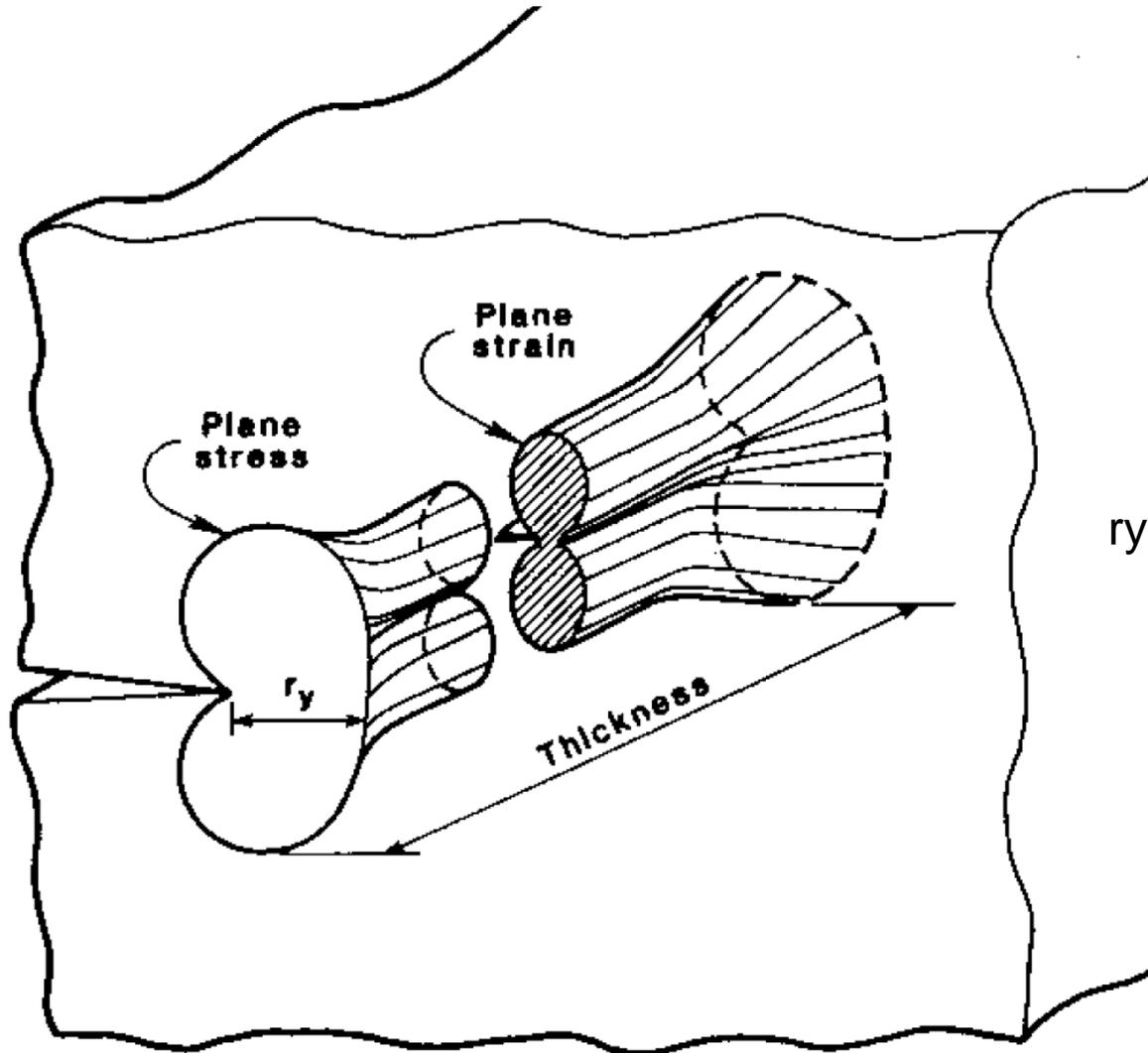
Plane strain in propagation direction,

as strain in propagation direction $\epsilon_{LL} = 0$!



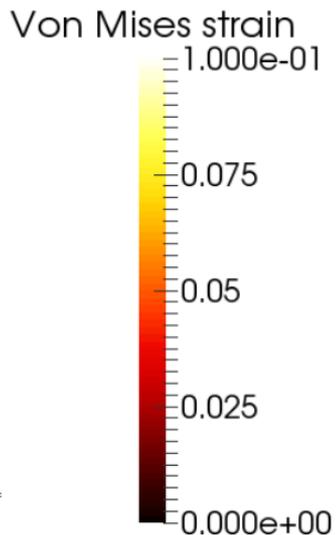
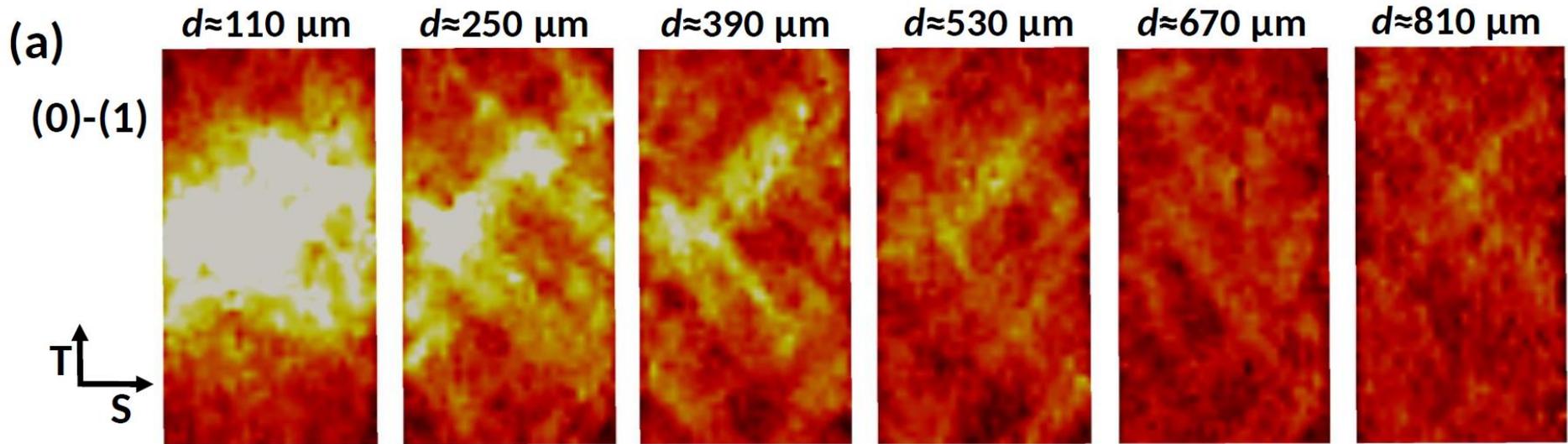
Reminder

- Plane strain direction in a thick sample: thickness direction!



r_y = plastic zone size

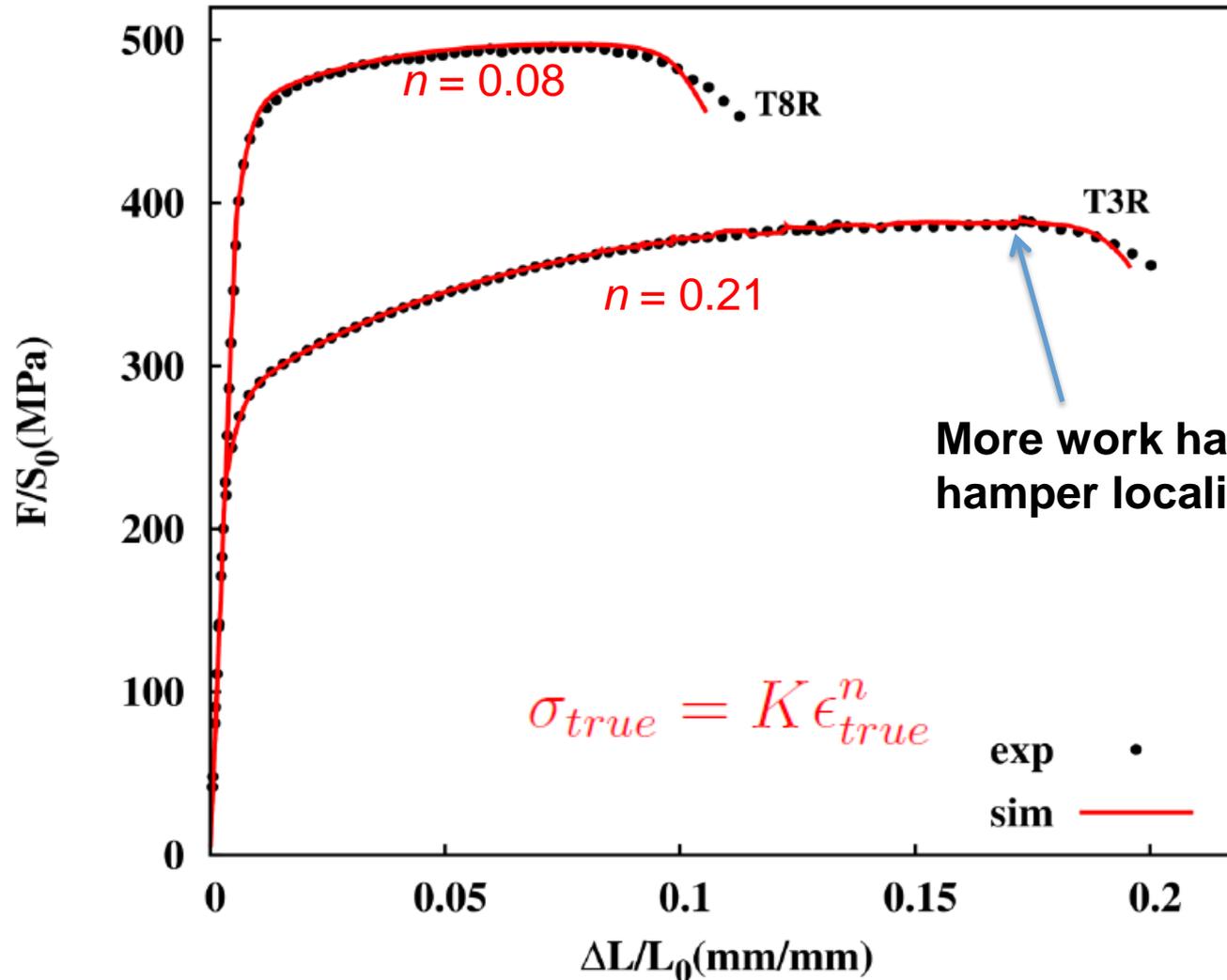
Cuts normal to propagation direction



Bands appearance is favoured by plane strain conditions,
i.e. structural effect!

- *In situ* observations at micrometre scale of notch-microstructure interactions are possible via *in situ* synchrotron laminography
- Evolution of (pre-existing) voids can be observed and measured:
 - Symmetric void growth on the flat crack
 - Void reorientation on the slant crack
- For materials with initial image contrast simultaneous strain and damage measurement are possible
 - Uncertainty about 1% strain for AA
- Strain concentration in the slant bands precedes the onset of damage
 - The bands cannot be reproduced with von Mises plasticity or the Gurson model
- **Do other materials show a similar slant strain concentration bands ?**

- Al-Cu-Li alloy



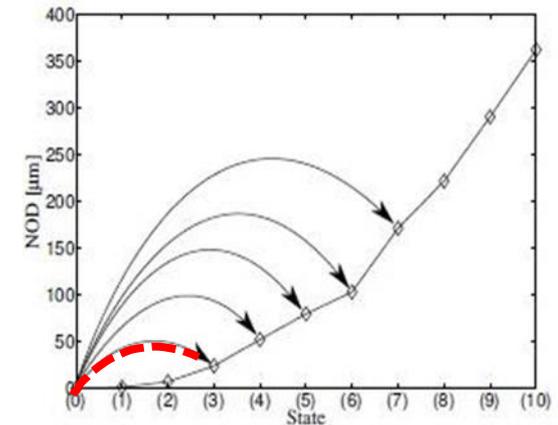
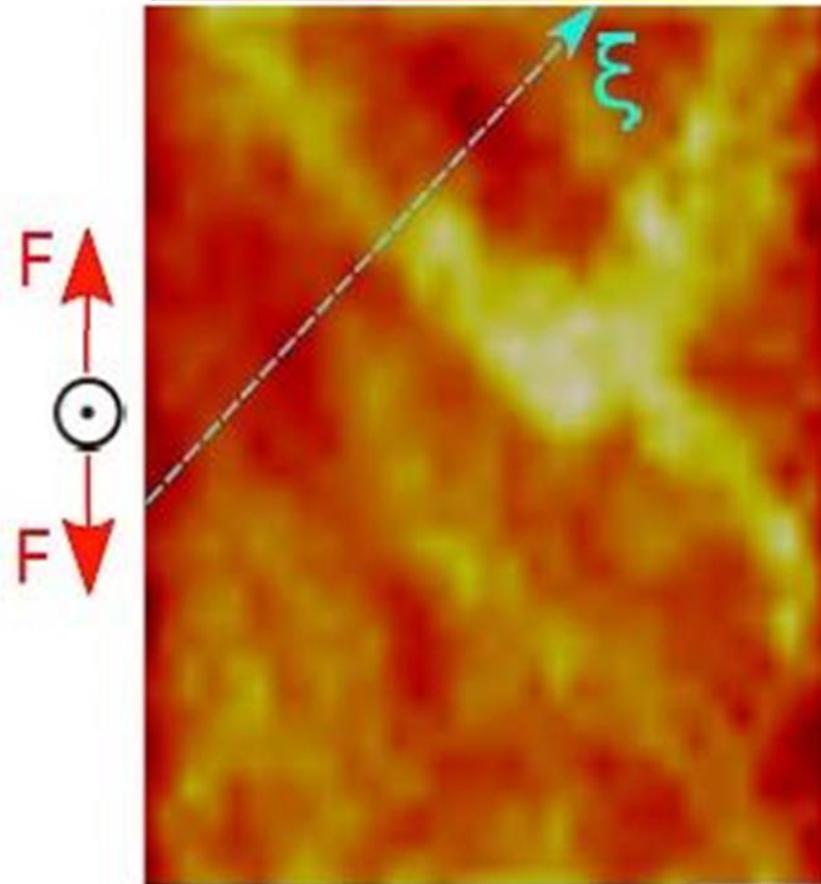
More work hardening should hamper localization (Rice 1976)!

Strain field far ahead of notch for 2198 T3

Cumulated von Mises strain
(0) - (3)

0.00e+00 0.003 0.006 0.009 1.40e-02

Direct correlations



⊙ crack propagation direction

200 μm

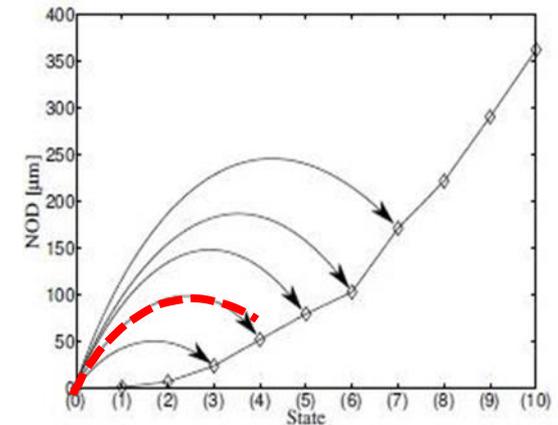
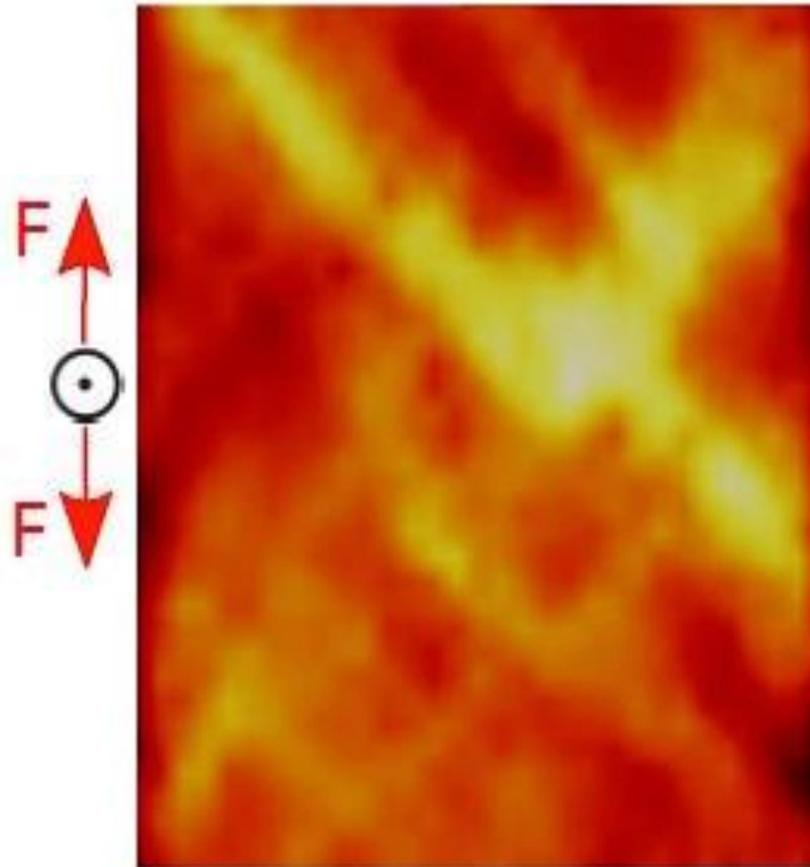
- early strained band despite high work hardening

Strain field far ahead of notch for 2198 T3

Cumulated von Mises strain
(0) - (4)

0.00e+00 0.008 0.016 0.024 3.86e-02

Direct correlations



⊙ crack propagation direction

200 μm

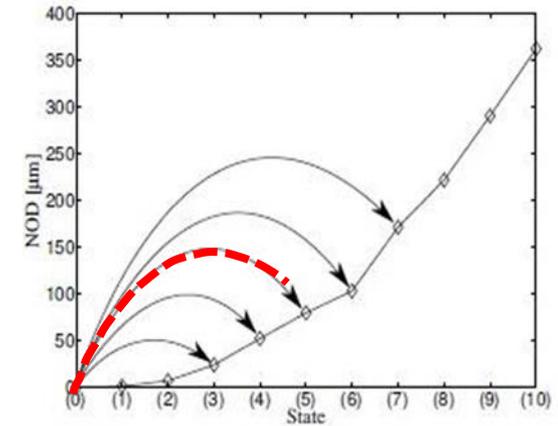
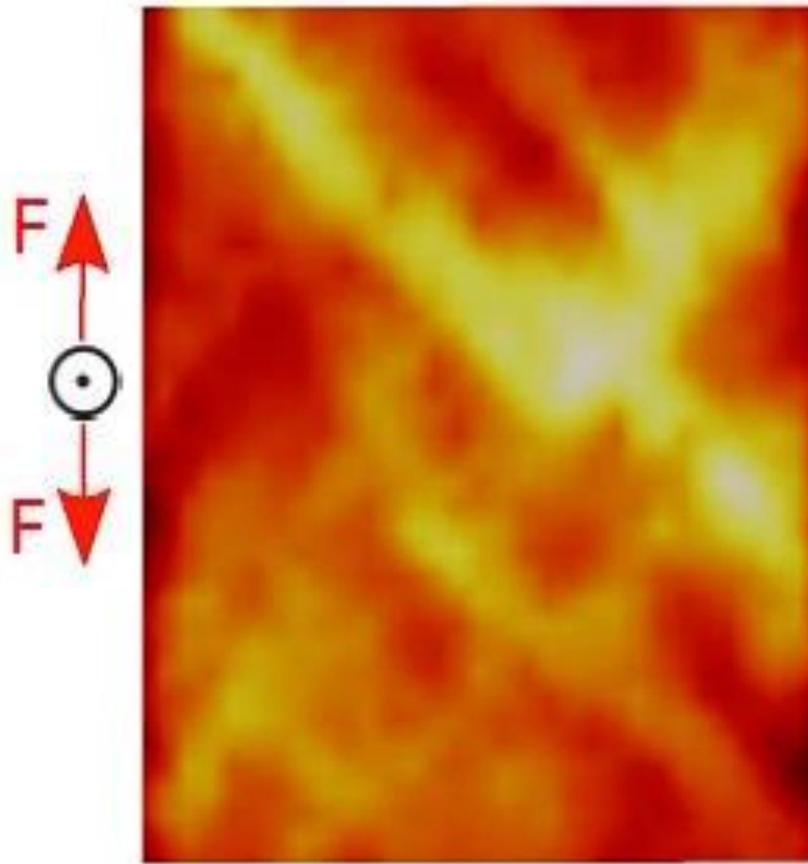
- early strained band despite high work hardening

Strain field far ahead of notch for 2198 T3

Cumulated von Mises strain
(0) - (5)

0.00e+00 0.01 0.03 5.30e-02

Direct correlations



⊙ crack propagation direction

200 μm

- early strained band despite high work hardening

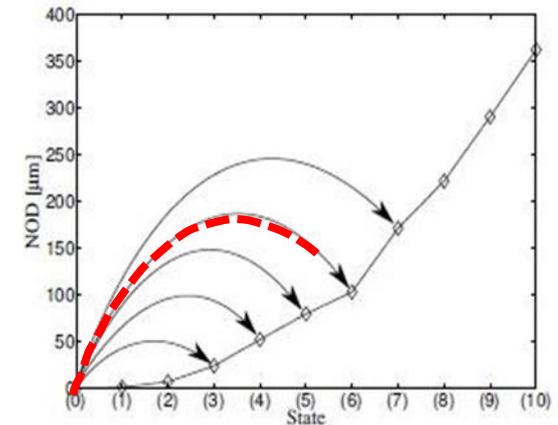
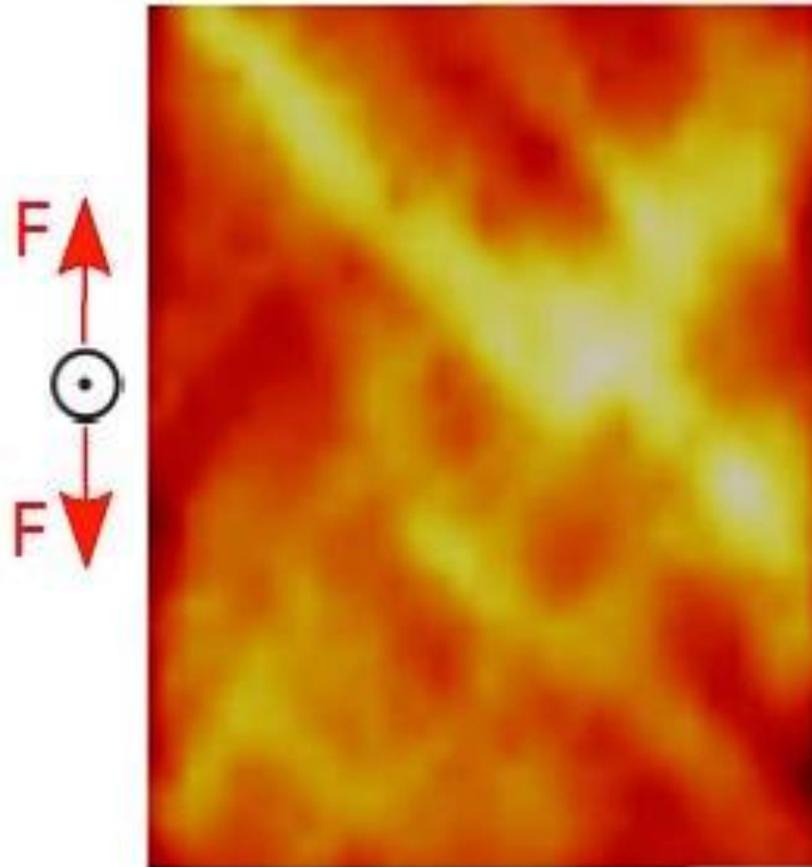
Strain field far ahead of notch for 2198 T3

Cumulated von Mises strain

(0) - (6)

0.00e+00 0.01 0.03 0.04 6.61e-02

Direct correlations



⊙ crack propagation direction

200 μm

- early strained band despite high work hardening

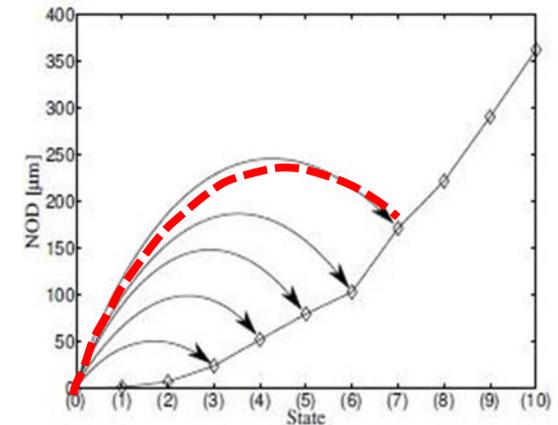
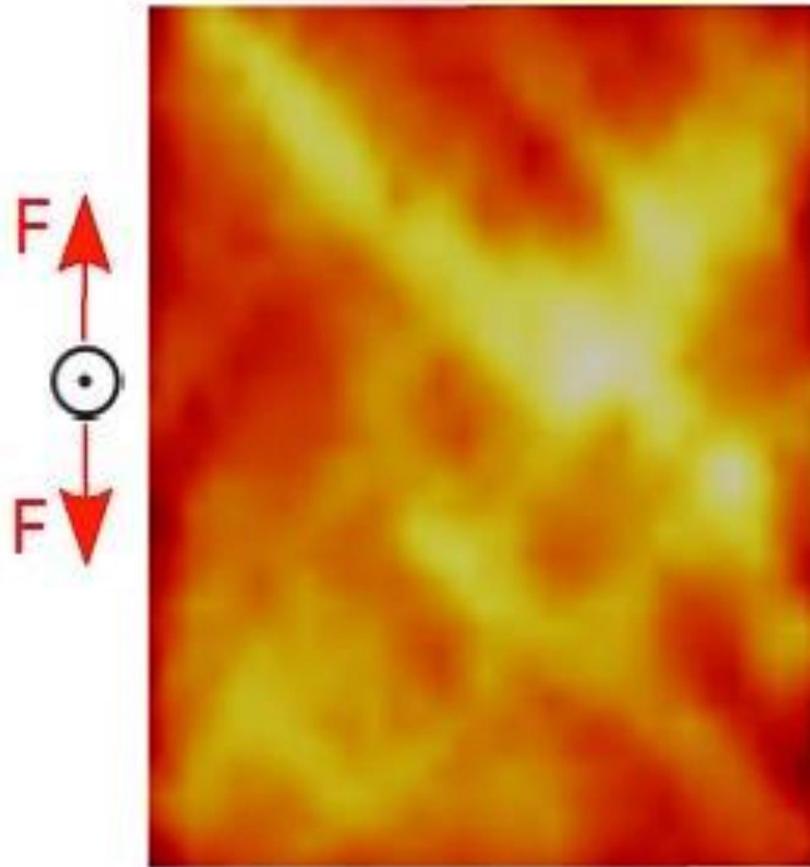
Strain field far ahead of notch for 2198 T3

Cumulated von Mises strain

(0) - (7)

0.00e+00 0.025 0.05 0.075 1.06e-01

Direct correlations



⊙ crack propagation direction

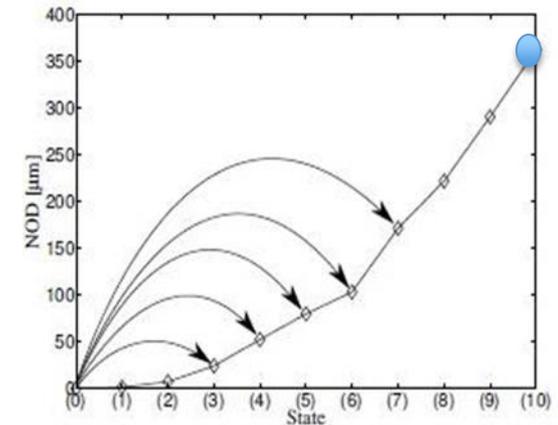
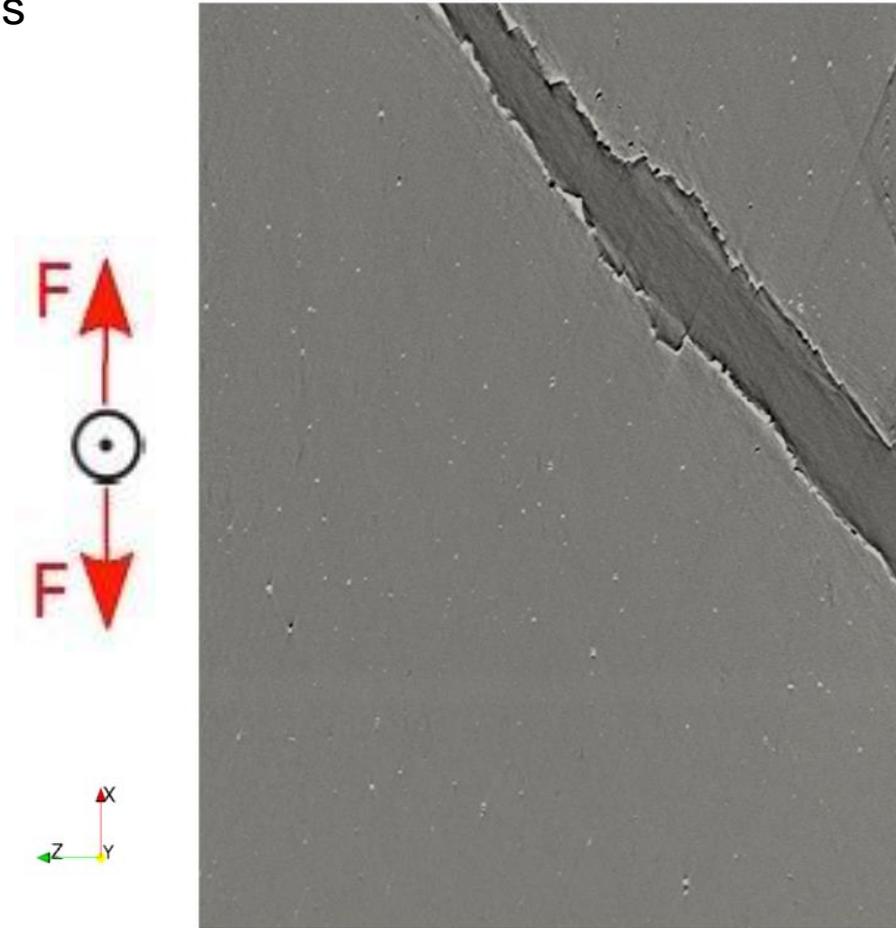
200 μm

- early strained band despite high work hardening

Strain field far ahead of notch for 2198 T3

Final fracture

Direct correlations



⊙ crack propagation direction

200 μm

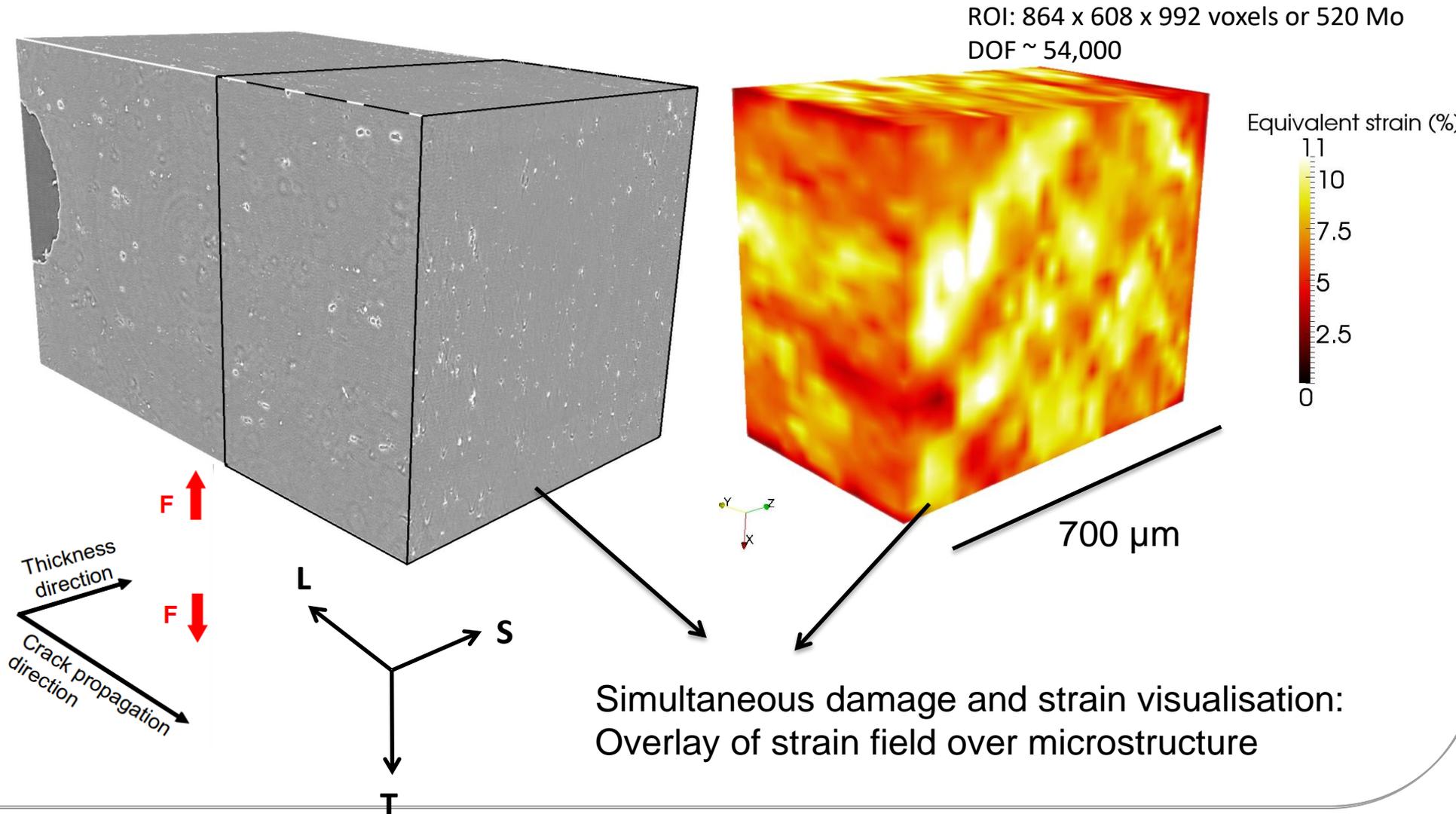
- early strained band despite high work hardening

Strain field for AA2139 T3

Perspectives: Effect of material (2139 T3)

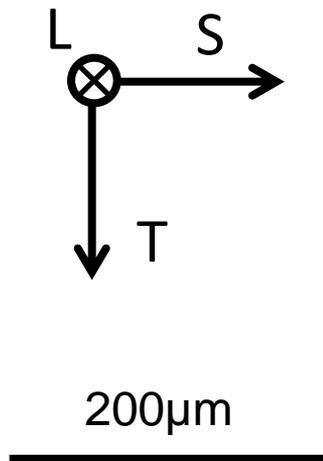
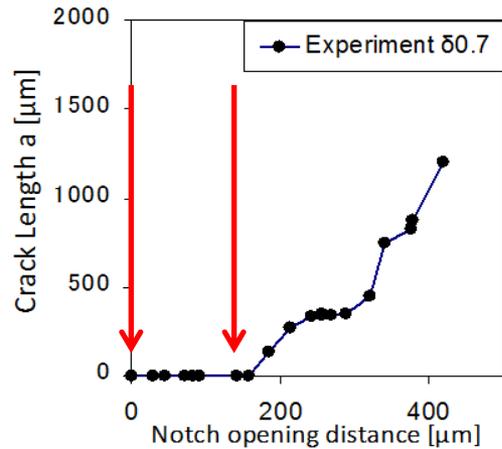
○ Effect of strain hardening and initial porosity

- Initial microstructure and incremental strain field

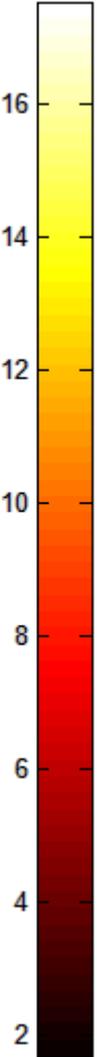
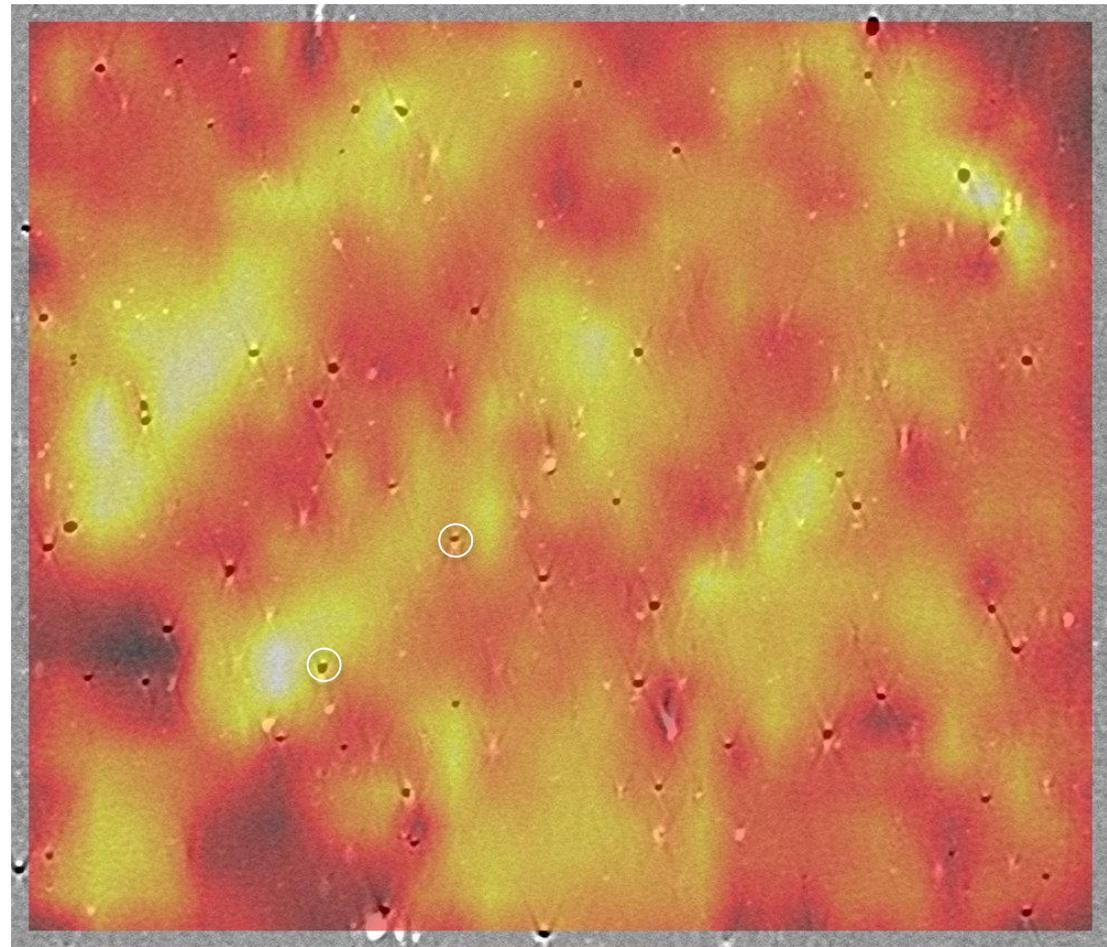


2139 T3: overlay strain + microstructure

Incremental von Mises equivalent strain: $p_{eq}(u_{b-i})$

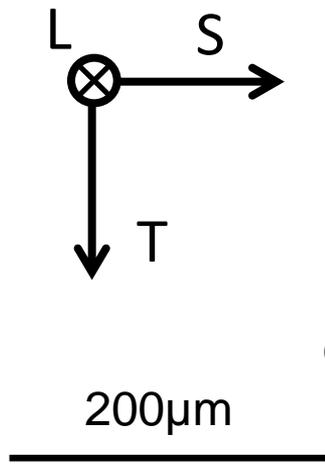
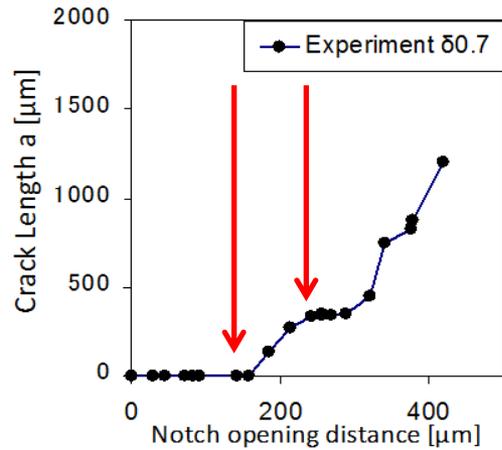


⊗ Crack Propagation

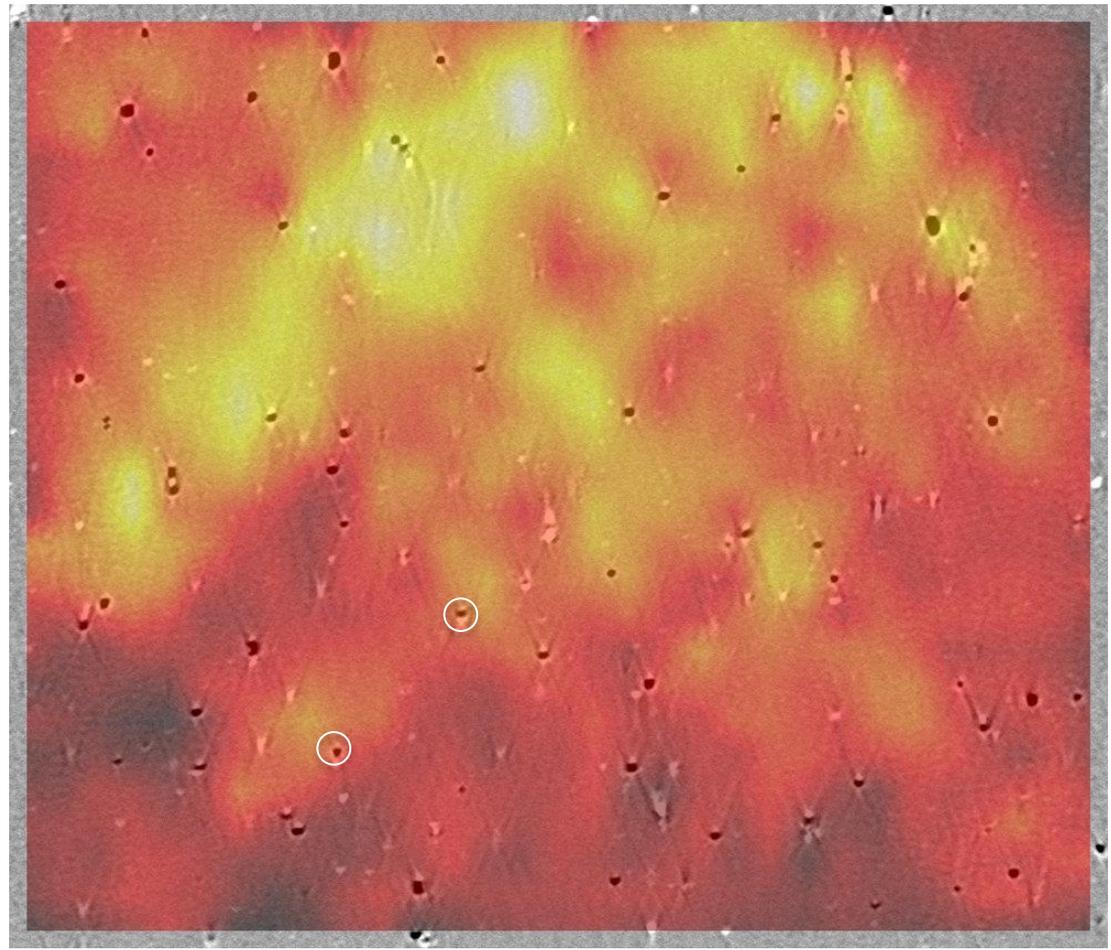


2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{i-m})$



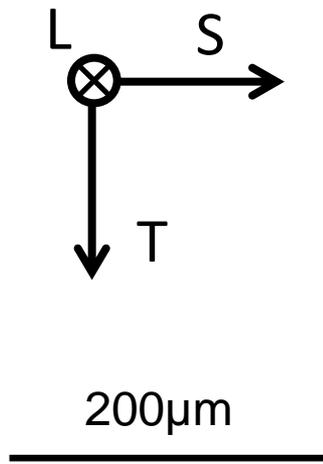
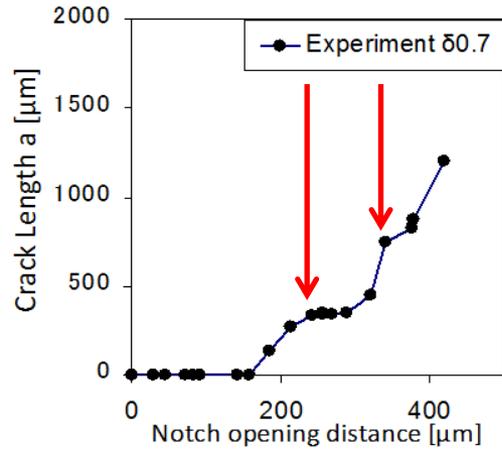
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direction
↓



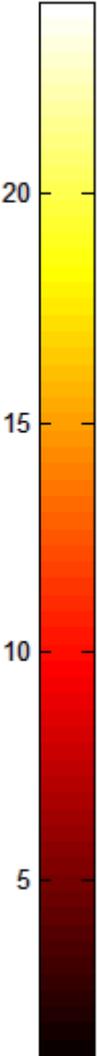
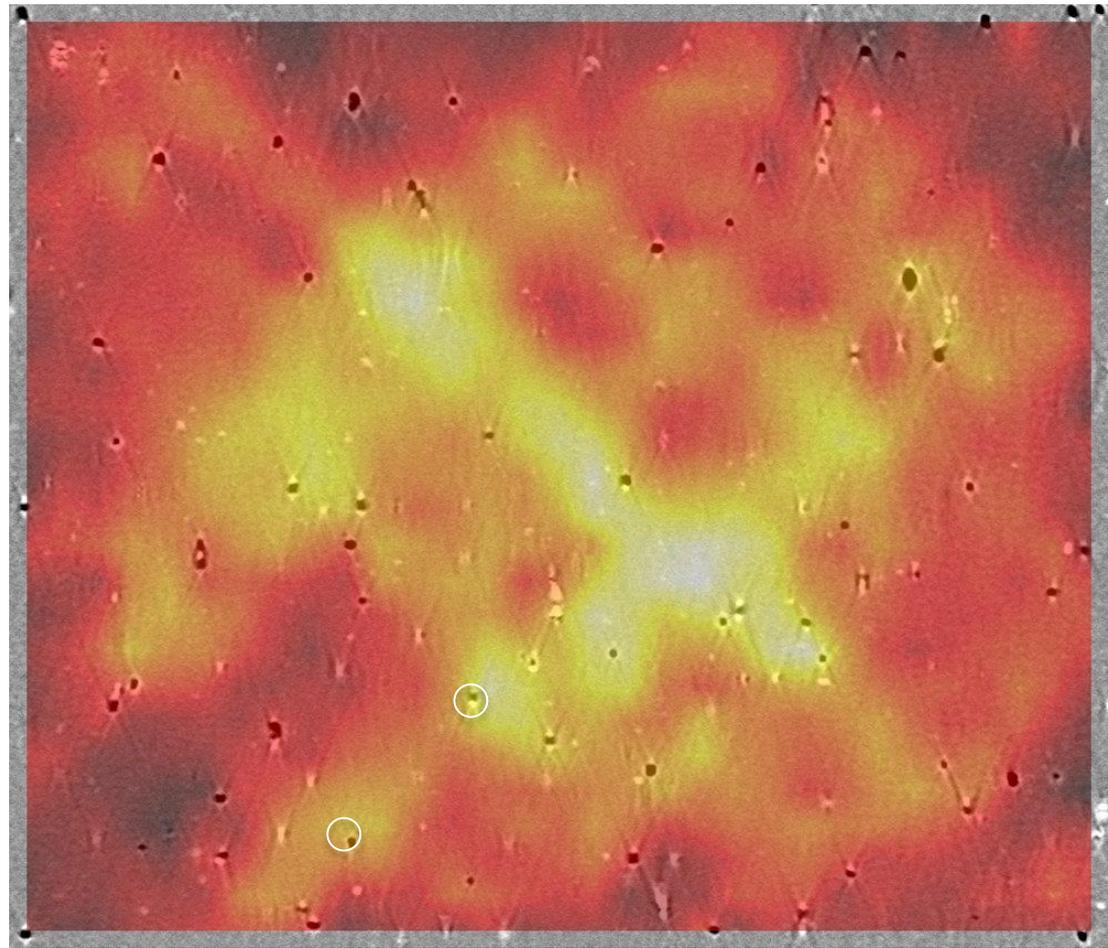
⊗ Crack Propagation

2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{m-r})$

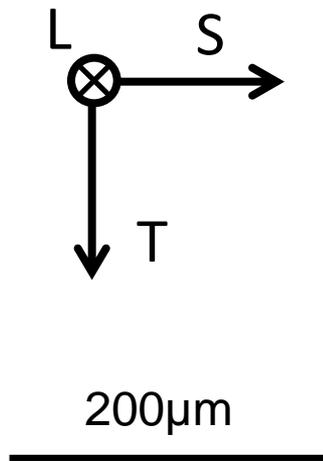
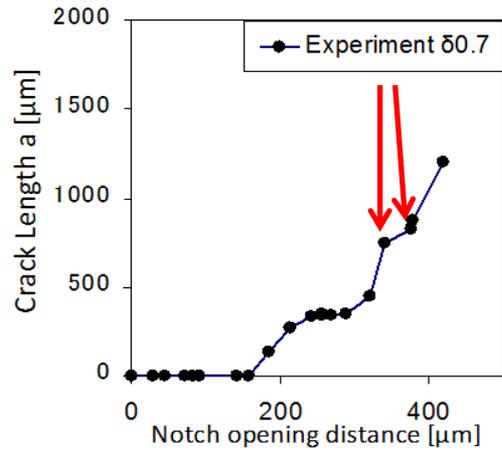


⊗ Crack Propagation

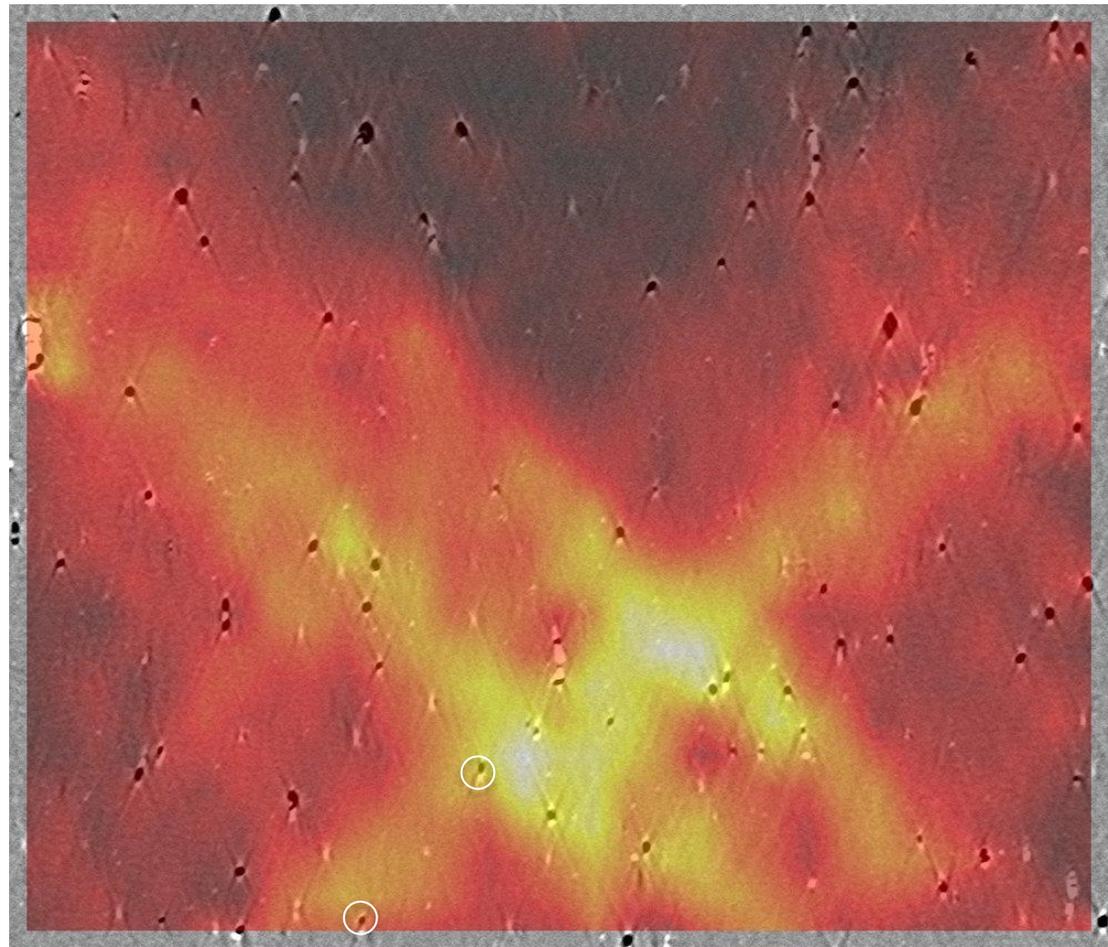


2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{r-s})$

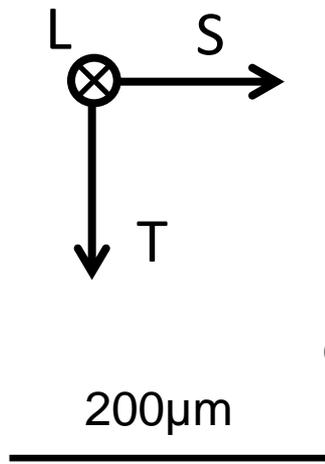
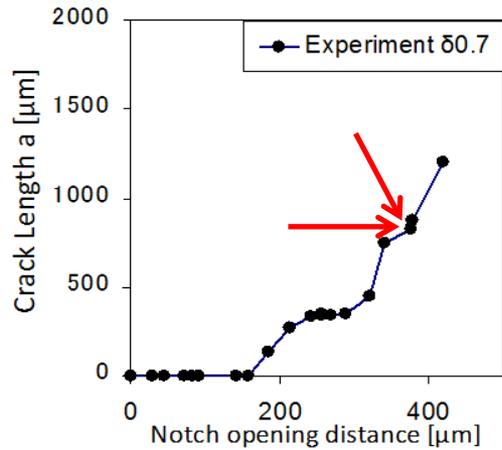


⊗ Crack Propagation



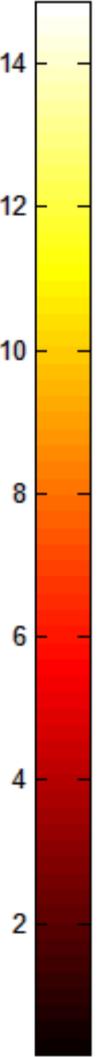
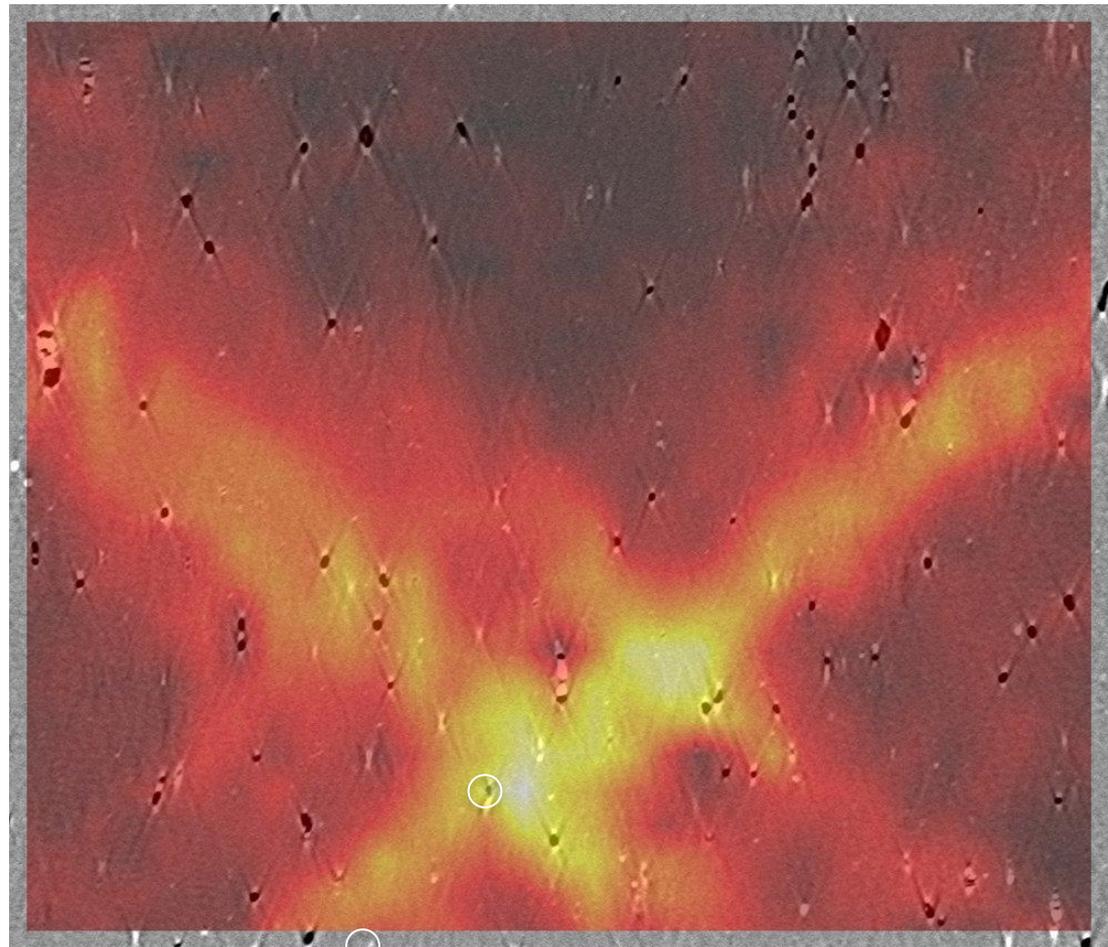
2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{s-t})$



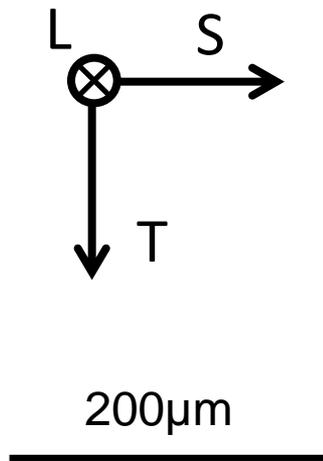
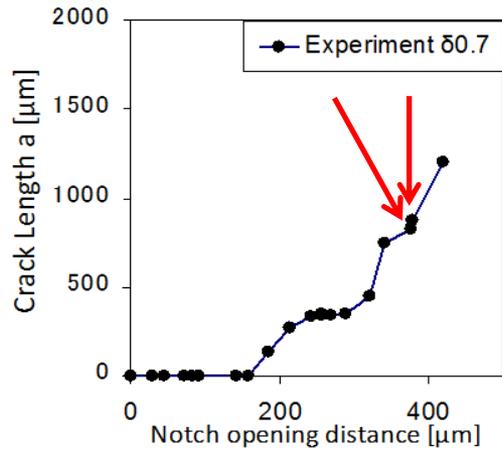
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direction
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⊗ Crack Propagation

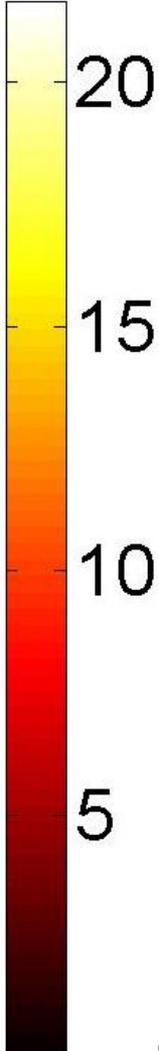
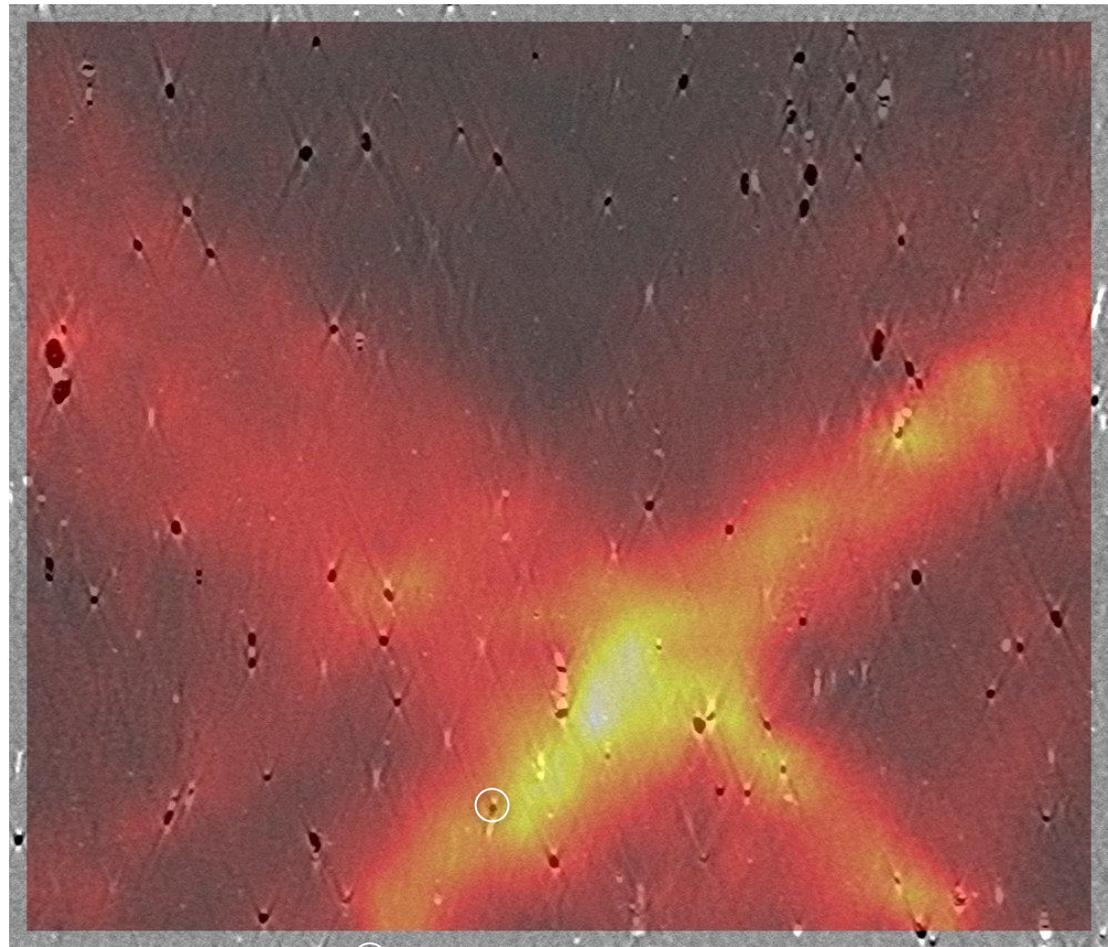


2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{t-u})$

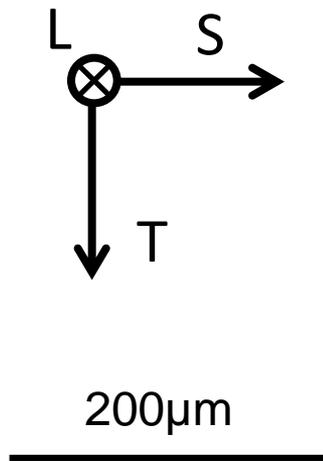
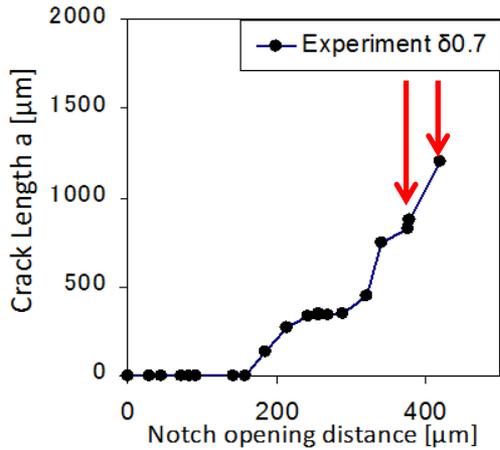


⊗ Crack Propagation

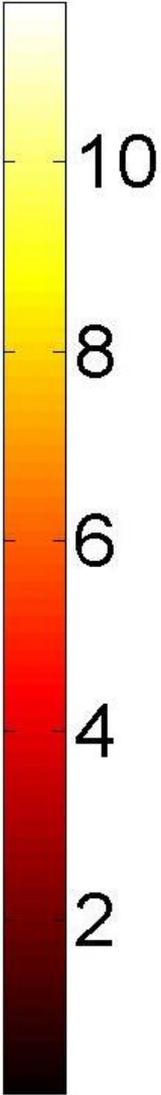
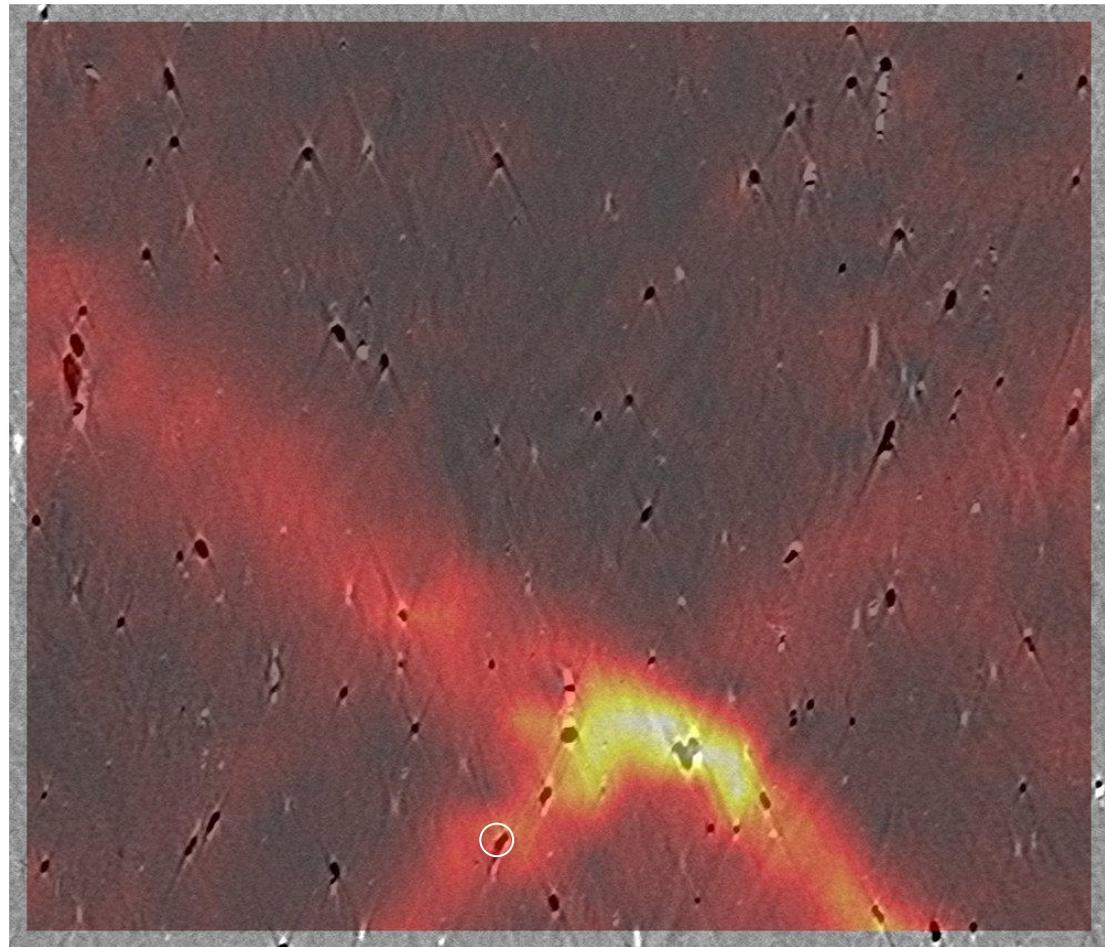


2139 T3: overlay strain + microstructure

Incremental strain: $p_{eq}(u_{u-v})$

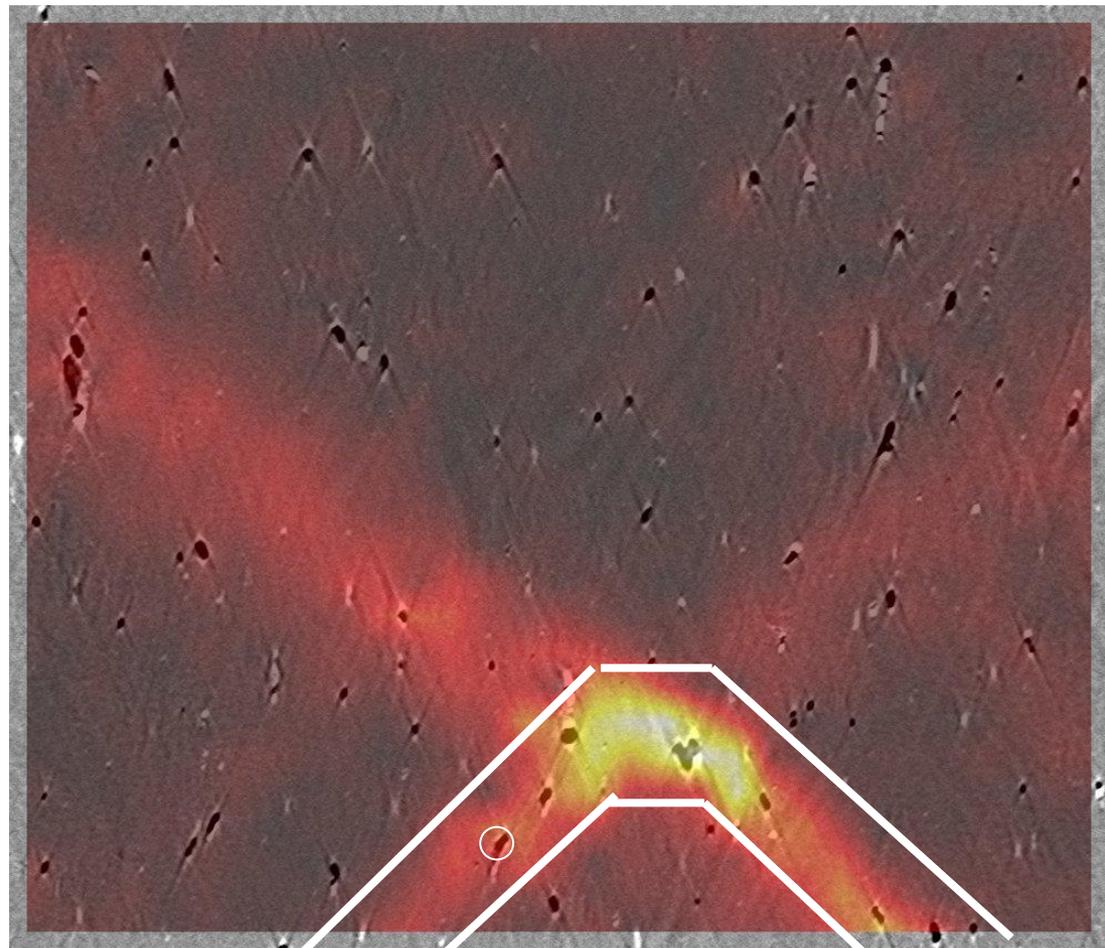
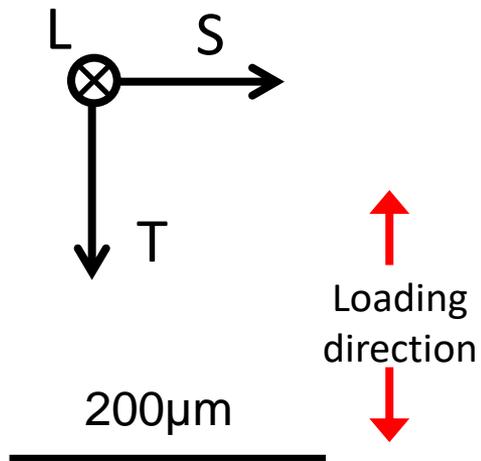
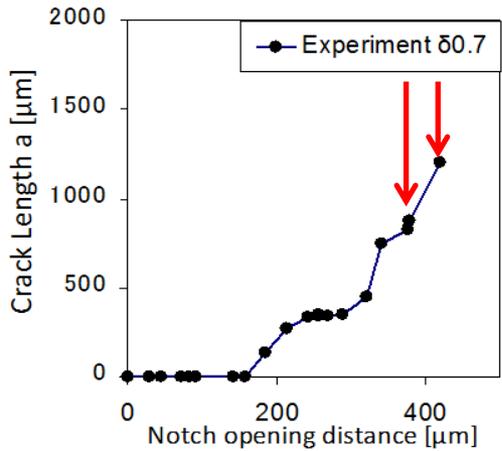


⊗ Crack Propagation



Incremental strain field: 2139 T3

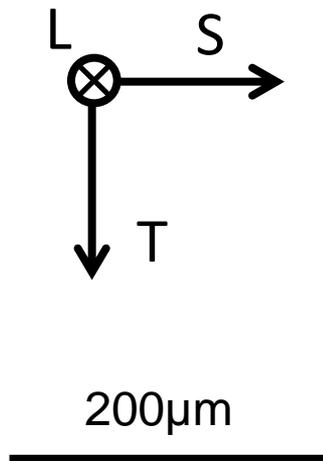
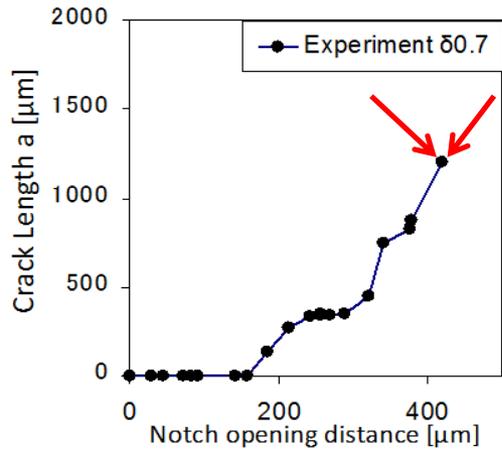
Incremental strain: $p_{eq}(u_{u-v})$



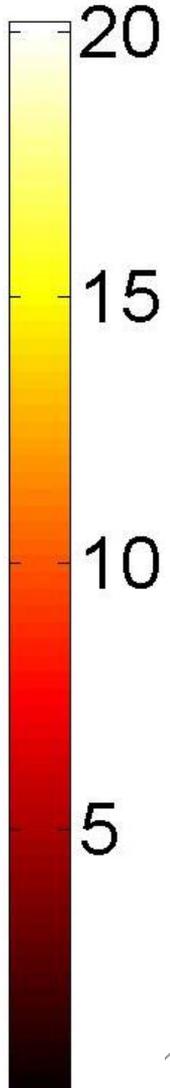
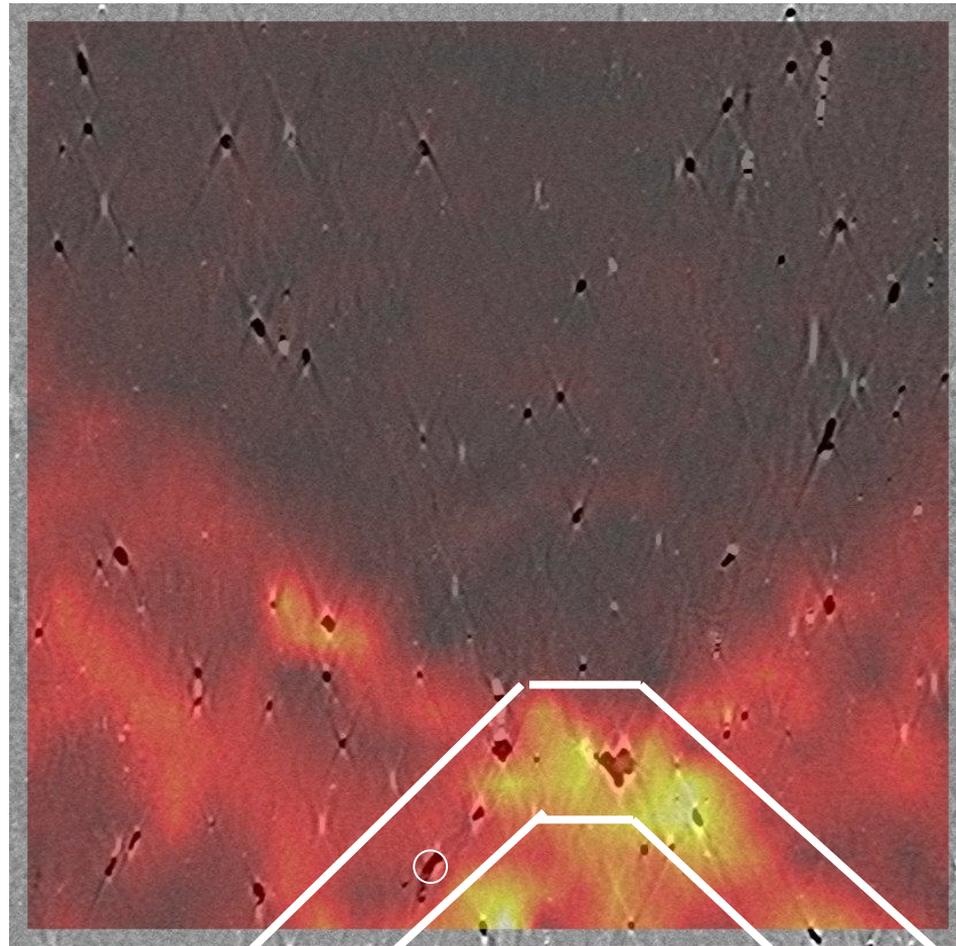
⊗ Crack Propagation

2139 T3: overlay strain + microstructure

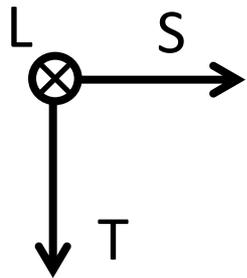
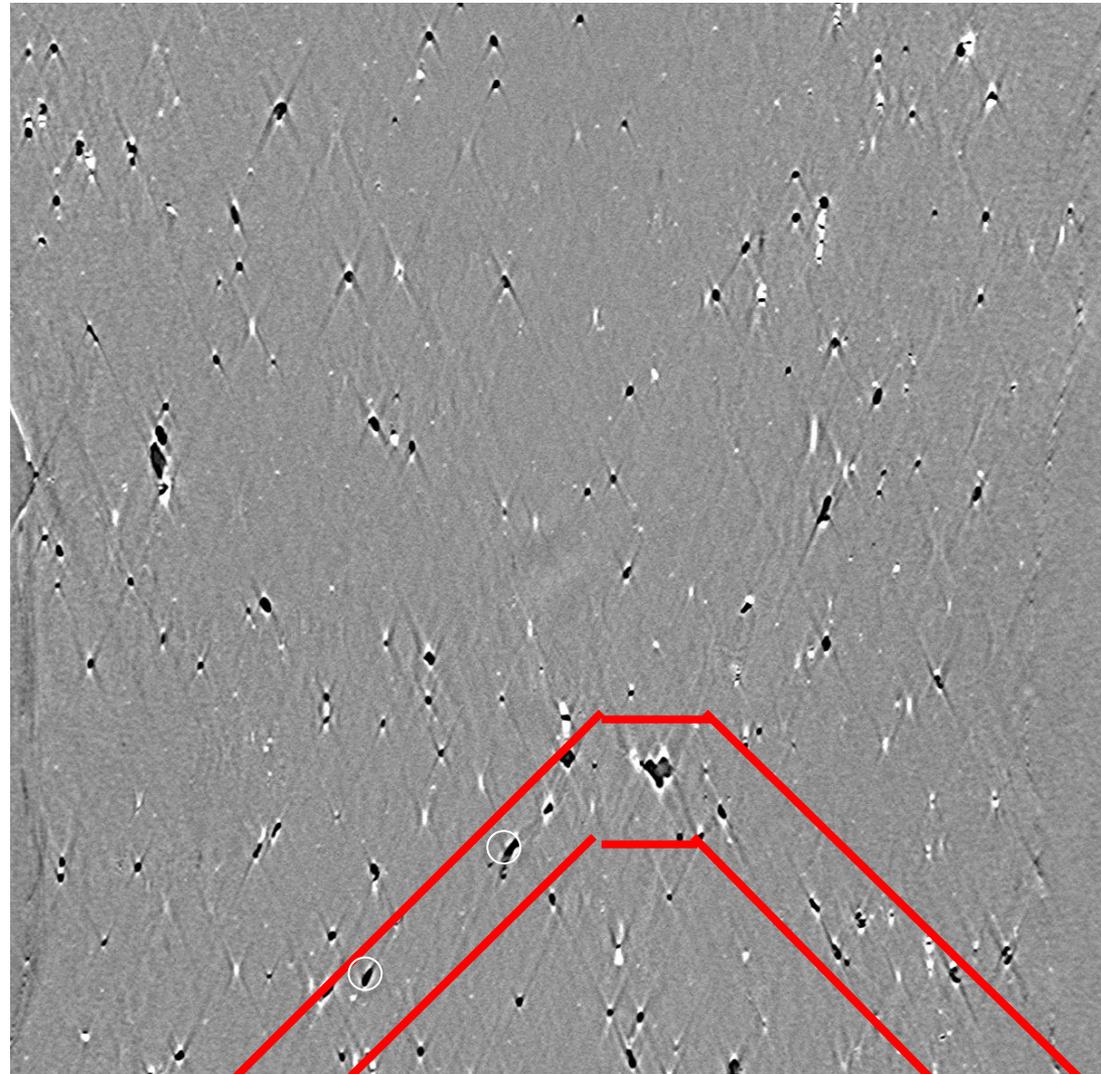
Incremental strain: $p_{eq}(u_{v-w})$



⊗ Crack Propagation



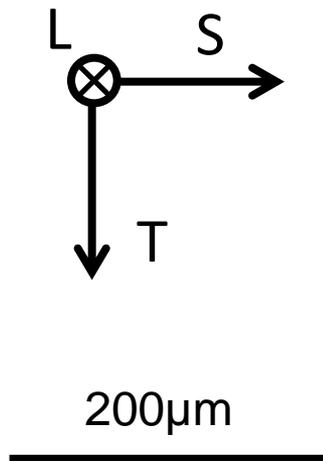
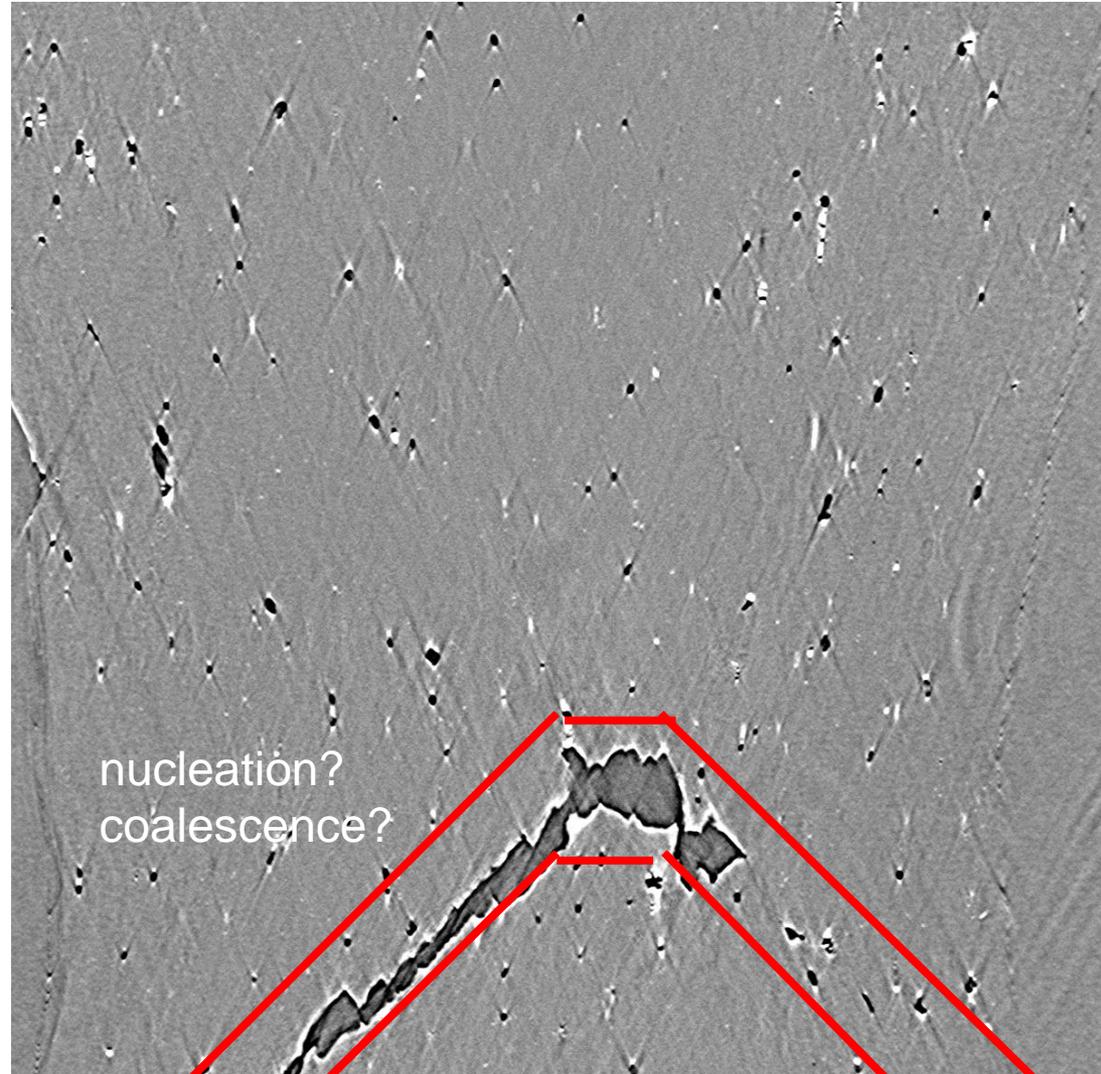
Microstructure: 2139 T3



200 μ m

⊗ Crack Propagation

Microstructure: 2139 T3 at fracture

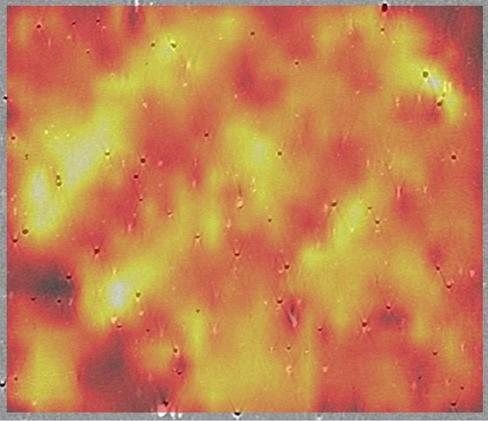
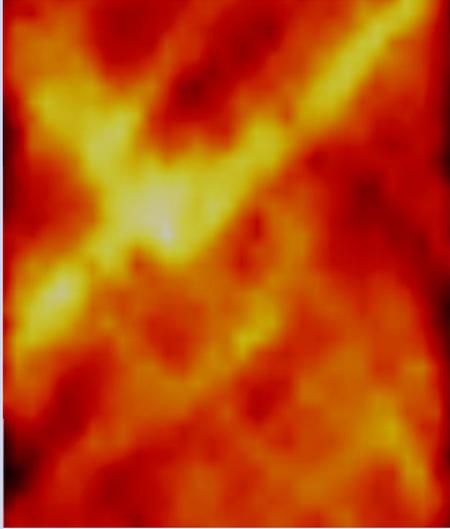
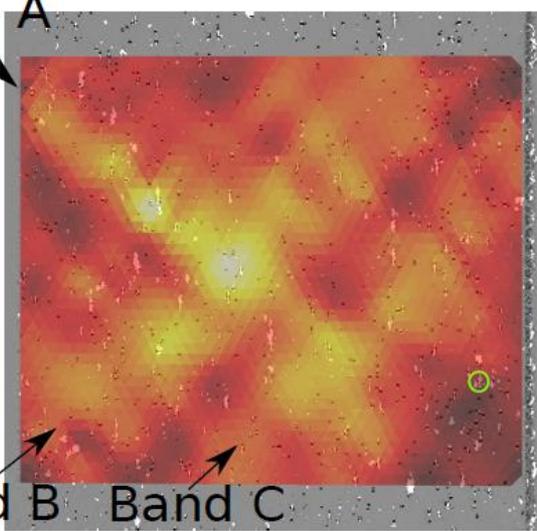
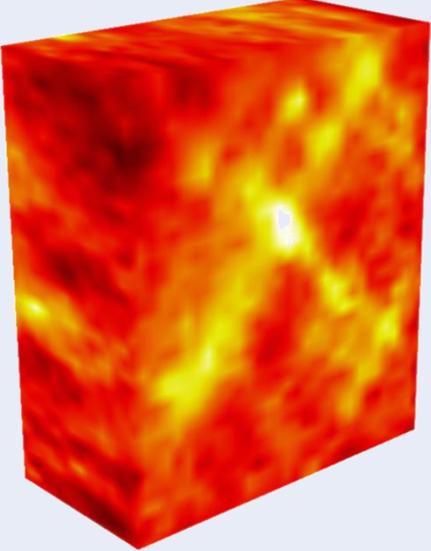


↑
Loading
direction
↓

⊗ Crack Propagation

Slant strain bands with intermittent activity precede slant fracture

Strain field for AA2139 T8

AA	2139 (initial voids)	2198 recrystallized (no voids)
T3	 <p data-bbox="991 287 1103 375">max 4.0 %</p> <p data-bbox="465 589 967 629">Morgeneyer et al. JMPS 2016</p>	 <p data-bbox="1740 287 1852 375">max 3.8 %</p> <p data-bbox="1740 508 1899 629">Buljac et al. Acta Mat 2014</p>
T8	 <p data-bbox="996 822 1174 1032">max 3.3% 1040 μm</p> <p data-bbox="465 1253 967 1293">Buljac et al. EJM- A submitted</p>	 <p data-bbox="1760 896 1852 985">max 5 %</p> <p data-bbox="1271 1246 1818 1286">Morgeneyer et al. Acta Mat 2014</p>

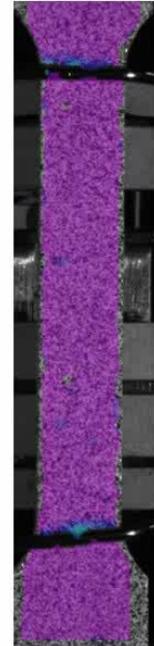
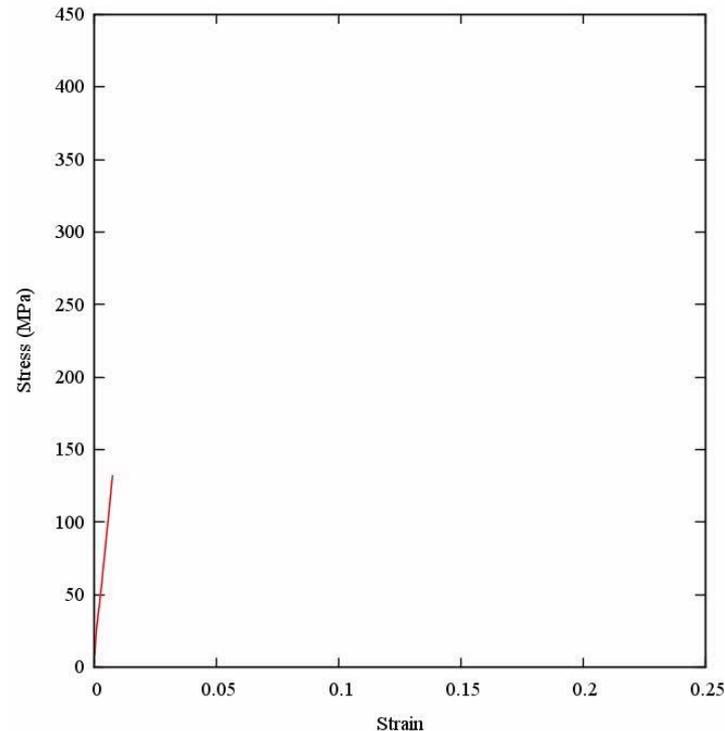
All four materials show early slant concentration bands in the slant fracture region

- Plasticity needs to be reproduced numerically to predict slant fracture
 - Possible localisation origin:
 - Strain rate effects : **Portevin Le Chatelier !**
 - Crystal plasticity and texture
 - Other yield behavior / criteria (e.g. Mohr Coulomb)???

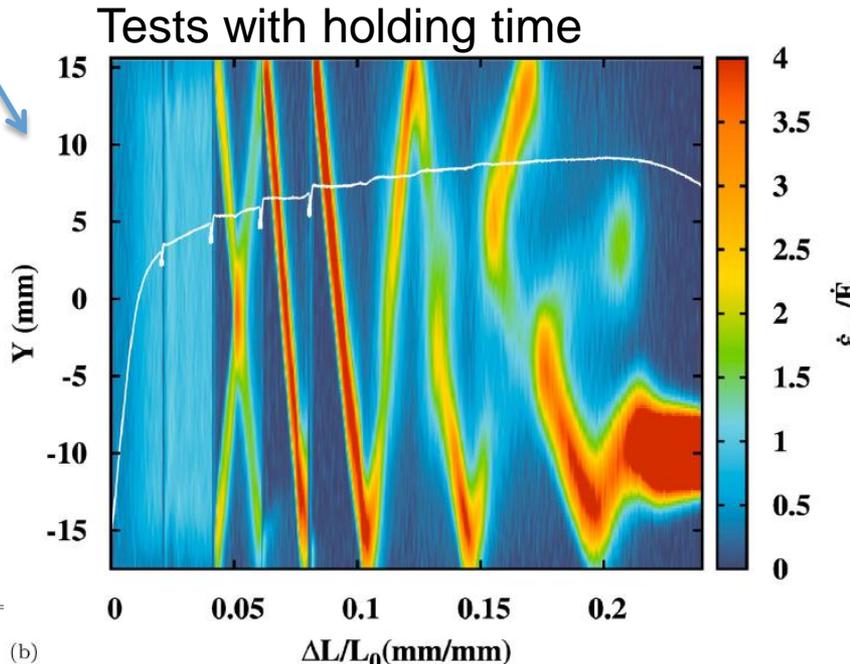
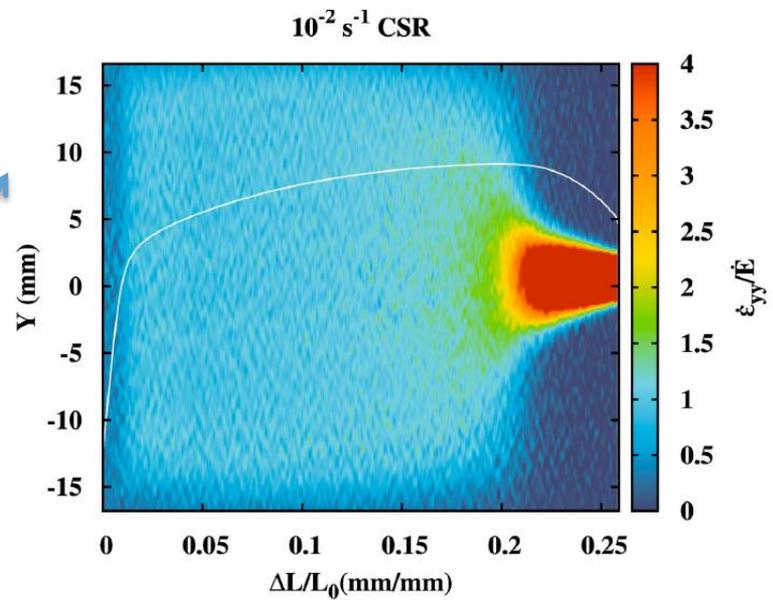
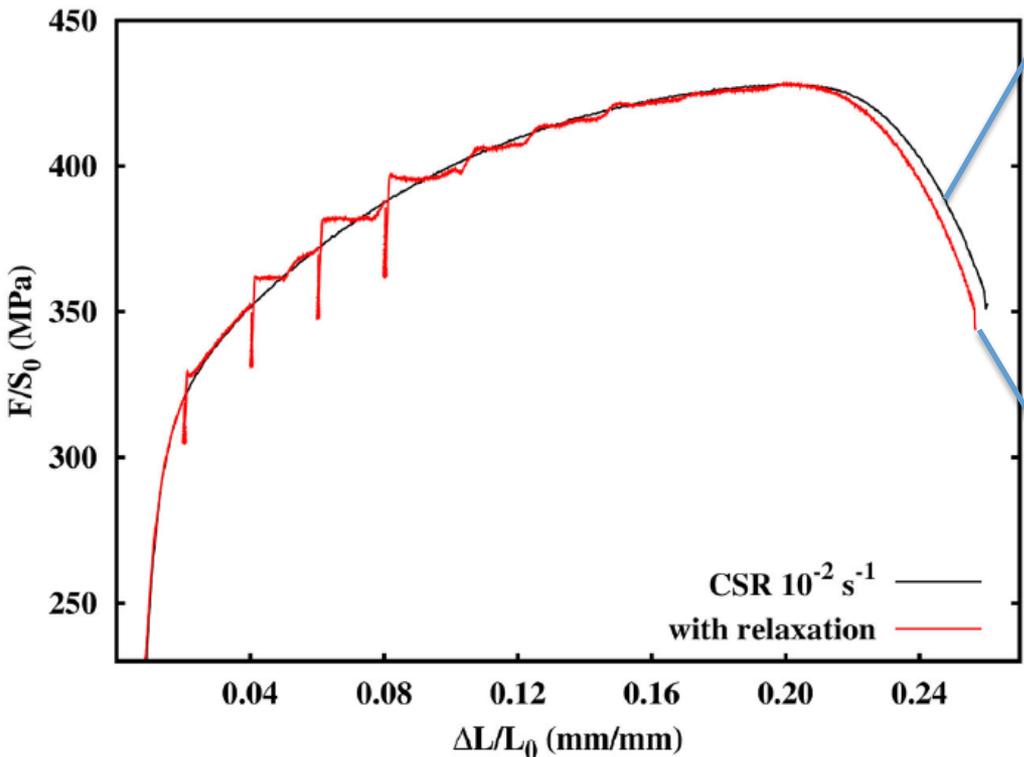
Portevin Le Chatelier Effect?

- Steel (Interstitial atoms: C, N),
Aluminium alloy (Substitutional atoms:
Cu, Li, Mg)
- Dynamic strain ageing: interaction
between mobile dislocations and solute
diffusion
- Macroscopic manifestation:
 - Temperature and strain rate related;
 - Negative strain rate sensitivity (nSRS);
 - Serrated yielding and localized
deformation.
- Consequences
 - Unsightly surface marking;
 - Loss of toughness and ductility.

2198 T3

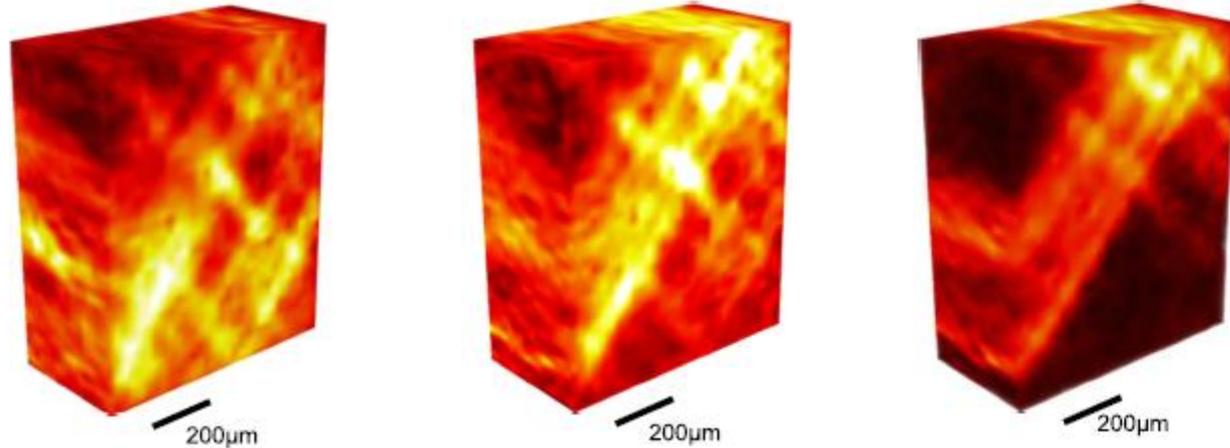


PLC effect in 2139 T3 triggered by interrupted loading

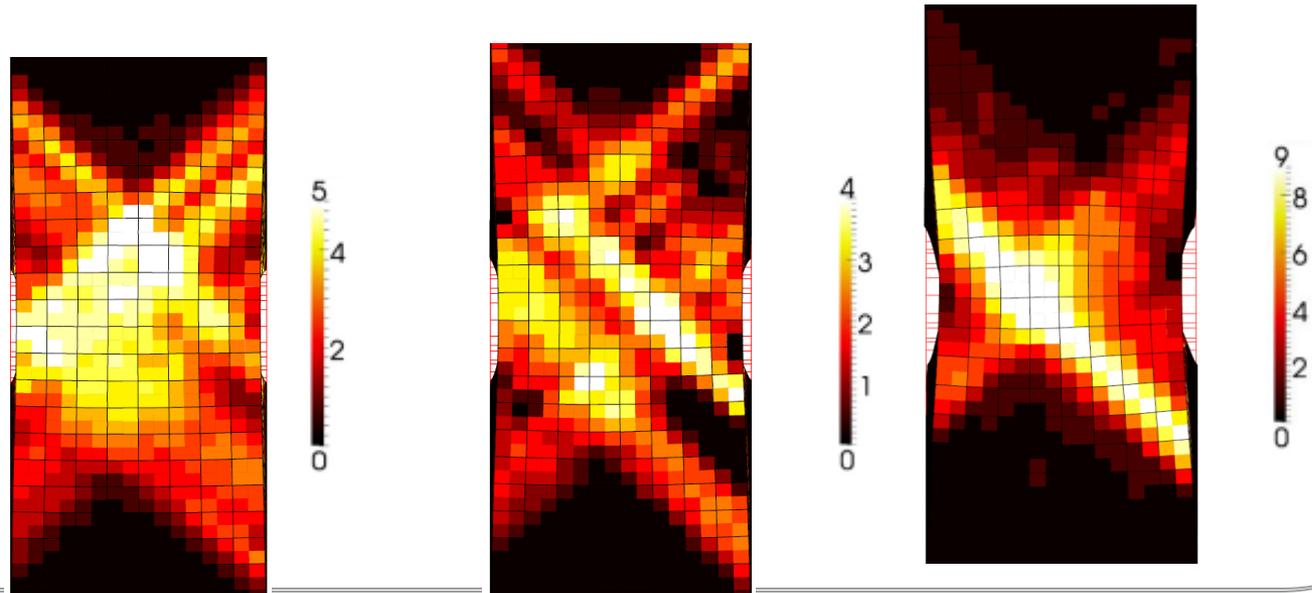


McCormick + Rousselier model simulations

- Incremental fields : experiment (equivalent strain)

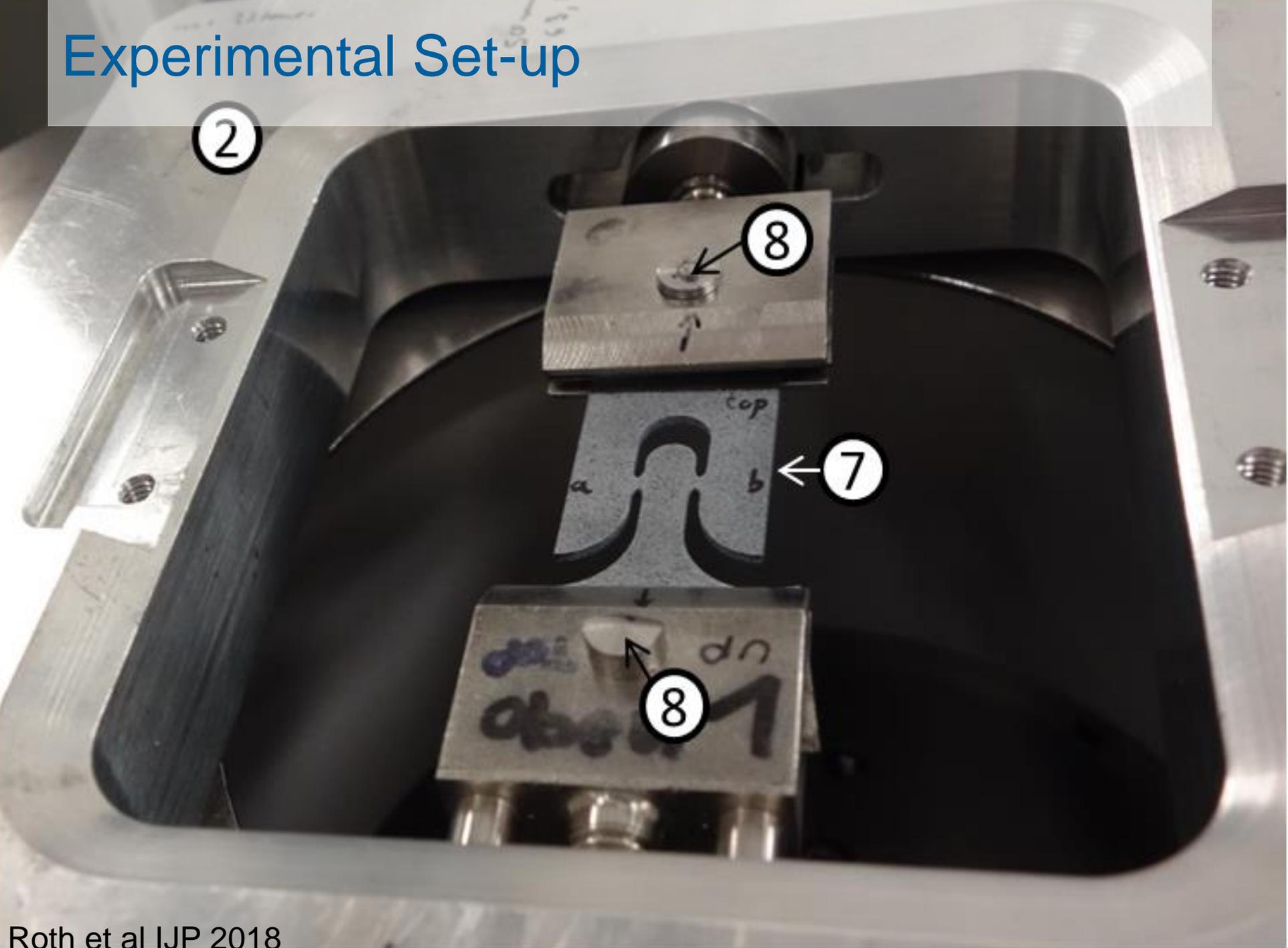


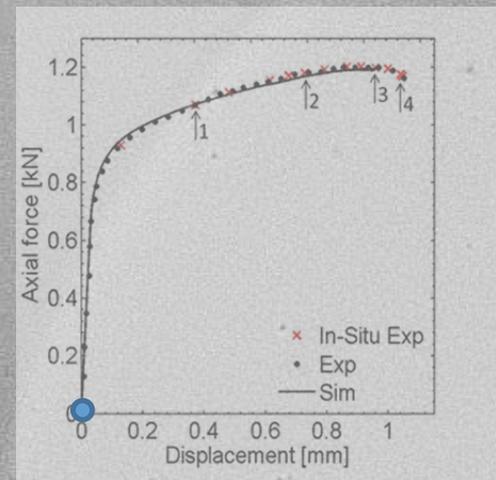
- Simulations



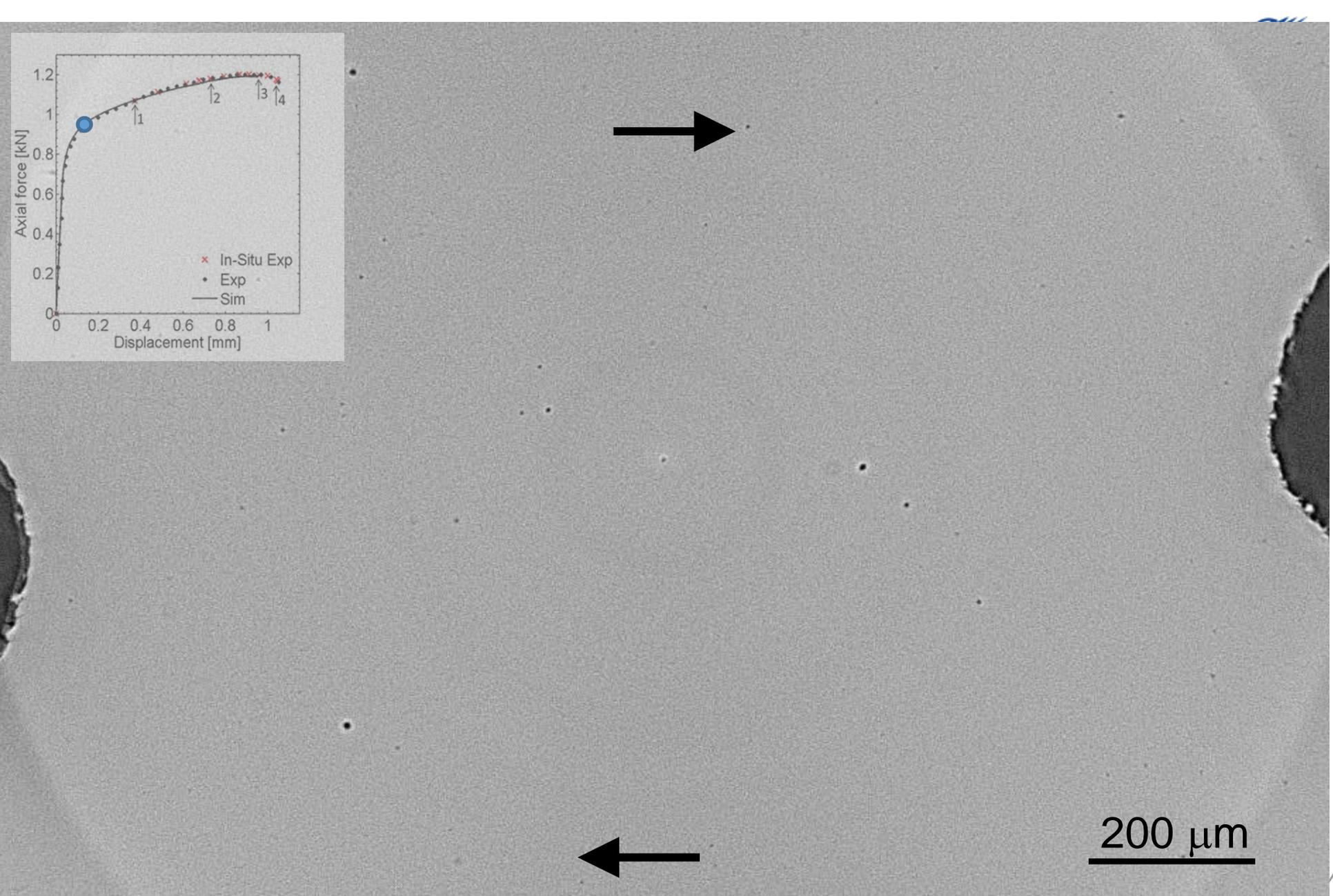
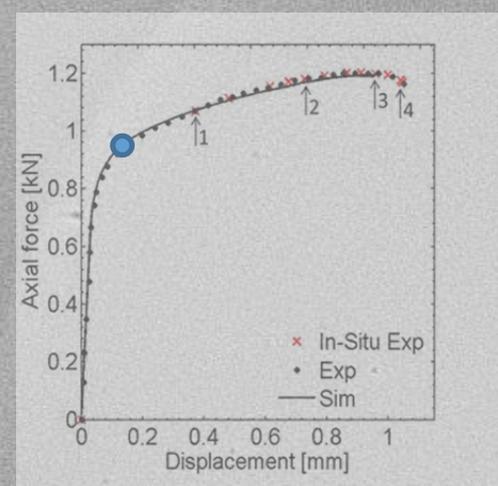
Changing loading conditions : shear loading

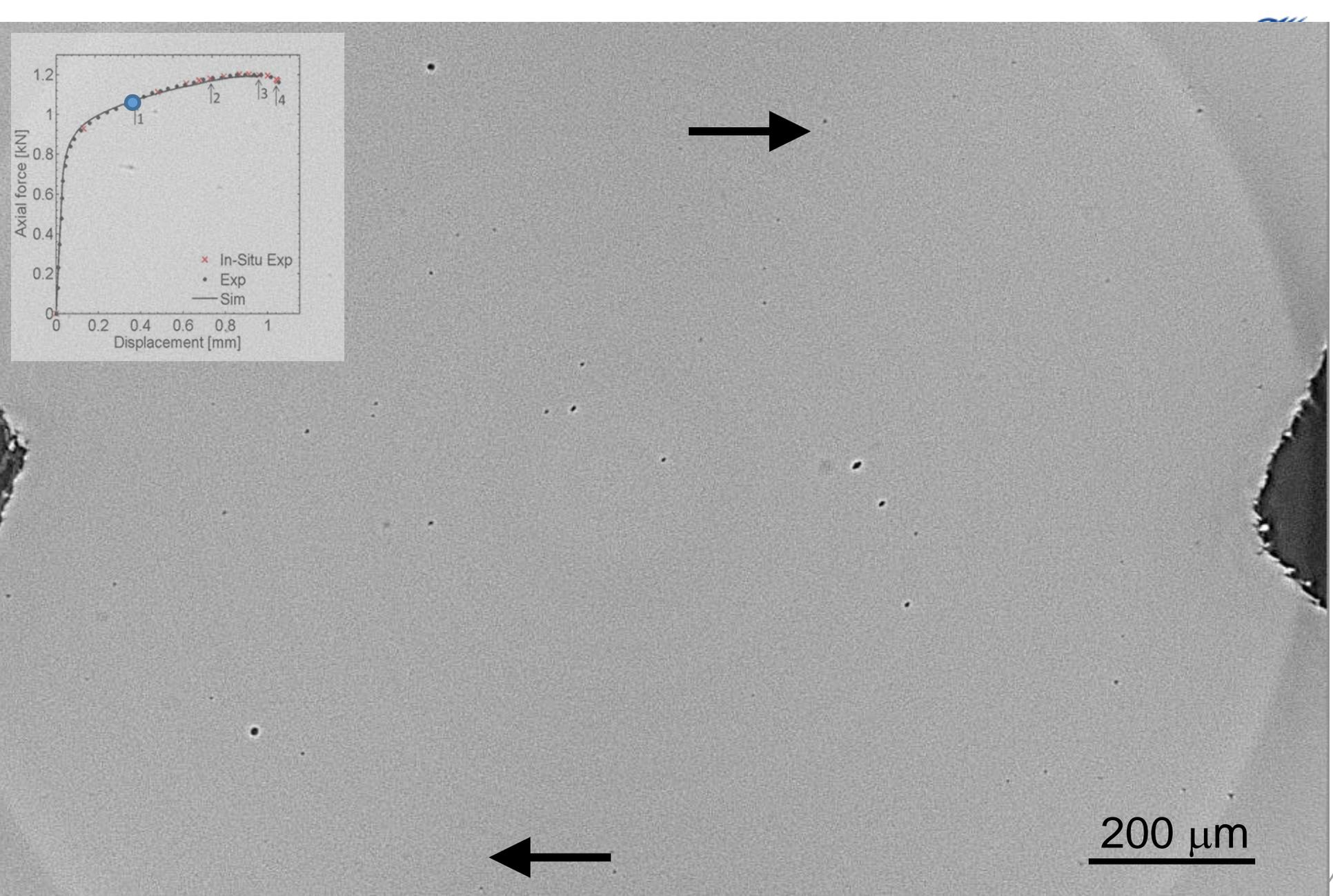
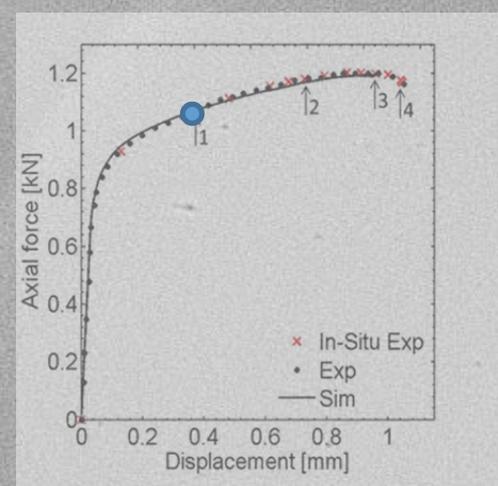
Experimental Set-up

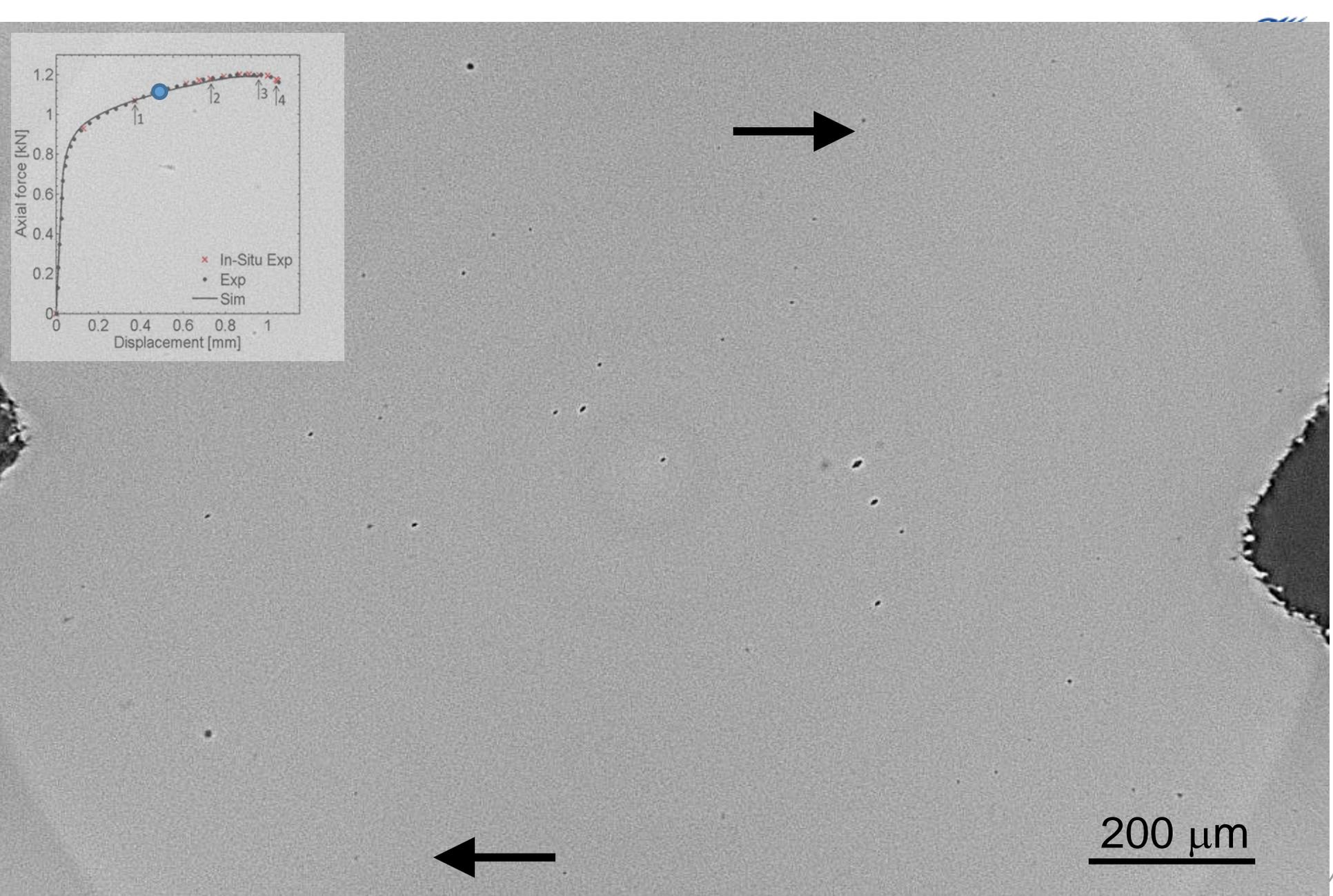
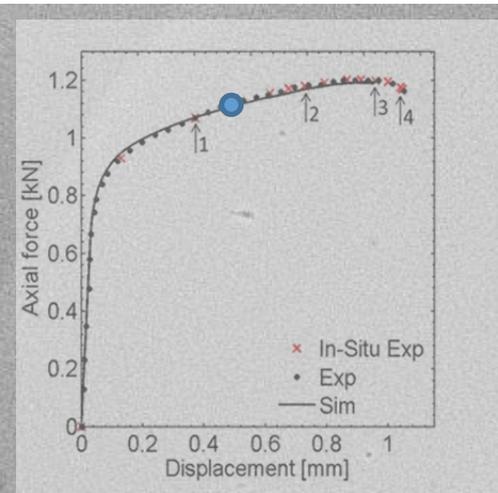




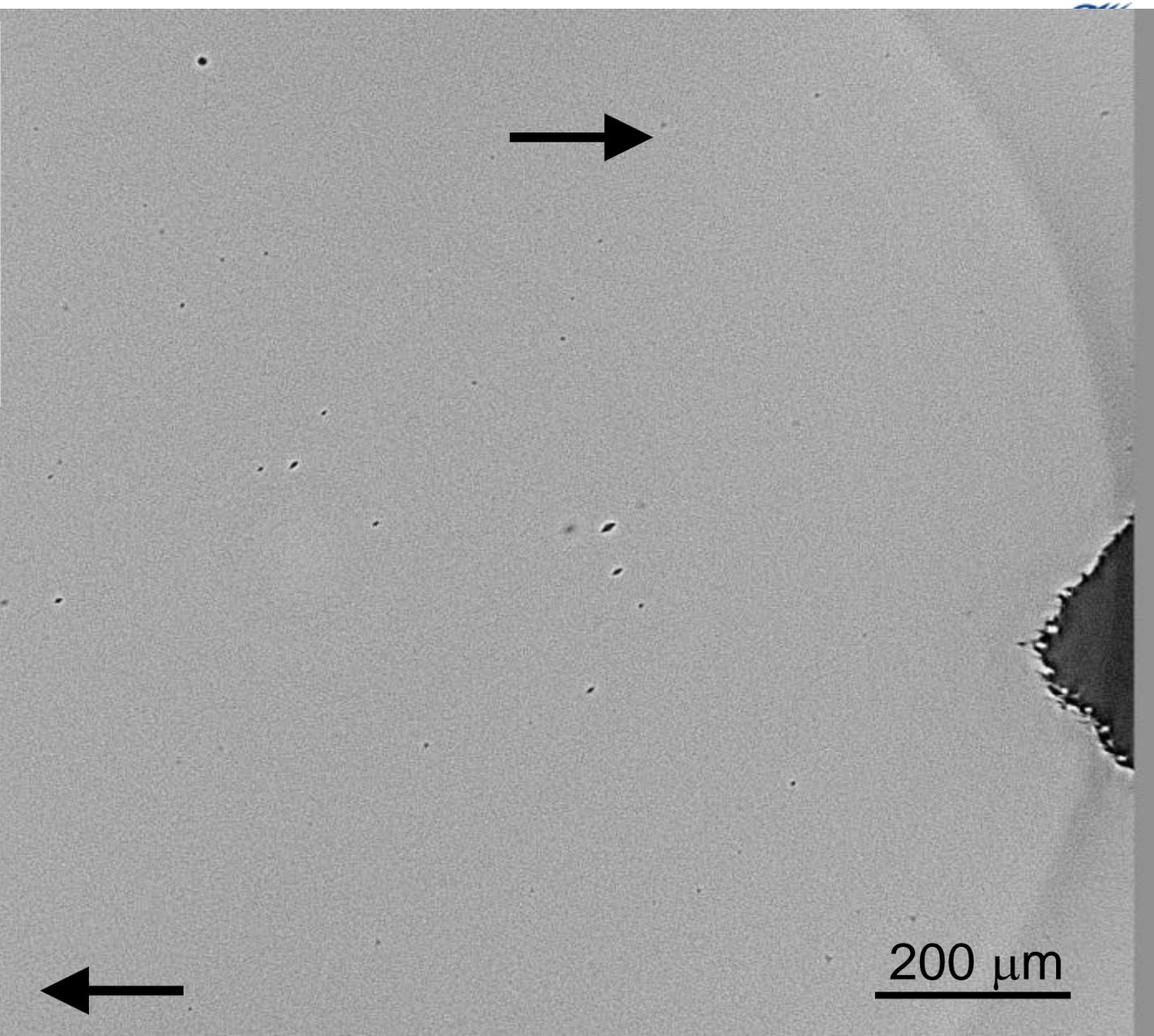
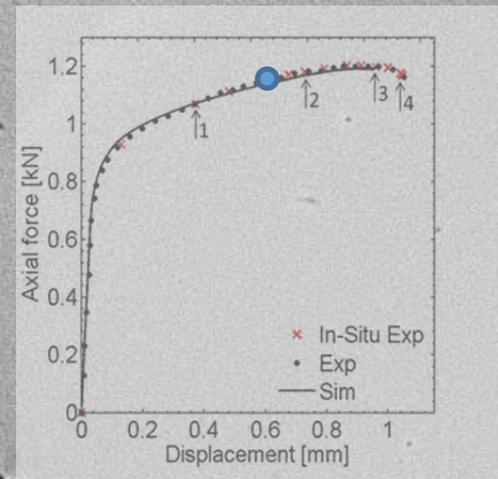
200 μm



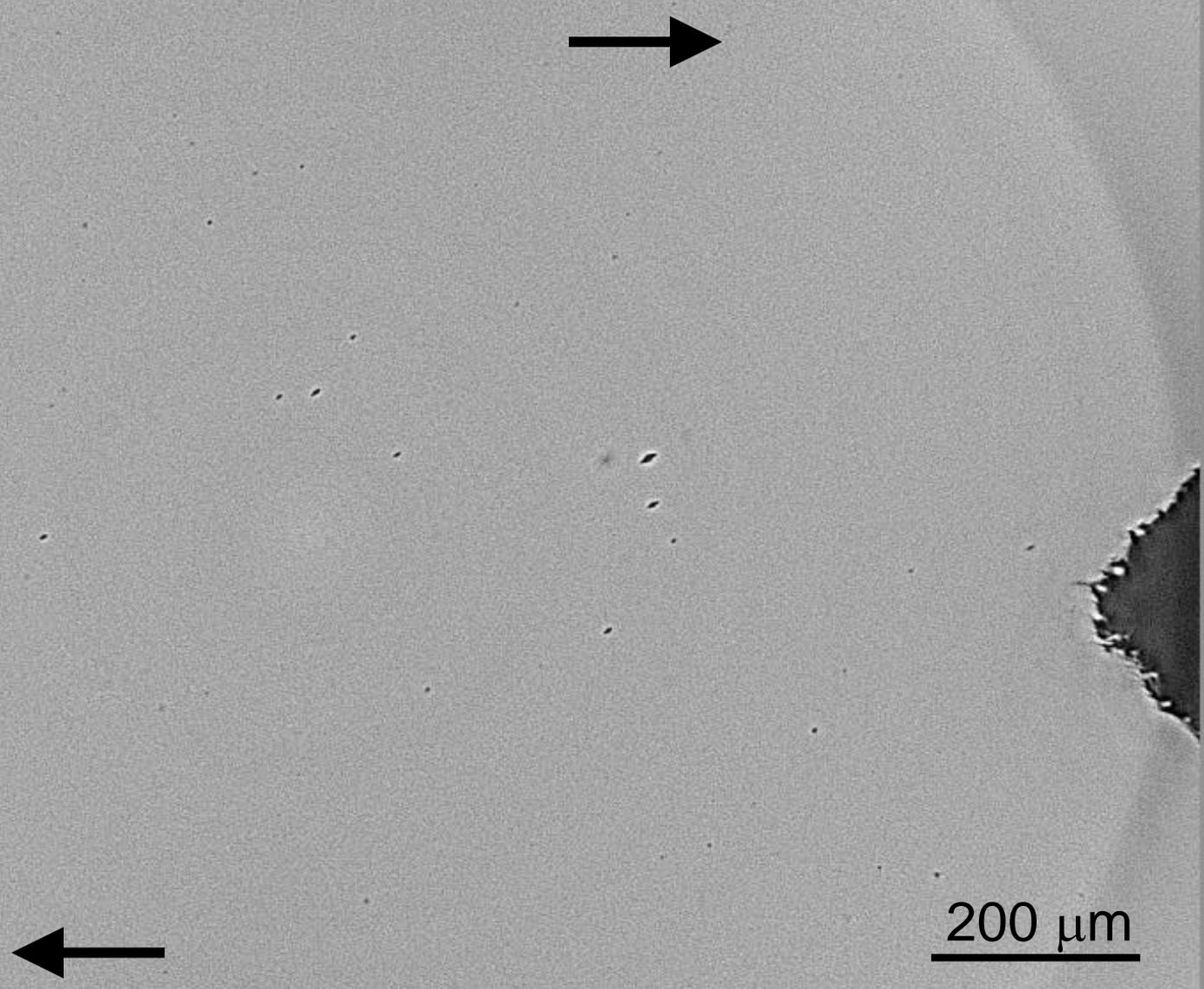
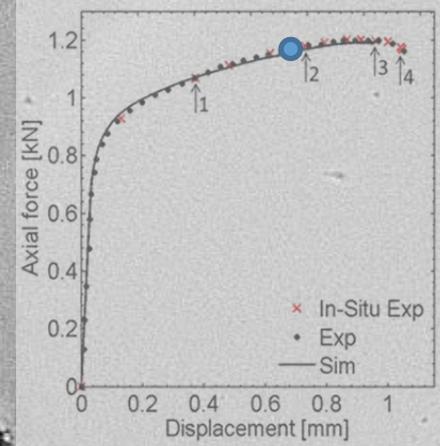




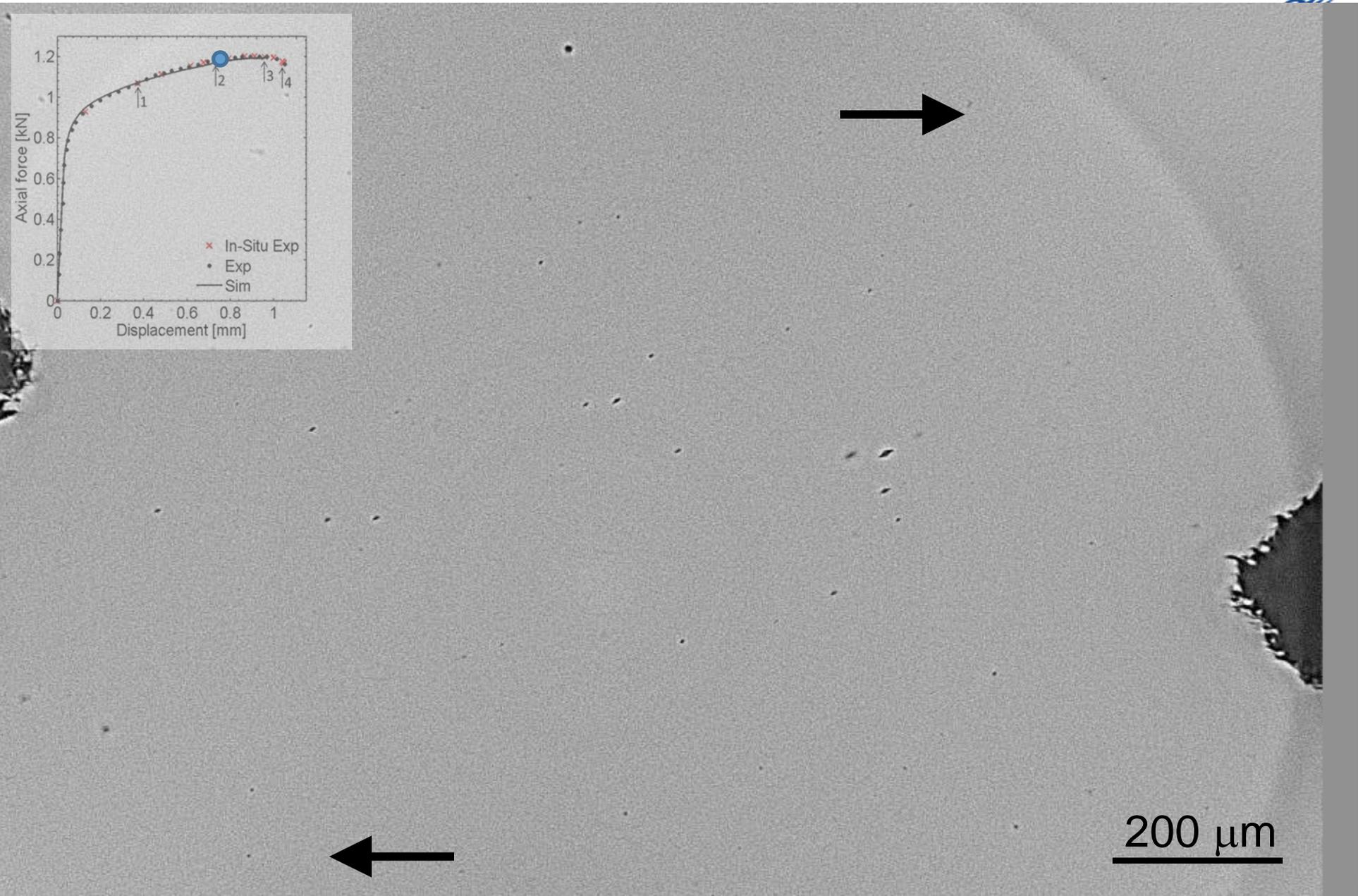
200 μm

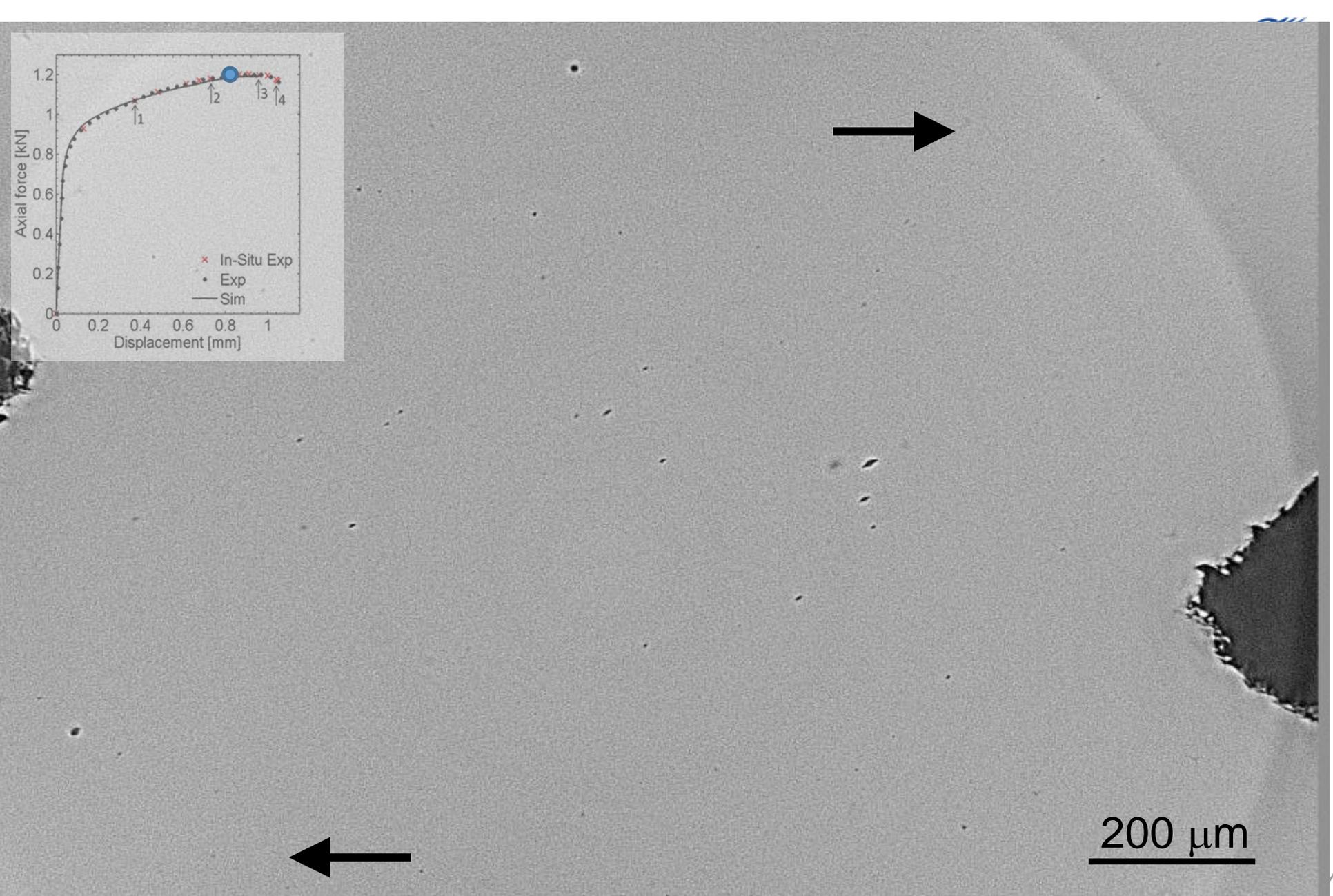
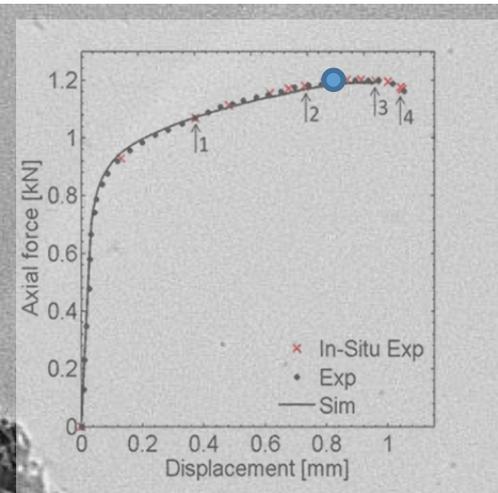


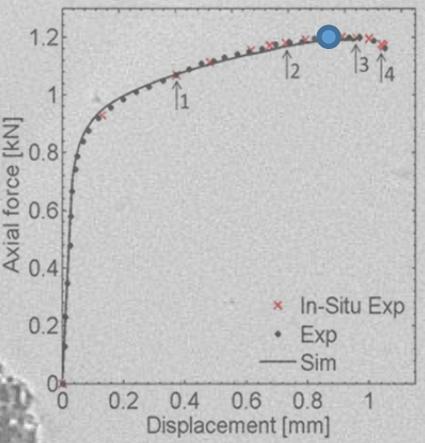
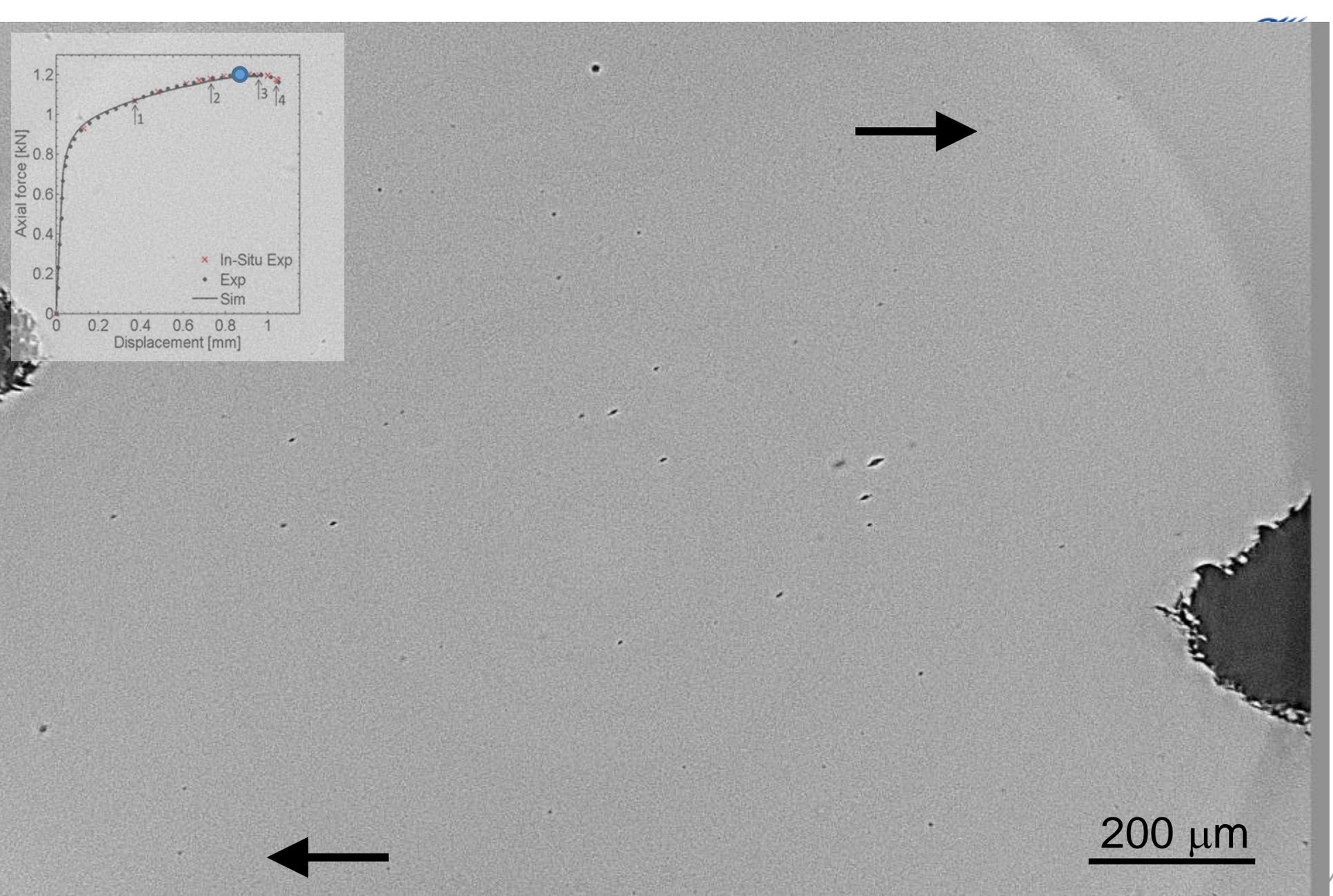
200 μm

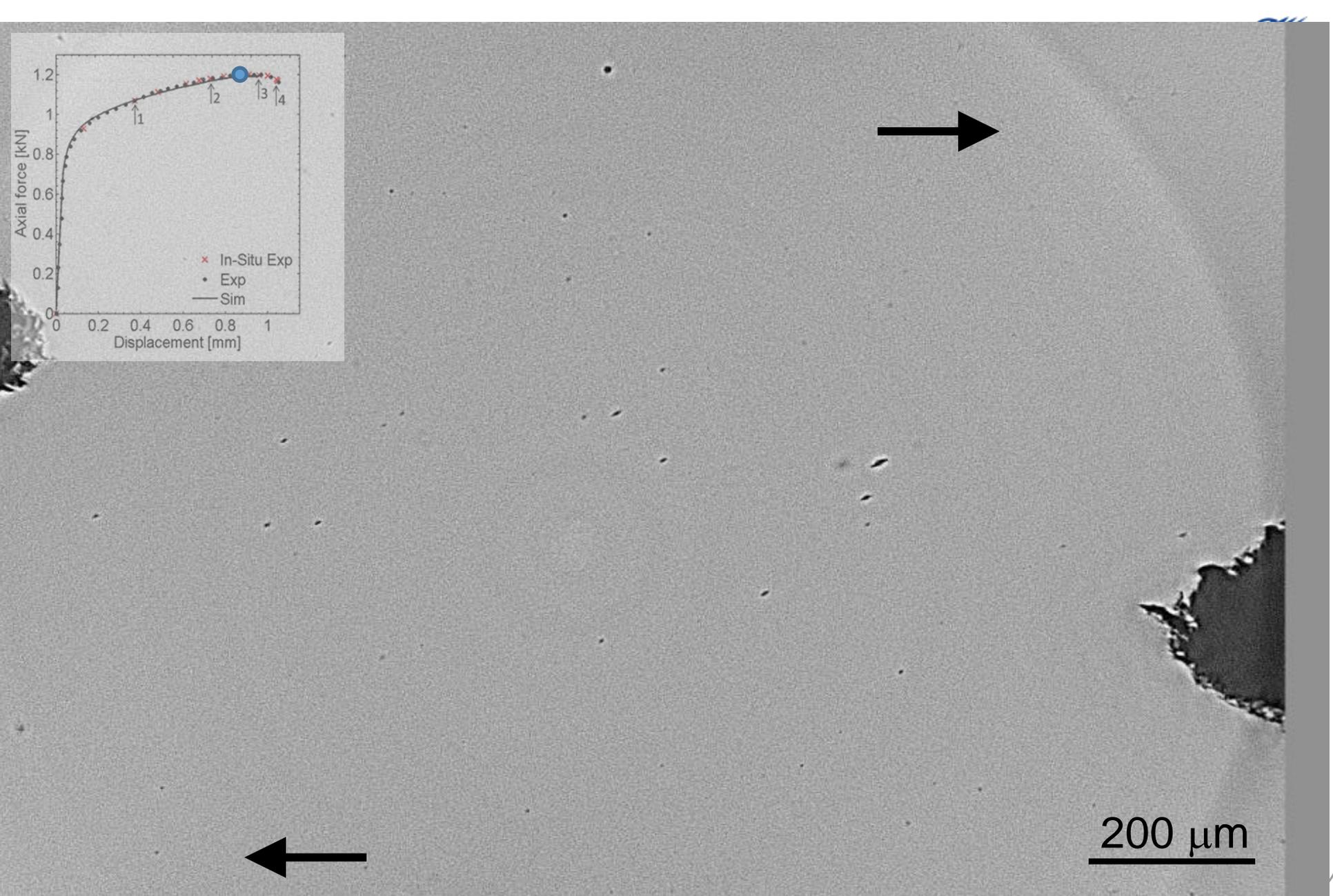
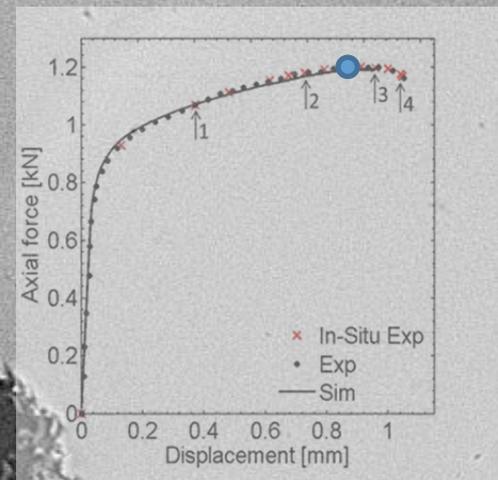


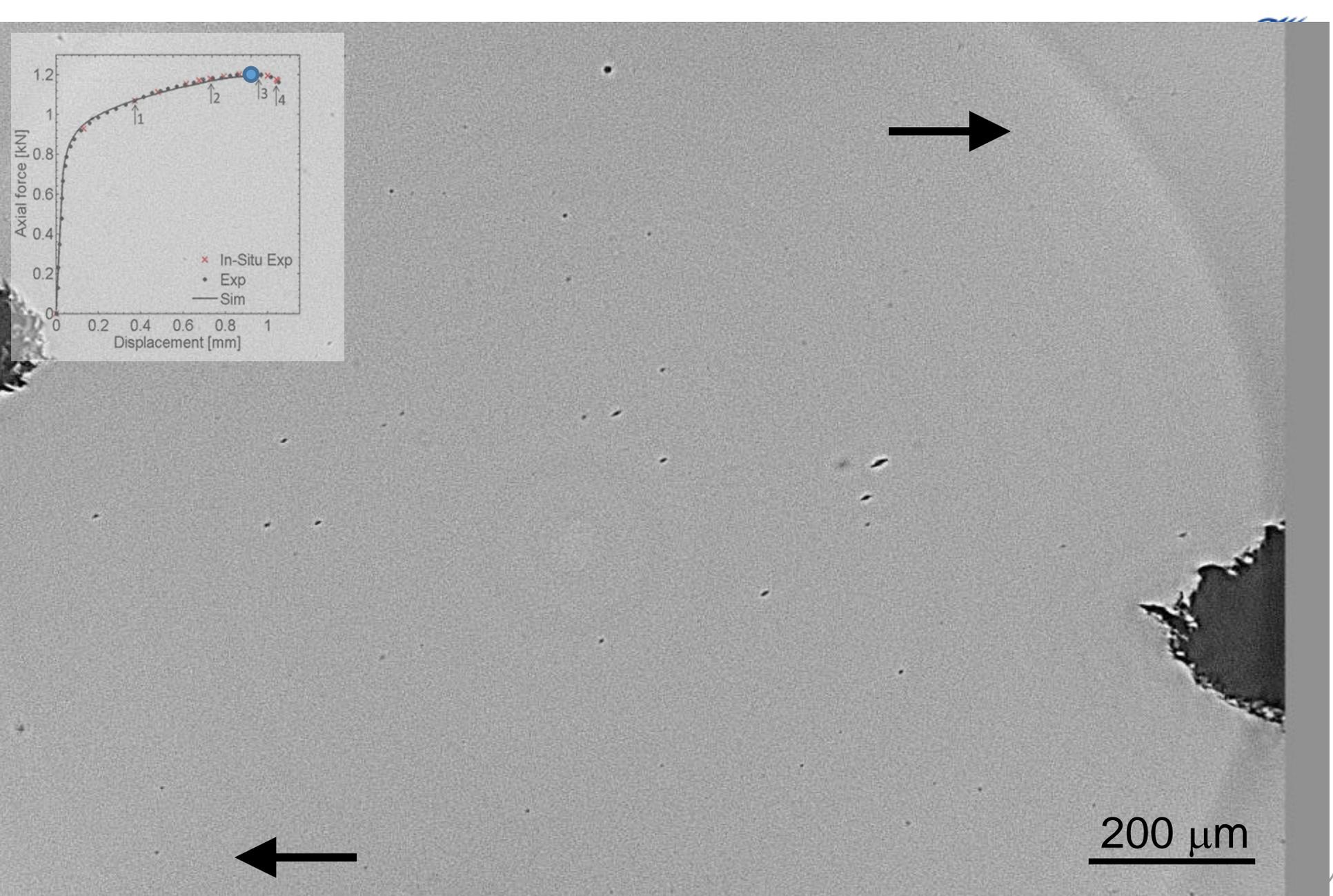
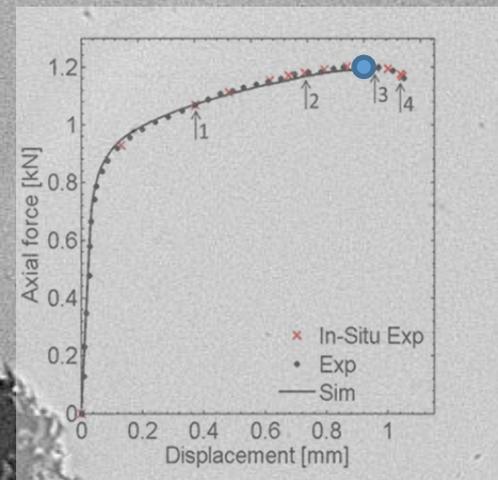
200 μm

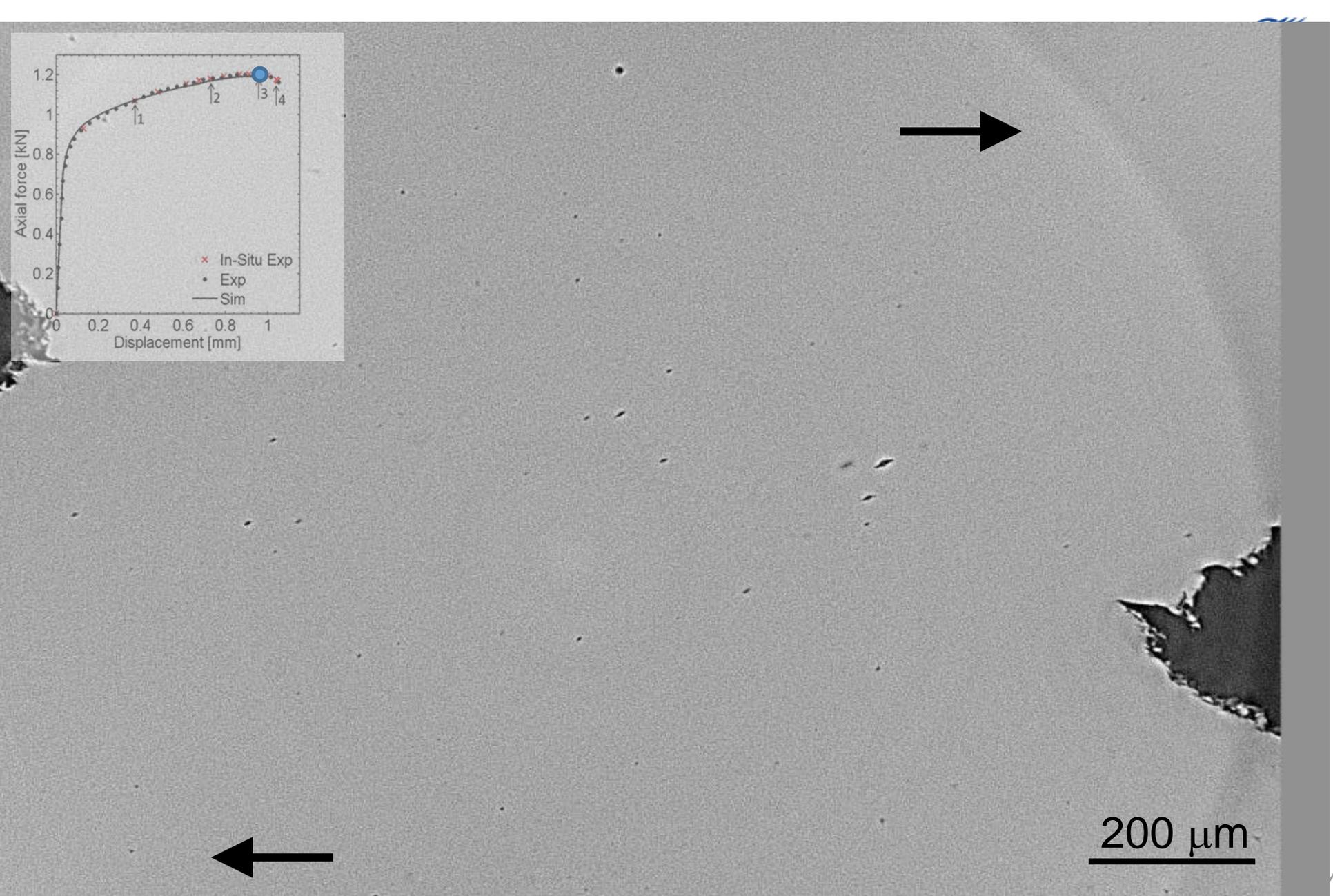
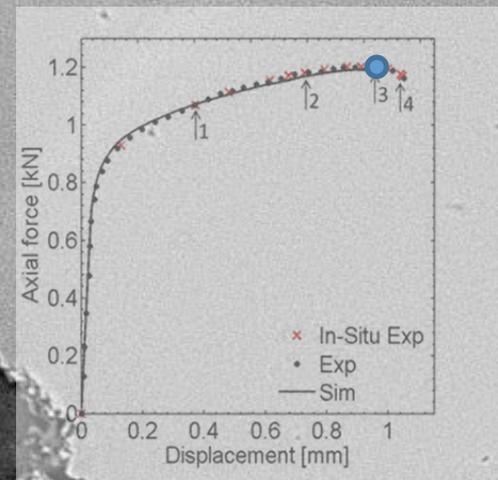


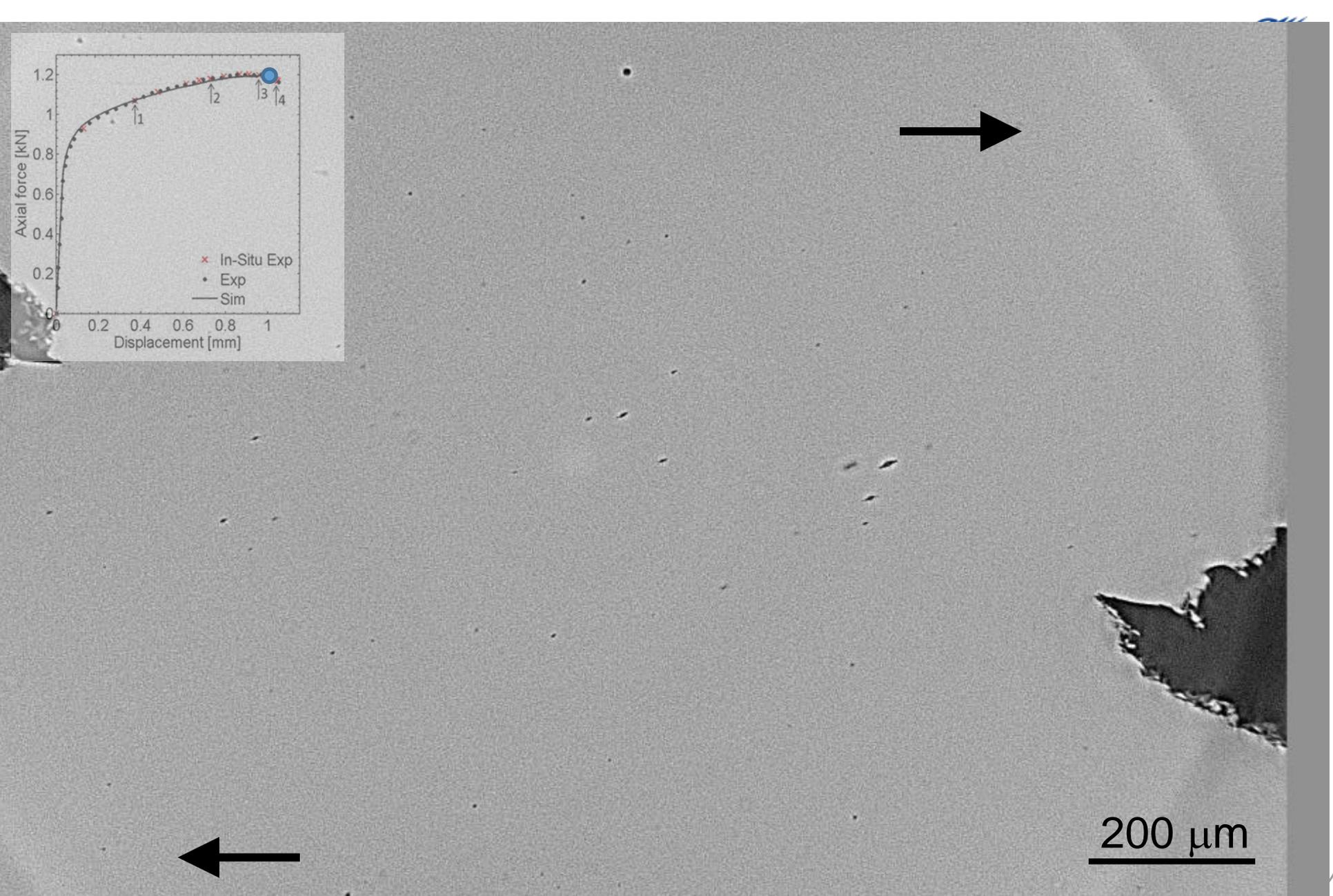
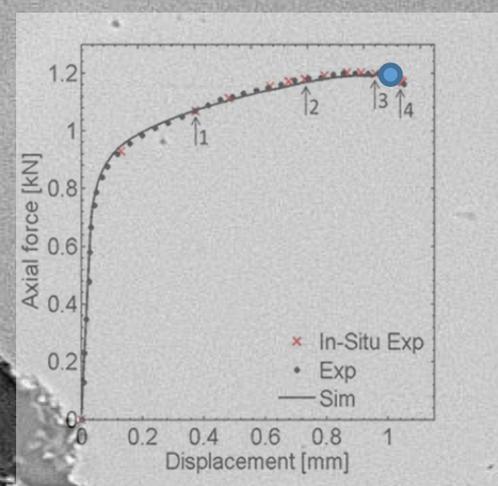


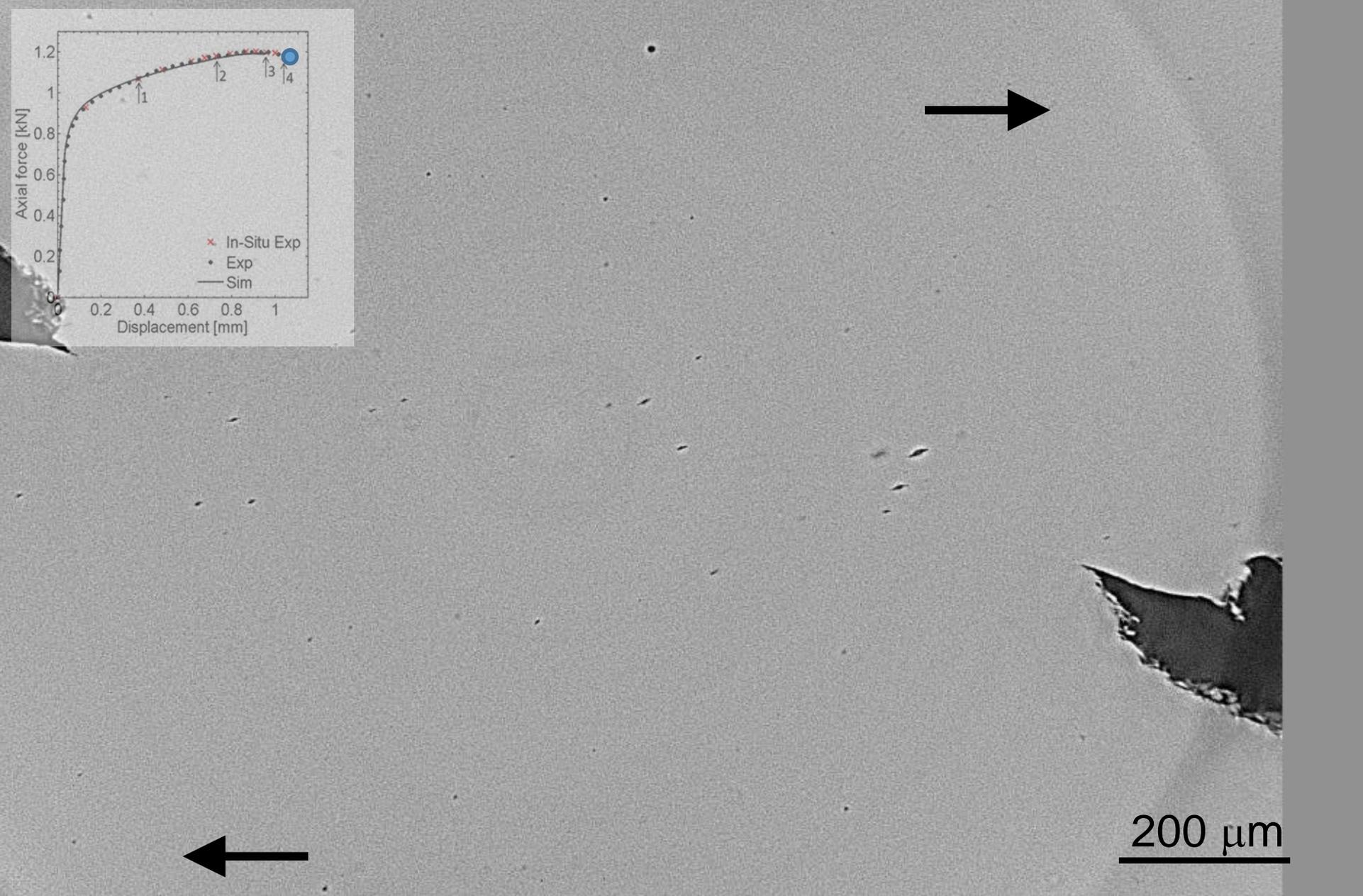


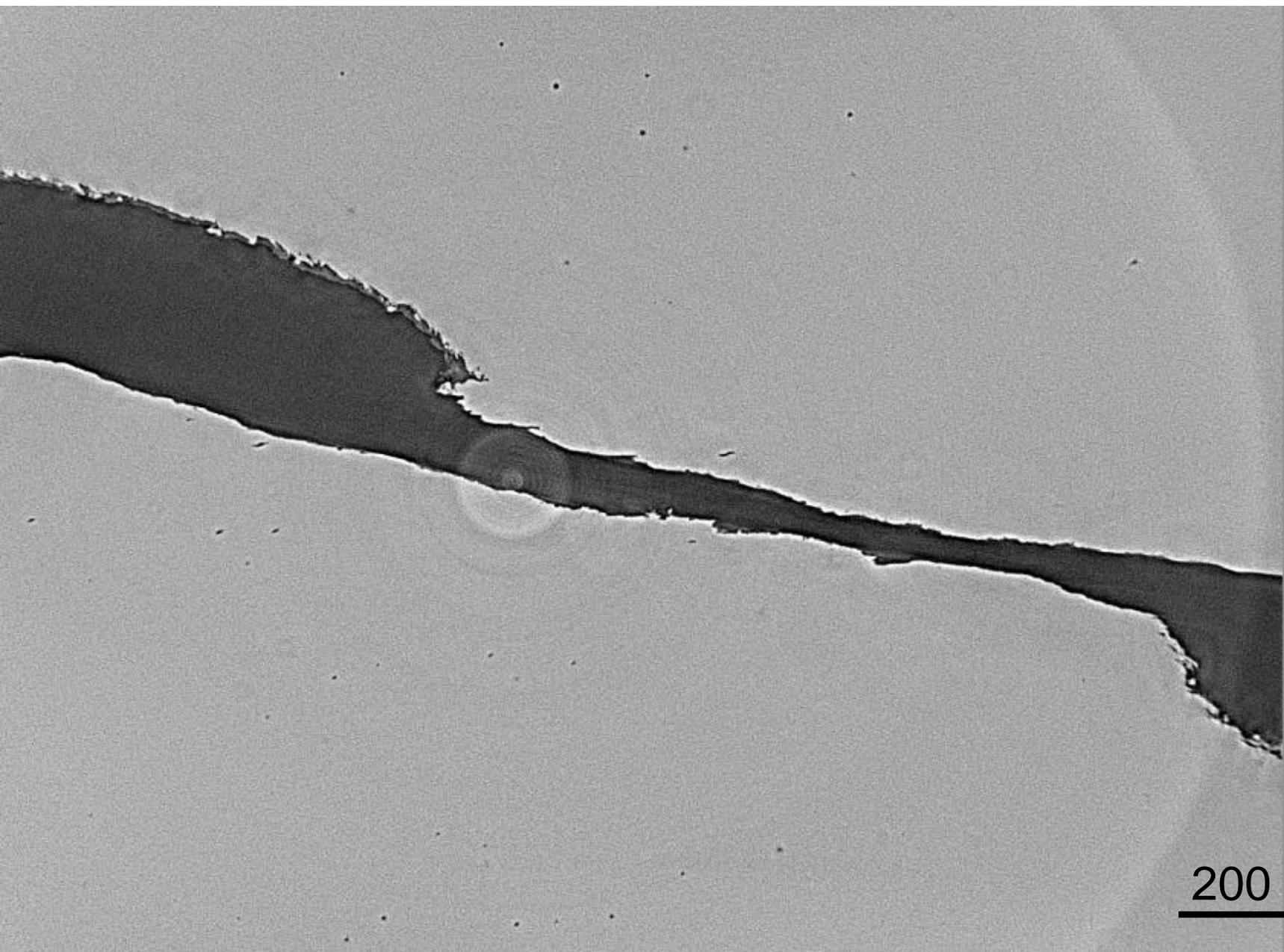






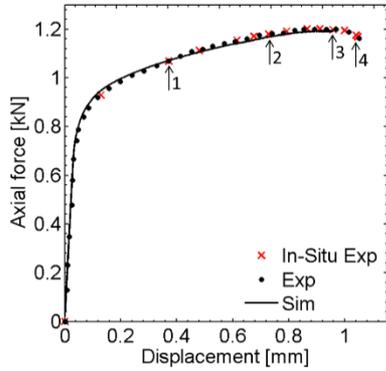






200 μm

Damage mechanism



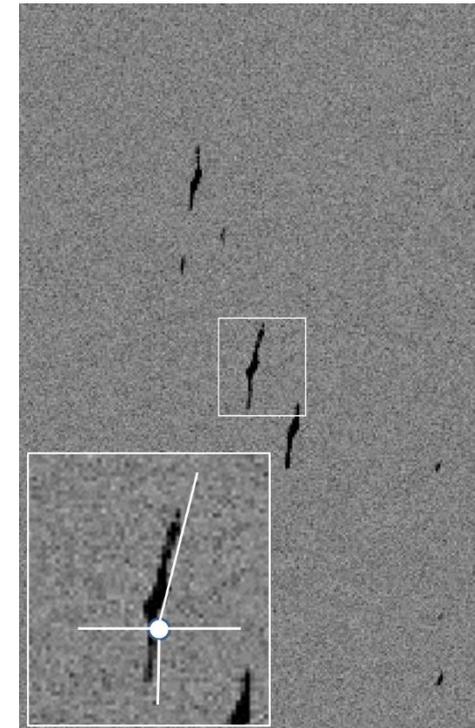
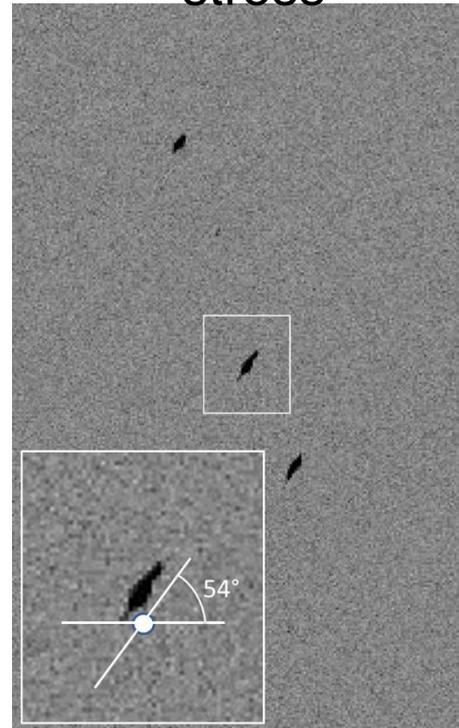
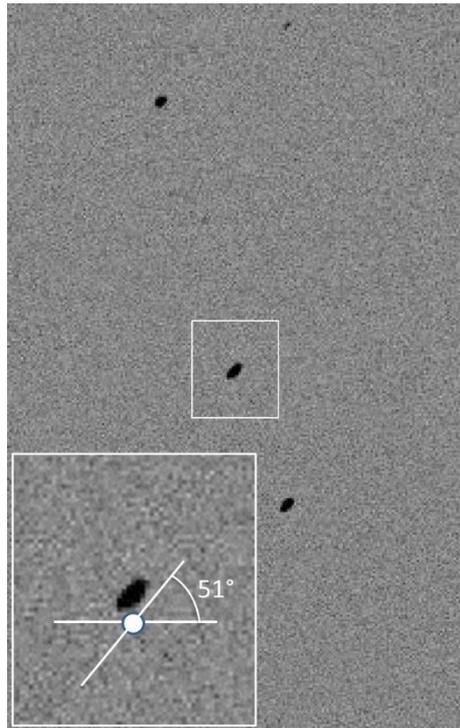
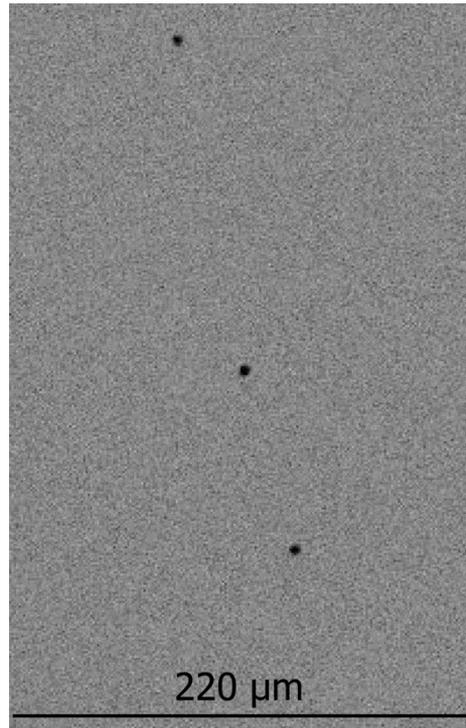
Particle debonding



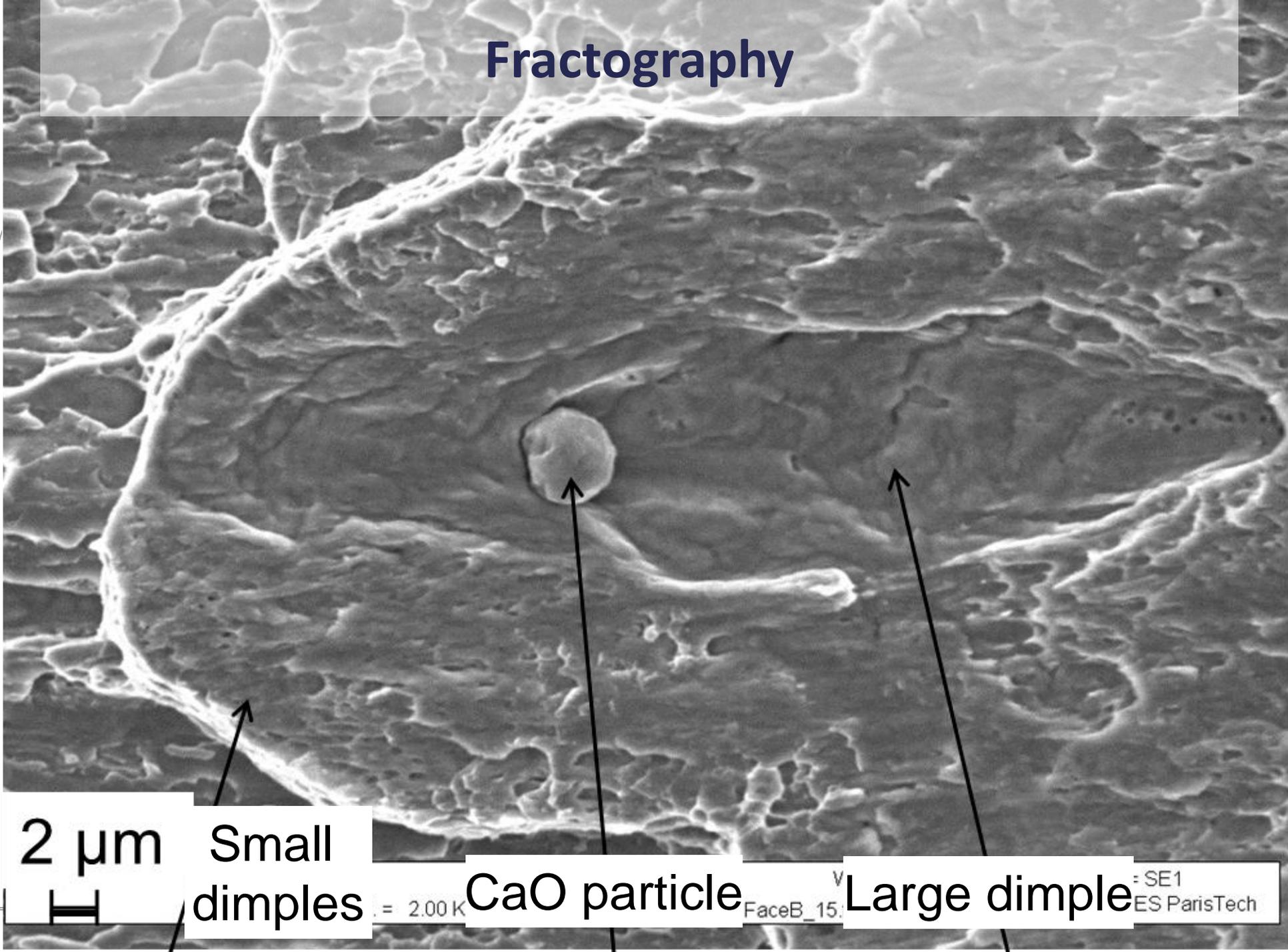
Void growth along direction of max. principal stress



Nucleation of microcracks



Fractography



2 μm



Small dimples

CaO particle

Large dimple

= 2.00 kV

FaceB_15

= SE1
ES ParisTech

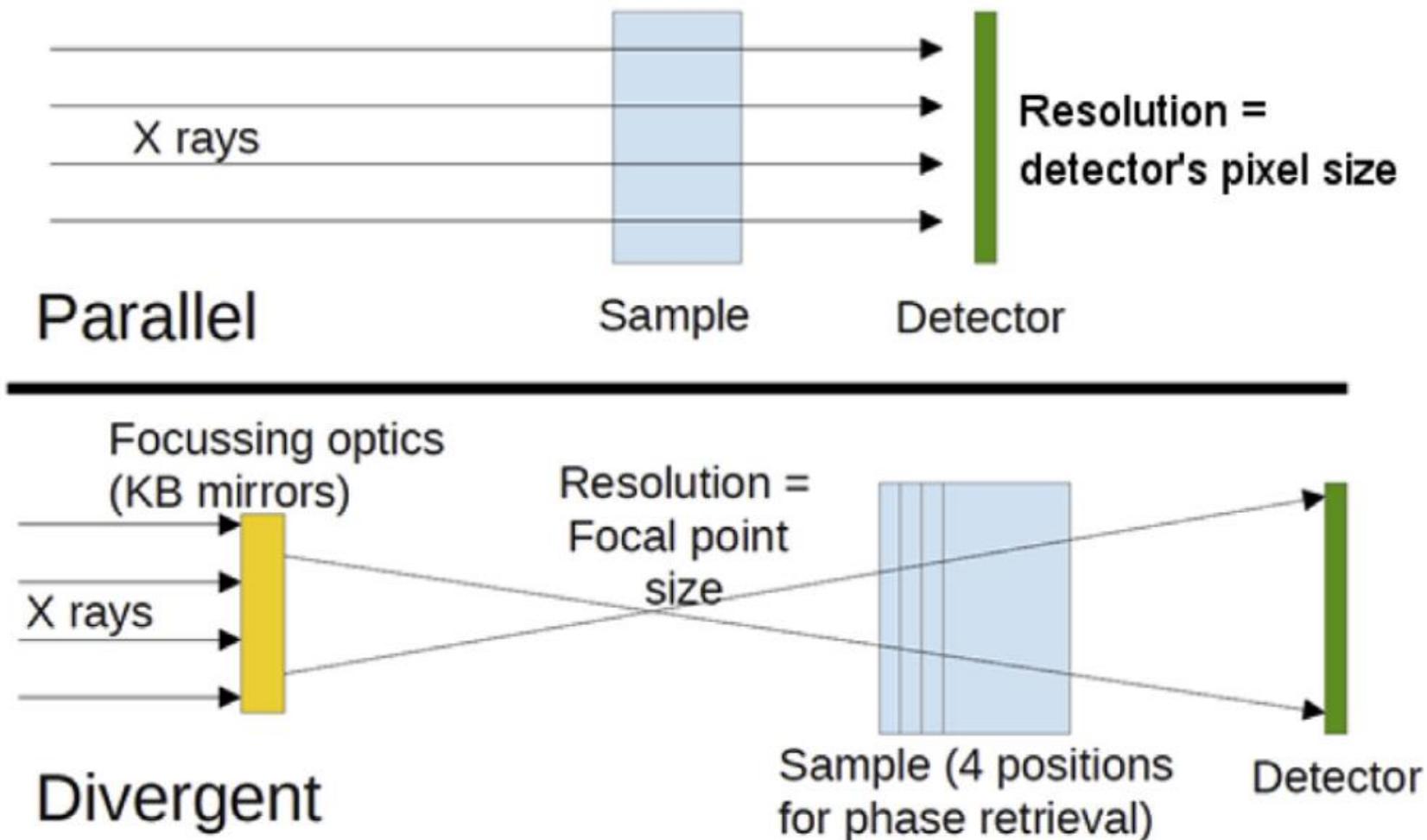
Conclusions on FB600 shear

- Voids nucleate and grow around particles even under shear-dominated loading conditions
- Elongated voids grow along the direction of maximum principal stress (simulation)
- Localization bands form along the direction of maximum shear (simulation)
- Shear localization possible at low void volume fractions (less than 0.02%)

Increasing resolution

Nucleation/Coalescence observed by enhanced resolution

○ ESRF ID22, ID16: Magnified Holotomography (P. Cloetens)



50 nm voxels

[Morgeneyer et al. Polymer 2014]

Outlook: cooperations welcome!

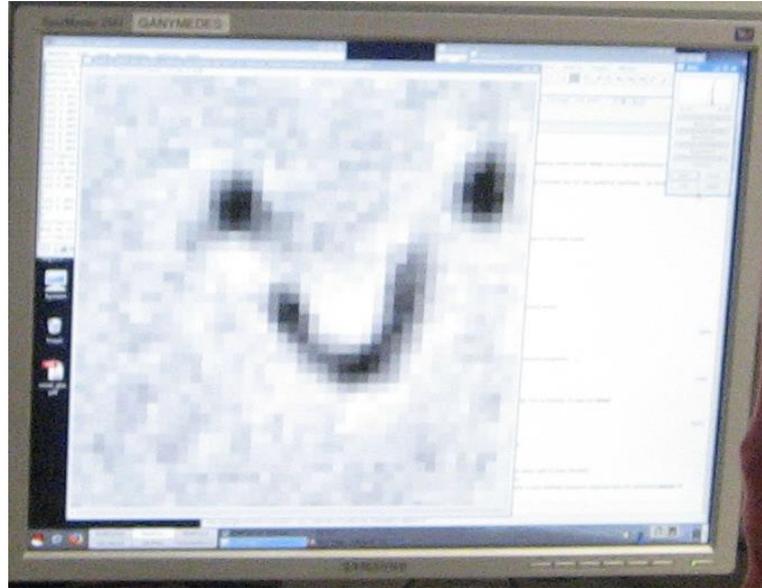
- Mechatronic *in situ* machine
 - 10 kN capacity with load cell
 - Tension-compression
 - Various stress states
 - Leading to fast scanning (1h per sample)

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- A. Buljac, F. Hild, L. Helfen, T.F. Morgeneyer, "On deformation and damage micromechanisms in strong work hardening 2198 T3 aluminium alloy", *Acta Materialia*, 149 (2018) 29-45
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- S. Ren, T. F. Morgeneyer, M. Maziere, S. Forest and G. Rousselier "Portevin-Le Chatelier effect triggered by complex loading paths in an Al-Cu aluminium alloy" *Philosophical Magazine*, 2019, accepted for publication

Questions?

Mg₂Si particle in AA6061 observed by synchrotron laminography:



Acknowledgements



'COMINSIDE' project



Beamtime ma1006,
mi1149, me1366